

SINGLE CRYSTAL SURFACE ENGINEERING

W. Kurz

C. Bezençon

S. Mokadem

J.-D. Wagnière

**Swiss Federal Institute
of Technology
Lausanne (EPFL)**

J.-M. Drezet

CALCOM Lausanne

M. Hoebel

M. Konter

**ALSTOM Power
Baden**

SX - SURFACE ENGINEERING

- What is SX surface engineering ?
- Why ?
- How does it work ?

THE CONCEPT OF SX - SURFACE ENGINEERING

- SX superalloys are generally used for HPT turbine blades.
- Mechanical properties \uparrow when
 1. Cr (oxidation) \downarrow
 2. C, B, Zr (grain boundaries) \downarrow

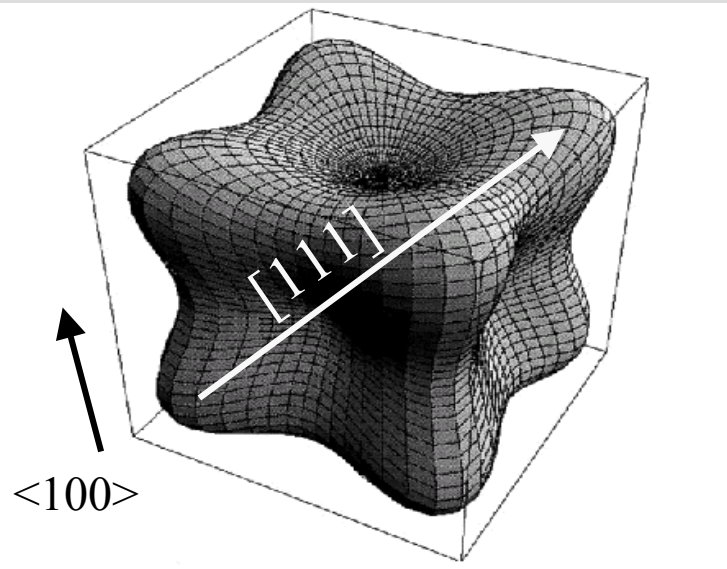
THE CONCEPT OF SX - SURFACE ENGINEERING - I COATING

Low Cr superalloys require protection against oxidation and corrosion :

- Present day technique: plasma spray Cr rich polycrystalline **NiCrAlY** coating.
- Ni has high crystal (E) anisotropy → incompatibility stresses in thermal cycling → premature cracking.

SOLUTION: SX COATING ON SX BLADE

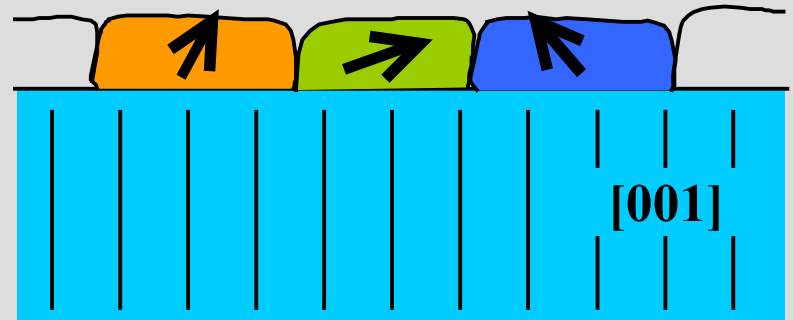
E-MODULUS ANISOTROPY OF Ni



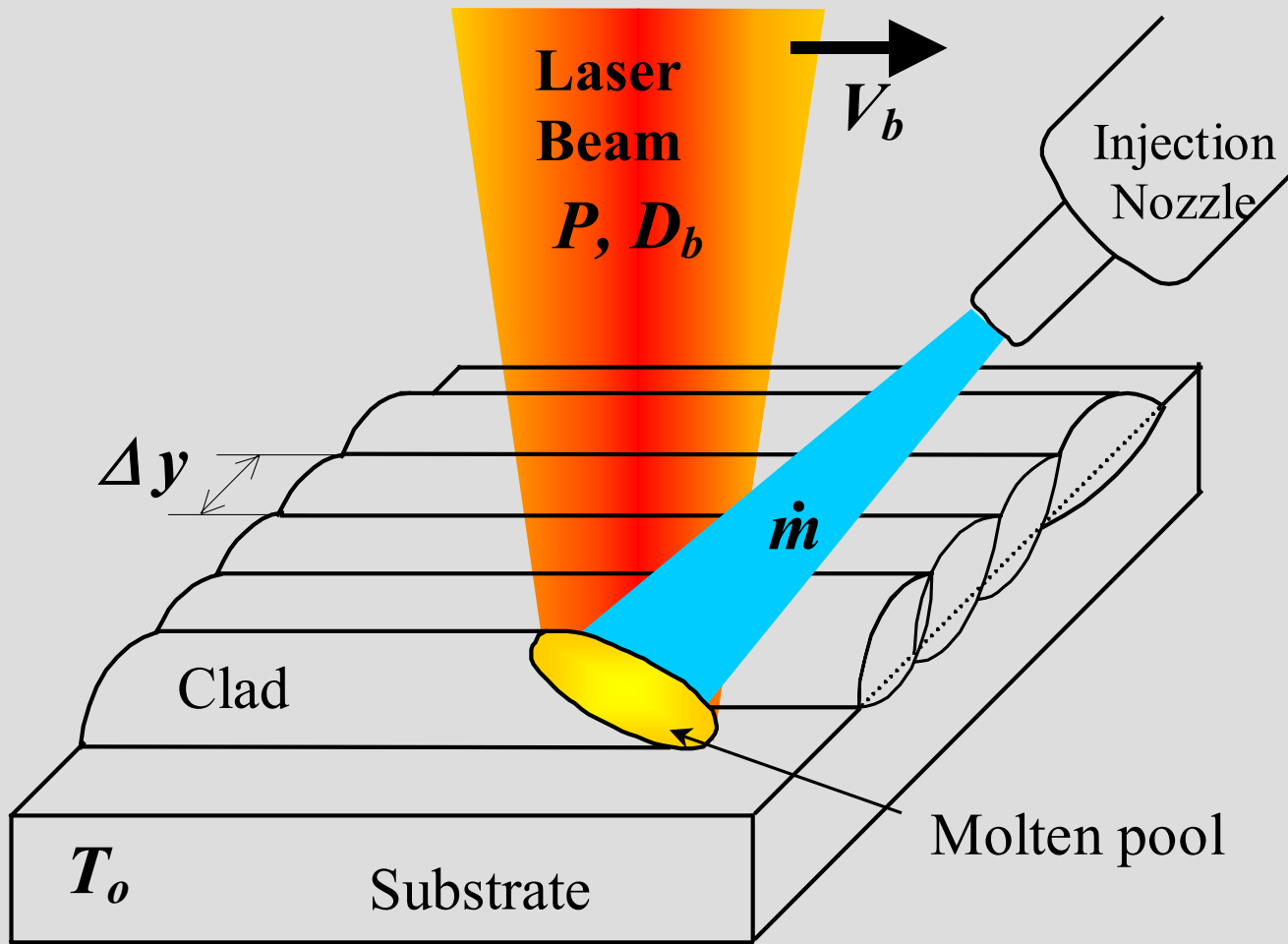
Phase γ - Siebörger et al.

