

# Density Functional Description of $\text{CoSi}_2$ on Si (100)

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# Outline

- Schrödinger's equation and Density Functional Theory
- Approximations!
- Periodic Boundary Conditions
- How an STM works
- Results from Experiment
- Surface Energies and Chemical Potentials
- The  $\text{CoSi}_2$  Surface Reconstructions
- Simulated STM images
- Co on Si (100)
- Conclusions

# Density Functional Theory I

- What we are trying to solve is the Time Independent Schrödinger's equation:

$$-\frac{\hbar^2}{2m}\nabla^2\psi(\mathbf{r}) + V(\mathbf{r})\psi(\mathbf{r}) = E\psi(\mathbf{r})$$

- This is a many-body equation and is **VERY** expensive to solve exactly, e.g. CI, QMC. These methods can typically treat only a few atoms.
- For **any** materials problem of interest we need to employ another approach.

# Density Functional Theory II

- The Density functional theory of Hohenberg, Kohn and Sham<sup>1</sup> states:

*The total ground state energy of a system is a unique functional of the electron density,  $\rho$ . The minimum energy of this functional is the ground-state energy of the system and the density that yields this is the exact single particle density.*

- With this statement they replaced the many-electron problem with an equivalent set of self-consistent one-electron equations, the Kohn-Sham equations.

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<sup>1</sup>P. Hohenberg and W. Kohn, *Physical Review* **136B**, 864-871 (1964);  
W. Kohn and L.J. Sham, *Physical Review* **140A**, 1133-1138 (1965)



# Self-consistent Kohn-Sham Equations I

- The Kohn-Sham energy functional is:

$$E = 2 \sum_{\text{occ}} \varepsilon_i + E_{\text{ion-ion}} - \frac{e^2}{2} \int \frac{\rho(\mathbf{r})\rho(\mathbf{r}')}{|\mathbf{r} - \mathbf{r}'|} d\mathbf{r}d\mathbf{r}' + \int \rho(\mathbf{r})(E_{xc} - V_{xc})d\mathbf{r}$$

- The Kohn-Sham equations are:

$$\varepsilon_i \psi_i(\mathbf{r}) = \left[ -\frac{\hbar^2}{2m} \nabla^2 + V_{\text{ion}}(\mathbf{r}) + V_H(\mathbf{r}) + V_{xc}(\mathbf{r}) \right] \psi_i(\mathbf{r})$$

$$\rho(\mathbf{r}) = 2 \sum_i |\psi_i(\mathbf{r})|^2$$

$$V_H(\mathbf{r}) = e^2 \int \frac{\rho(\mathbf{r}')}{|\mathbf{r} - \mathbf{r}'|} d\mathbf{r}'$$

$$V_{xc}(\mathbf{r}) = \frac{\delta E_{xc}[\rho(\mathbf{r})]}{\delta \rho(\mathbf{r})}$$

- These Kohn-Sham equations represent the mapping of the interacting many-body equation onto a system of non-interacting electrons moving in an effective potential due to all the other electrons.

# Self-consistent Kohn-Sham Equations II

- The general scheme for solving the equations is:
  1. Make an initial guess at the charge density; typically a superposition of atomic charge densities.
  2. Construct the potentials with this.
  3. Solve for the eigenvalues and eigenvectors,  $\psi_i$ 's.
  4. Construct a new density from these.
  5. Mix the old and new densities together to form a new input density.
  6. Repeat from step 2 until the input and output densities agree within some criteria.

# Approximations:-

## Exchange-Correlation Functional

- The only thing we do not know is what the exact exchange-correlation functional is.
- If we knew the exact exchange-correlation functional then we could solve the problem exactly. **Density functional theory is not an approximation.**
- We do not, however, know what the functional is so we must make a **guess** for this.
- The guess we make in general is the so called *local-density approximation*:

The exchange-correlation per electron at a point in space is the same as for a homogeneous electron gas of the same electron density.

$$E_{xc}[\rho(\mathbf{r})] = \int \varepsilon_{xc}(\mathbf{r})\rho(\mathbf{r})d\mathbf{r}$$

- This is still the most widely used functional although a lot of calculations now use the generalised gradient approximation<sup>2</sup>.

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<sup>2</sup>A.D. Becke, *Physical Review A* **38**, 3098 (1988)

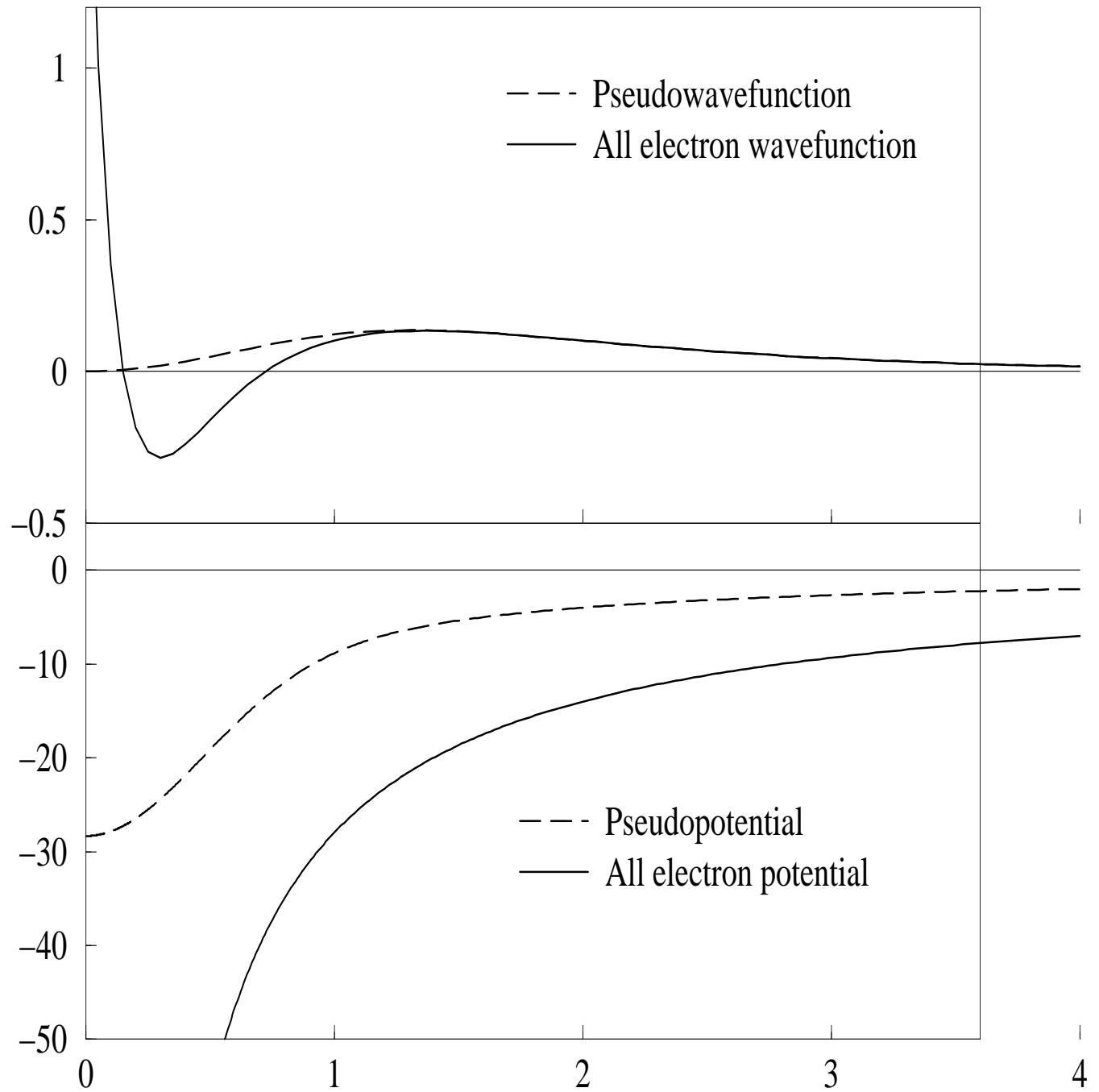
# Approximations:- Pseudopotentials I

- When atoms bond together it is the valence (outer) electrons that take part in the bonding.
- What pseudopotentials do is replace the full potential (due to the nucleus) with one due to a combination of the nucleus and the core electrons.
- So pseudopotentials are an effective potential constructed so as to mimic the potential a valence electron feels due to the combination of the ion core and the core electrons <sup>3</sup>.

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<sup>3</sup>M.L. Cohen and V. Heine, *Solid state physics* **24**, 37 (1970)

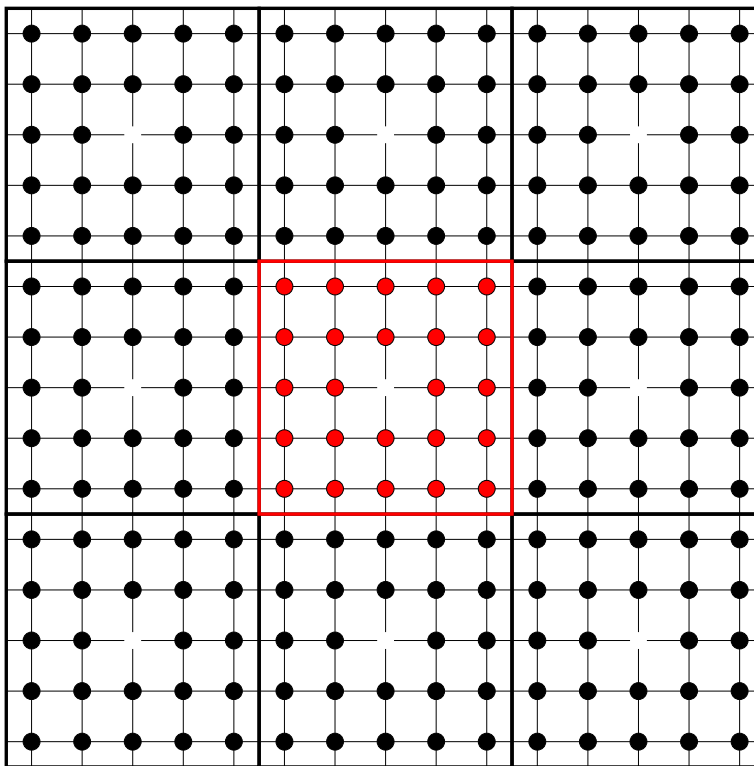
# Approximations:- Pseudopotentials II



# Periodic Boundary Conditions I

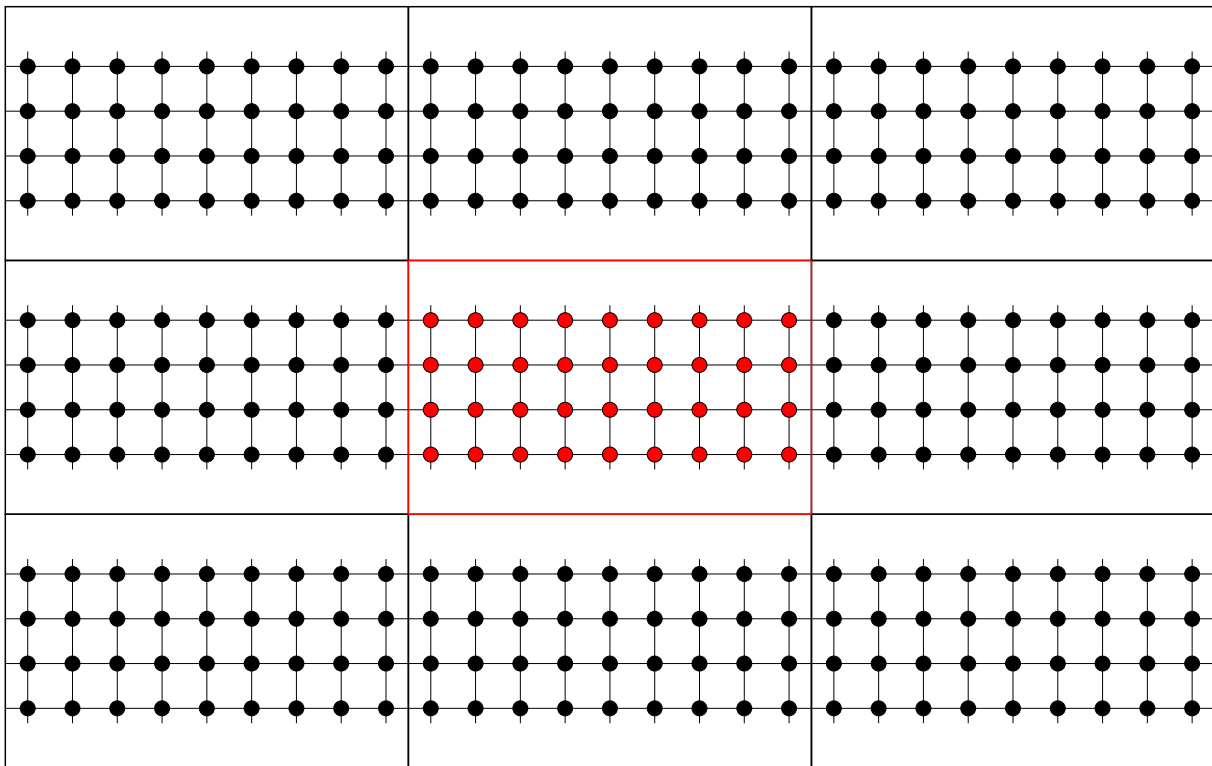
- In order to perform calculations of interest we need to perform calculations on systems which are not periodic; such as surfaces and vacancies.
- The methods we typically use can only deal with periodic systems.
- What we do is create a periodic cell from the nonperiodic system.

