

**TEXTURE ANALYSIS OF METALS IN MICRO-VERSION :
THE ELECTRON BACKSCATTER DIFFRACTION TECHNIQUE**

Leo Kestens

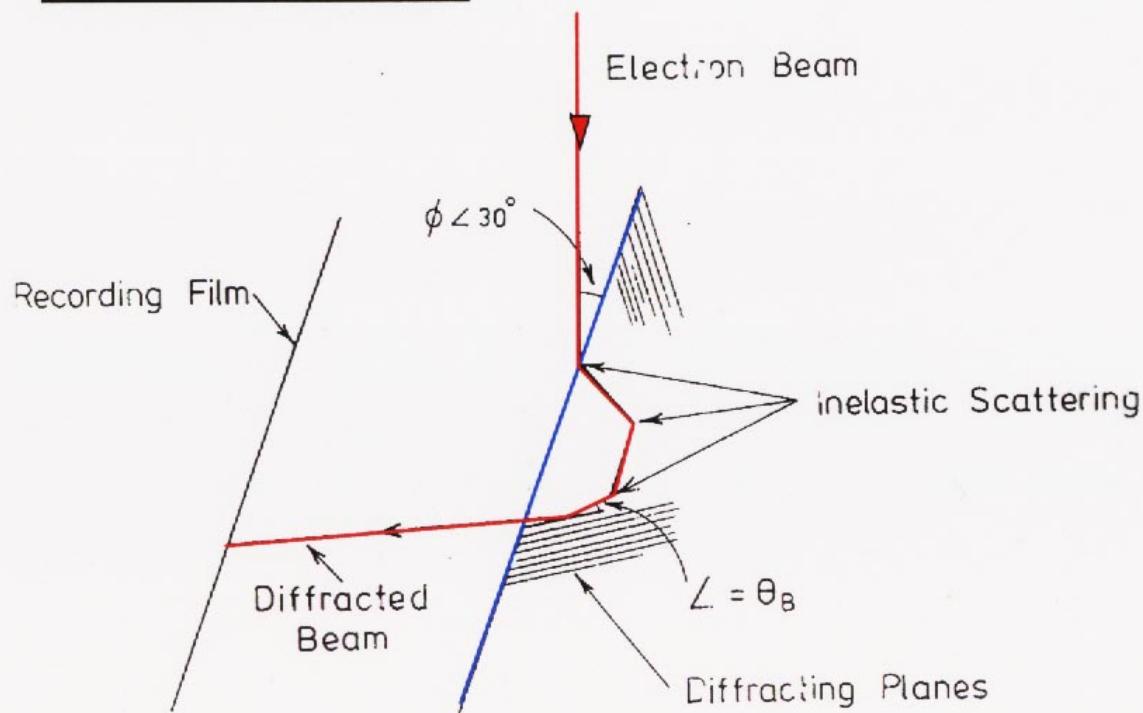
Laboratory for Iron and Steelmaking, Ghent University, Belgium



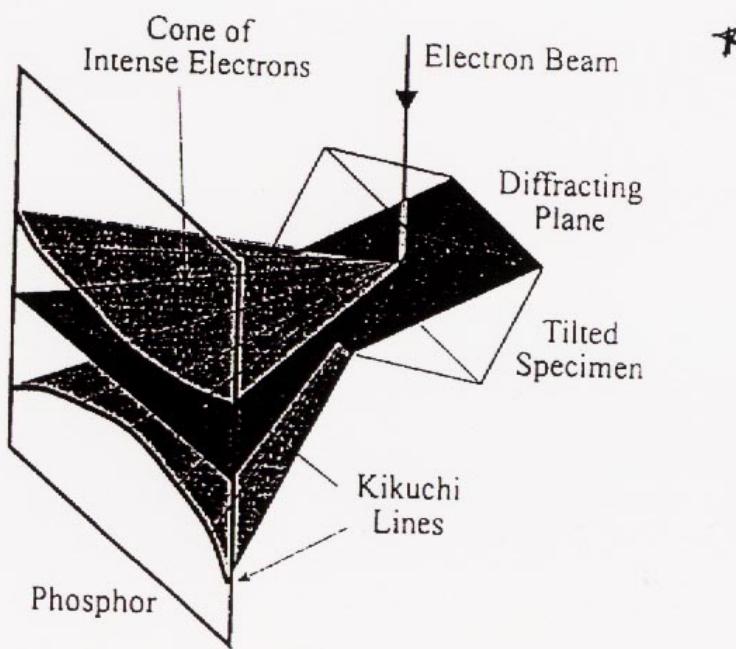
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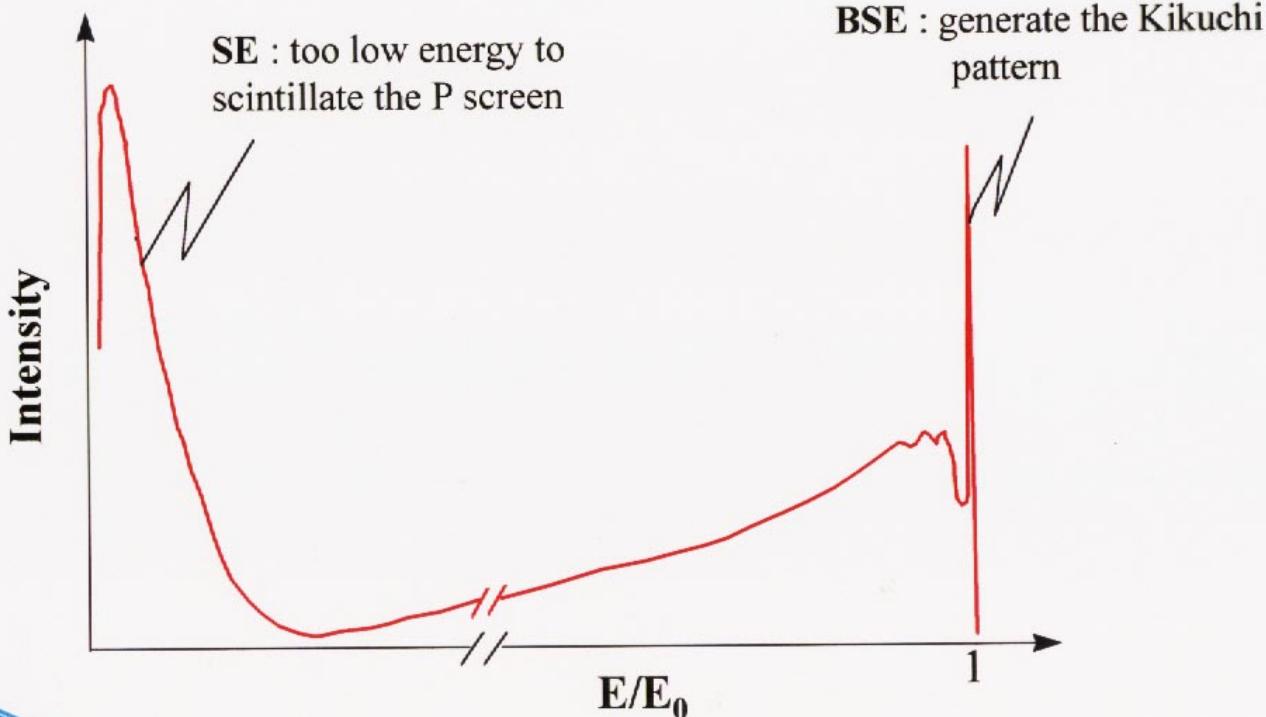
The Kikuchi Pattern

→ Electron BackScattering Diffraction (EBSD)