$$E_{[100]} \sim 125 \text{ [GPa]}$$

$$E_{[111]} \sim 310 \text{ [GPa]}$$



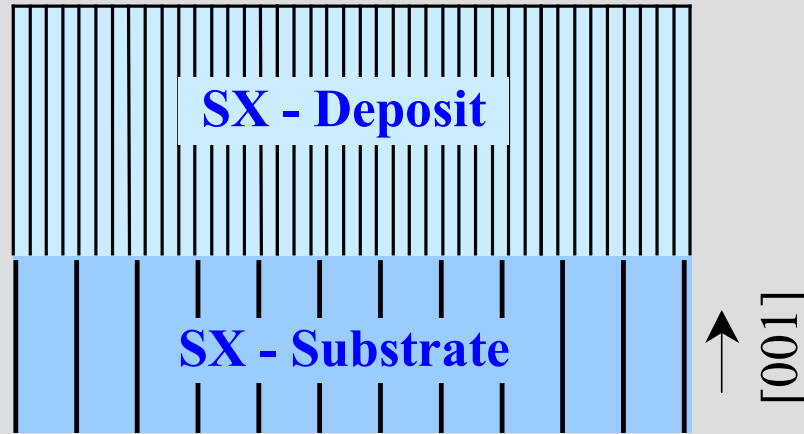
LASER CLADDING PROCESS



MICROSTRUCTURES / DEFECTS

Columnar →

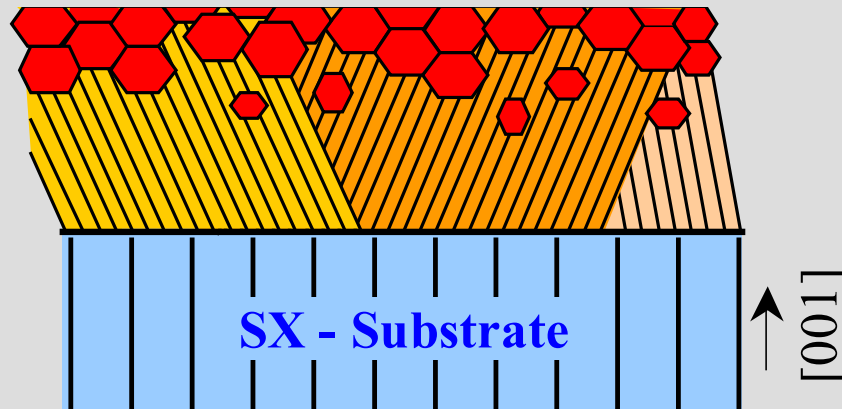
Epitaxial growth →



Ideal case

Equiaxed grains →

Non-epitaxial columnar grains →



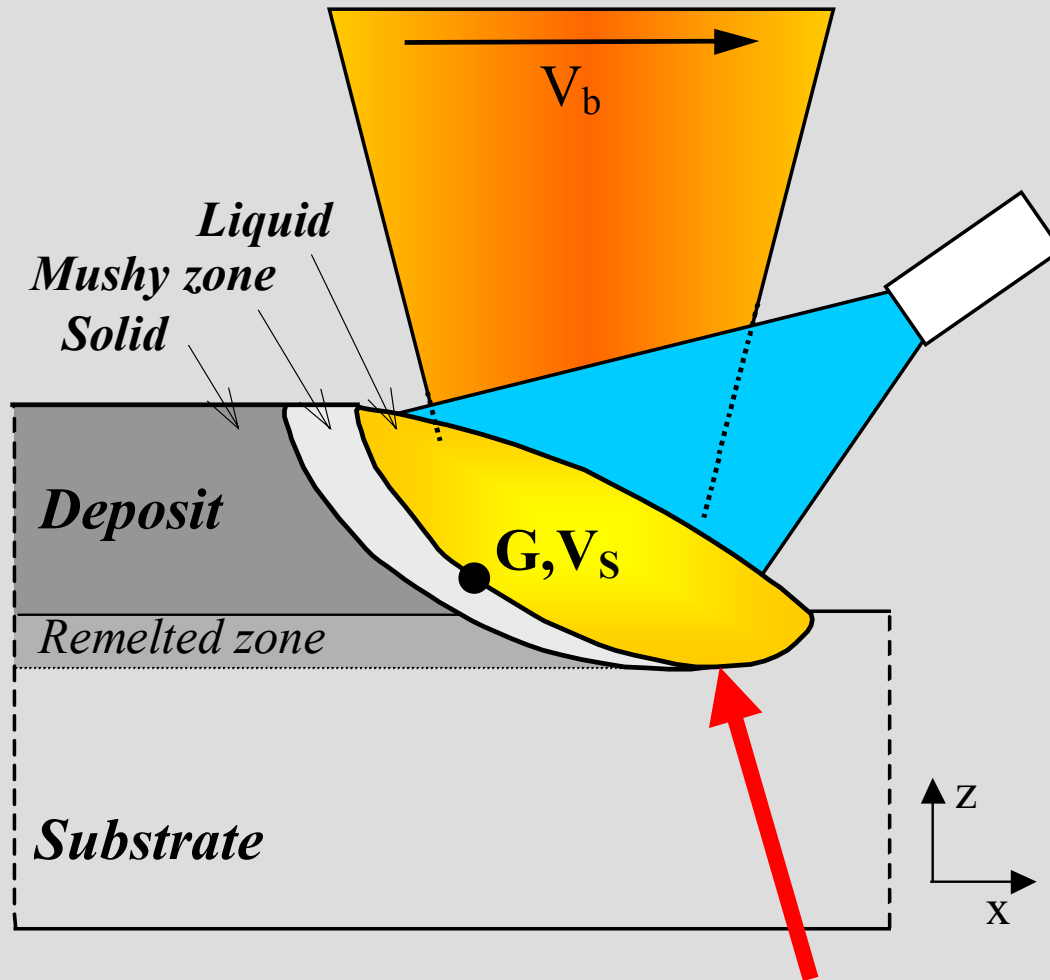
Defects to avoid

EPITAXY

Ensure at the surface

- metallic contact with the substrate
- similar crystal structure of primary phases

LASER CLADDING PROCESS



EPITAXY

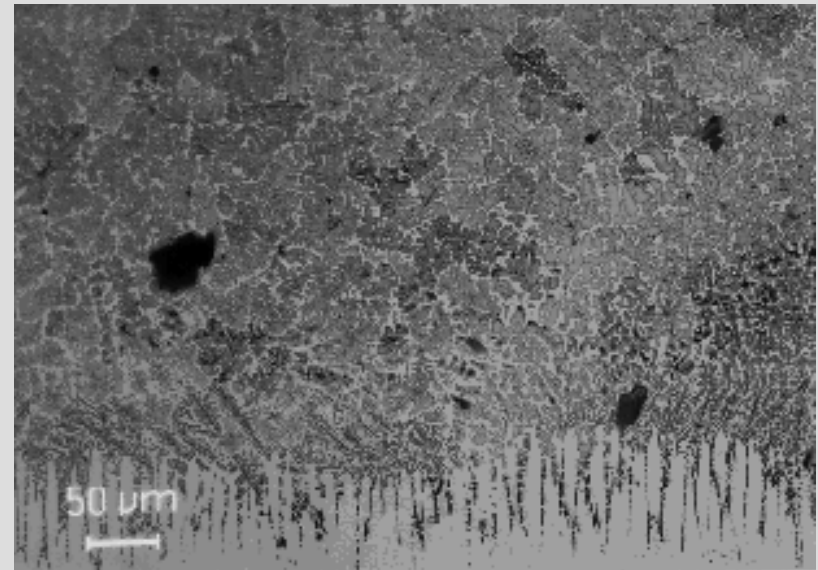
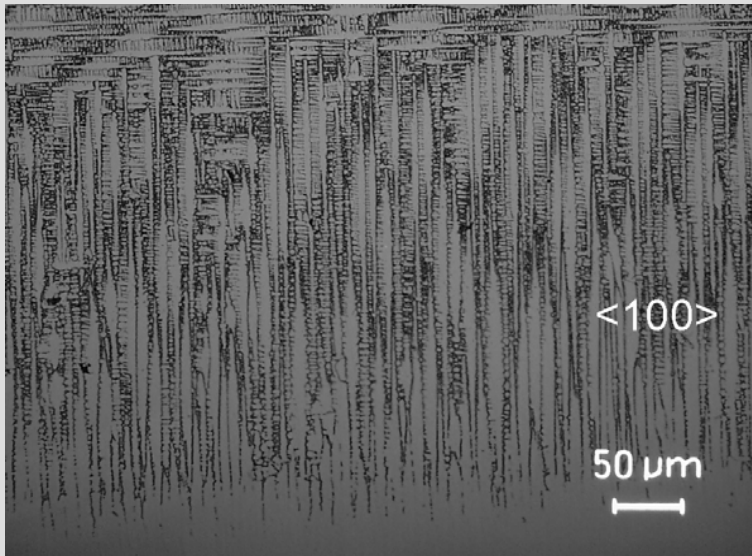
MCrAlY with

low Al

high Al

γ primary phase

β primary phase

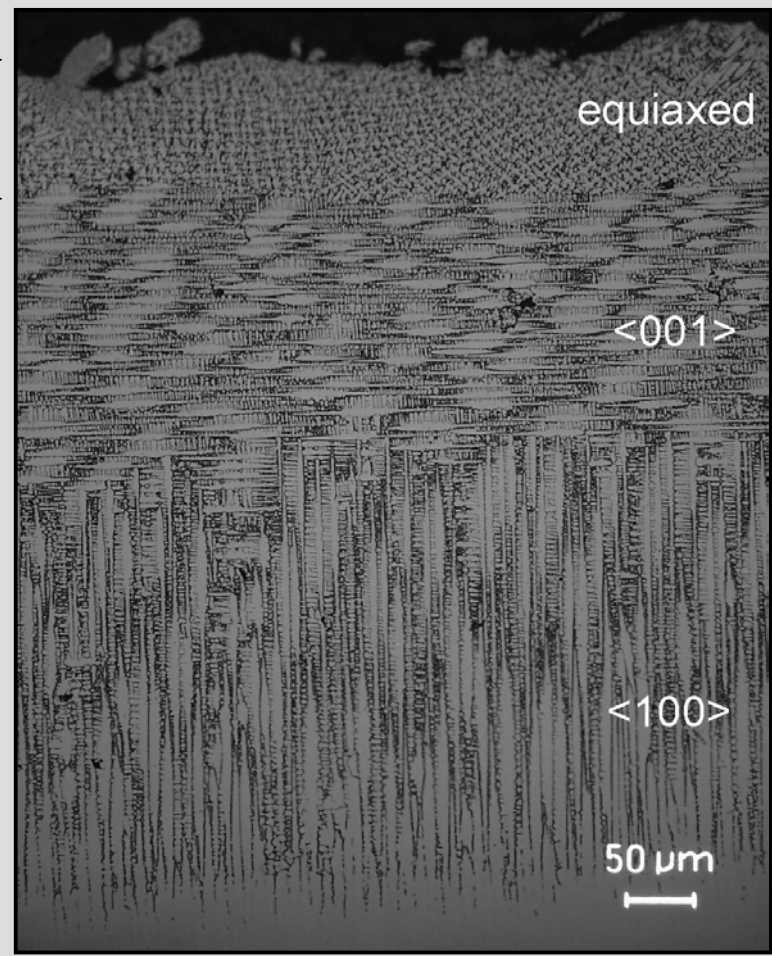
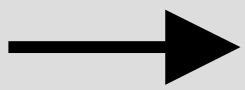


SINGLE CRYSTAL LASER CLADDING OF MCrAlY ON CMSX 4

Equiaxed

*Columnar
Dendritic
Growth*

Epitaxy



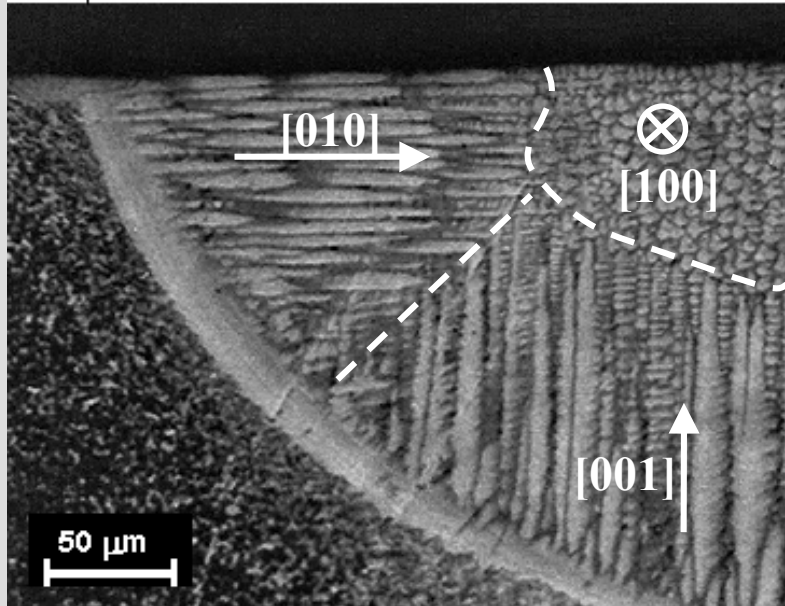
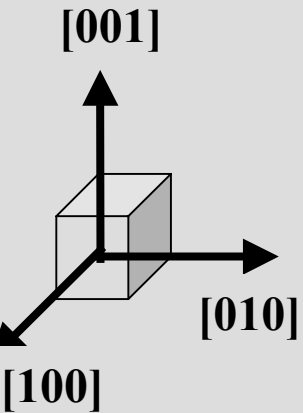
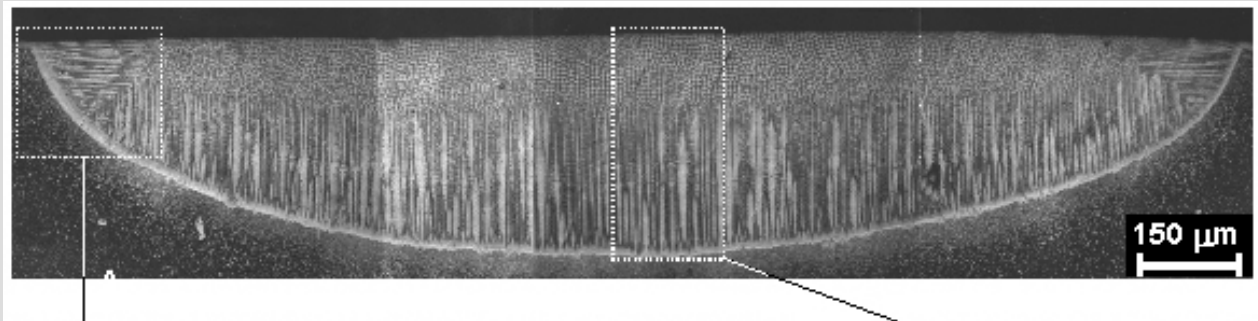
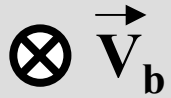
MCrAlY