Vacancy:



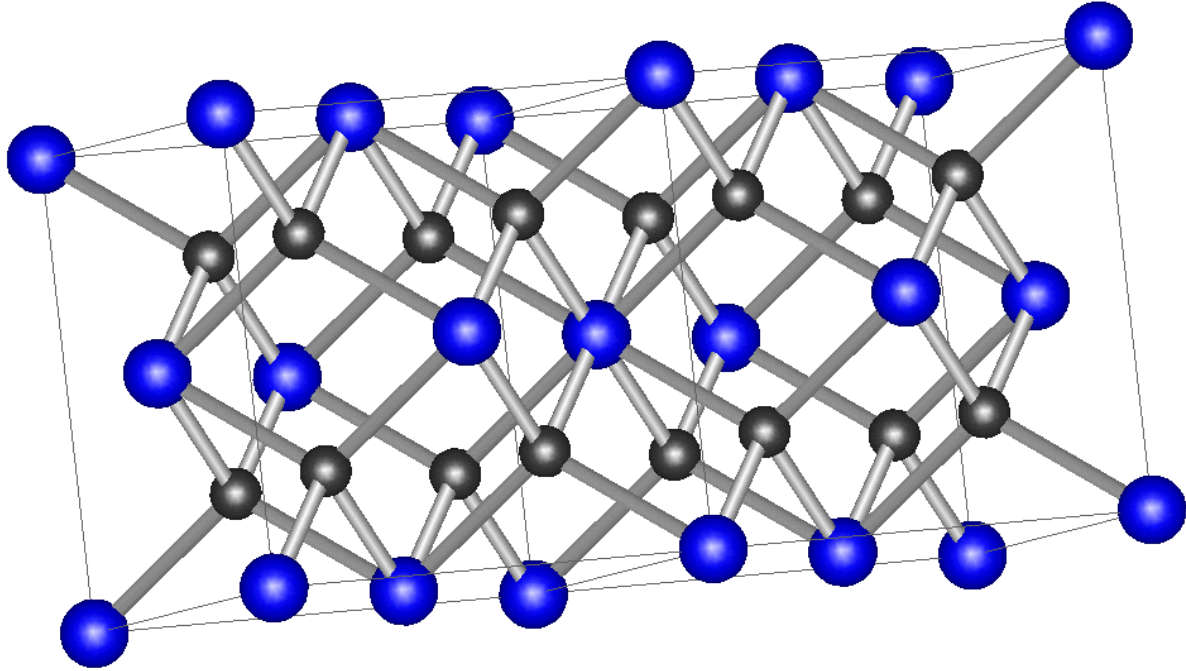
# Periodic Boundary Conditions II

Surface:



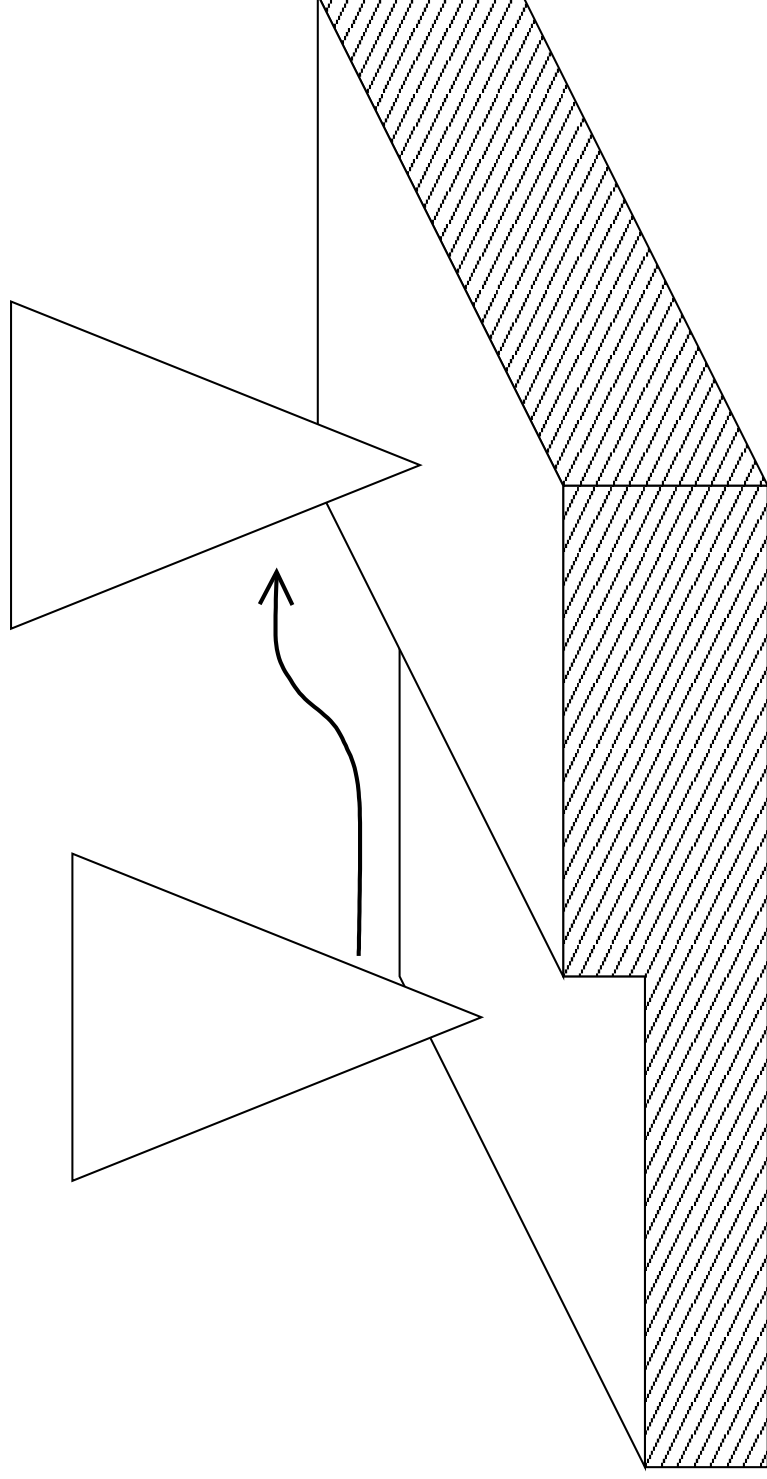
- Have to be careful that cell is large enough that there is little interaction between periodic images of the vacancy or surface.

