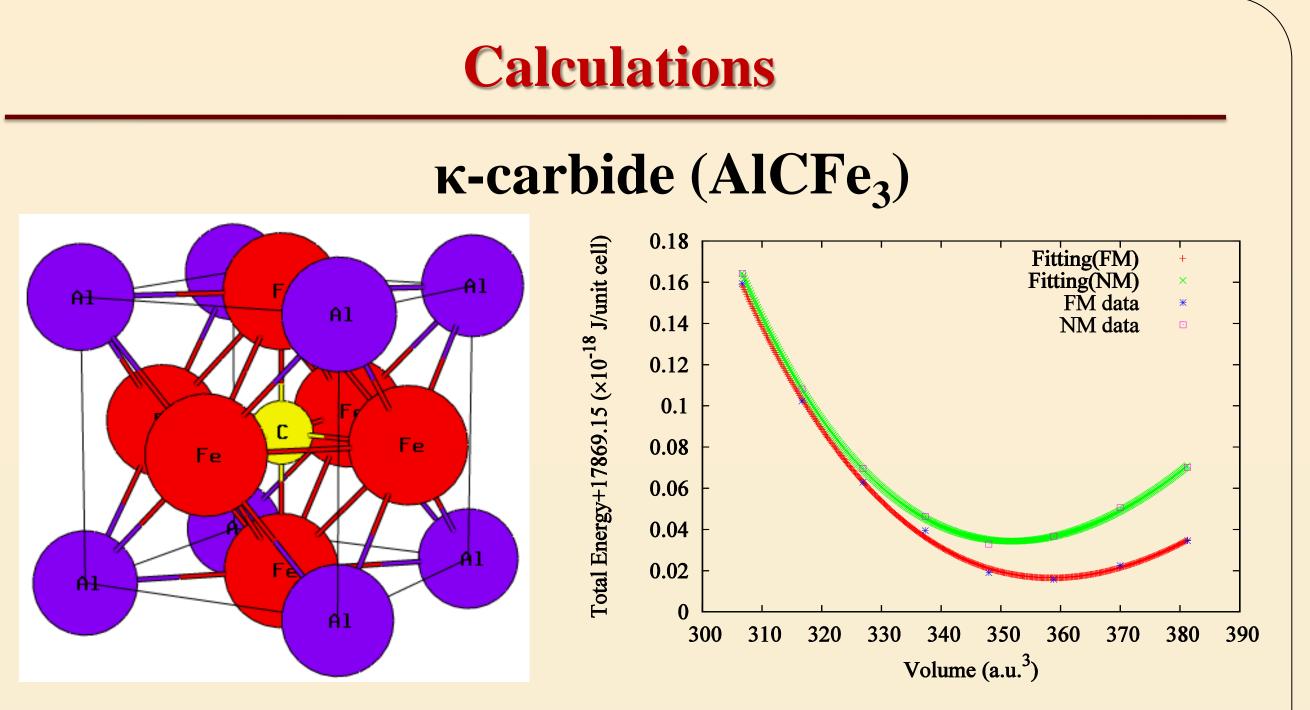


Formation energy of k-carbide using *ab initio* calculations

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Introduction

The compound Al(Fe,Mn)₃C with an anti-perovskite structure, known as κ -carbide, is easily found in strong, low-density steels containing substance concentrations of aluminium and manganese. The strength of Mn-Al-C steels is enhanced by the precipitation of κ -carbide, which is coherent with austenite, causes a shear band induced plasticity effect (SIP effect) [1]. The formation of κ -carbide in the austenite is an important method of controlling the final properties of these high-performance steels. However, there are a few calculations about formation energy by theoretical approach. In this research, we calculated the formation energy of Fe₃AlC in term of first-principles calculation by using the total-energy all-electron full-potential linearized augmented plane wave (FLAPW) method within the generalized gradient approximation (GGA).