CMSX-4

EPFL

DENDRITE BRANCH SELECTION

Laser
Remelting



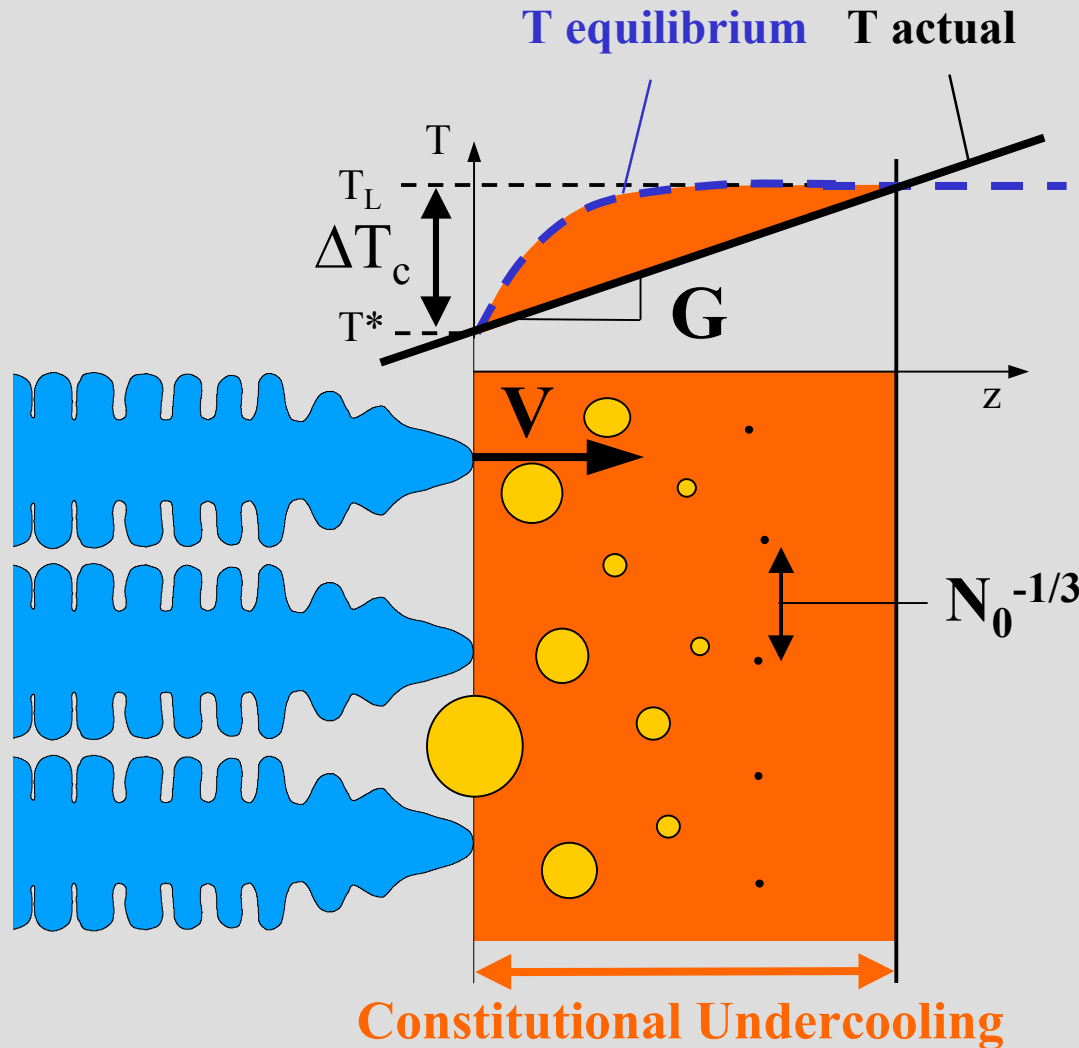
Dendrites grow
along
<100> direction

(direction closest to
heat flux is selected)

COLUMNAR-EQUIAXED TRANSITION CET

Avoid nucleation and growth in the
constitutionally undercooled region
ahead of the columnar dendritic zone

COLUMNAR TO EQUIAXED TRANSITION



Process

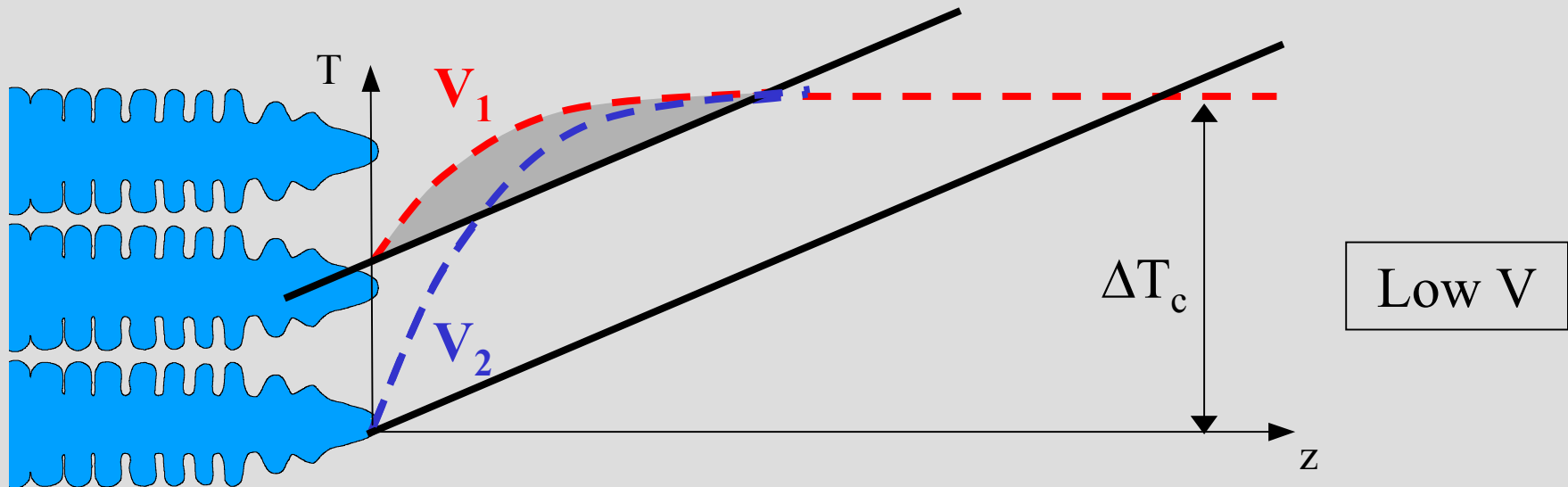
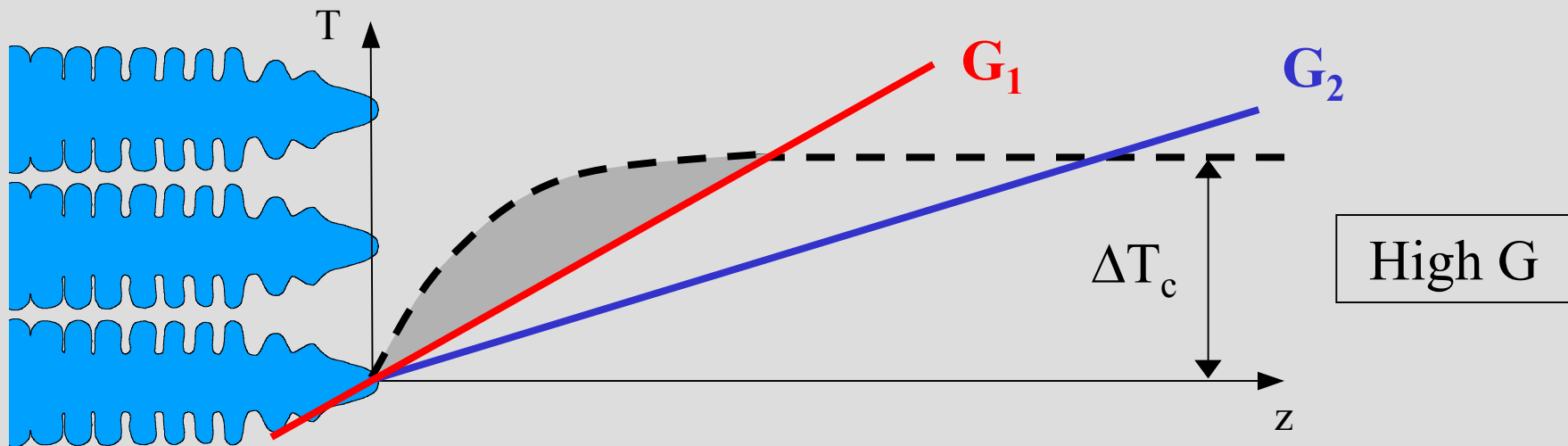
- Temperature Gradient **G**
- Solidification Velocity **V**

Alloy

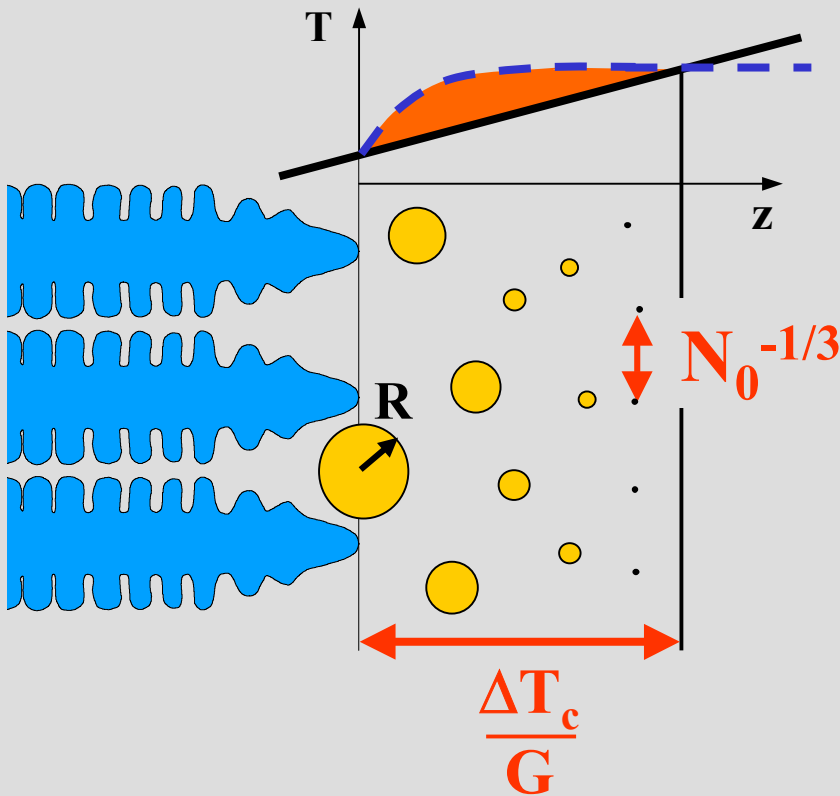
- Growth Undercooling ΔT_c
- Nucleation Site Density N_0
- Nucleation Undercooling ΔT_n