# CoSi<sub>2</sub> Structure

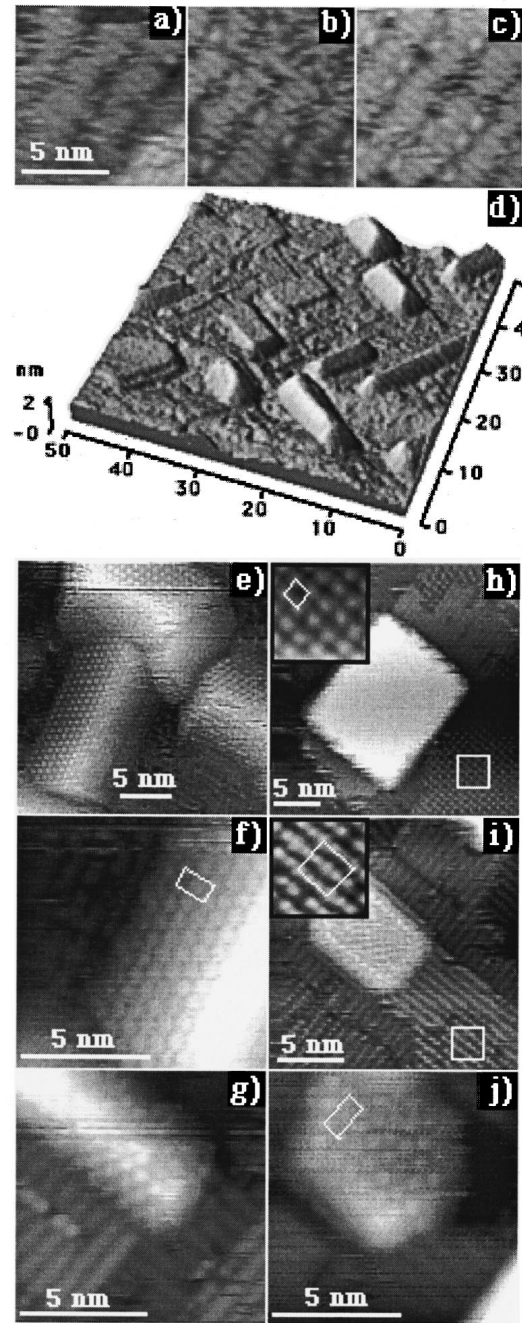




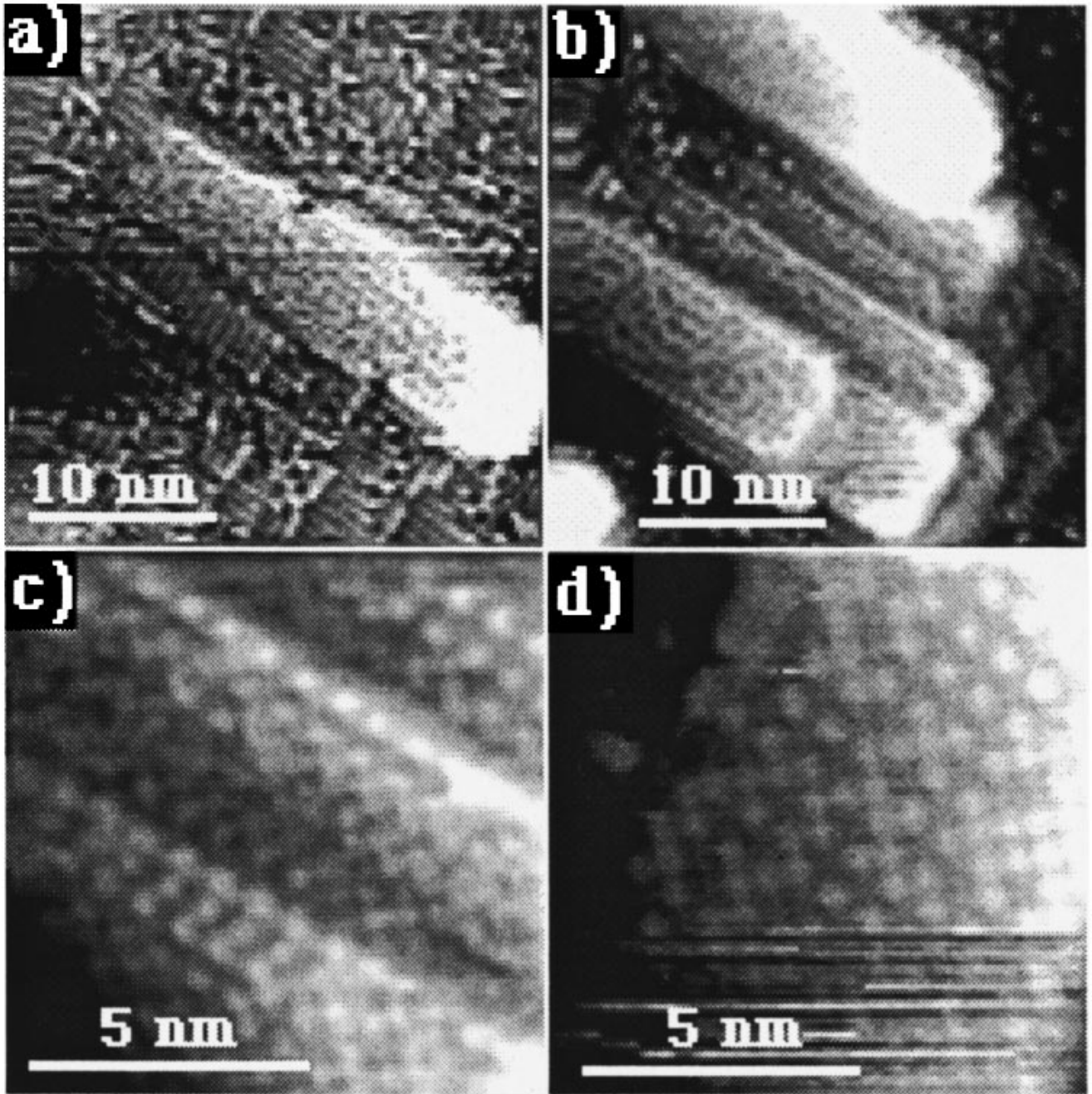
# Principal of an STM



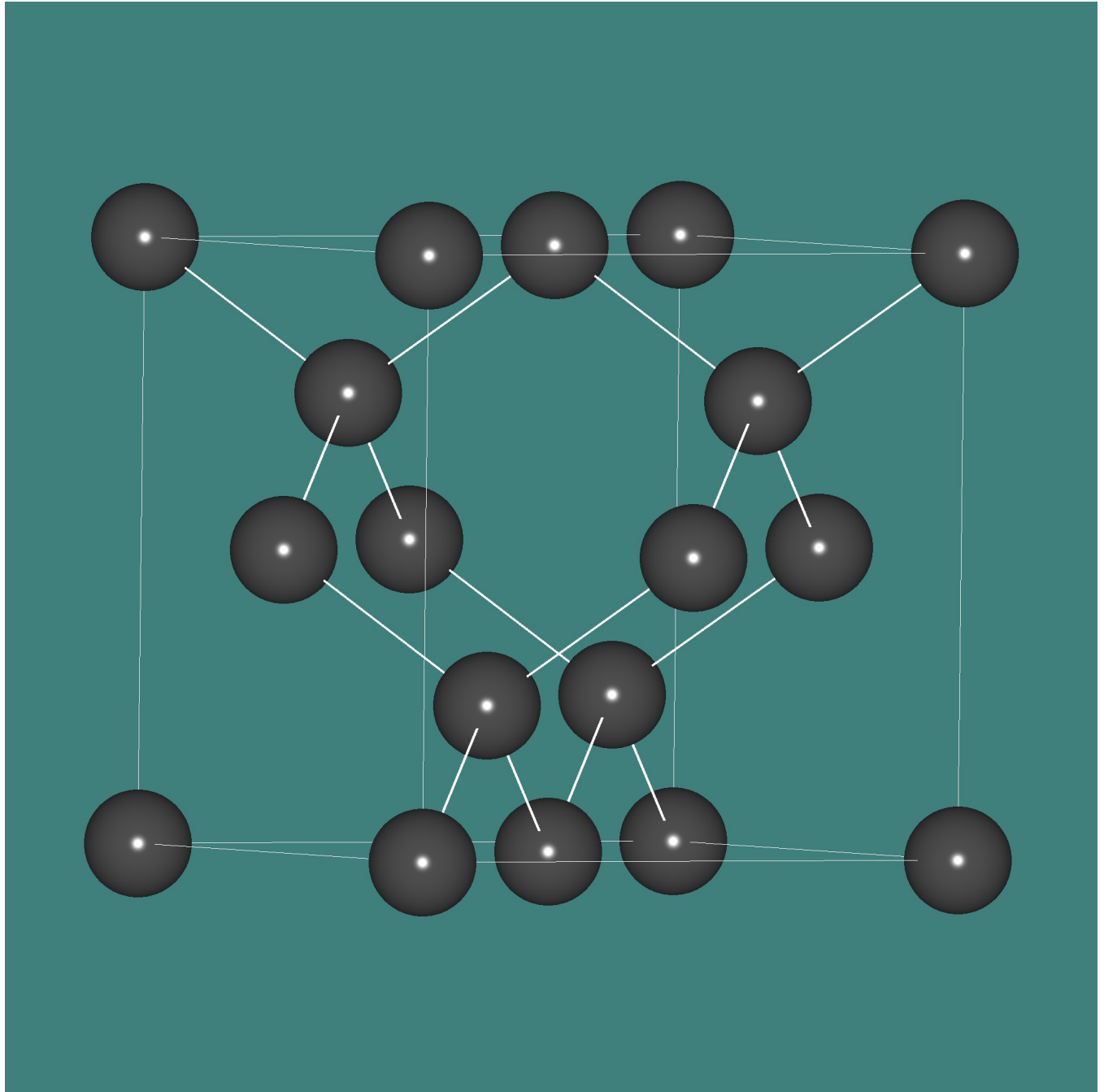
# STM image of $\text{CoSi}_2$ on Si (100)



# STM image of (100) $\text{CoSi}_2$ on Si (100)

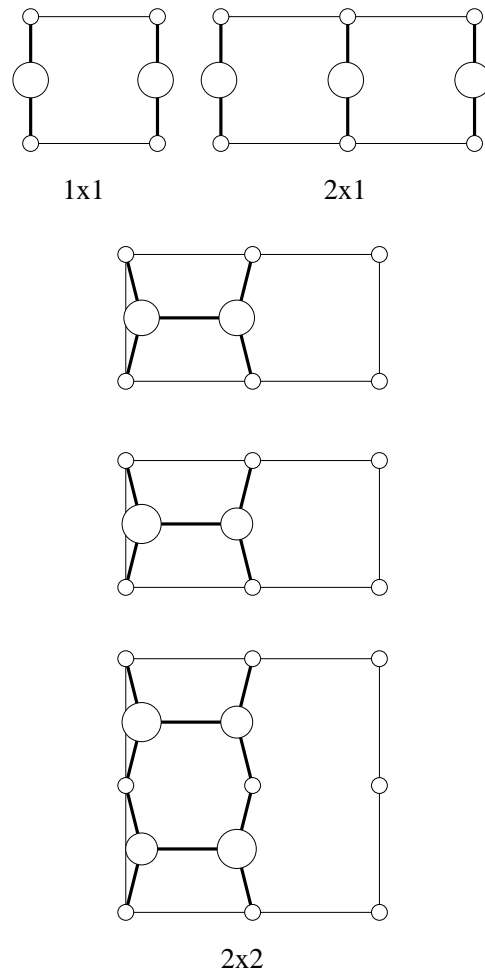


# Surface Reconstructions



# Surface Reconstructions

- Most simple metal surfaces do not reconstruct but lots of semiconductor surfaces do.
- Silicon does forming a dimerised surface thus reducing the number of dangling bonds.



- Reconstructions such as tilted dimers only happen if you break the symmetry.

# Surface Energies

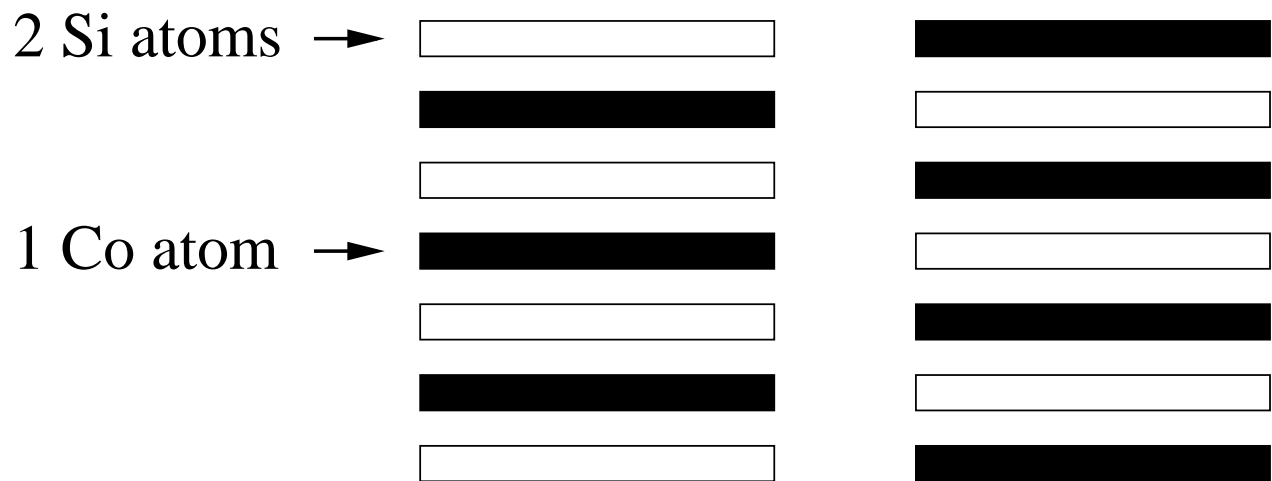
- When surfaces reconstruct they do so to lower their surface energy.
- This might be to try and saturate dangling bonds or it may be more preferable to put one atom species on the surface rather than another.
- Many other factors also influence what the most stable surface reconstruction is.
- An understanding of the energetics of the surface reconstructions is essential for understanding the equilibrium driving forces for a system.

# Chemical Potentials

- For systems containing more than one atomic species we need to address the problem of defining a **chemical potential**.

- Why?