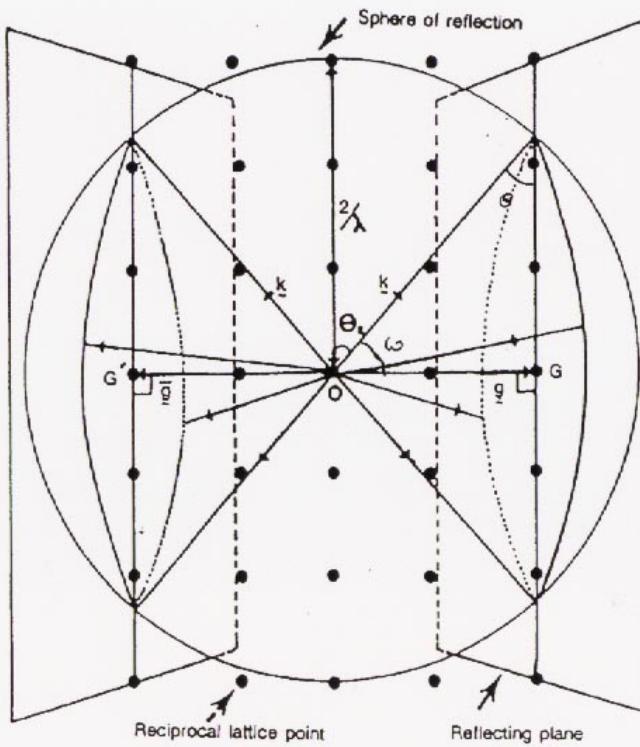
Kikuchi Lines = Image of crystal planes





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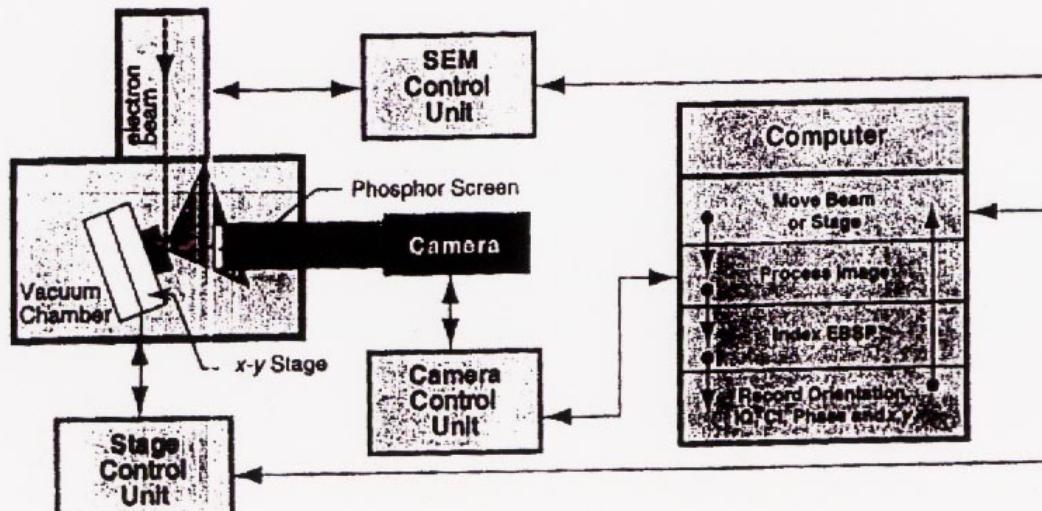
The formation of Kikuchi Lines



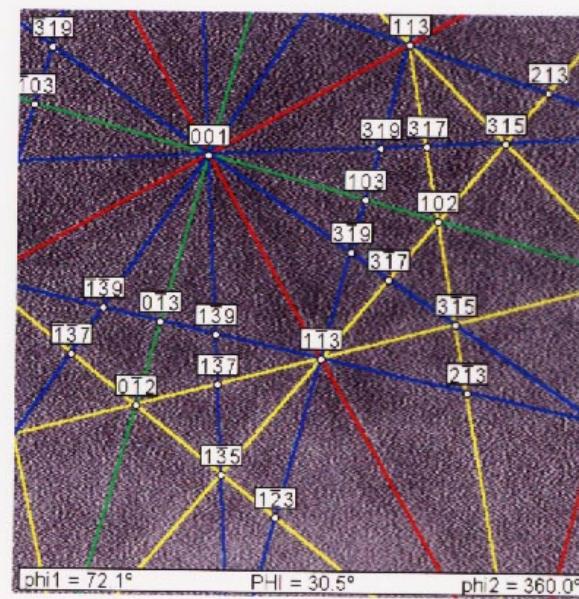
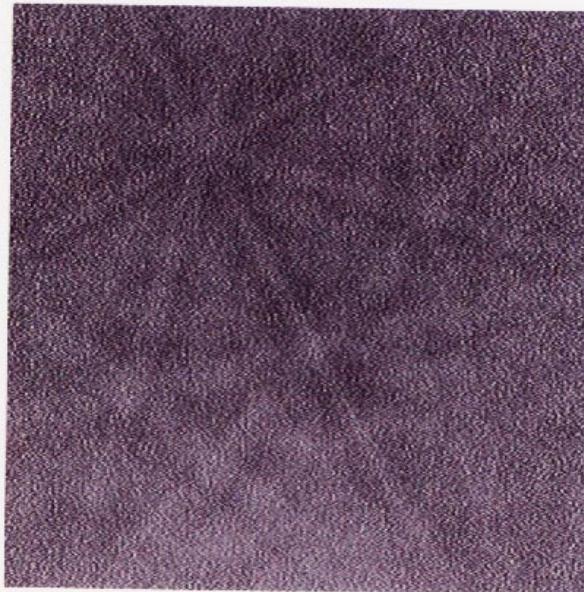
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Orientation Imaging Microscopy

= Automated EBSD acquisition + post processing of data

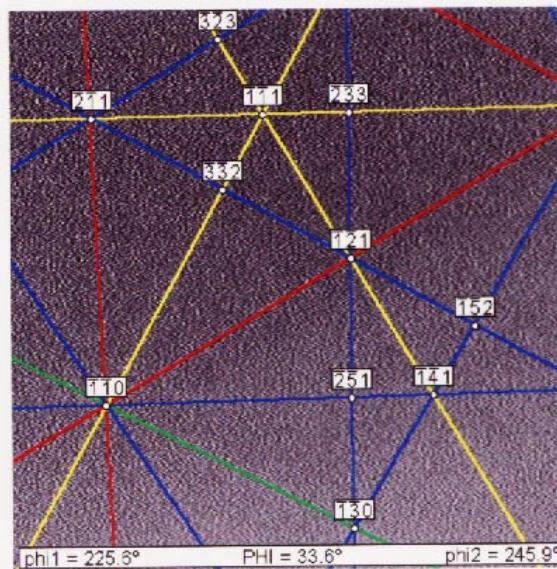
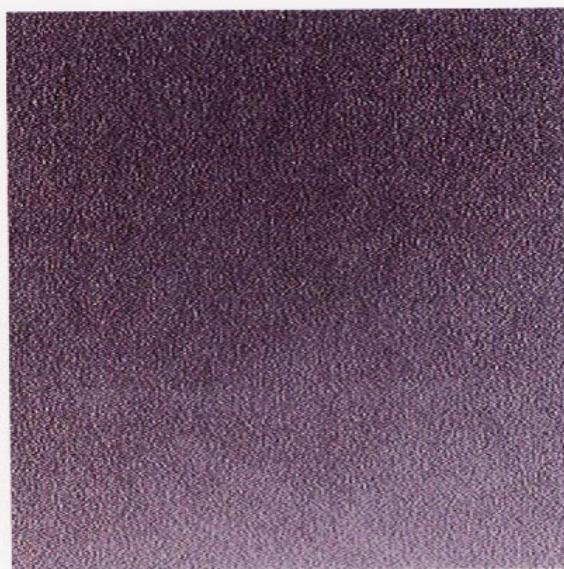


Example of experimental EBSD pattern (BCC steel)



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Example of experimental EBSD pattern
(residual austenite in TRIP steel)

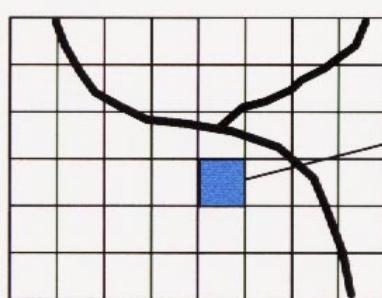


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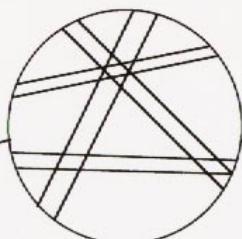
Orientation Imaging Microscopy (OIM)

= automated scanning version of Electron Backscattering Diffraction (EBSD)

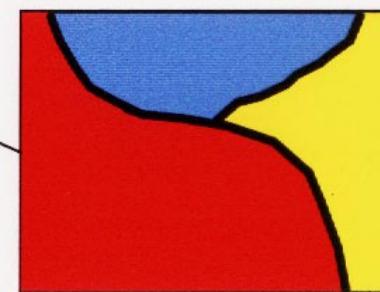
(triple junction)



sample



EBSD
orientation
crystal structure



Image

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OIM image contains crystallographic information of sample surface

- crystal structure (bcc, fcc, hcp)
- crystal orientation
- crystal defects: dislocation substructure
 - residual stresses
 - grain boundaries



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Image Quality Factor (IQ)

- Quantifies the band contrast of the EBSD pattern
- Depends on sample preparation and operation mode
- Also depends on material parameters :
crystal and structural defects !