Equilibrium lattice parameter is 3.761 Å

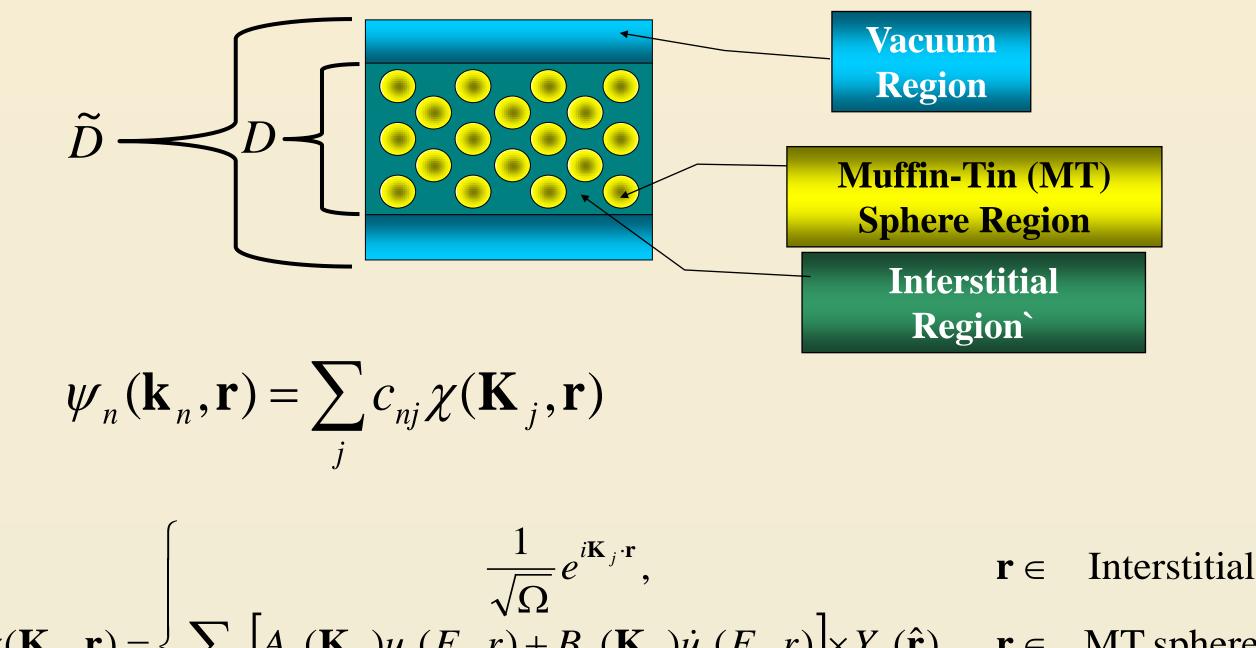
Method

Ab-initio calculation

A method of calculating atomic and molecular structure directly from the first principles of quantum mechanics, without using experiment data as parameters