- Surface reconstructions on the  $\text{CoSi}_2(100)$  surface



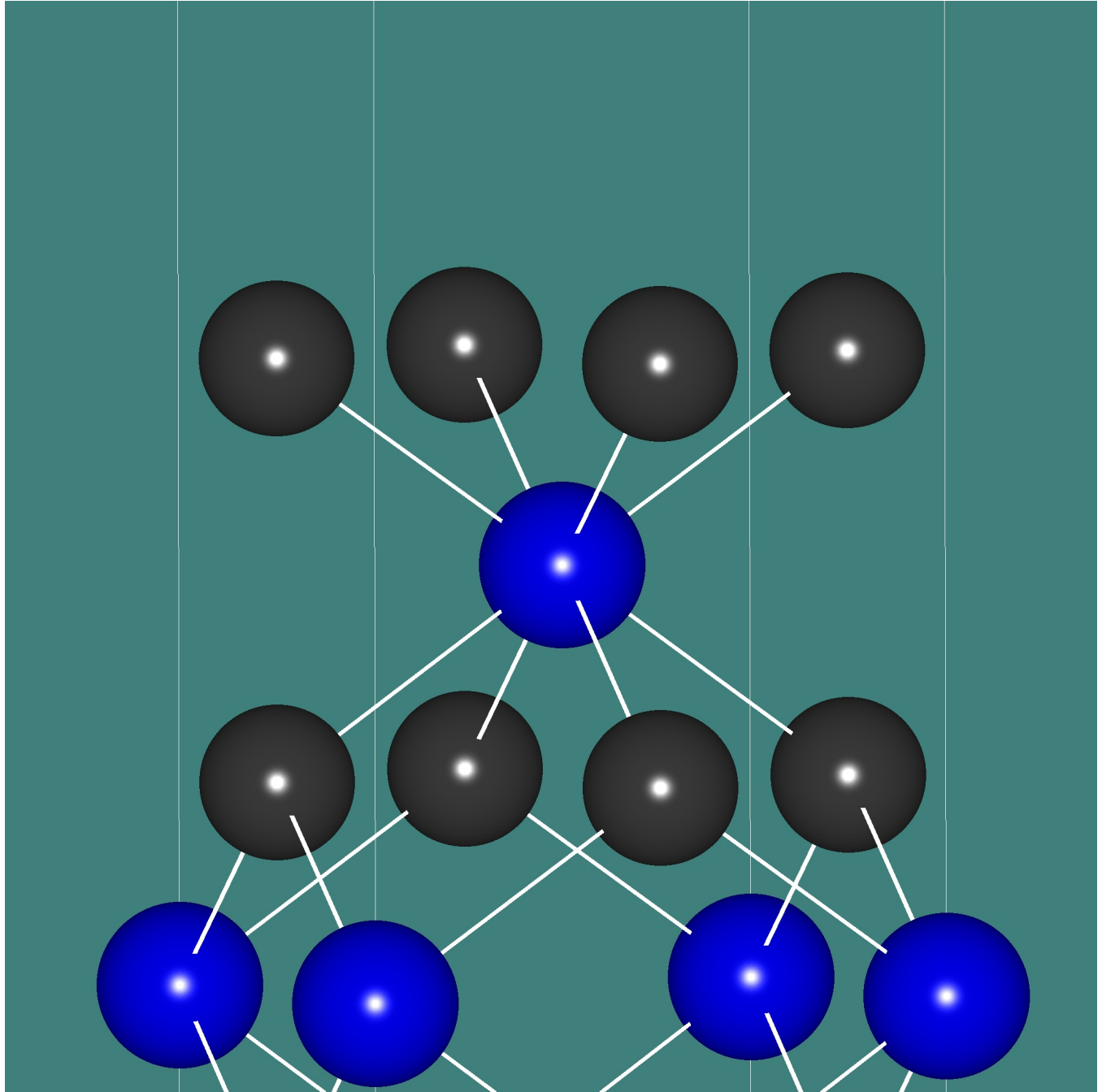
- Cell on the left has 3 Co's and 8 Si's the one on the right 4 Co's and 6 Si's; neither is stoichiometric.

- If we want to calculate the surface energy of either cell we need to define the chemical potential of the Co and Si atoms.