Confidence Index (CI)

- Quantifies the reliability of the pattern indexing
- $0 < CI < 1$

- $$CI = \frac{V_1 - V_2}{V_{total}}$$
 with V_1 = 1st solution
 V_2 = 2nd solution
 V_{tot} = total number of solution



Operational mode of microscope for OIM

AV = 20 kV

Specimen tilt = 70deg

Spot size = 5-7

Dynamic Focus

(Specimen Tilt Correction)

Hardware parts of OIM system

Phosphor screen + camera

Image processor (frame integration, background correction)

Beam Controller

PC (>200 MHz)



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Acquisition Time

- Quality of Diffraction Pattern
- Sample preparation
- Numbers of Integrated Frames
- Crystal Structure(s) present in sample

E.g.: Steel sample (fully recrystallized, single phase)

$\approx 0.5 - 1.0$ s per pixel

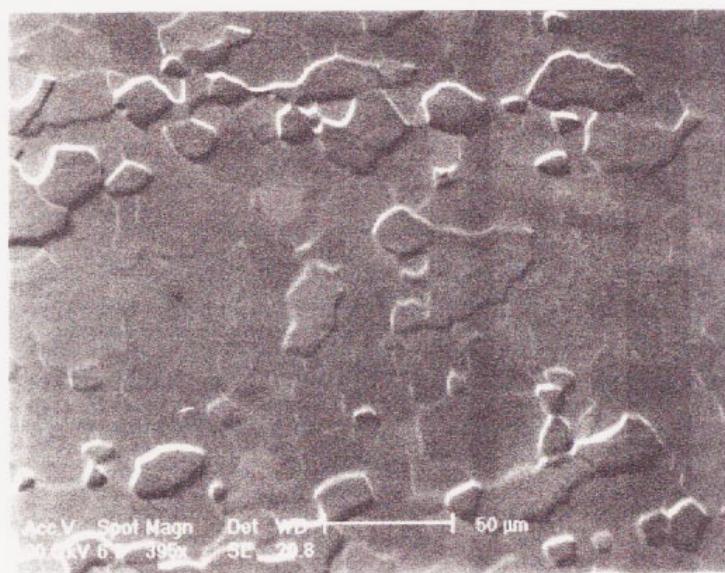
Typical scan ≈ 40.000 pixels $\longrightarrow 6 - 12$ h



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ULC steel (CEIT) : 90% cold rolled + annealed

(ferrite single phase polycrystal structure)



SE image on ESEM XL30

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OIM image of same sample site