COLUMNAR TO EQUIAXED TRANSITION

Effect of G and V



COLUMNAR TO EQUIAXED TRANSITION



$$\frac{\Delta T_c}{G} < N_0^{-1/3}$$

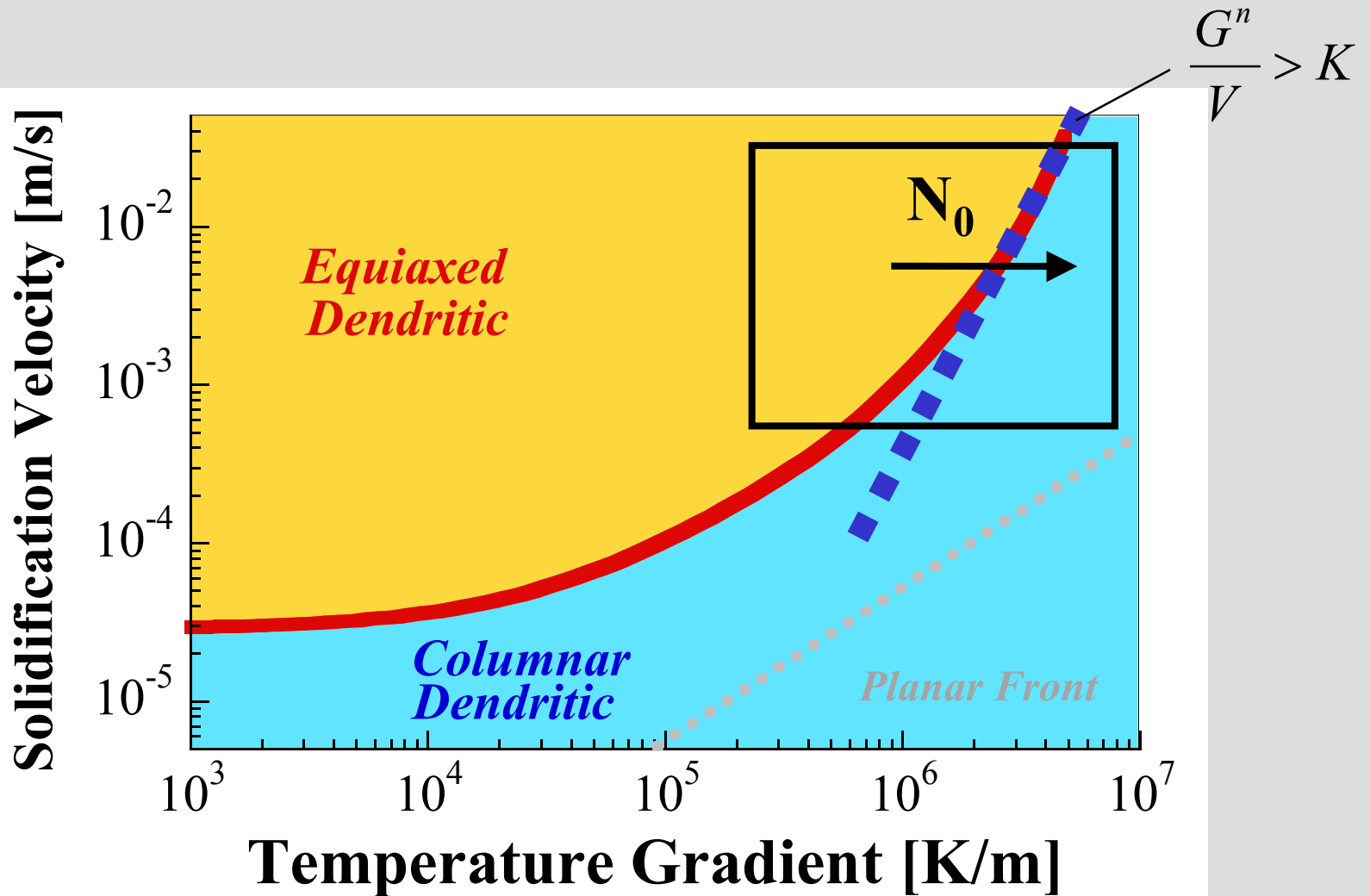
$$\Delta T_c = \Delta T_0 (a \cdot V)^{1/n}$$

$$\Delta T_0 (a \cdot V)^{1/n} < G \cdot N_0^{-1/3}$$

$$\frac{G^n}{V} > a \cdot (\sqrt[3]{N_0} \cdot \Delta T_0)^n$$

$$\frac{G^n}{V} > a \cdot \left\{ \sqrt[3]{\frac{4\pi \cdot N_0}{3\phi_c}} \cdot \frac{\Delta T_0}{1+n} \right\}^n \quad \Rightarrow \quad \boxed{\frac{G^n}{V} > K}, \text{ for a columnar structure}$$

MICROSTRUCTURE SELECTION MAP



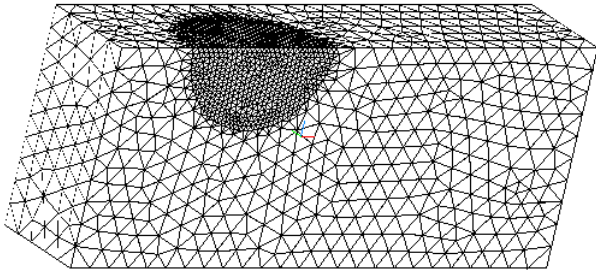
PROCESSING PARAMETERS

Calculation of

- temperature gradient, G
- interface velocity, V

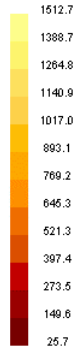
as a function of position in the melt pool