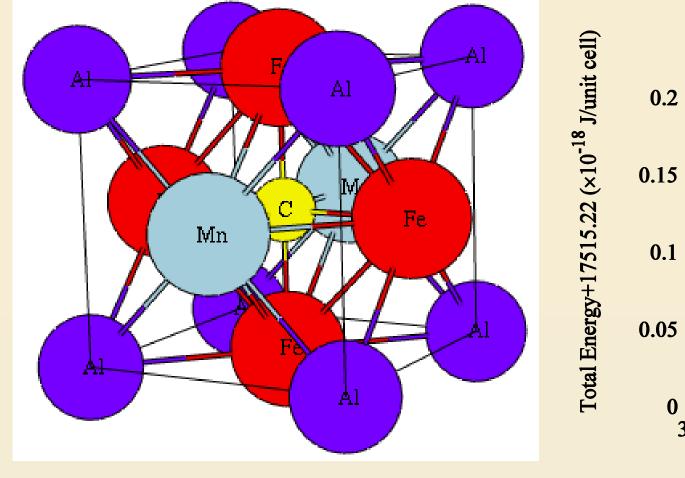
FLAPW method

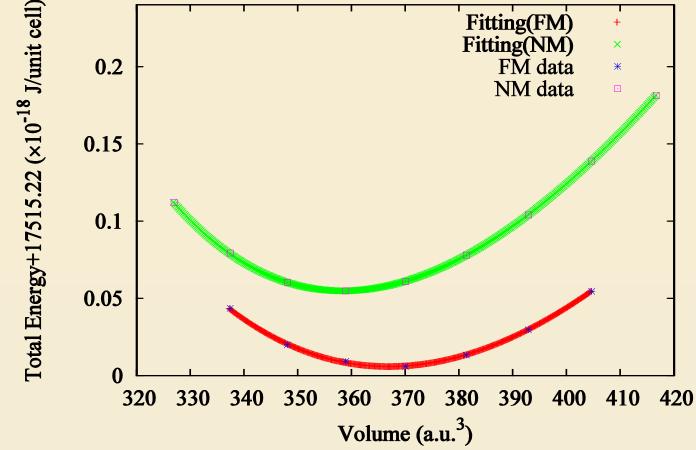
An all-electron method which within density functional theory is universally applicable to all atoms of the periodic table and to systems with compact as well as open structures



Formation energy for NM = -26.1 kJ/atom-molfor FM = -28.2 kJ/atom-mol

κ-carbide (AlCMnFe₂)





Equilibrium lattice parameter is 3.788 Å

Formation energy for NM = 456.8 kJ/atom-mol for FM = 450.5 kJ/atom-mol

AlCMnFe₂ with another octahedral site

 $\chi(\mathbf{K}_{j},\mathbf{r}) = \begin{cases} \sum_{j} \left[A_{L}(\mathbf{K}_{j})u_{l}(E_{l},r) + B_{L}(\mathbf{K}_{j})\dot{u}_{l}(E_{l},r) \right] \times Y_{L}(\hat{\mathbf{r}}), & \mathbf{r} \in MT \text{ sphere} \\ \sum_{q} \left[A_{q}(\mathbf{K}_{j})u_{q}(E_{v},z) + B_{q}(\mathbf{K}_{j})\dot{u}_{q}(E_{v},z) \right] \times e^{i\mathbf{K}_{q}\cdot\mathbf{r}_{\parallel}}, & \mathbf{r} \in Vacuum \end{cases}$

 $\mathbf{K}_{i} = \mathbf{k} + \mathbf{G}_{i}$ $\mathbf{K}_{q} = \mathbf{K}_{j\parallel} + \mathbf{G}_{j}$

Formation energy (Formation enthalpy)

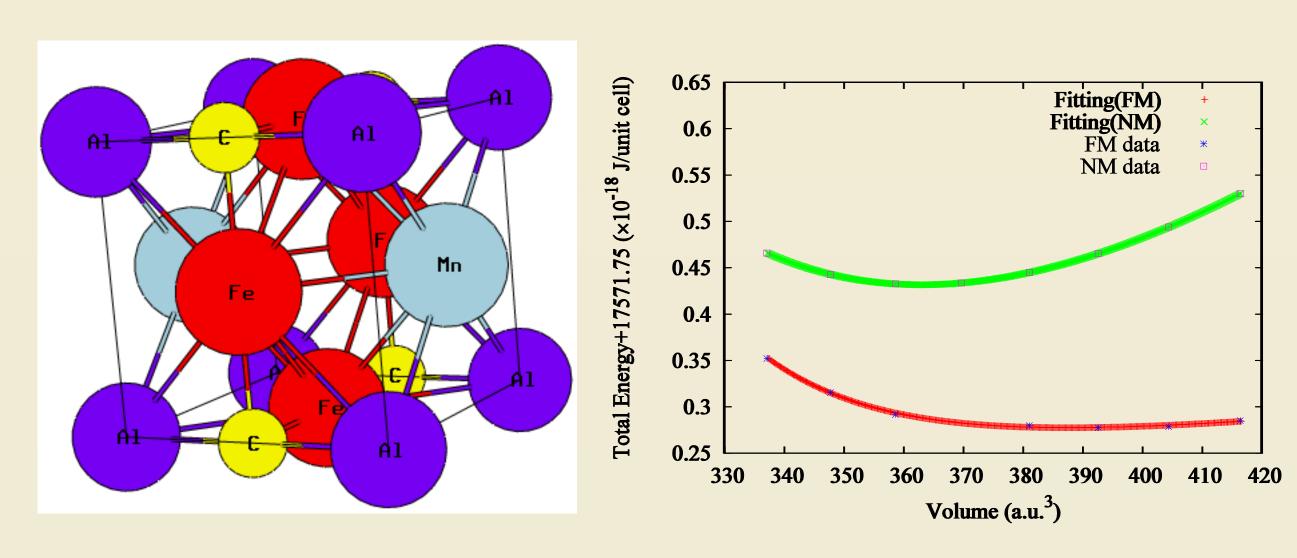
 $\Delta H(Al_a C_b Mn_c Fe_d) = U(Al_a C_b Mn_c Fe_d) - a U(fcc Al)$ $-b U(Graphite C) - c U(\alpha - Mn) - d U(bcc Fe)$