$$\mu_{\text{CoSi}_2} = \mu_{\text{Co}} + 2\mu_{\text{Si}} + H_{\text{CoSi}_2}$$

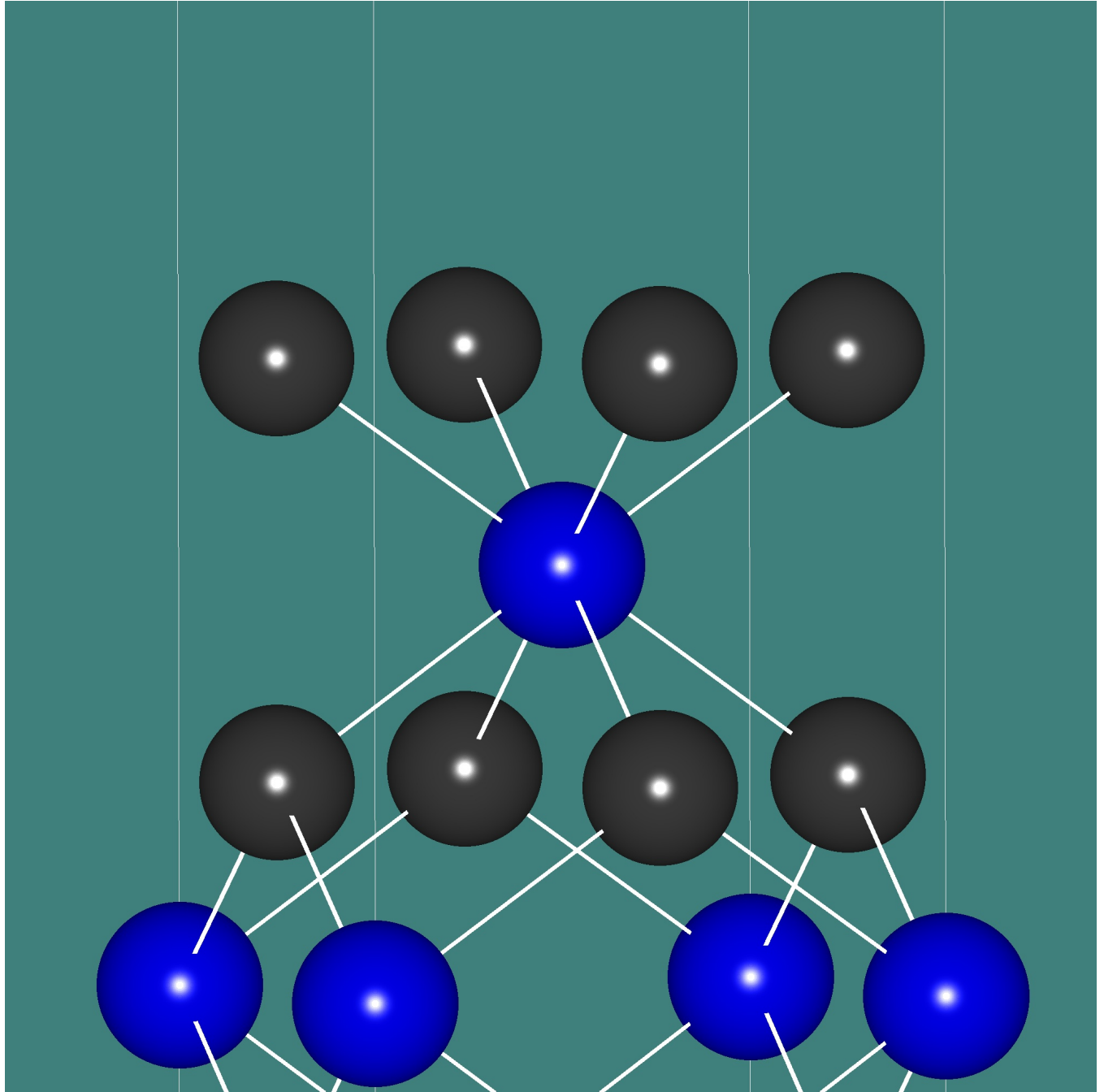


# $p(1 \times 1)$ Si terminated Surface

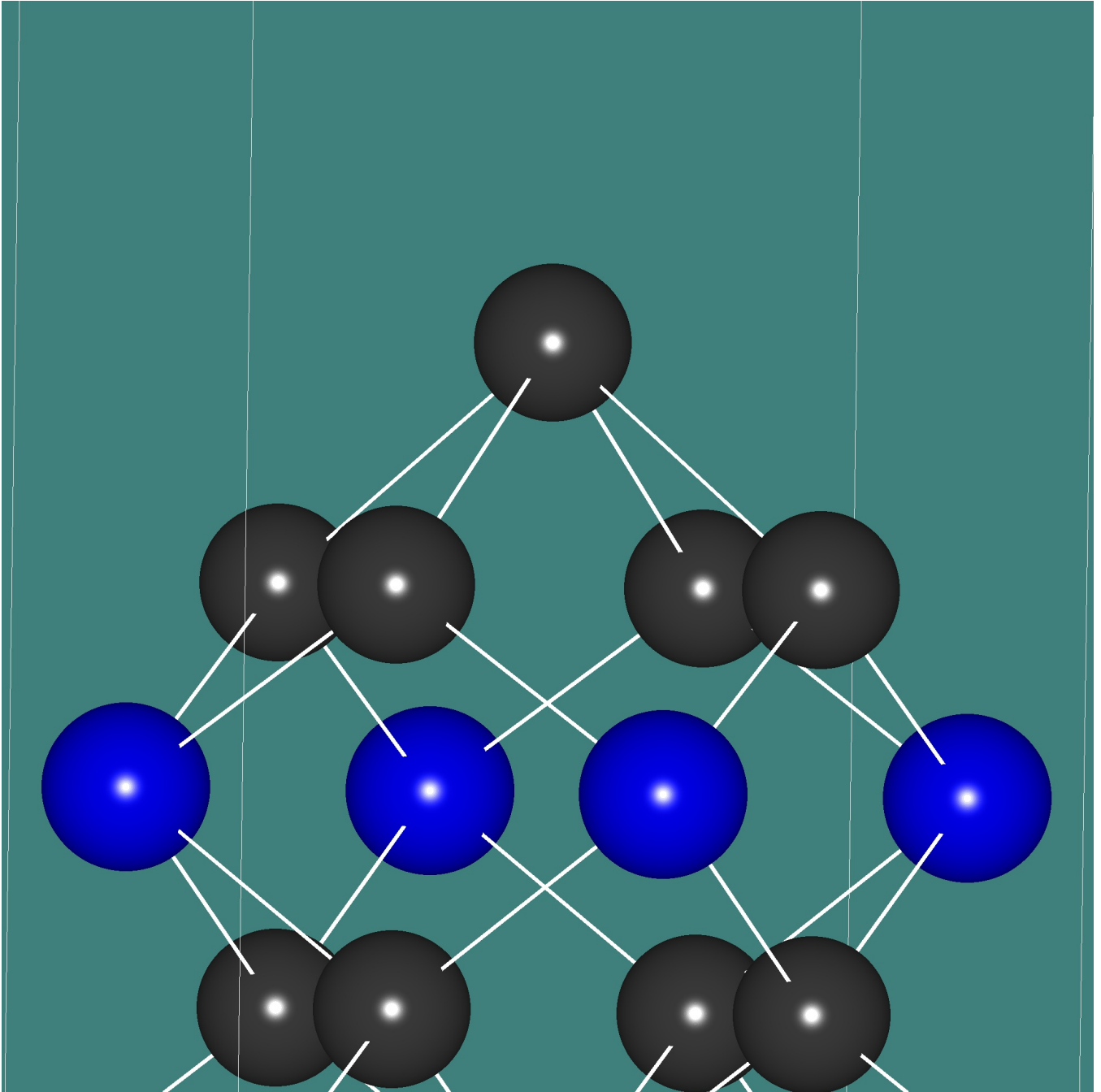




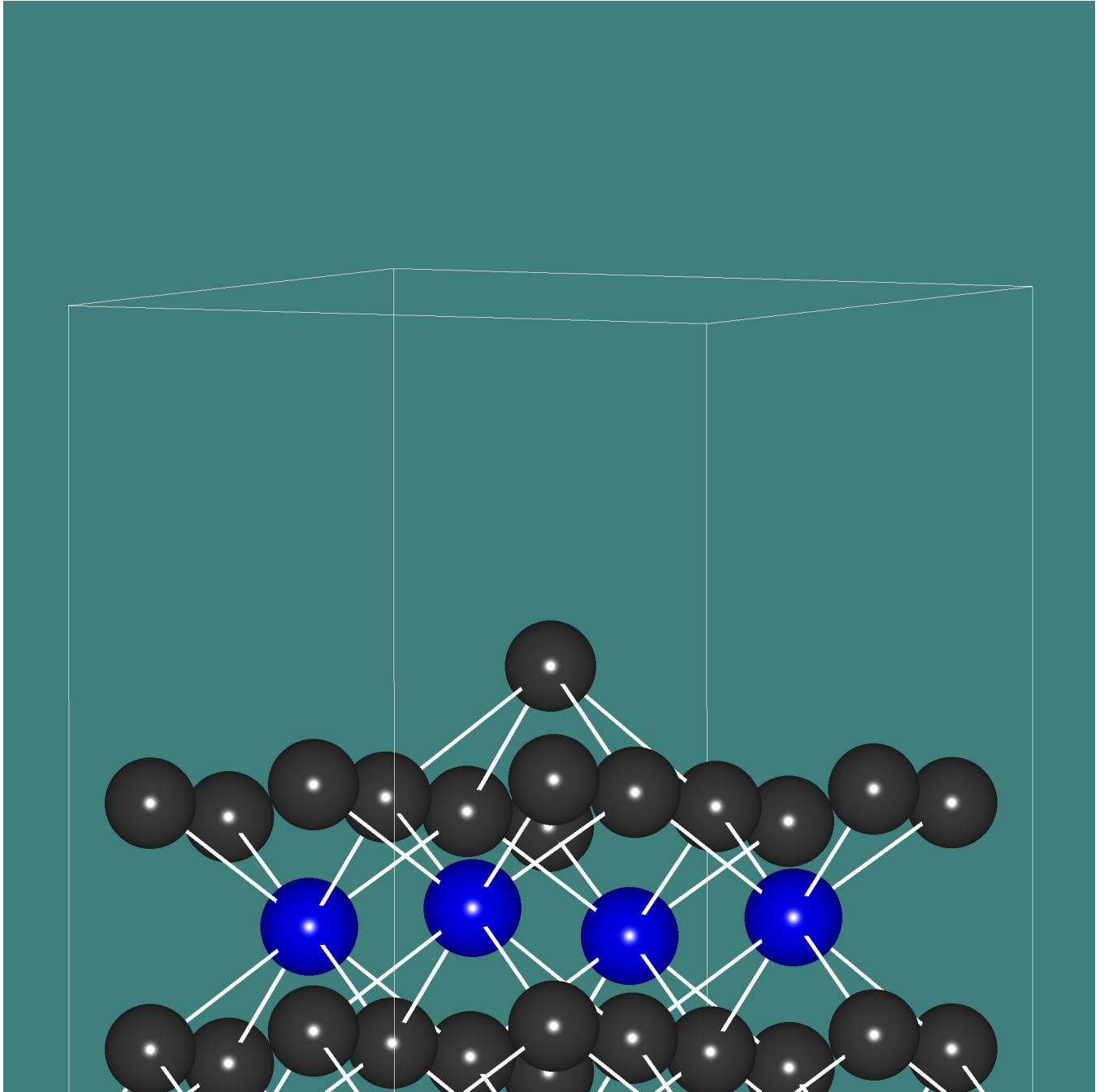
# $p(1 \times 1)$ Co terminated Surface



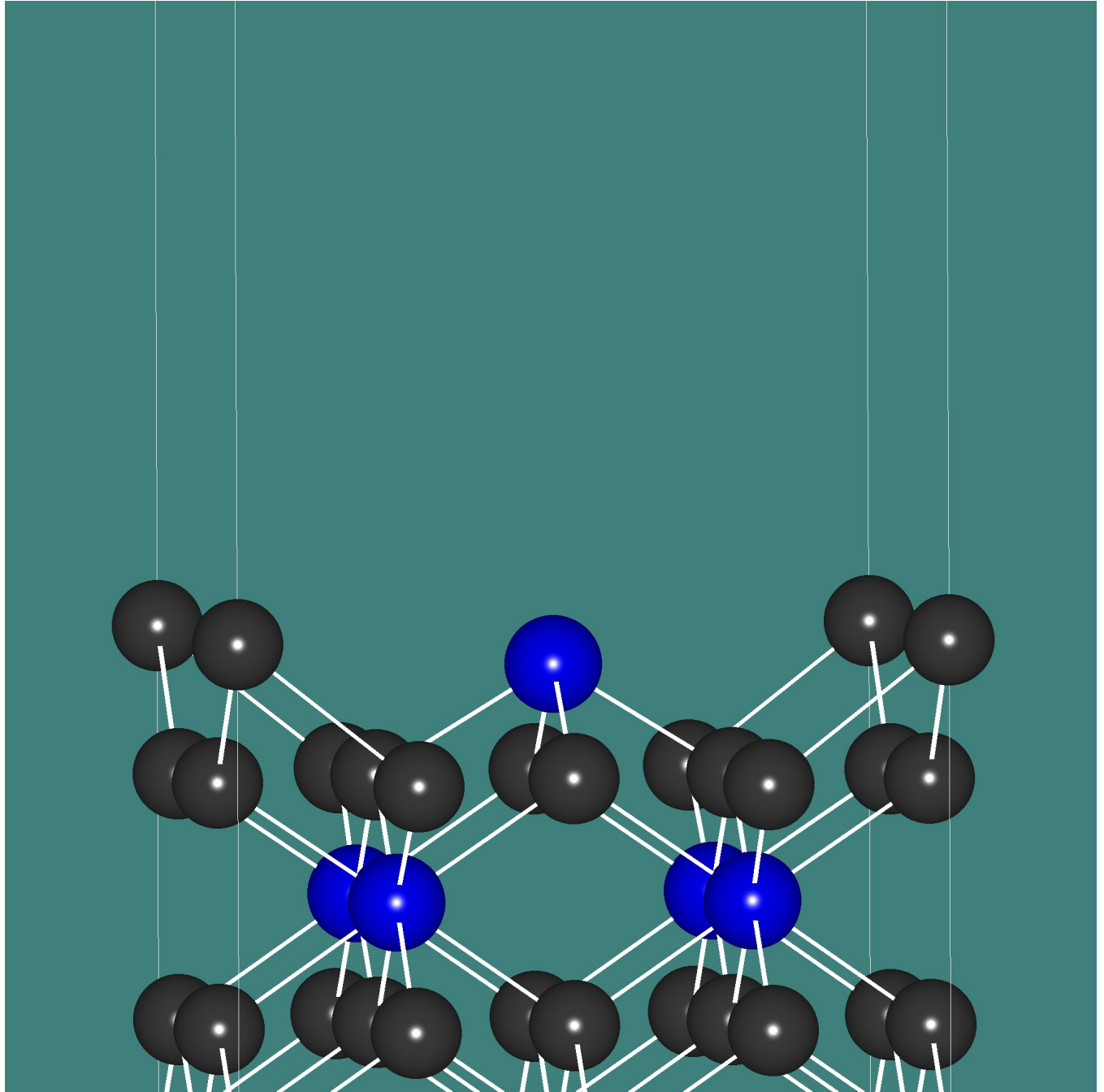
# c(2×2) Si terminated Surface



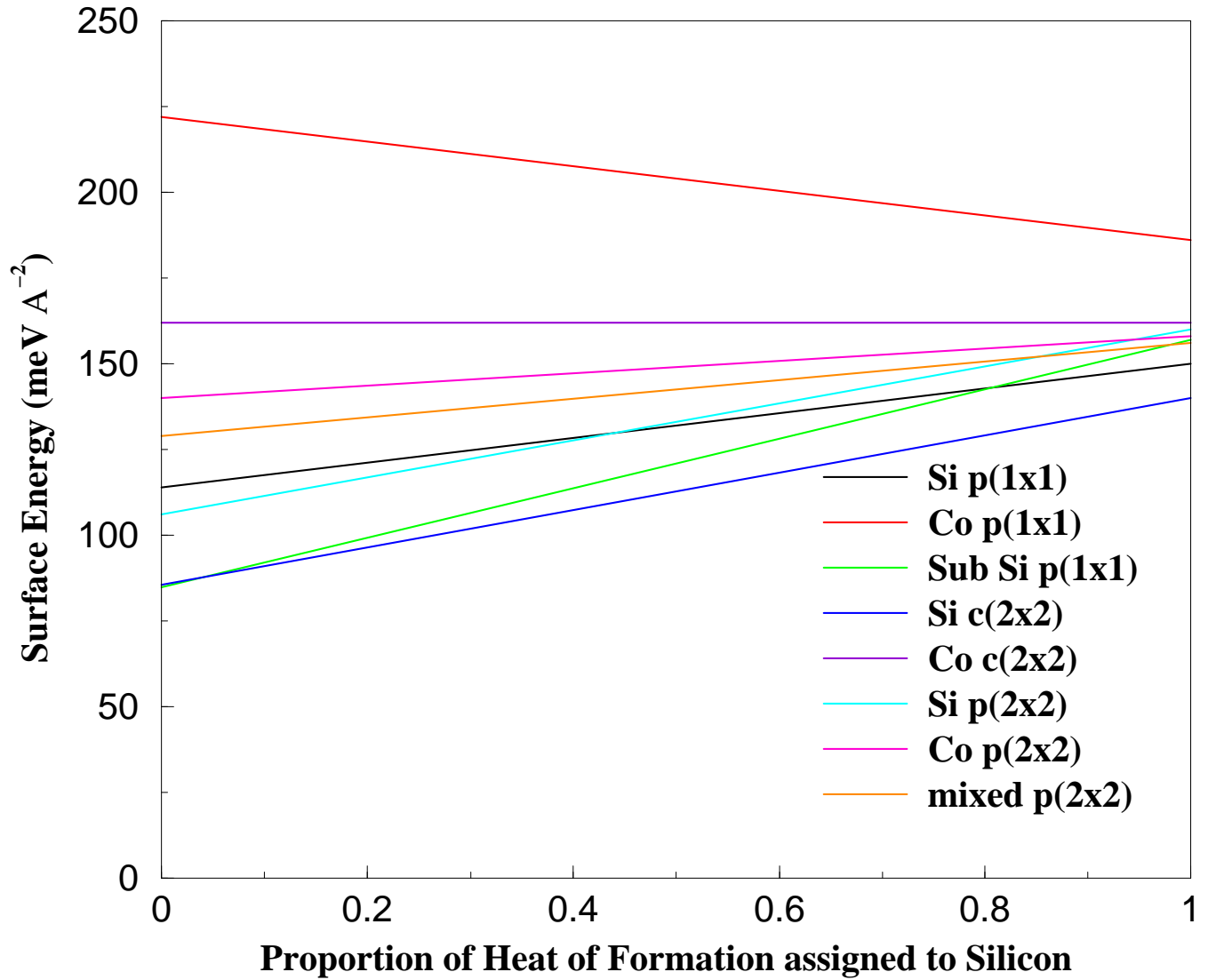
# $p(2 \times 2)$ Si terminated Surface



# $p(2 \times 2)$ Mixed terminated Surface

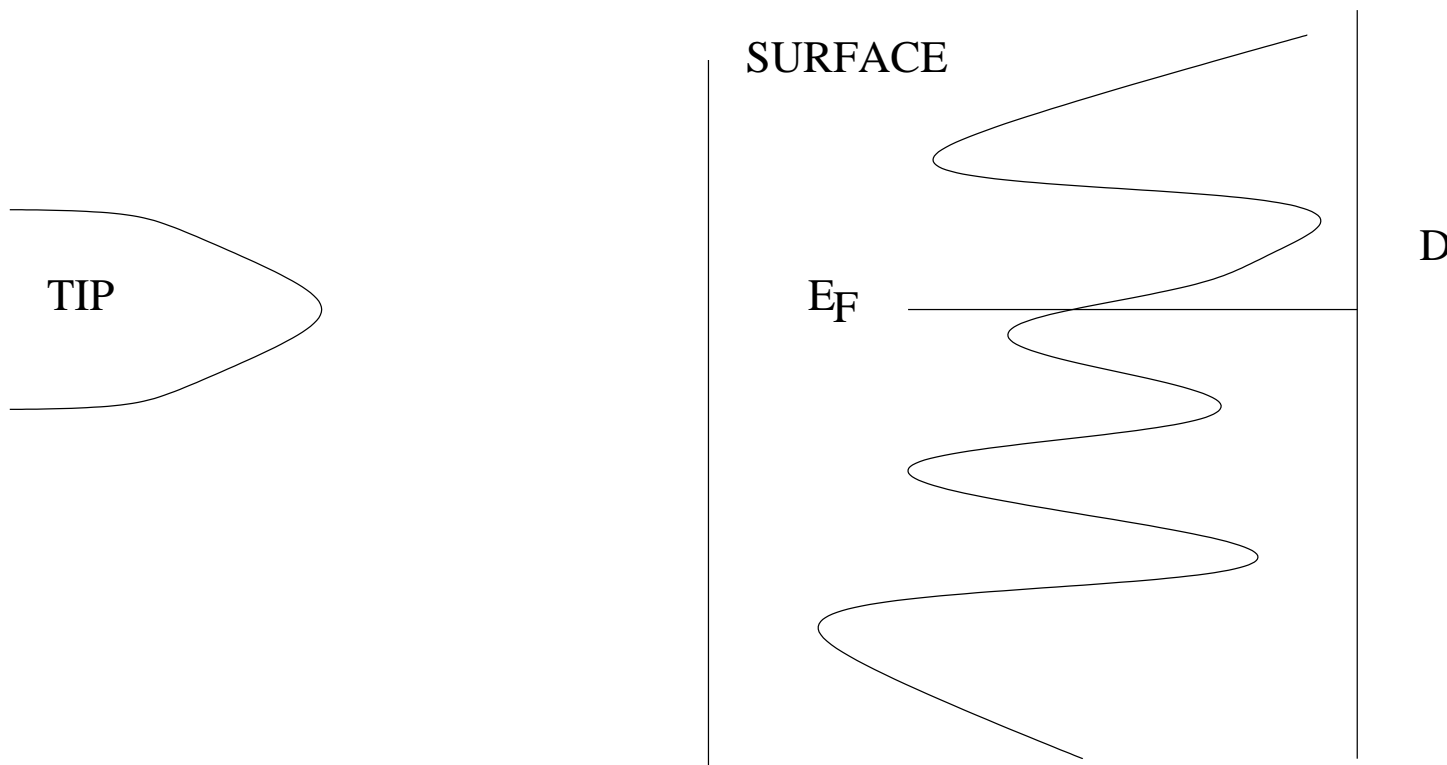