Greyscale defined by image quality factor of diffraction pattern

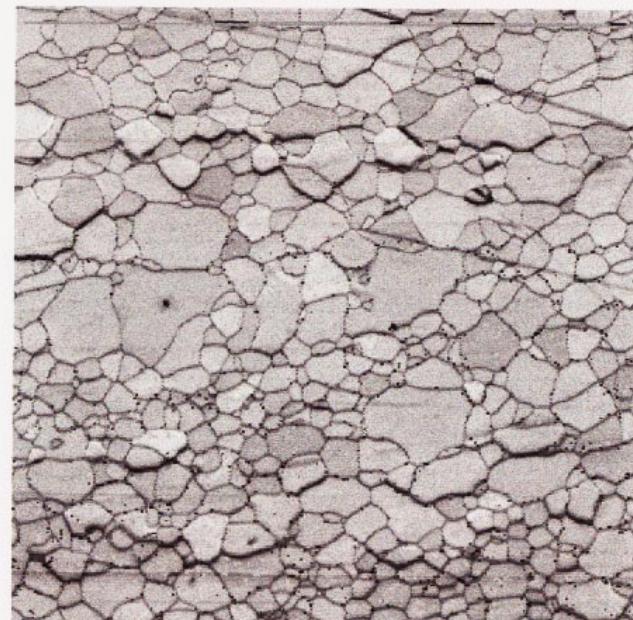


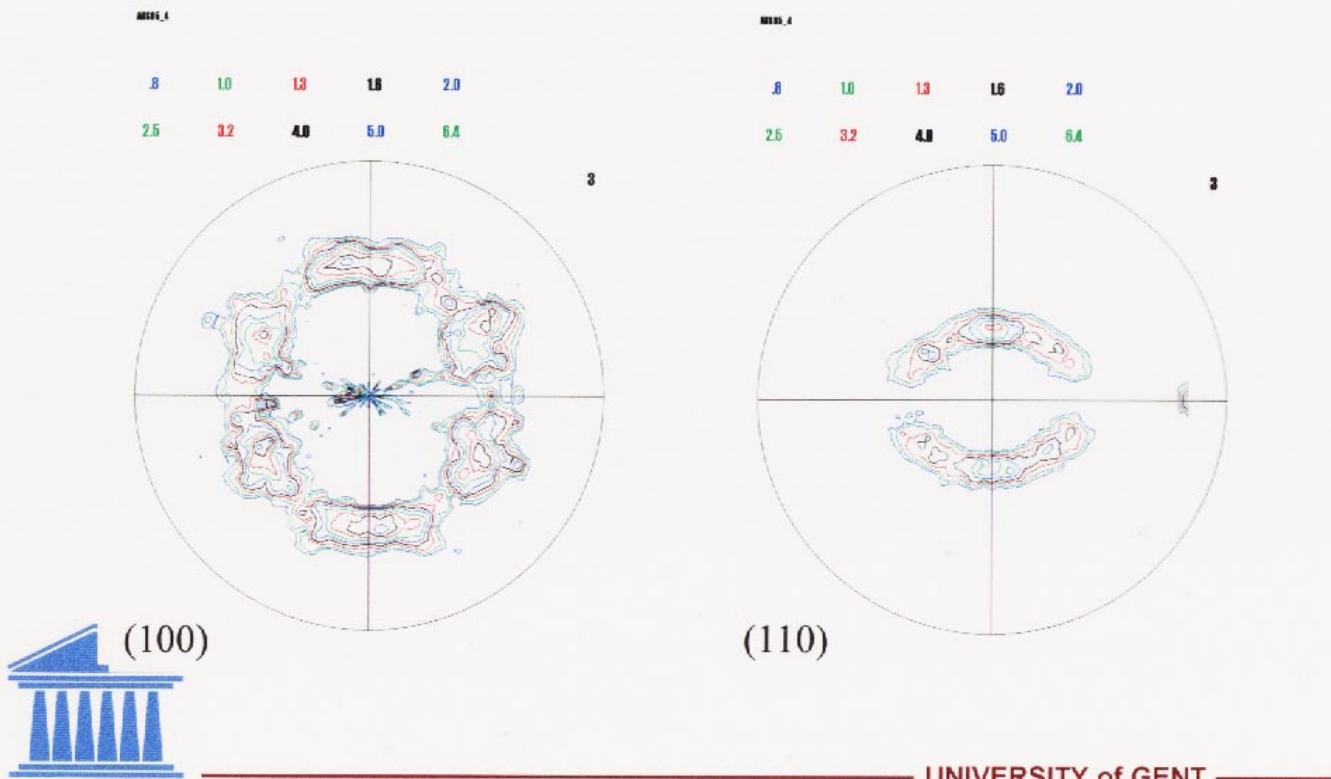
Image Quality
1.74 - 174.20



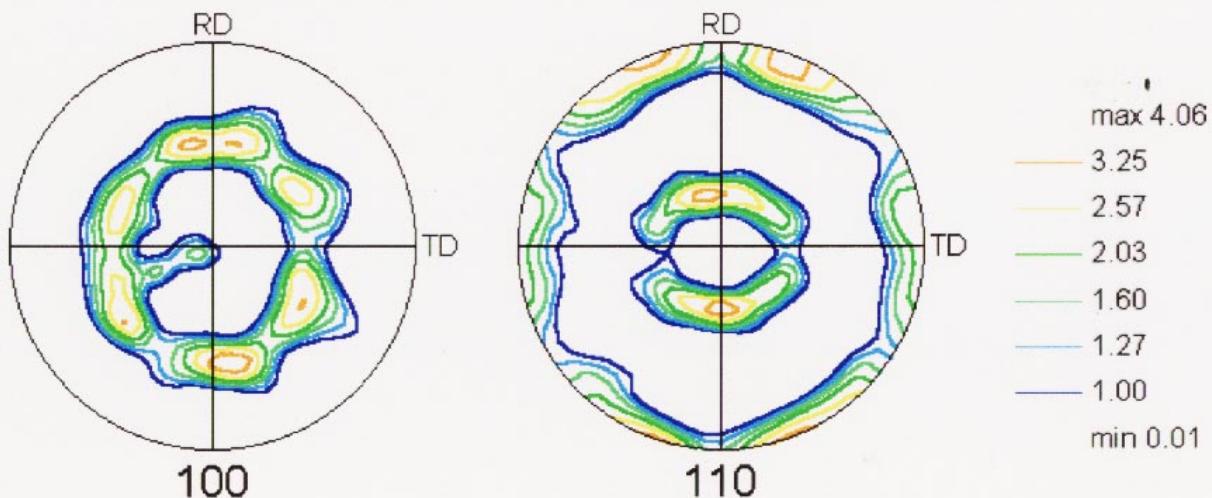
70 μm

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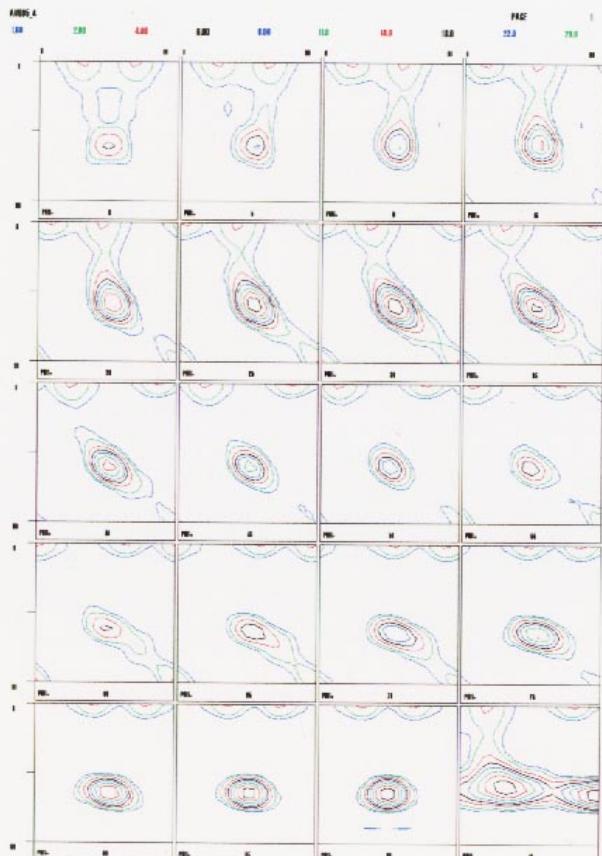
Pole figures obtained by X-ray diffraction



Pole figures obtained by OIM

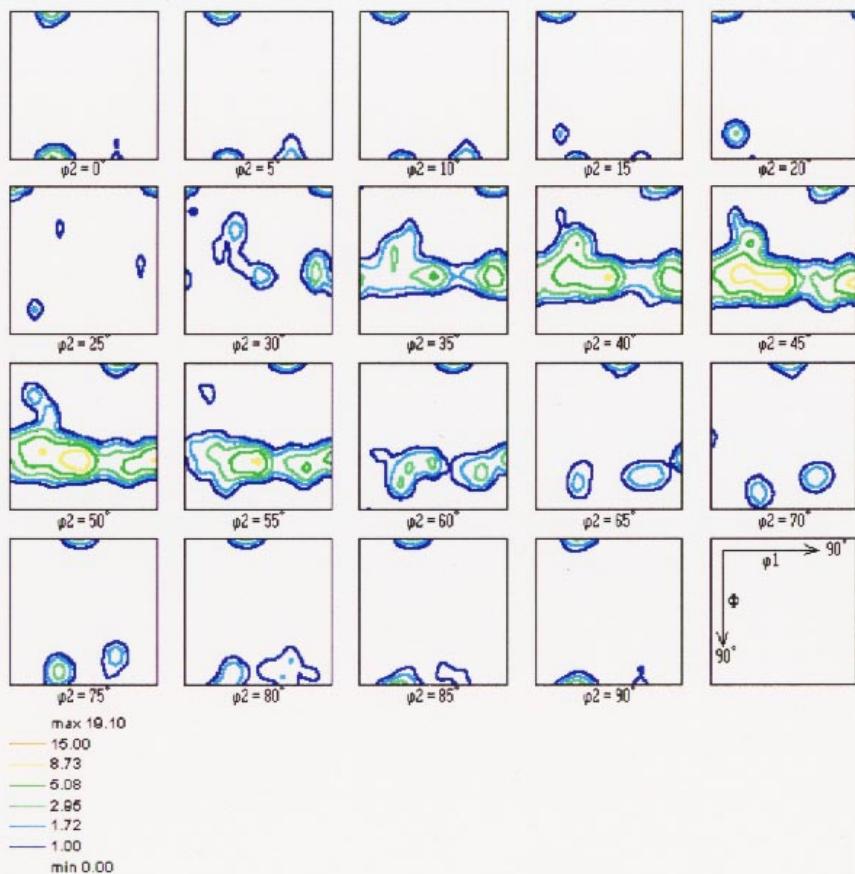


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ODF obtained from X-ray diffraction measurement

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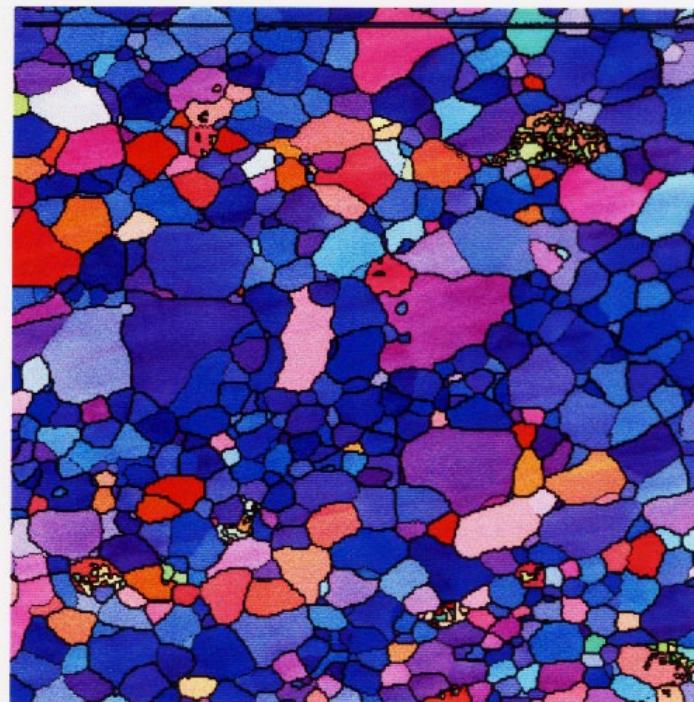
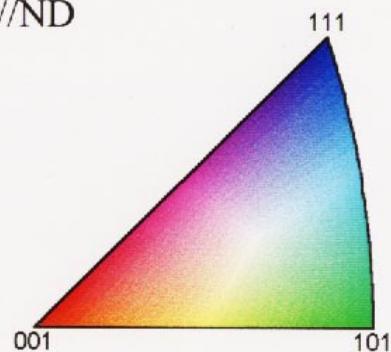
**ODF obtained from
OIM measurement
as calculated by
TSL software**

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OIM image of same sample site