Mesh



.iclad

0.0006081



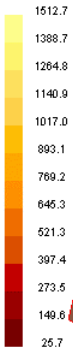
.iclad

temperature

Time : 200 s

0.0004598

Temperature



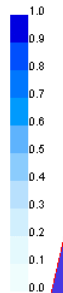
.iclad

temperature

Time : 200 s

0.000516

Solid fraction



.iclad

fraction of solid

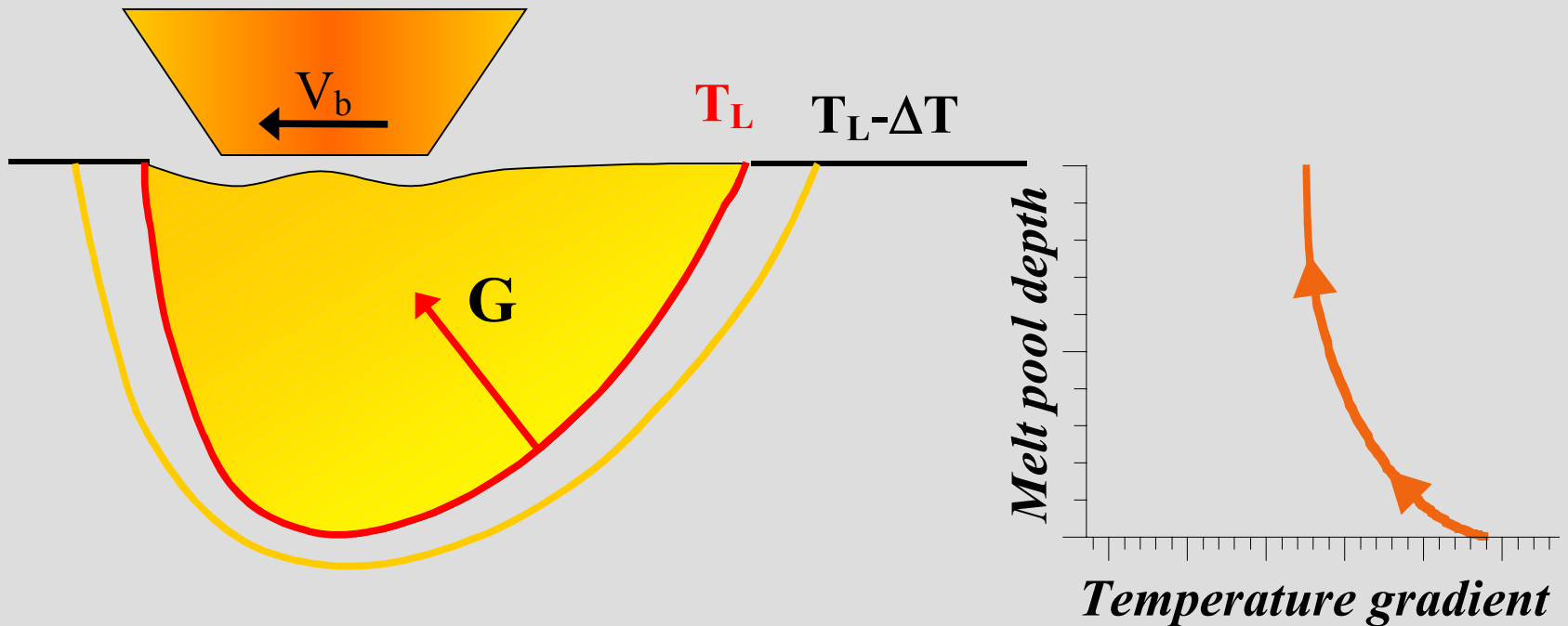
Time : 200 s

0.000470

SOLIDIFICATION CONDITIONS

Temperature Gradient

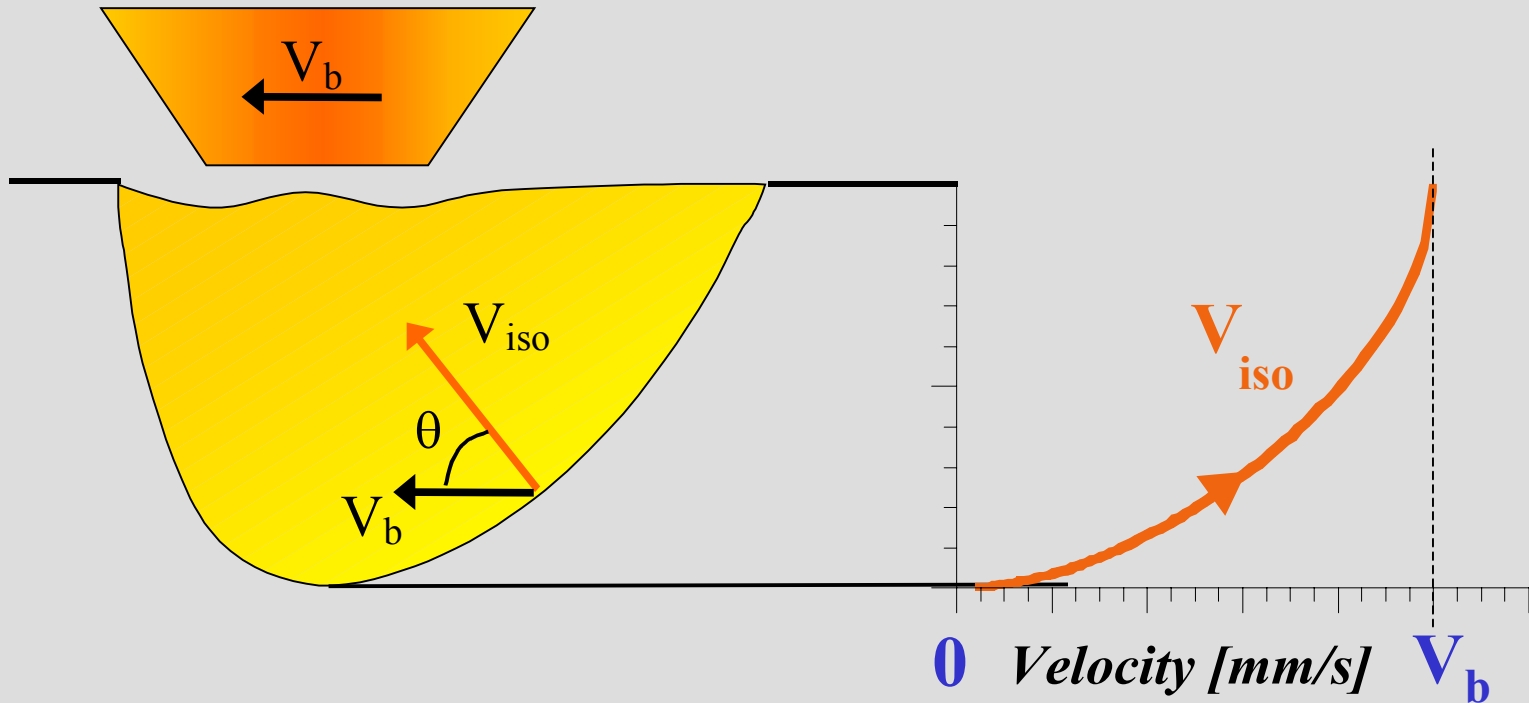
Rosenthal Solution of the Heat-flux equation