# Phase Diagram for $\text{CoSi}_2$ Surface

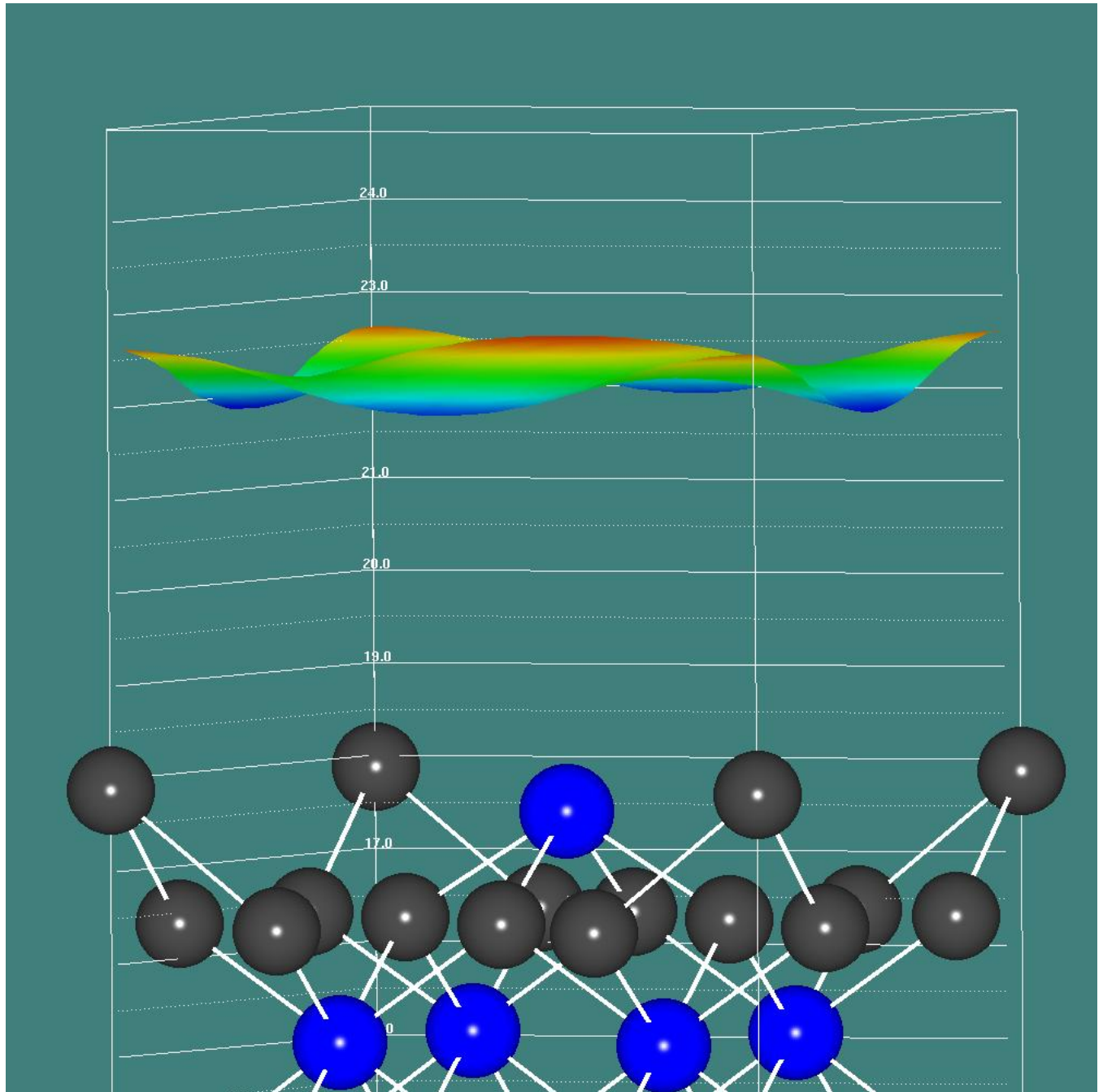


# STM Image Simulation

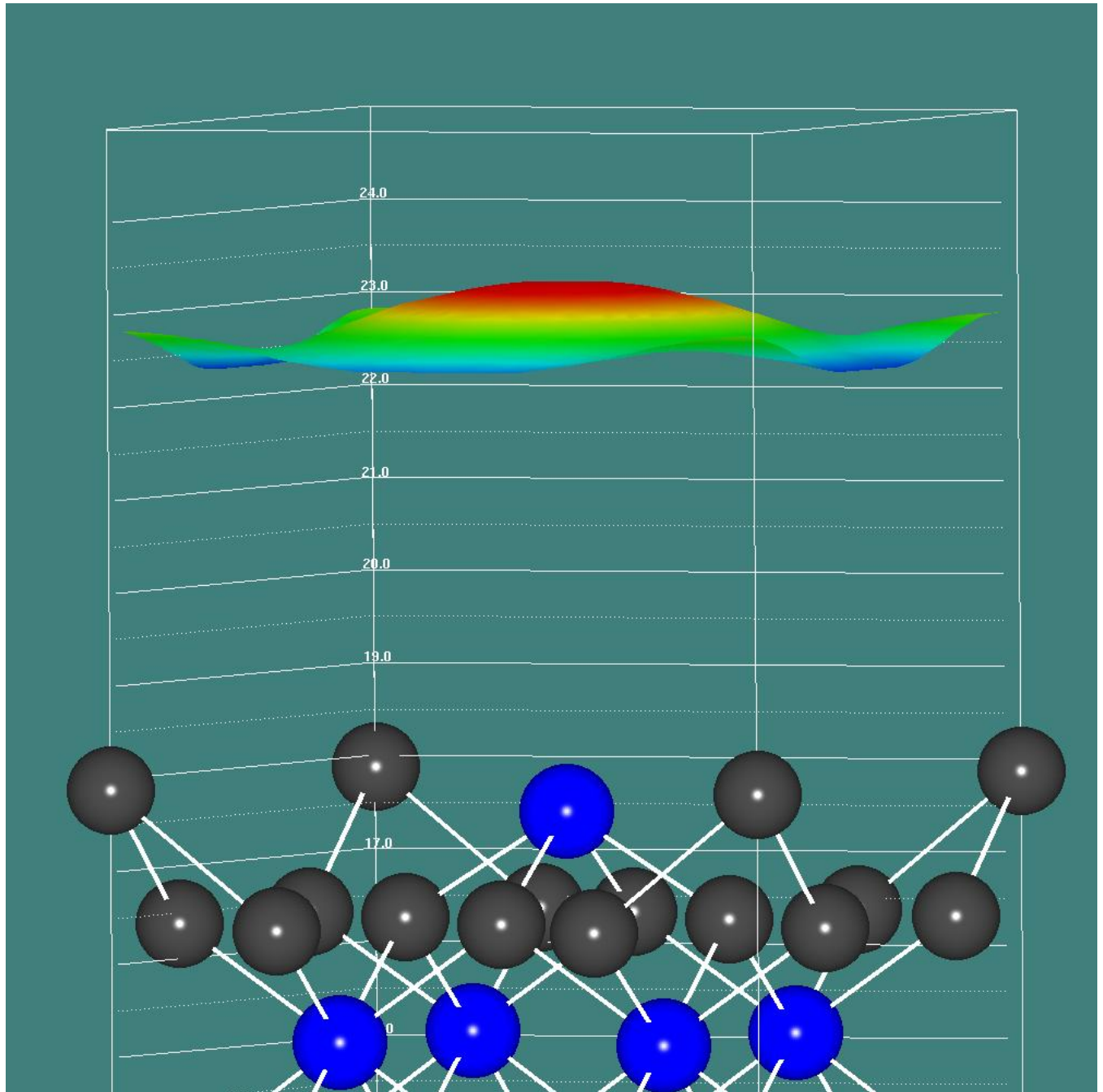
- What an STM tip sees is the **surface electronic structure**, this is **not** always the same as the underlying **physical surface**.
- A simple model can be used to produce simulated STM images from an electronic structure calculation.
- By taking eigenstates within the energy window of the tunnelling a simulated STM image can be produced that gives at least a qualitative description of what is observed.