Colors defined by orientation of crystal direction // ND

//ND

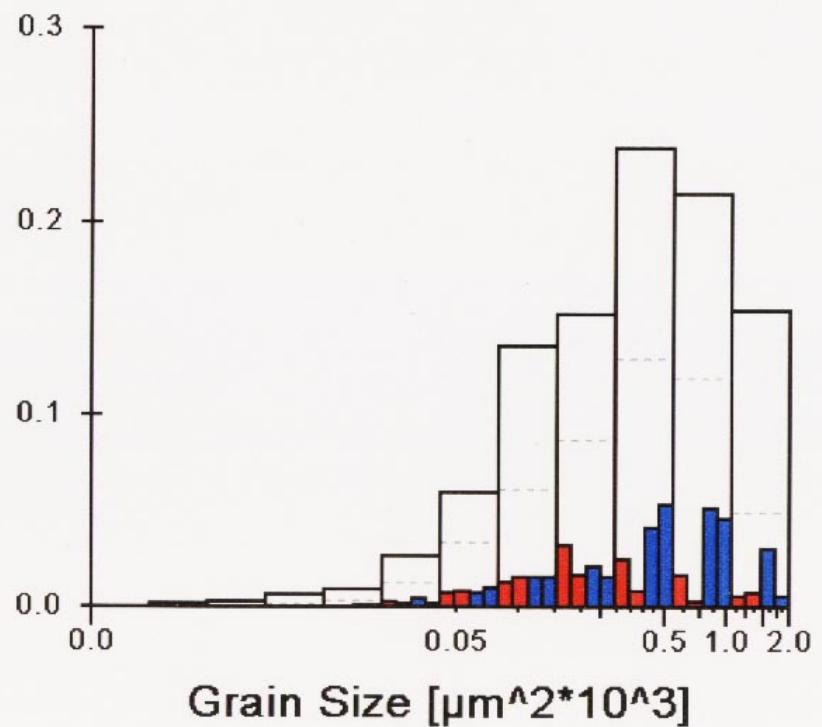


70 μm

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Grain Size Distribution

Area Fraction

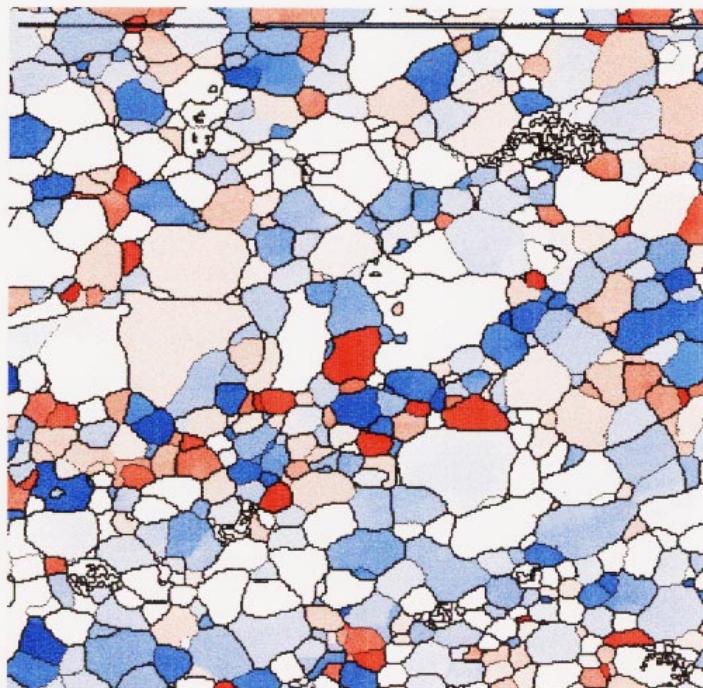


blue = $\{111\}<211>$

red = $\{111\}<110>$

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Presence of Specific Orientations in the Microstructure

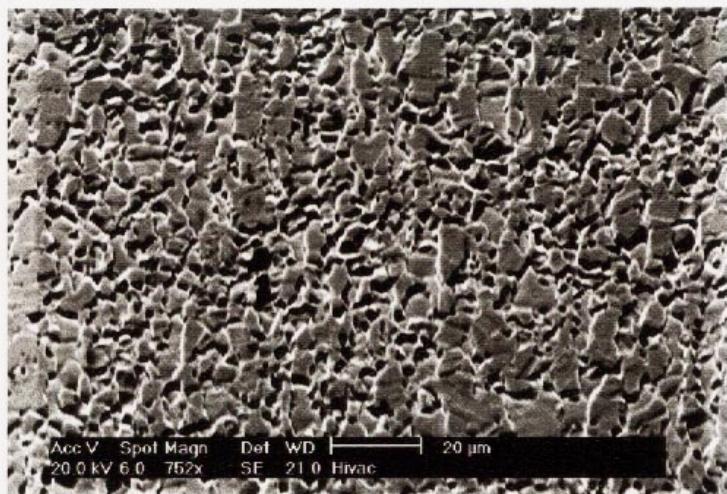


blue = $\{111\} <211>$
(43% volume fraction)

red = $\{111\} <110>$
(27% volume fraction)

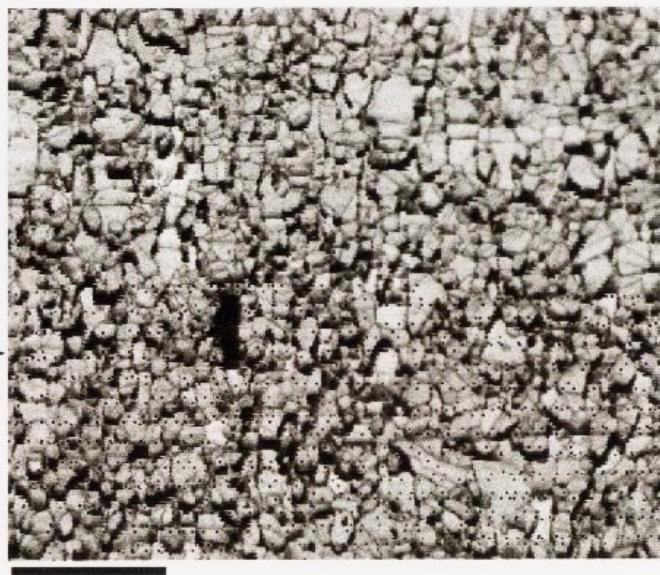
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Duplex Stainless Steel (Ferrite + Austenite)

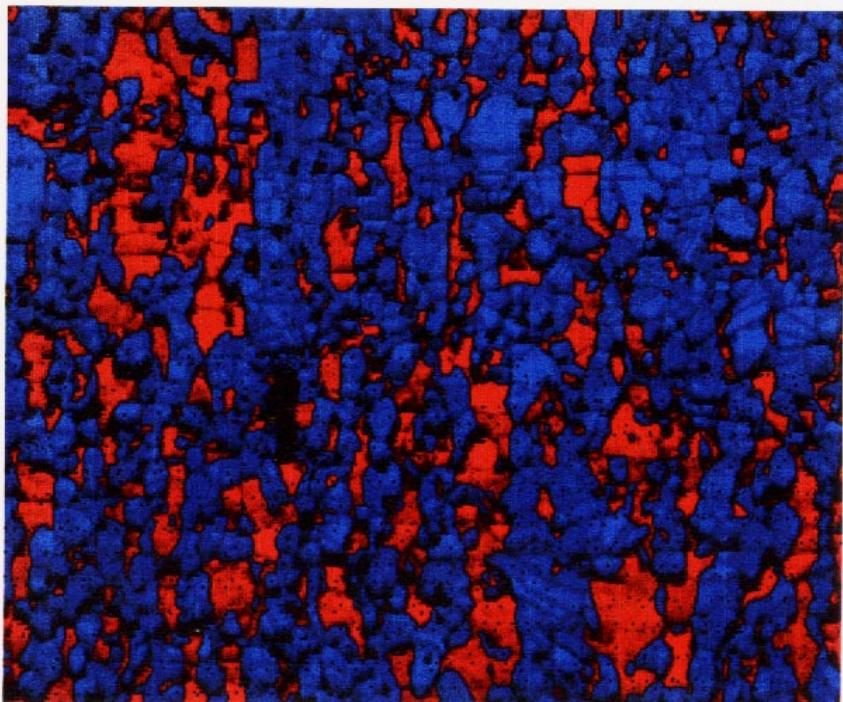


← SE image in SEM

OIM image
Greyscale = Image Quality factor →



Phase Distribution in Duplex Stainless Steel



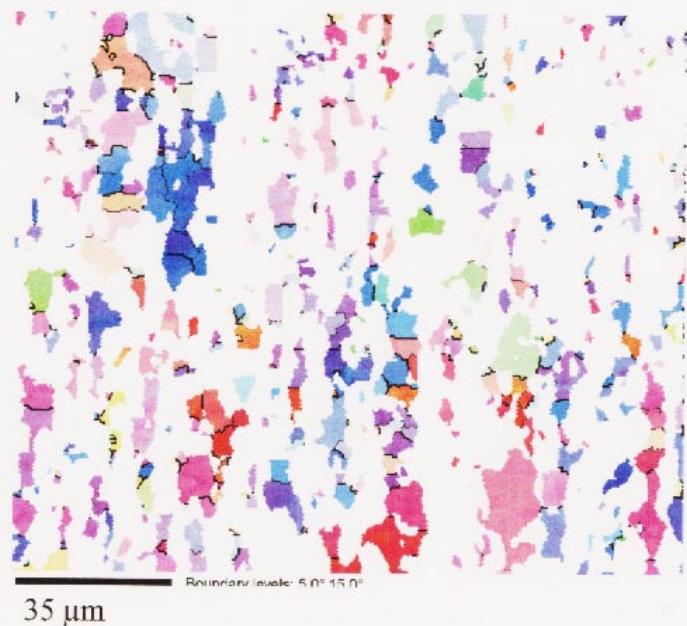
35 µm

blue = austenite
(71% volume fraction)

red = ferrite
(29% volume fraction)

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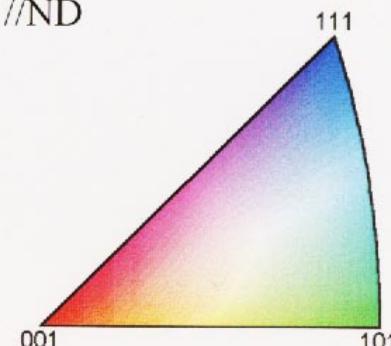
Texture of Ferrite Phase