Where, ΔH is formation enthalpy and U is internal energy of specimen.

Conclusion

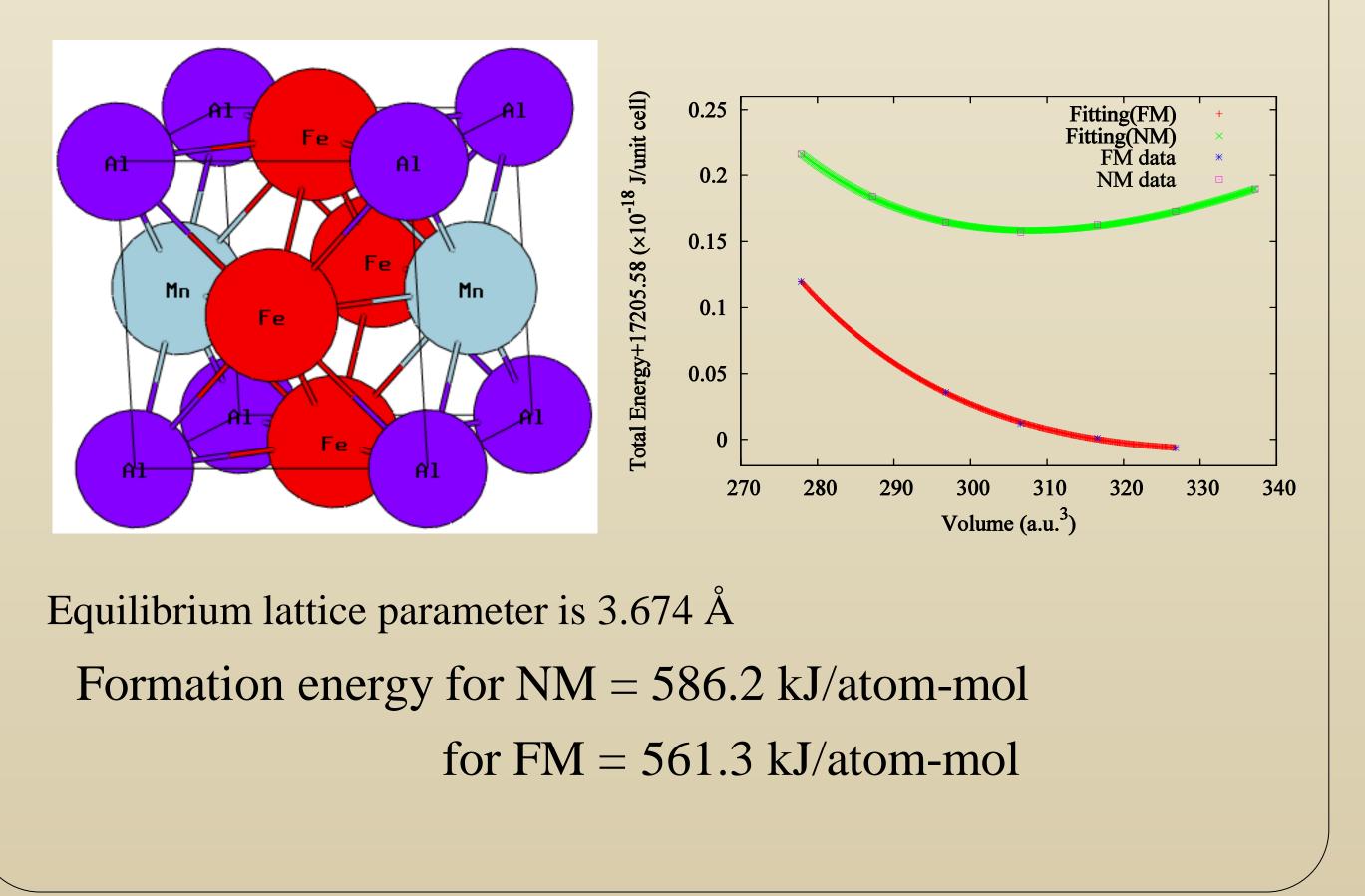
The calculated equilibrium lattice parameters are 3.736 Å and 3.761 Å for the nonmagnetic (NM) and ferromagnetic (FM) cases, respectively, which are comparable to the measured one of FM Fe₃AlC, 3.781 Å [2]. The formation enthalpies of Fe₃AlC are calculated to be -26.1 and -28.2 kJ/atom-mol, for the NM and FM cases, respectively. These values are little smaller than the results of Ohtani et al. [3], but the formation enthalpy difference between the NM and FM cases are the same.