# Simulated STM Image:- Relaxed $V=0.5$

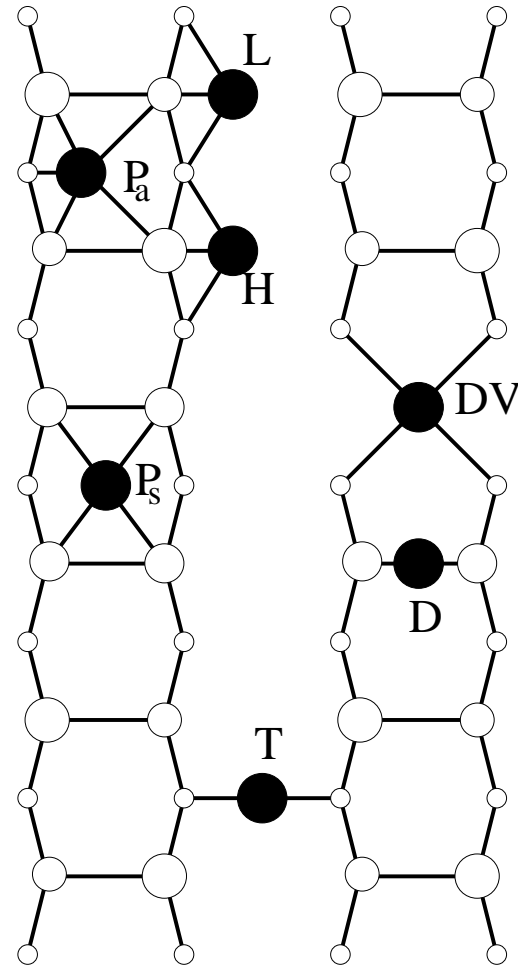


# Simulated STM Image:- Relaxed $V=1$



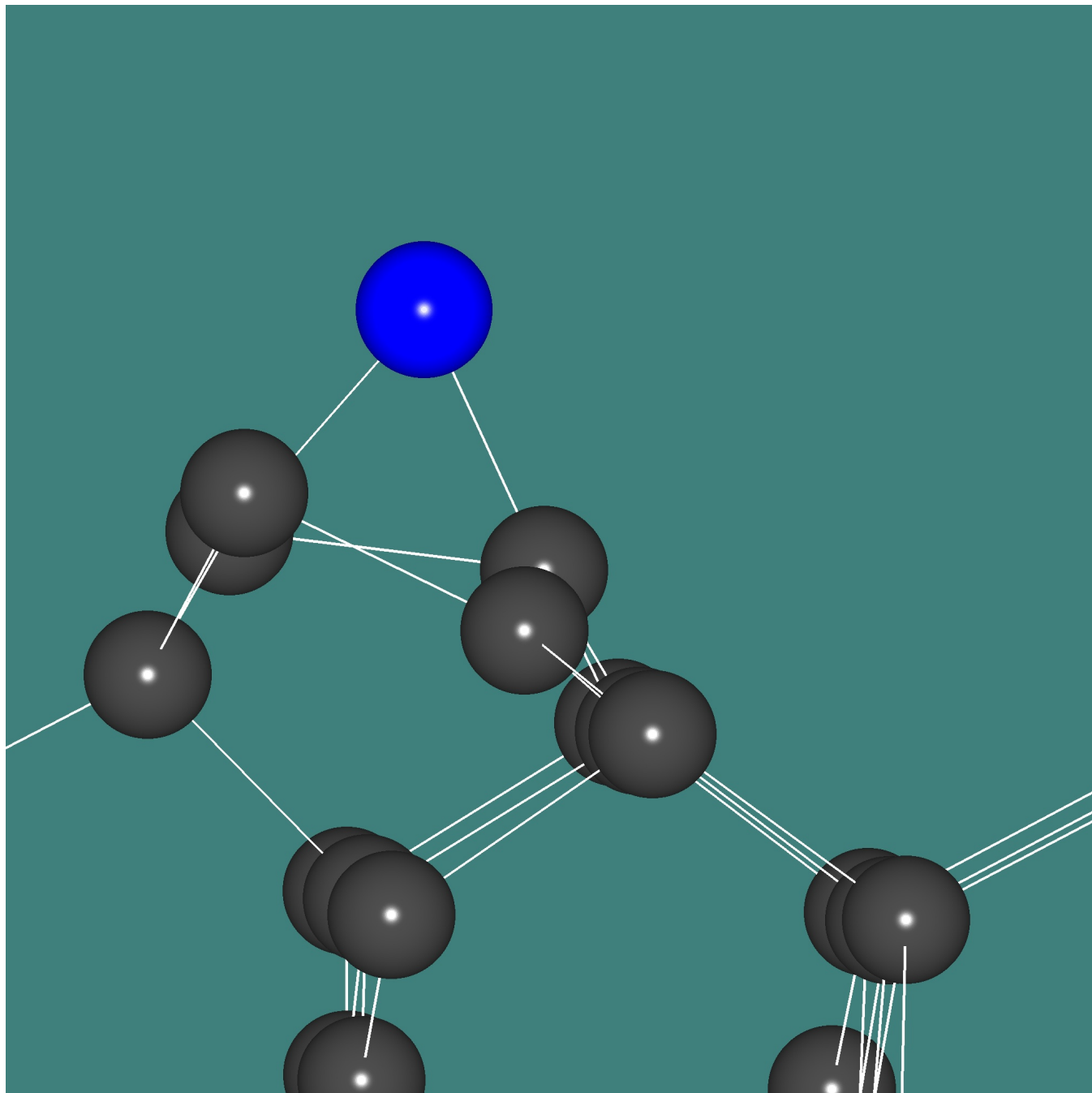


# Co on Si (100):- Initial bonding Site

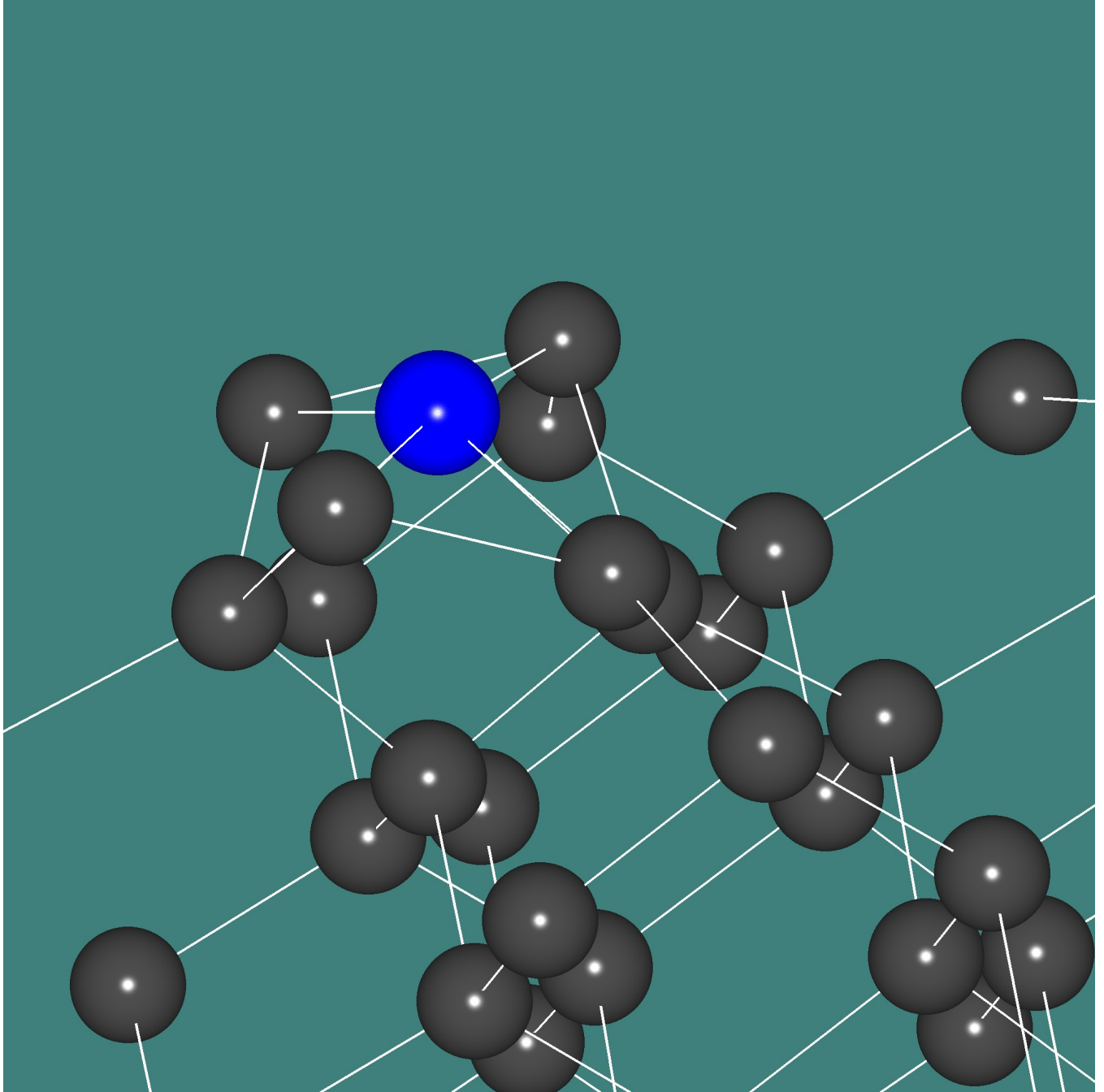


- The picture above shows a number of possible bonding sites for a Co atom on a Si(100) surface.
- To find the real site in which the Co atom sits it is necessary to perform calculations for all the sites to see which is most stable.