— $5^\circ < \omega < 15^\circ$

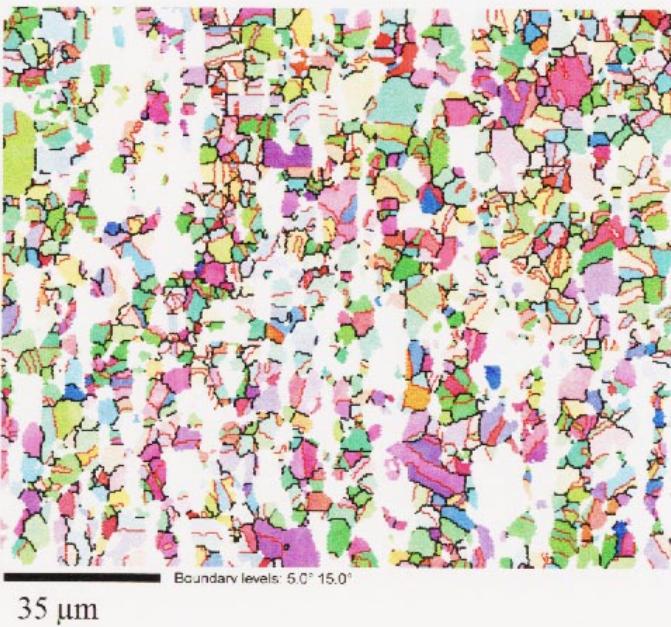
— $15^\circ < \omega < 180^\circ$

//ND

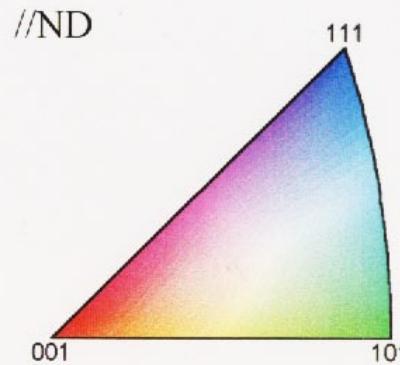


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Texture of Austenite Phase



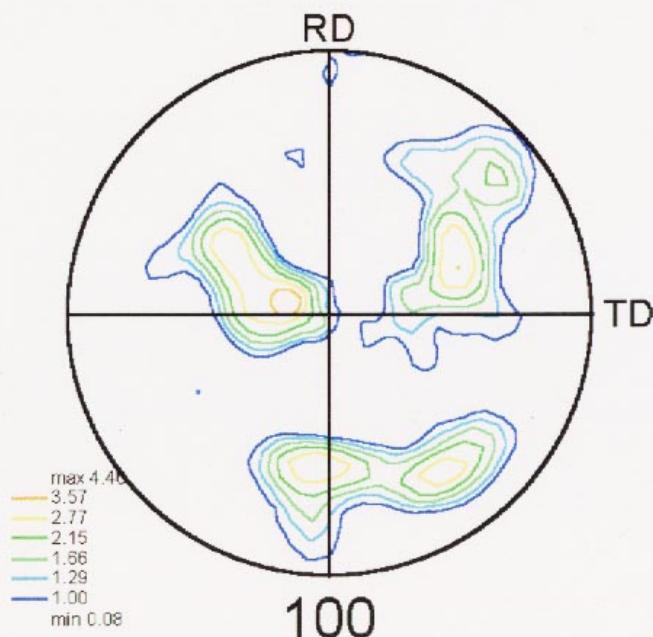
- $5^\circ < \omega < 15^\circ$
- $15^\circ < \omega < 180^\circ$
- twin misorientation



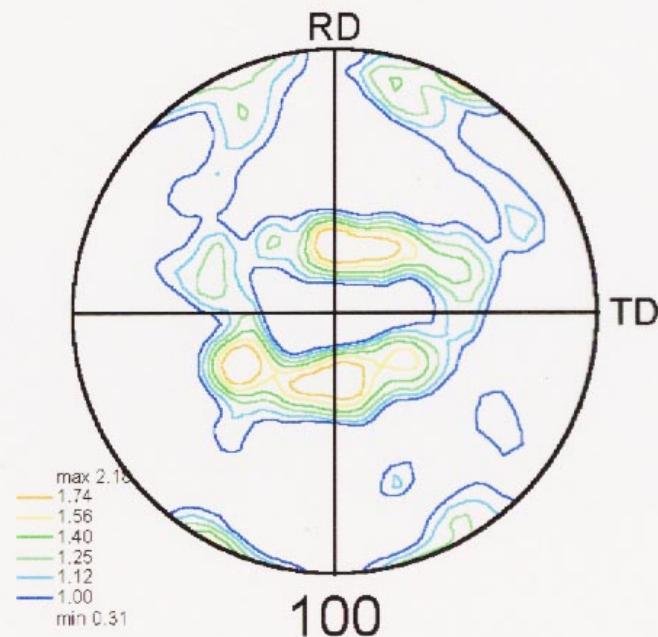
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Comparison of Austenite and Ferrite Texture

Ferrite



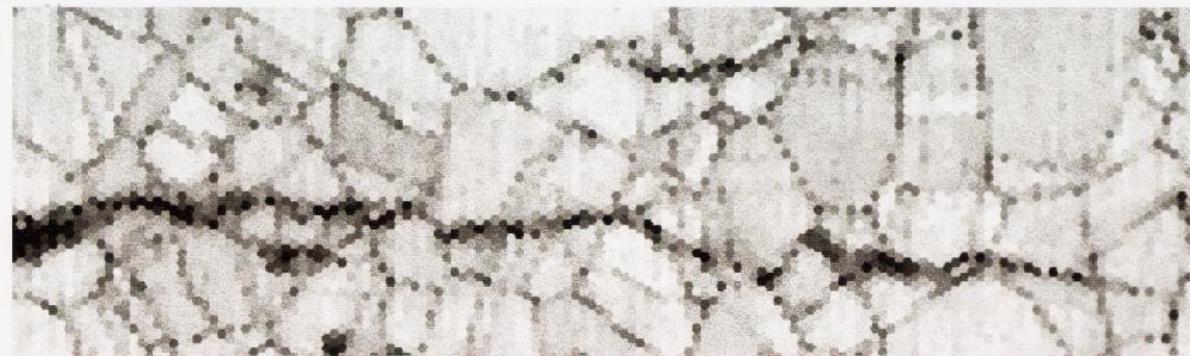
Austenite



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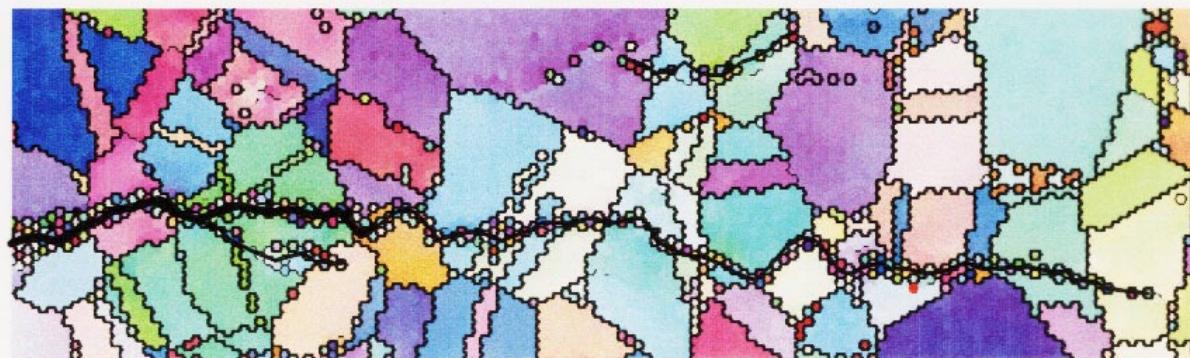
Microcrack in Aluminium

(courtesy: TexSem ltd.)