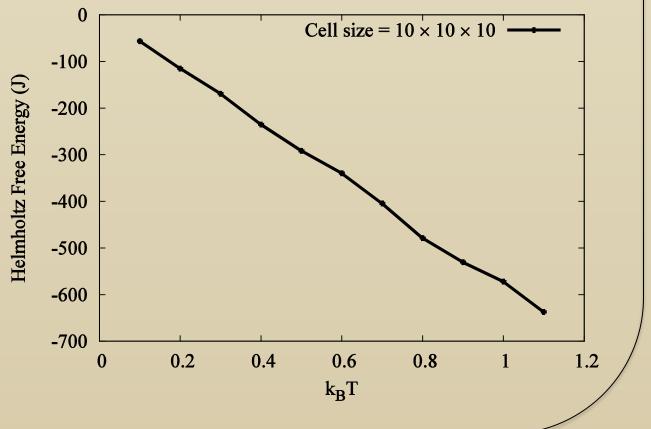
For Fe₂MnAlC, which has three octahedral site for carbon, we also calculated formation enthalpy with and without carbon. Clearly, these three types have higher energy than Fe₃AlC. With these formation enthalpy, we assume that Fe₂MnAlC with carbon in body center is only κ -carbide and the others are austenite. Then, set $10 \times 10 \times 10$ matrix which has κ -carbide or austenite in each cell Cell size = $10 \times 10 \times 10$ ----and calculate Helmholtz free energy with -100 gy (J) changing temperature by Monte Carlo. -200 However, we cannot see any sign of -300 phase transformation. -400 In future, we will calculate the Gibbs free -500 energy with calculated enthalpy. Also, -600 we will improve this Monte Carlo method.



Equilibrium lattice parameter is 3.854 Å Formation energy for NM = 505.0 kJ/atom-molfor FM = 483.8 kJ/atom-mol







References

[1] G. Frommeyer and U. Brüx, Steel Res. Int. 77, 627 (2006).

[2] L. S. Palatnik, I. A. Tananko and Yu. G. Bobro, *Kristallografiya* 9, 163 (1964).

[3] H. Ohtani, M. Yamano, and M. Hasebe, ISIJ International 44, 1738 (2004).