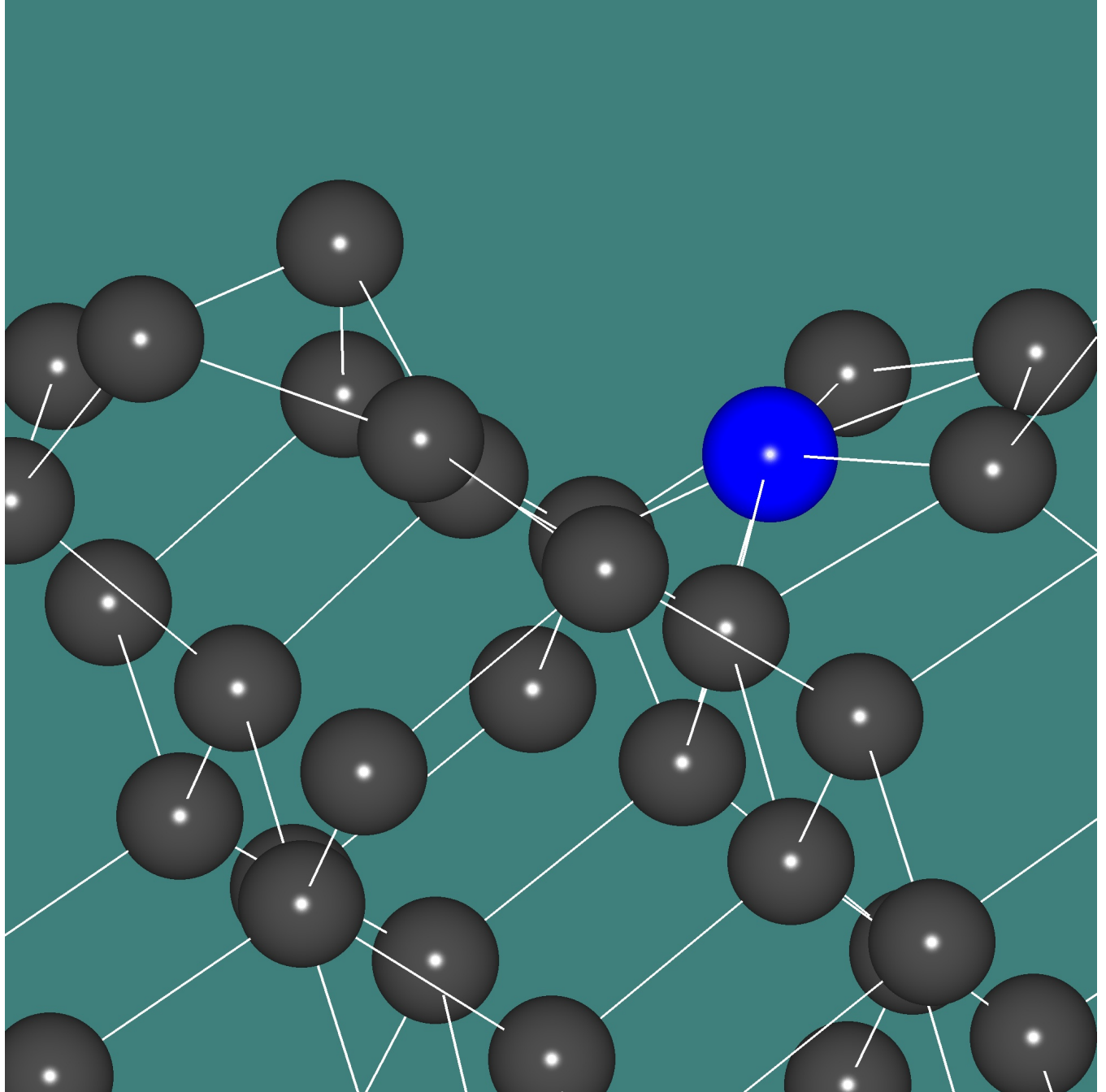
# Dimer Site



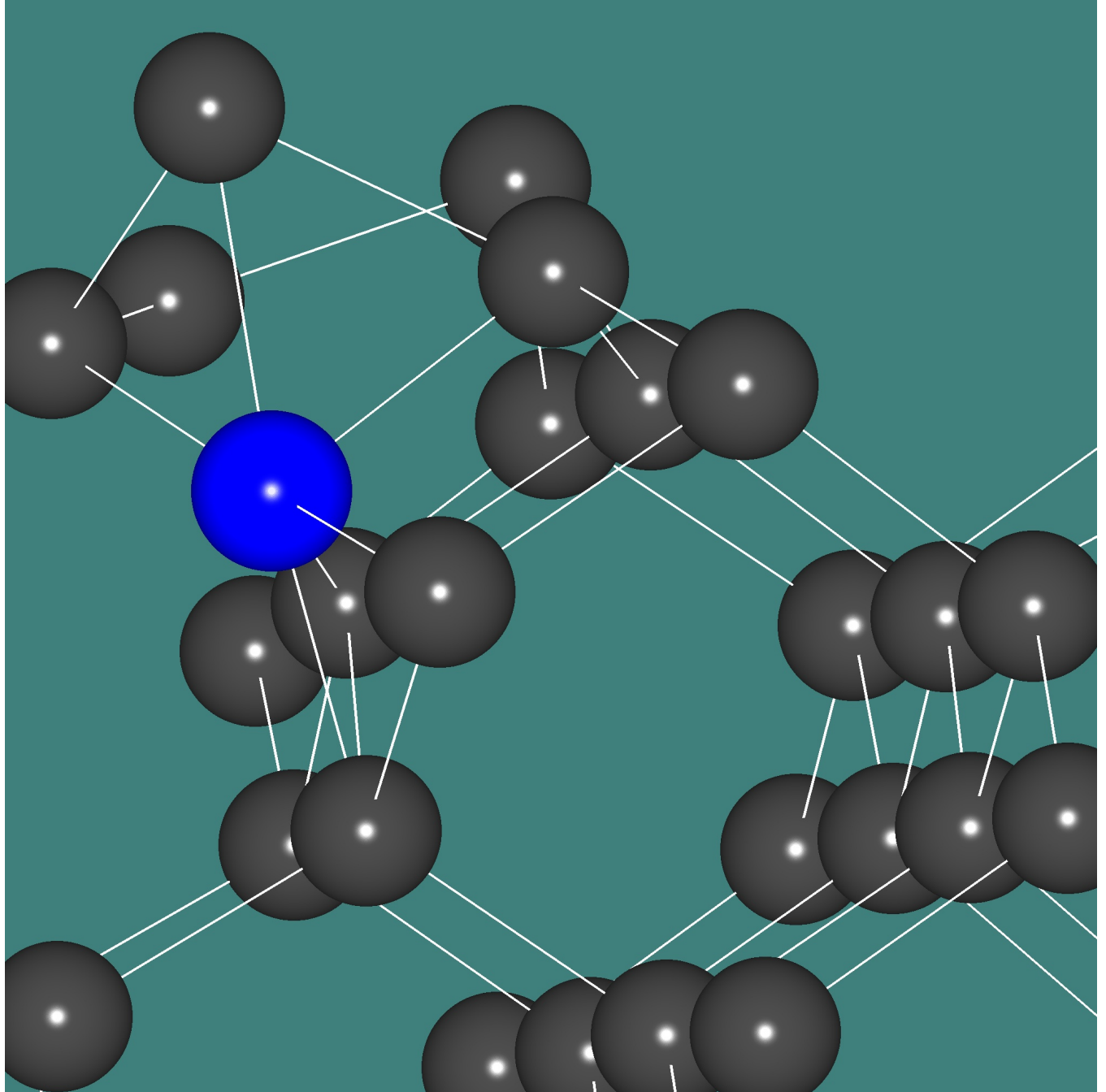
# Pedestal Site



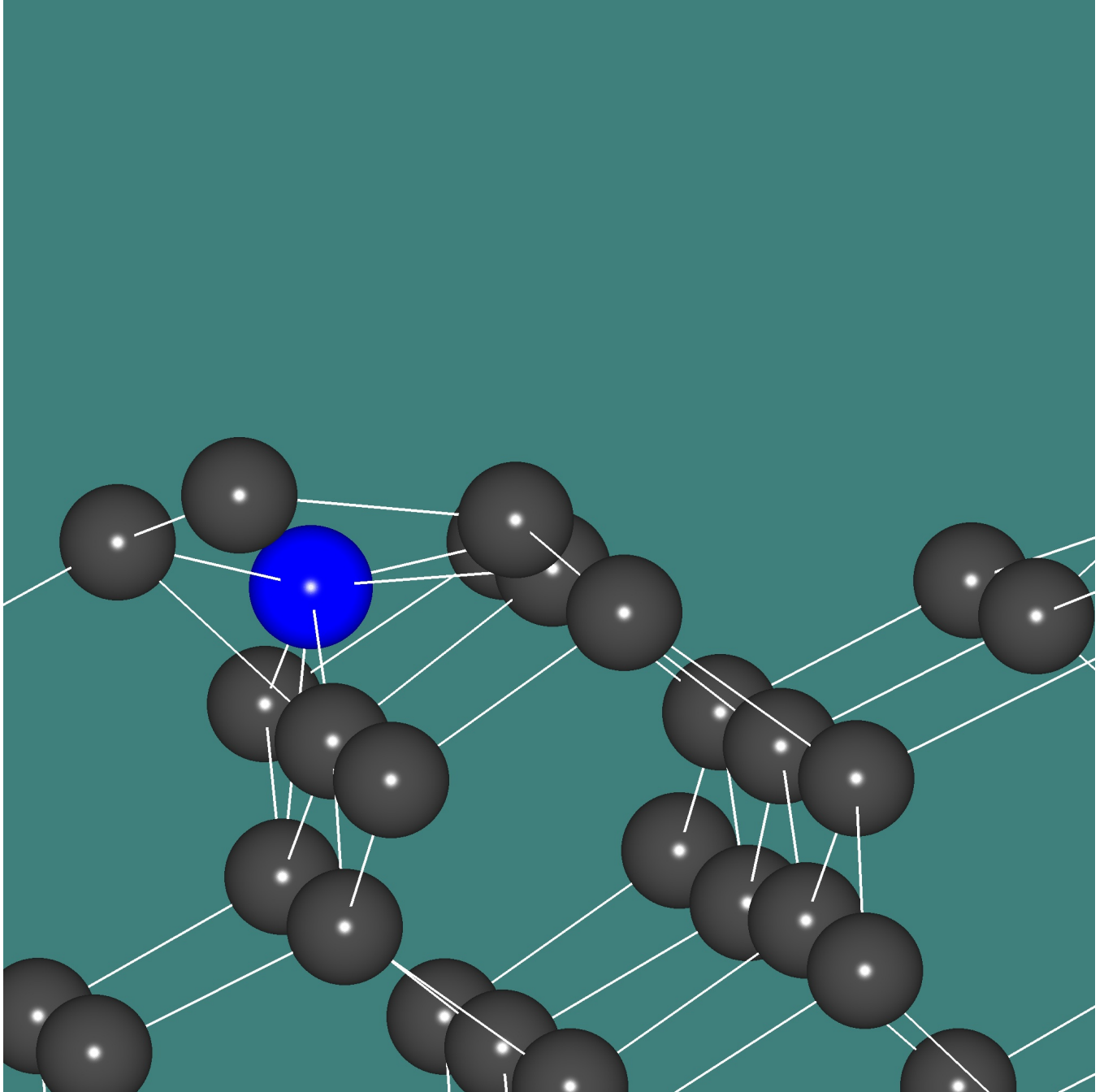
# Low Site



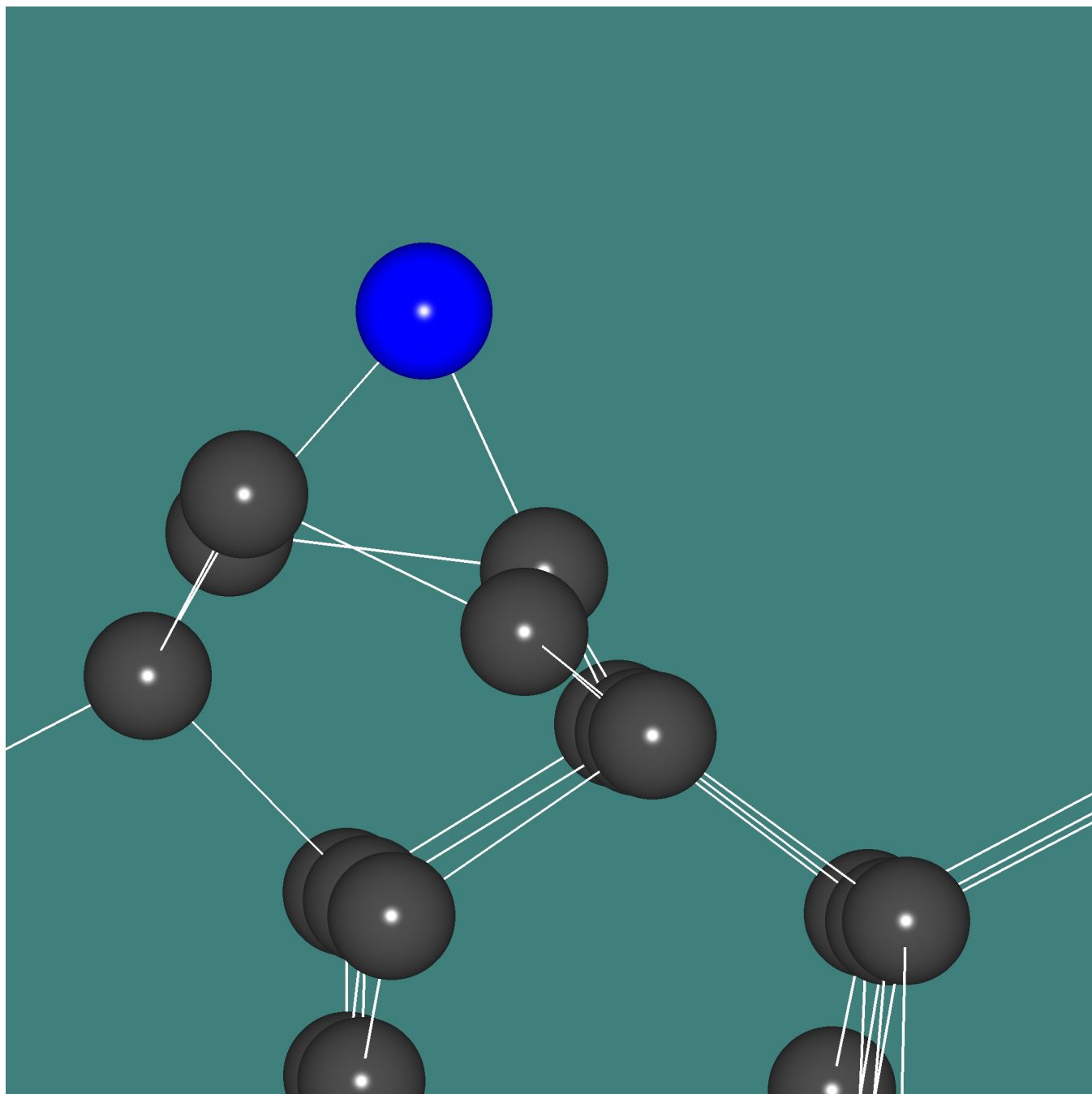
# Under Dimer Site



# Dimer Vacancy Site



# Dimer Site



# Conclusions

- The calculations confirm the models that have been proposed that suggest that the surface is always **silicon terminated**.
- The calculations indicate that it is likely that the observed  $c(2\times 2)$  (100) surface reconstruction is one due to an extra half layer of silicon.
- The unusual bonding configuration for the silicon leads to a situation where saturating more dangling bonds is offset by the decrease in strength of the bonds due to the loss of the ability to relax.
- STM images show **surface electronic structure** which is **not** always the same as the **physical structure**.
- When Co bonds to Si (100) it wants to sit in a packed environment with lots of neighbours.