SOLIDIFICATION CONDITIONS

Solidification velocity

Isotherm velocity : $V_{iso} = V_b \cdot \cos \theta$

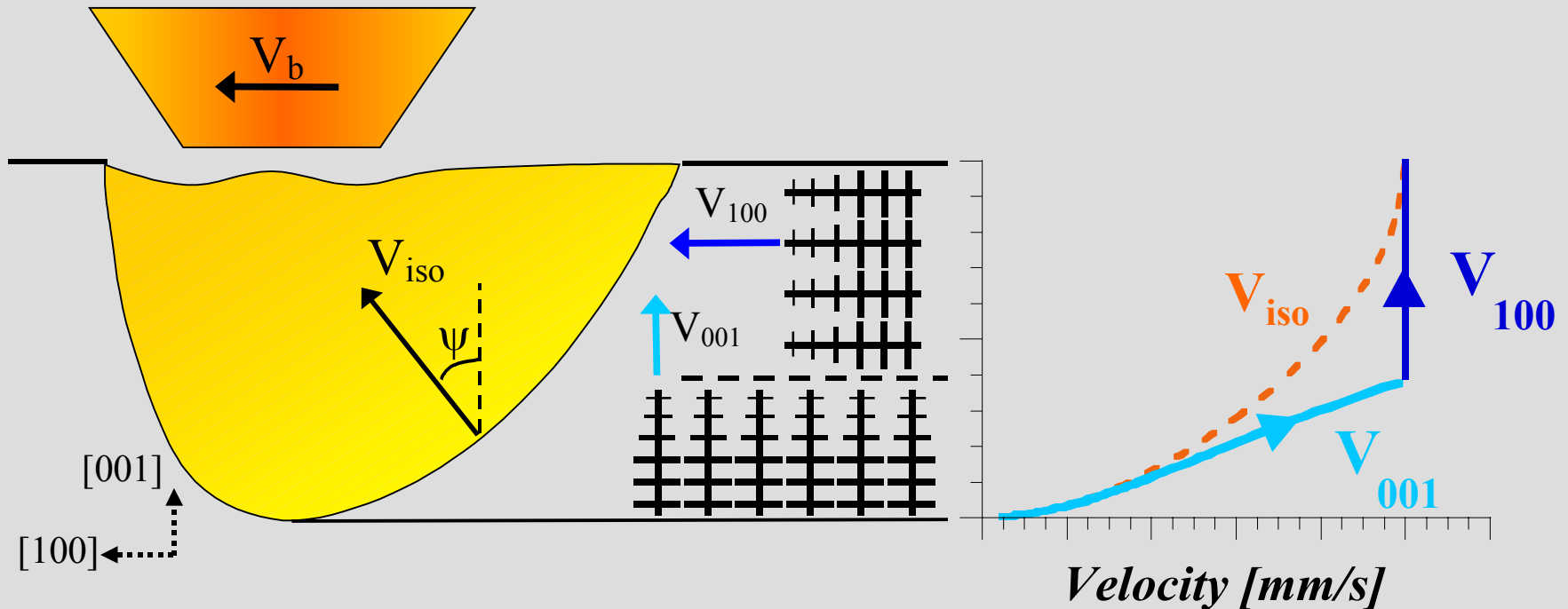


SOLIDIFICATION CONDITIONS

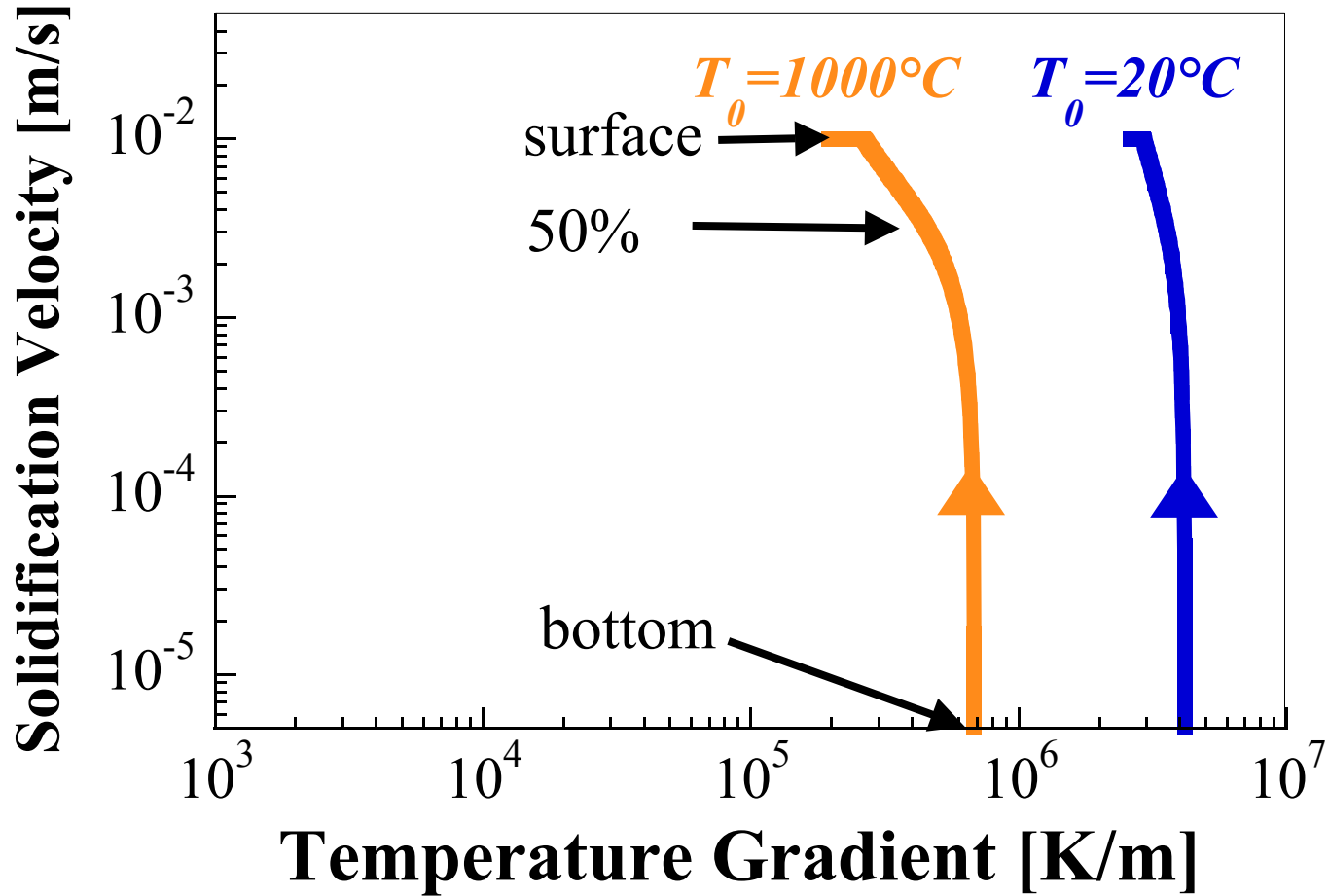
Solidification velocity

Isotherm velocity : $V_{iso} = V_b \cdot \cos \theta$

Dendrite velocity : $V_{hkl} = \frac{V_{iso}}{\cos \psi}$



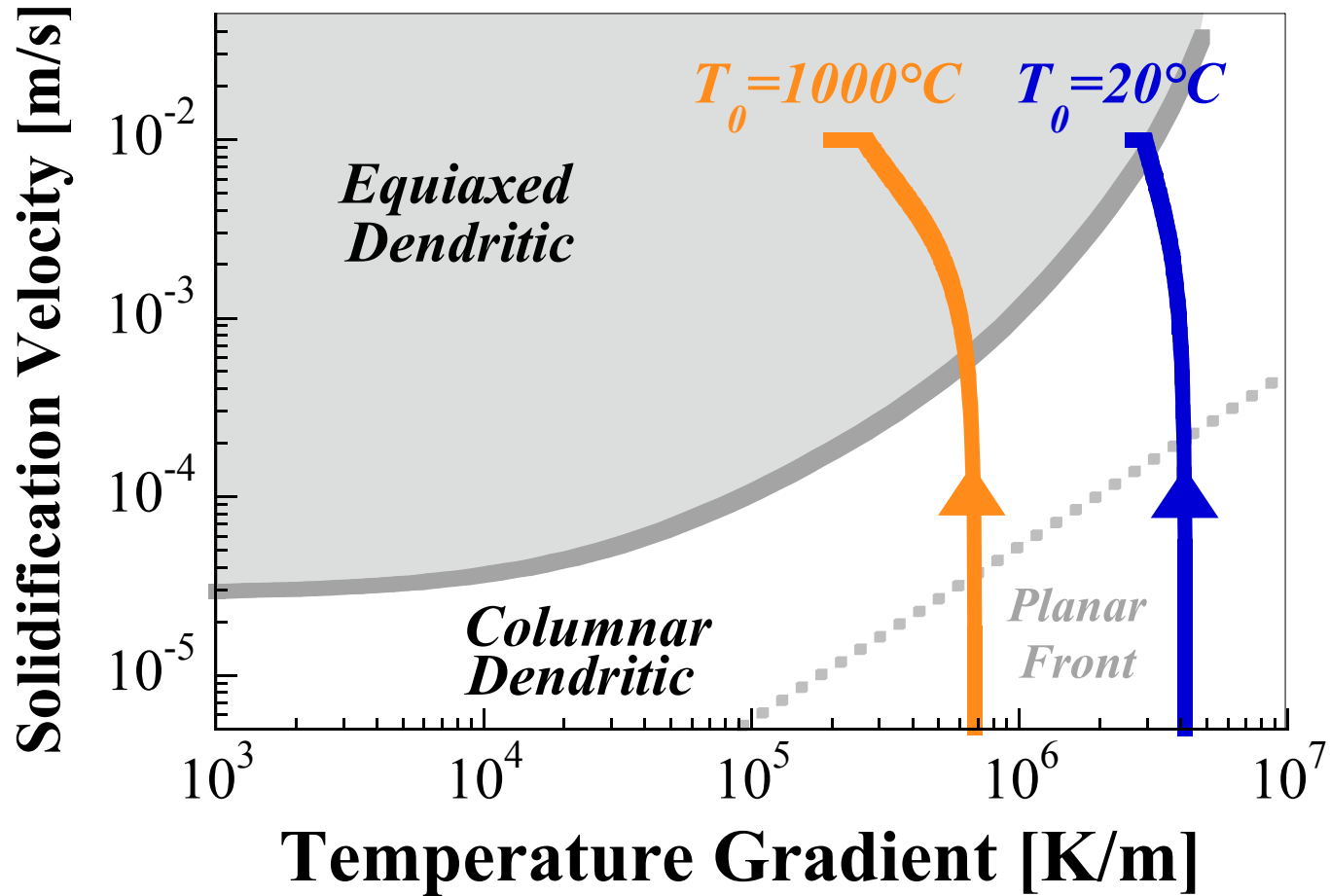
SOLIDIFICATION CONDITIONS



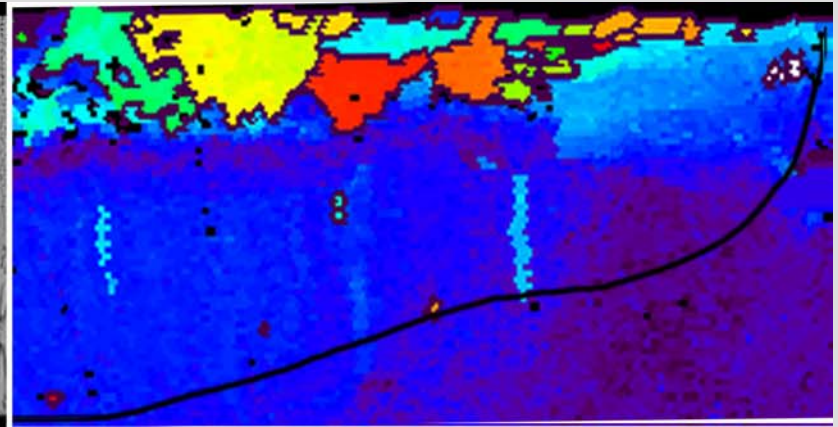
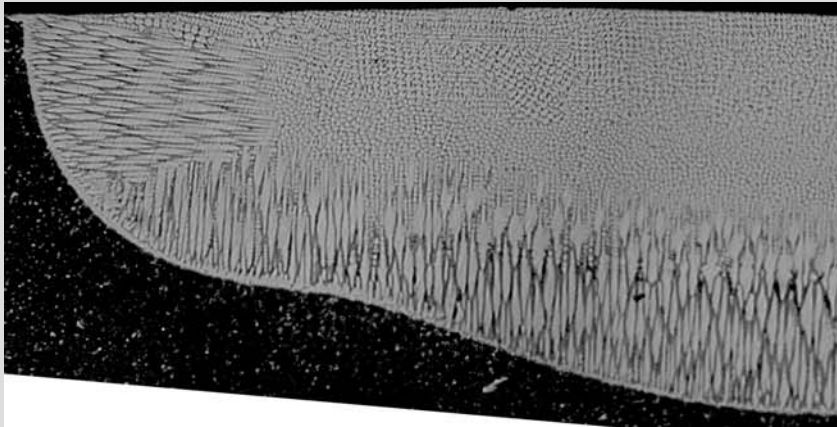
**COMBINATION OF
PARAMETERS CHARACTERISTIC FOR
MICROSTRUCTURES
&
PROCESSING**

MICROSTRUCTURE-PROCESSING MAP

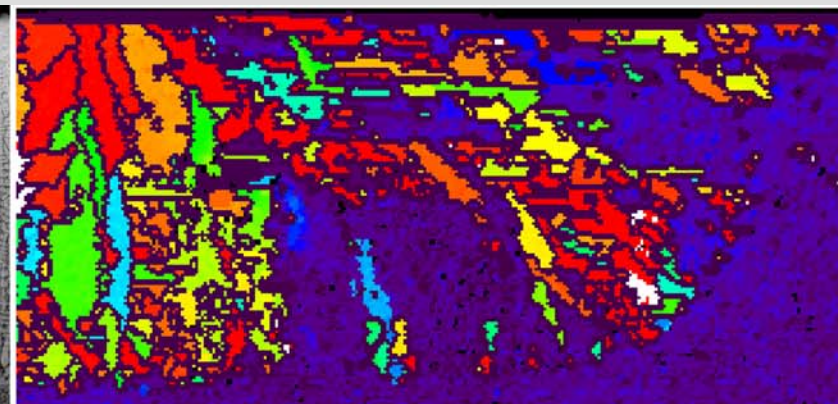
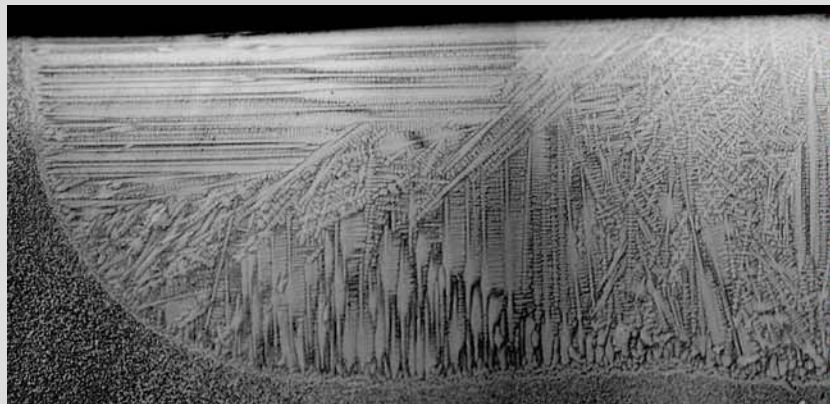
PROCESSING-MICROSTRUCTURE MAP



$$T_0 = 20^\circ\text{C}$$



$$T_0 = 1000^\circ\text{C}$$

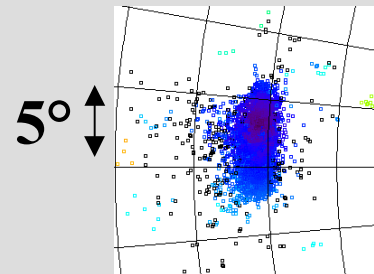
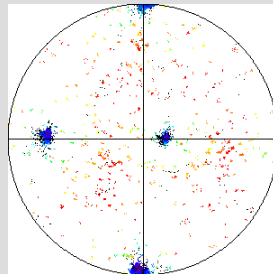
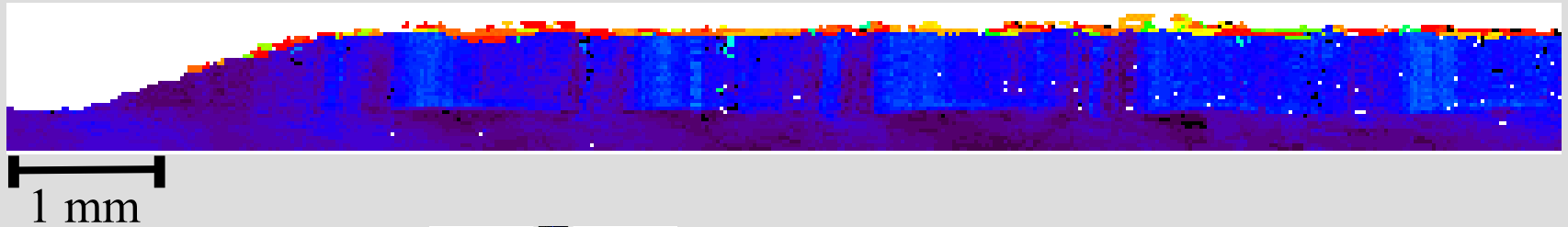
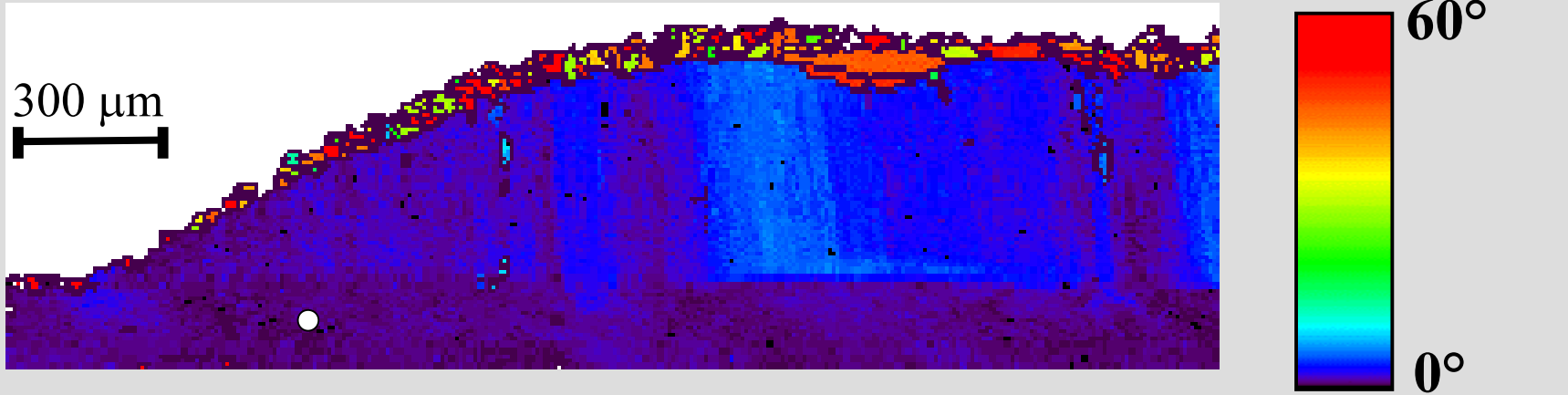