■

10.00 μm = 5 steps

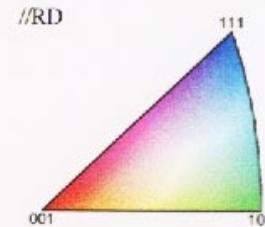
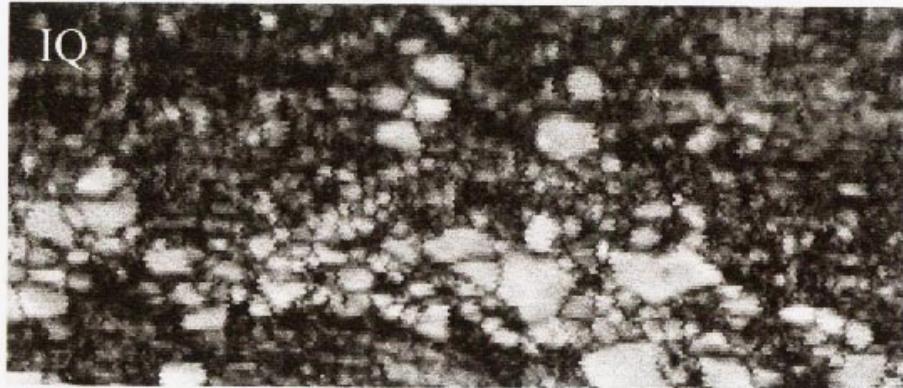


■

10.00 μm = 5 steps

f GENT

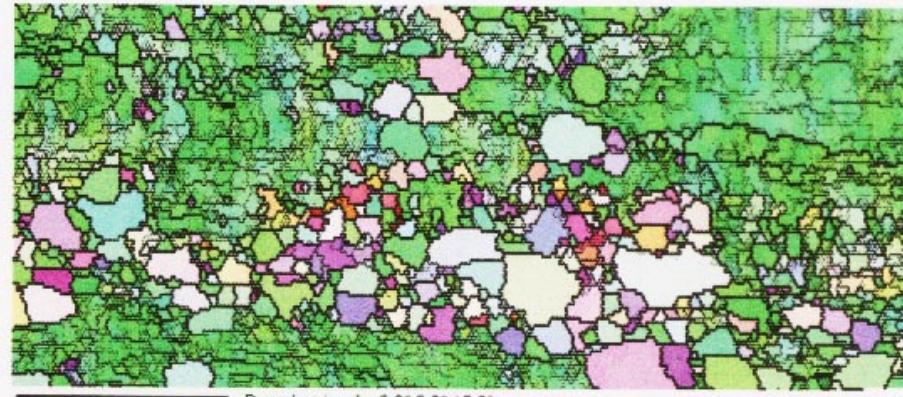
Partially recrystallized steel sample



20 μm

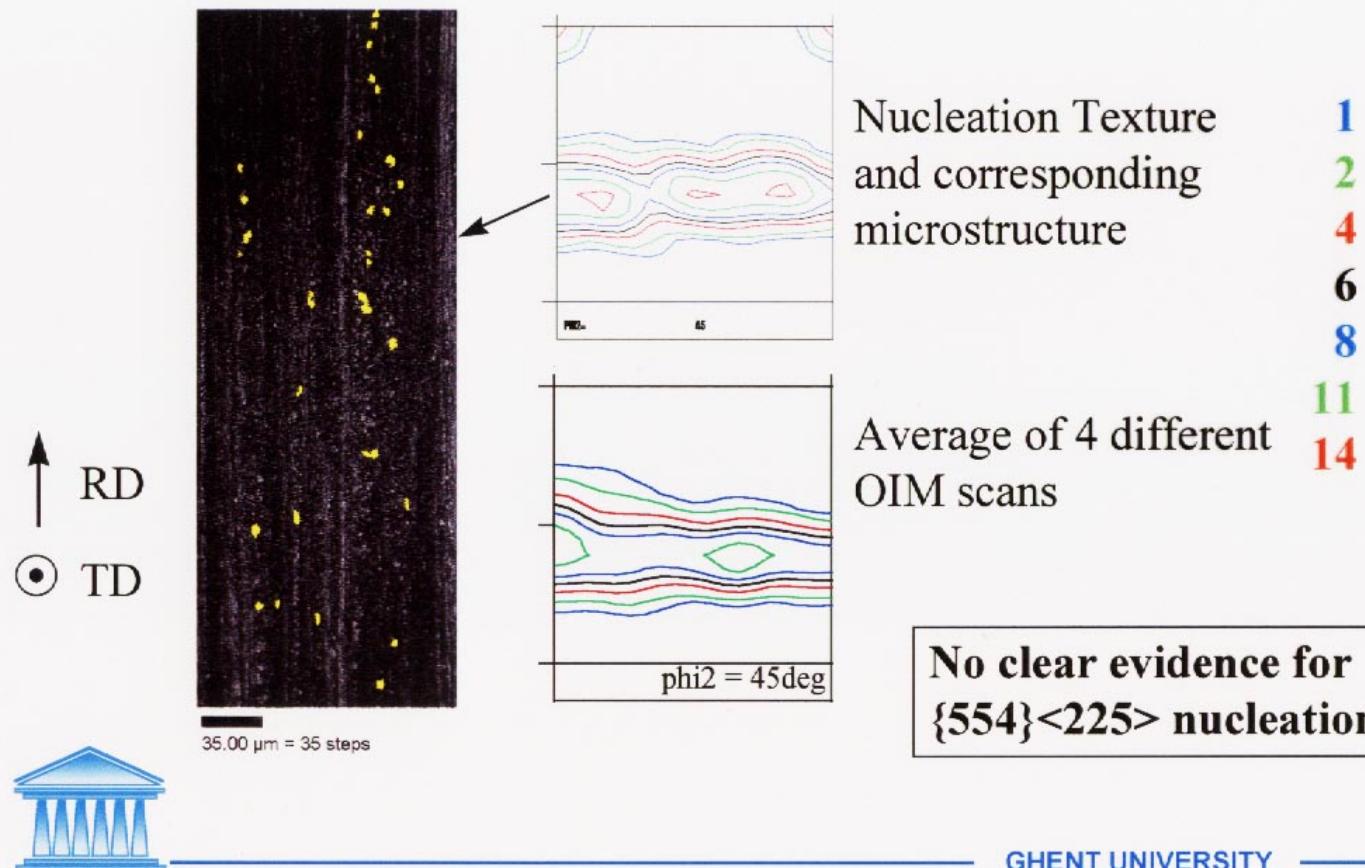
RD

ND



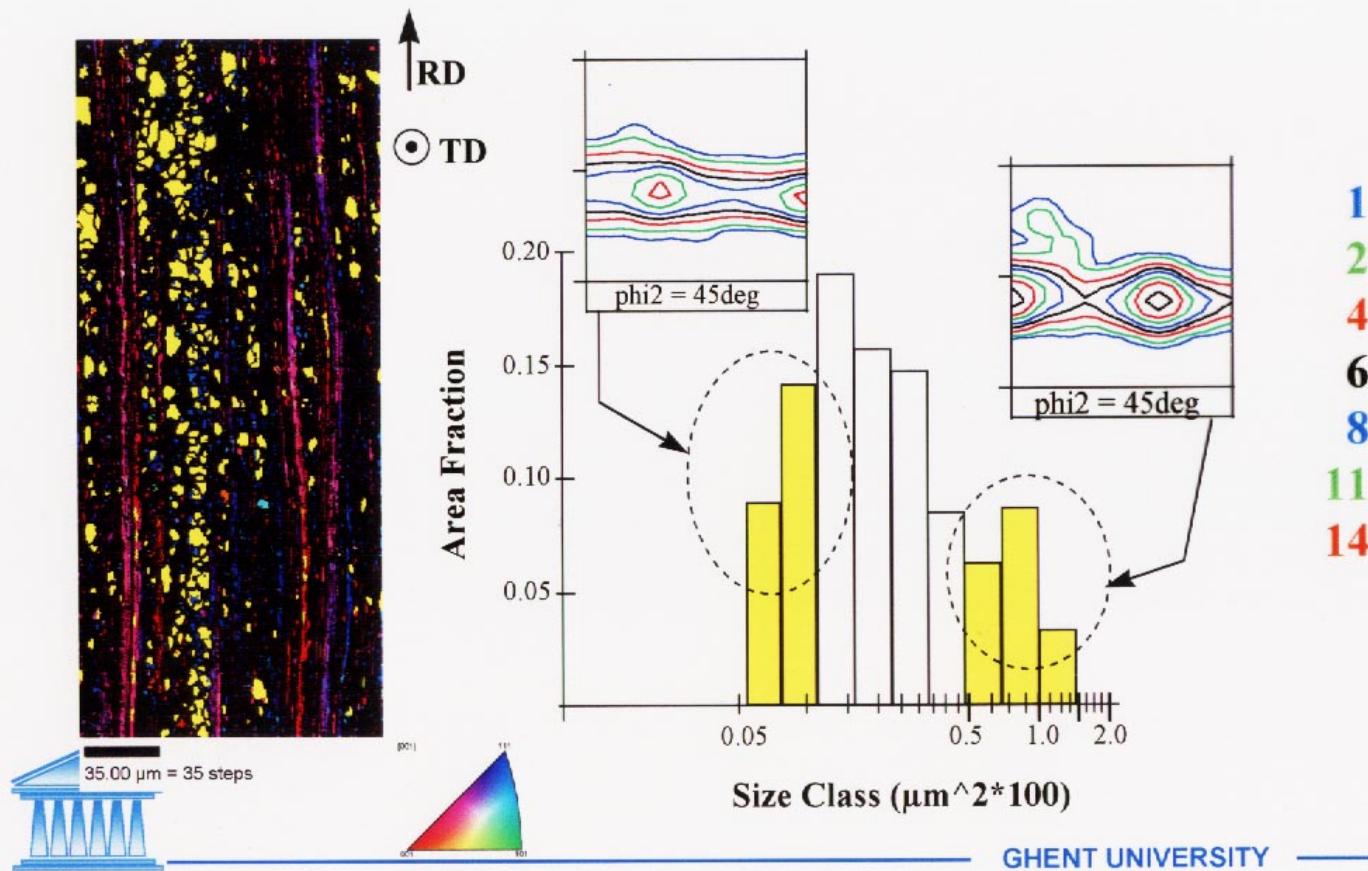
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OIM scan of Early Nucleation Structure



