Comments on "Bainite formation kinetics in high carbon alloyed steel"

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Scripta Materialia 59 (2008) 1275–1276

Abstract

We examine here the physical aspects and input parameters of a recently published model on the overall transformation kinetics of the bainite reaction in steels.

Keywords: Bainite, Avrami theory, kinetics, steels

A kinetic model for the development of bainite in steels has recently been published which purports to be physically based and uses only two fitting parameters [1]. The purpose of this note is to explore this further.

The kinetic equation apparently based on Avrami theory [2–4], is an adapted version quoted from [5] who claim that the general form is:

$$f = 1 - \exp\{-b(t - t_0)^n\}$$
(1)

where f is the fraction of transformation, b and n are constants and t_0 is the "incubation time" which is measured experimentally. This is in fact a mistake because t_0 is incorrectly taken to be the incubation time detected in an overall transformation measurement [5]. Avrami theory does not lead to the term t_0 in the equation for f, but deals instead with an incubation time τ for each particle.

It is important to emphasise the difference between t_0 and τ . The former is the time corresponding to an arbitrary detectable volume fraction of transformation and hence depends on the precision of the measuring technique. τ is the time at which a particular particle comes into existence; τ does not explicitly appear in the Avrami equation; once established, the latter permits t_0 to be *calculated* in a manner consistent with the detection limit of the experimental technique used. This mistake regarding t_0 is carried through in equation 5 of [1] and means that t_0 is an additional fitting parameter for *each* isothermal experiment; the three t_0 values needed in [1] are presumably from the experimental data in their Fig. 1.

In conducting the analysis to find the fitting parameter K_2 in order to make an empirical adjustment to the activation energy for nucleation, the form of the Avrami type equation used is correctly stated as $f = 1 - \exp\{-k\{T\}t^n\}$ with n = 2, which is inconsistent with their equation 5 which has the term $(t - t_0)^2$.

The growth rate equation used is based on the work of Bosze and Trivedi [6] for the diffusion– controlled growth of parabolic cylinders (plates). The solution leads to a constant lengthening rate but parabolic thickening. Therefore, the volume of a particle in the absence of soft–impingement must vary with $(t - \tau) \times (t - \tau) \times (t - \tau)^{\frac{1}{2}}$. Given the assumed constant nucleation rate, this would lead to $n = 3\frac{1}{2}$ and not the 2 assumed in [1]. Notice that $k\{T\}$ in the nucleation rate analysis is said to be proportional to the product of the nucleation rate and the lengthening rate $\dot{N}_{\alpha} \times v_{0\alpha}$, but this assumes only one–dimensional growth. Plates actually grow in three dimensions.

Further fitting parameters include v' the so-called deviation from the maximum growth rate, the nucleation site density and interfacial energy and the cross-section of the bainite plate which is known to vary with temperature [7].

In summary, it is difficult to accept that this model has predictive capabilities [8] but it may be useful in the quantitative description of known experimental data.

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