LASER REMELTING

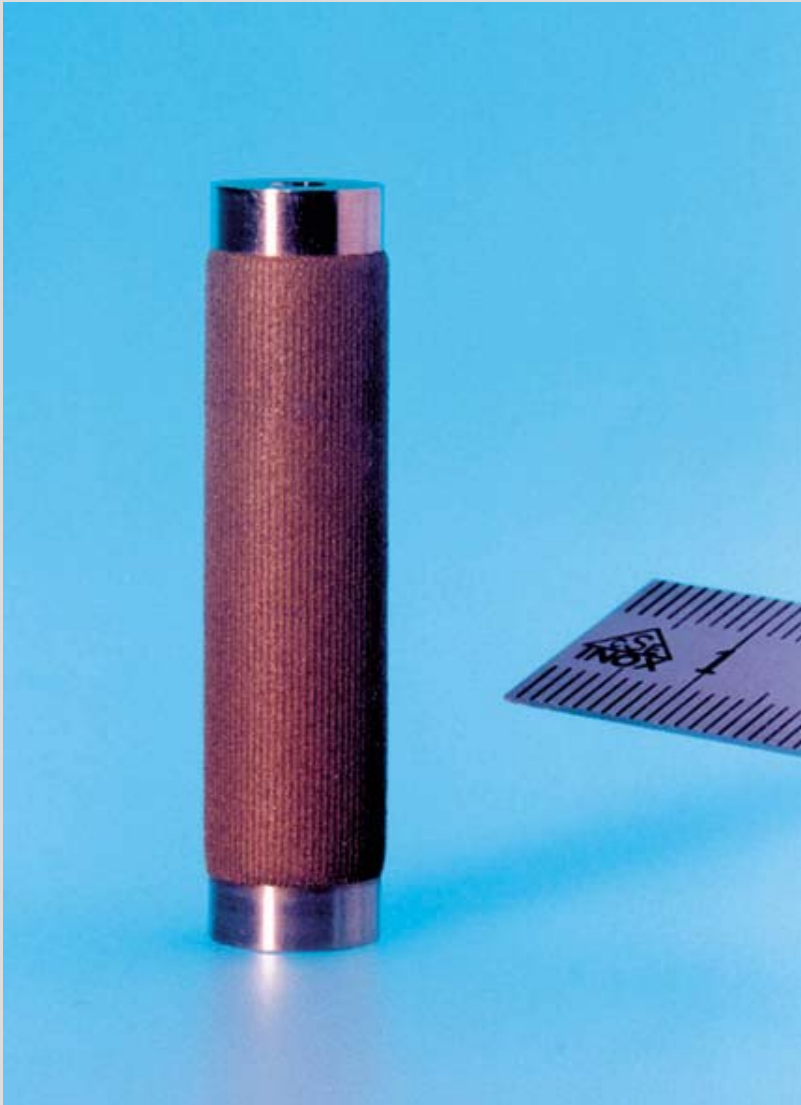
EPFL

LASER CLADDING MCrAlY

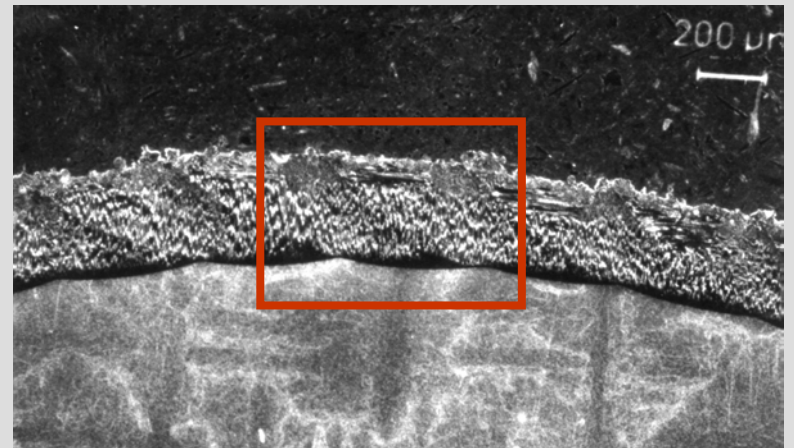
Orientation analysis (EBSD)



SX CLADDING OF MECH. TEST SPECIMEN

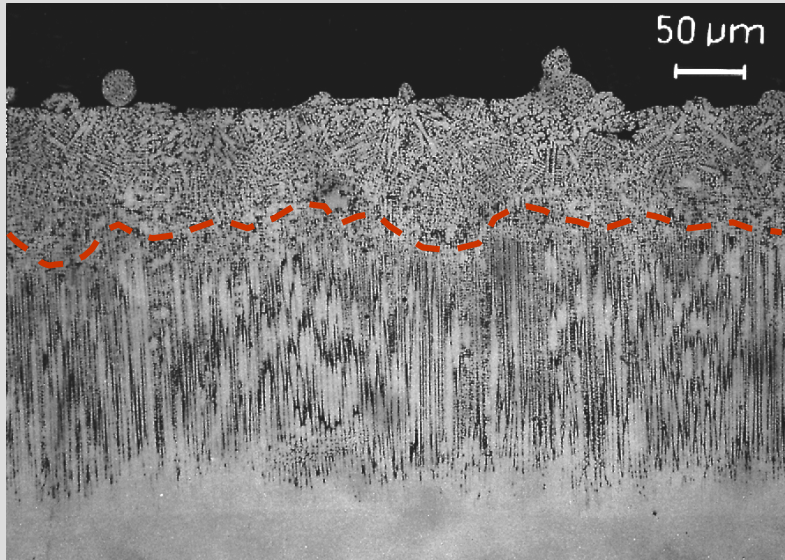


Transverse Section

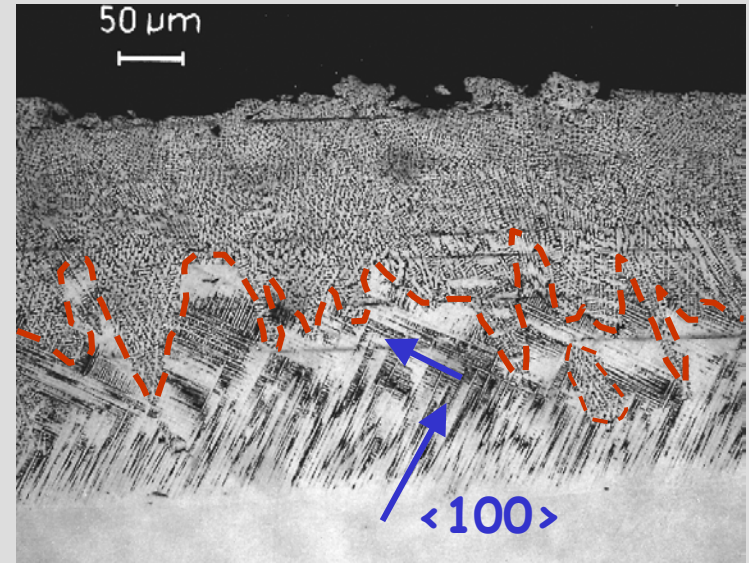
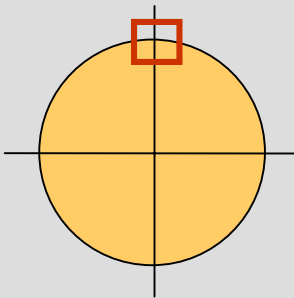


MCrAlY - 0,3 mm

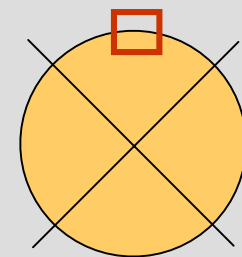
MICROSTRUCTURE OF TEST SPECIMEN



$[100]$



$[110]$



REPAIR ENGINEERING

High price of SX components,
typically 5 k€ per blade,
asks for life time extension techniques

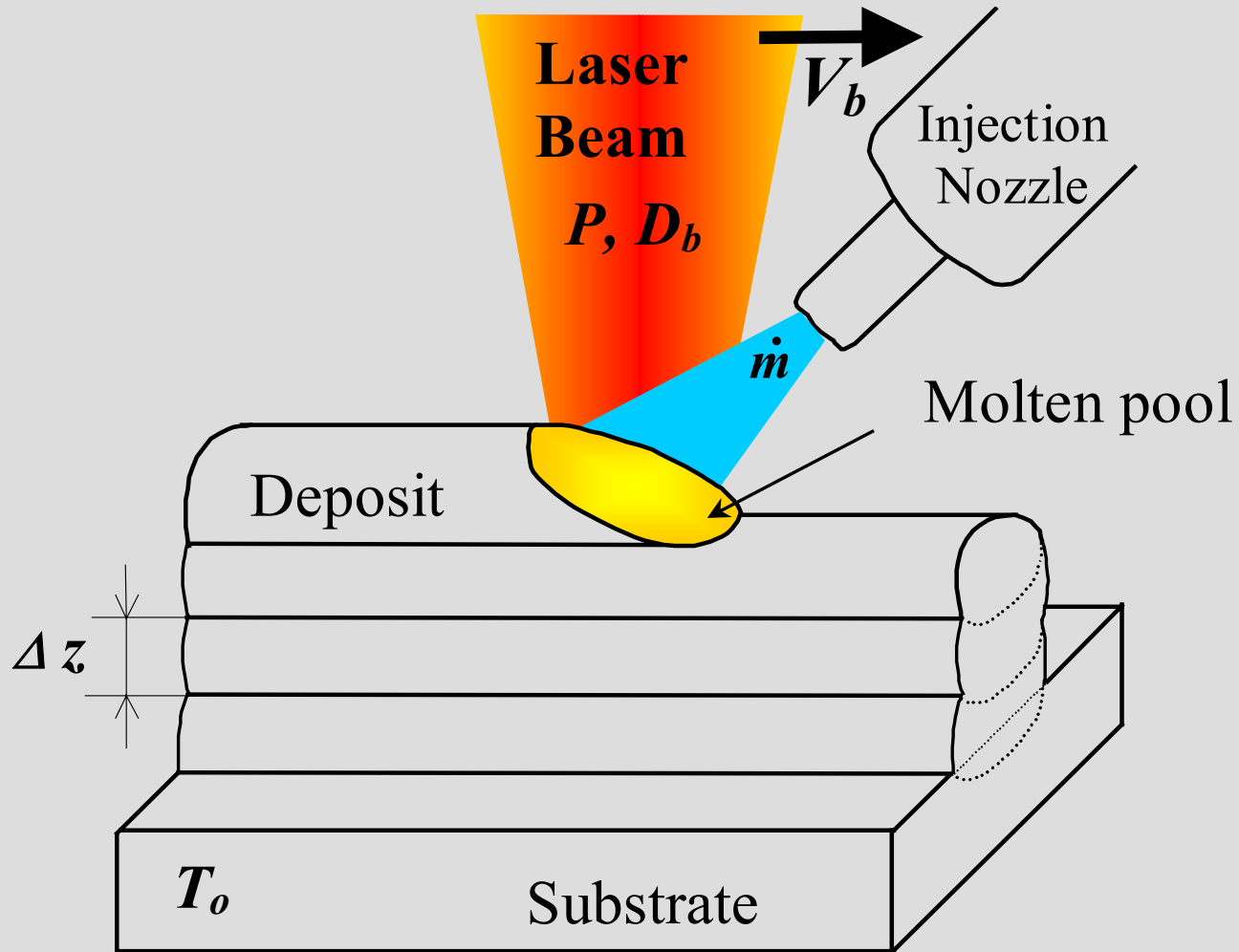
THE CONCEPT OF SX - SURFACE ENGINEERING - II REPAIR ENGINEERING

Low C, B, Zr superalloys form (at high T) mechanically strong crystals - but are very weak at grain boundaries:

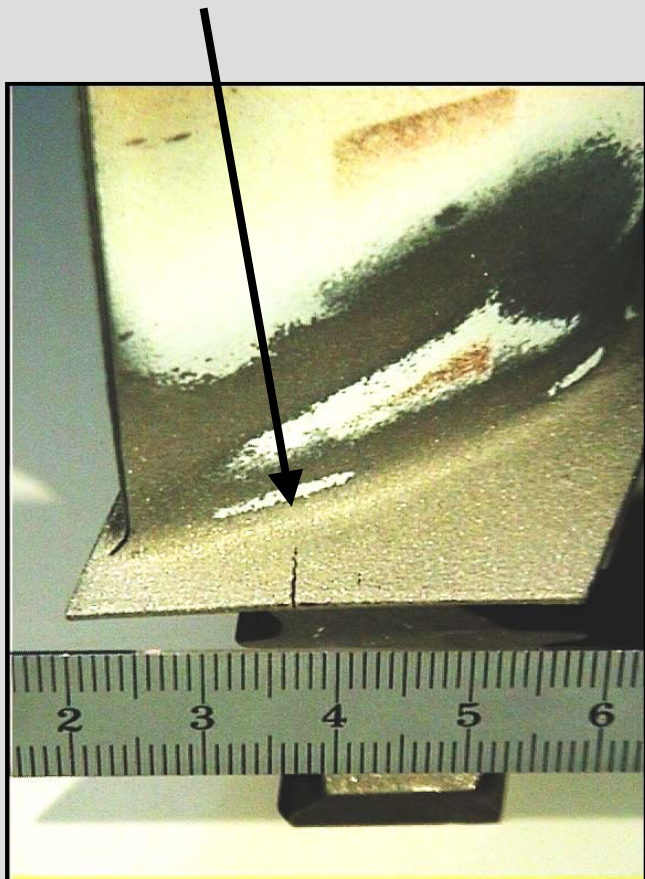
- avoid formation of g.b.

SOLUTION: SX REPAIR OF SX BLADES

LASER METAL FORMING

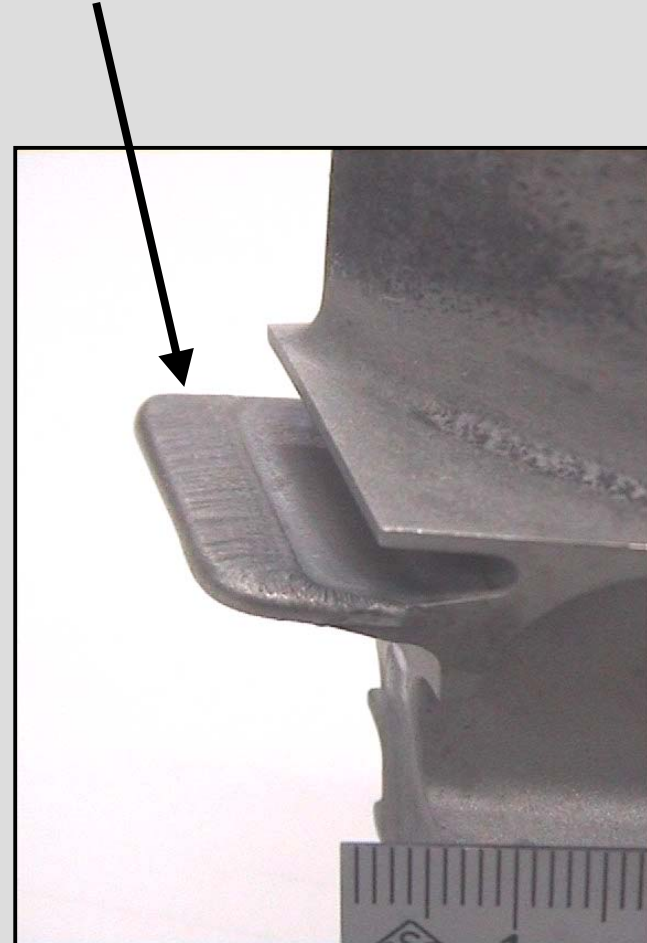
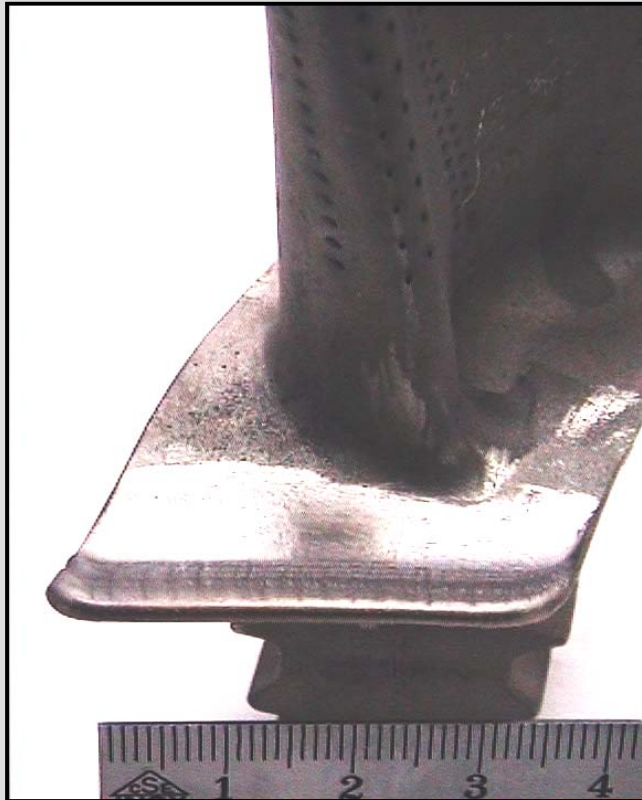


Platform crack due to thermo-mechanical fatigue



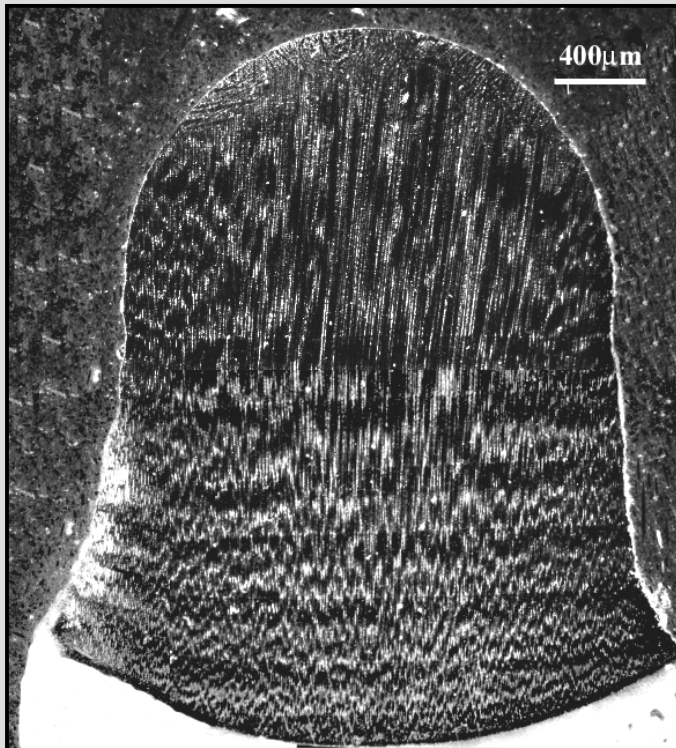
After cutting/polishing

AFTER LASER METAL FORMING

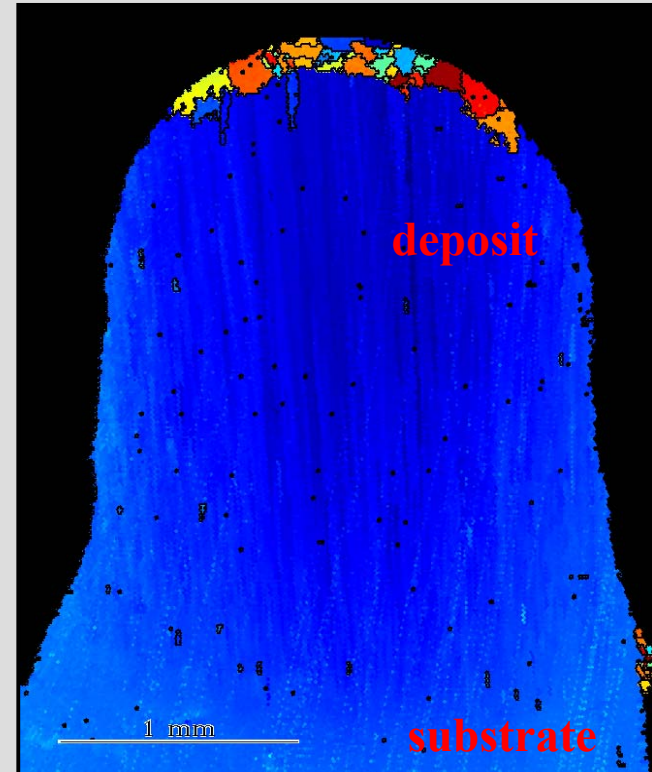


LASER METAL FORMING

Microstructure Analysis



8 laser traces



0°



54°

CONCLUSIONS

- Epitaxial Cladding,
 - Epitaxial Laser Forming,
 - (also Epitaxial Welding),
- can be achieved on SX components. It requires **solidification theory** for a close control of
- macroscopic heat flux (epitaxy) and
 - microstructure development to control CET.

CONCLUSIONS

Today there is industrial activity to transfer the concepts into production for stationary gas turbines and aircraft engines.

Ideal ground for application of long term university research to industrial problems.