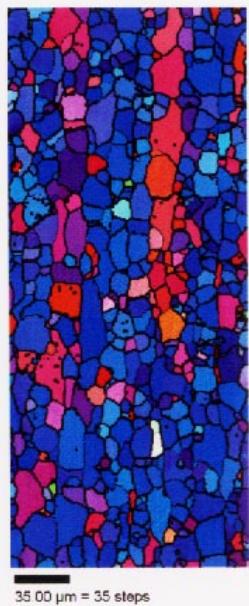
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Intermediate Recrystallization Stage

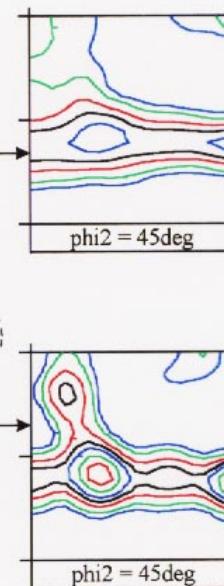
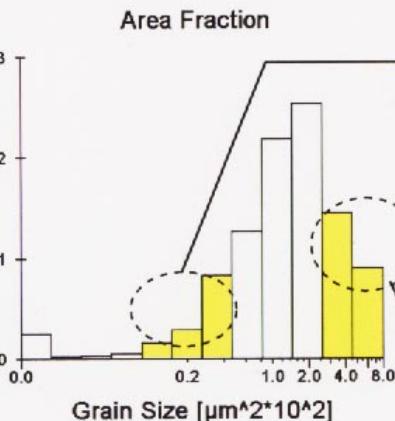


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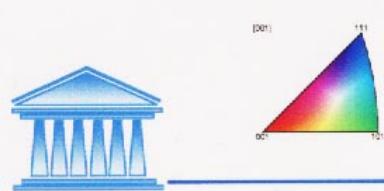
OIM scan of Fully Recrystallized Structure Quenched from 720°C



↑ RD
● TD

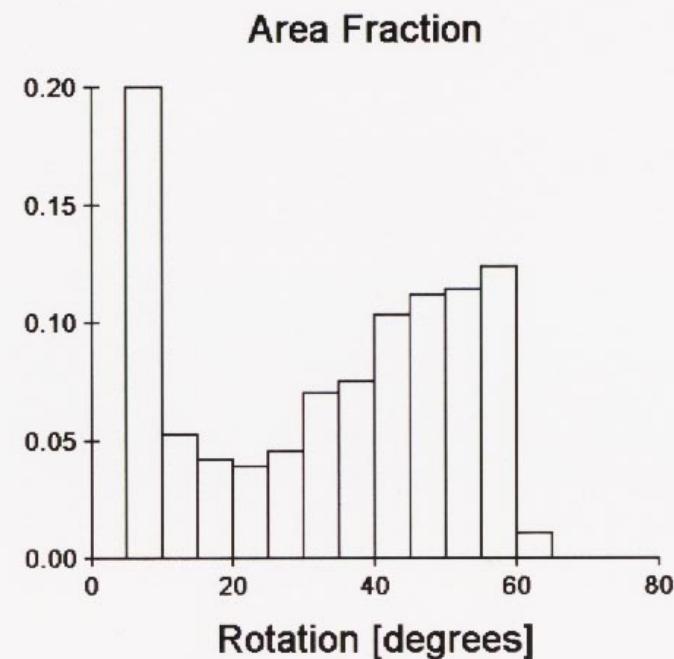
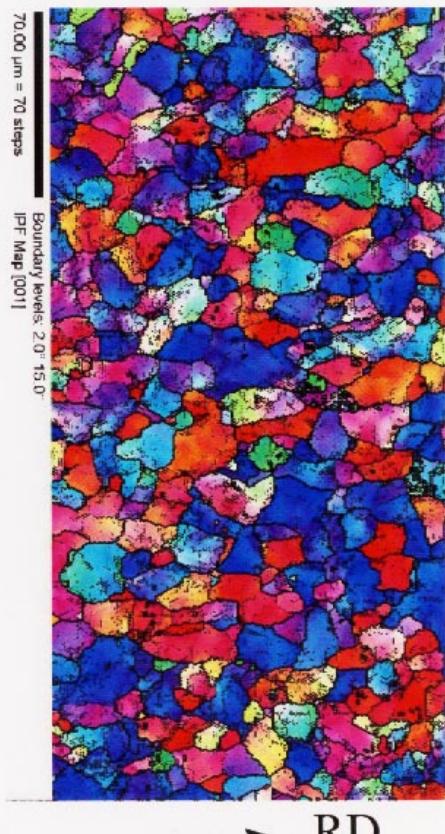


1
2
4
6
8
11
14



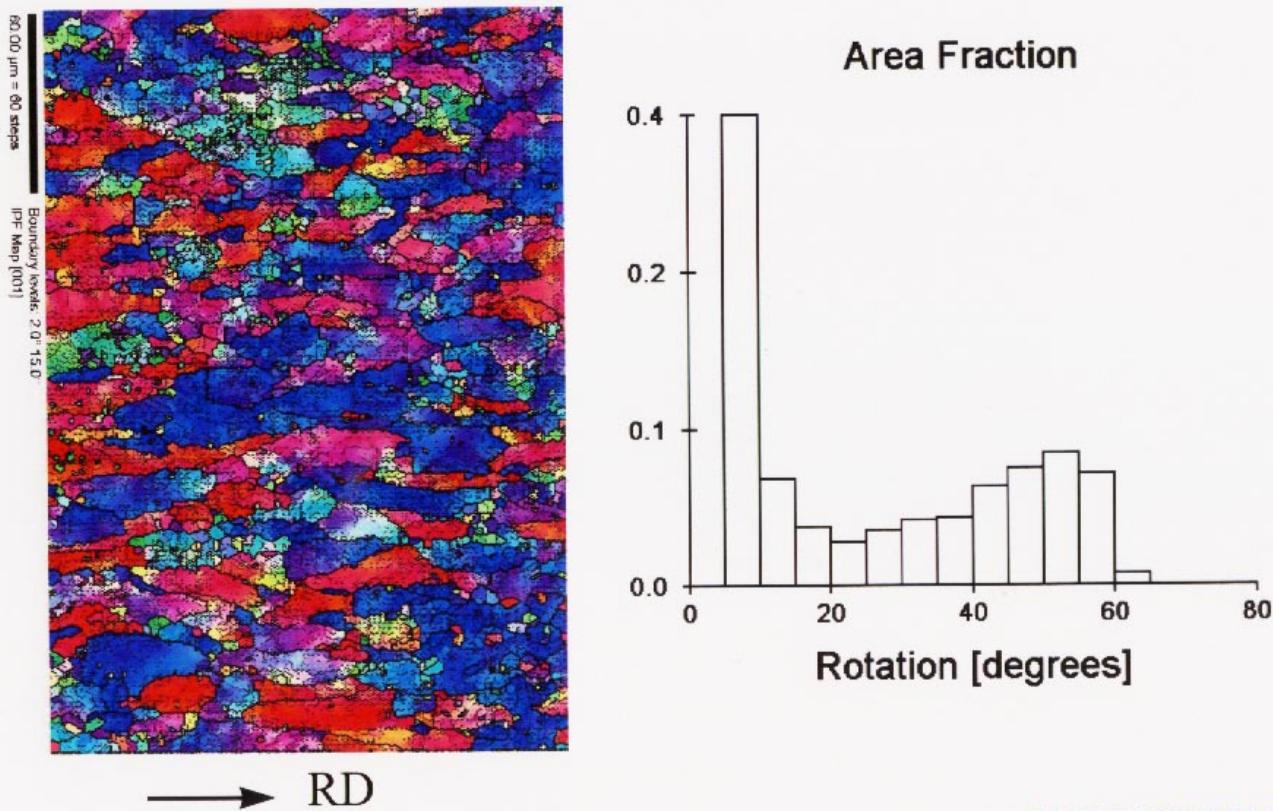
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Deformation (sub-)structure : ~~45%~~¹⁰ cold rolled steel sheet



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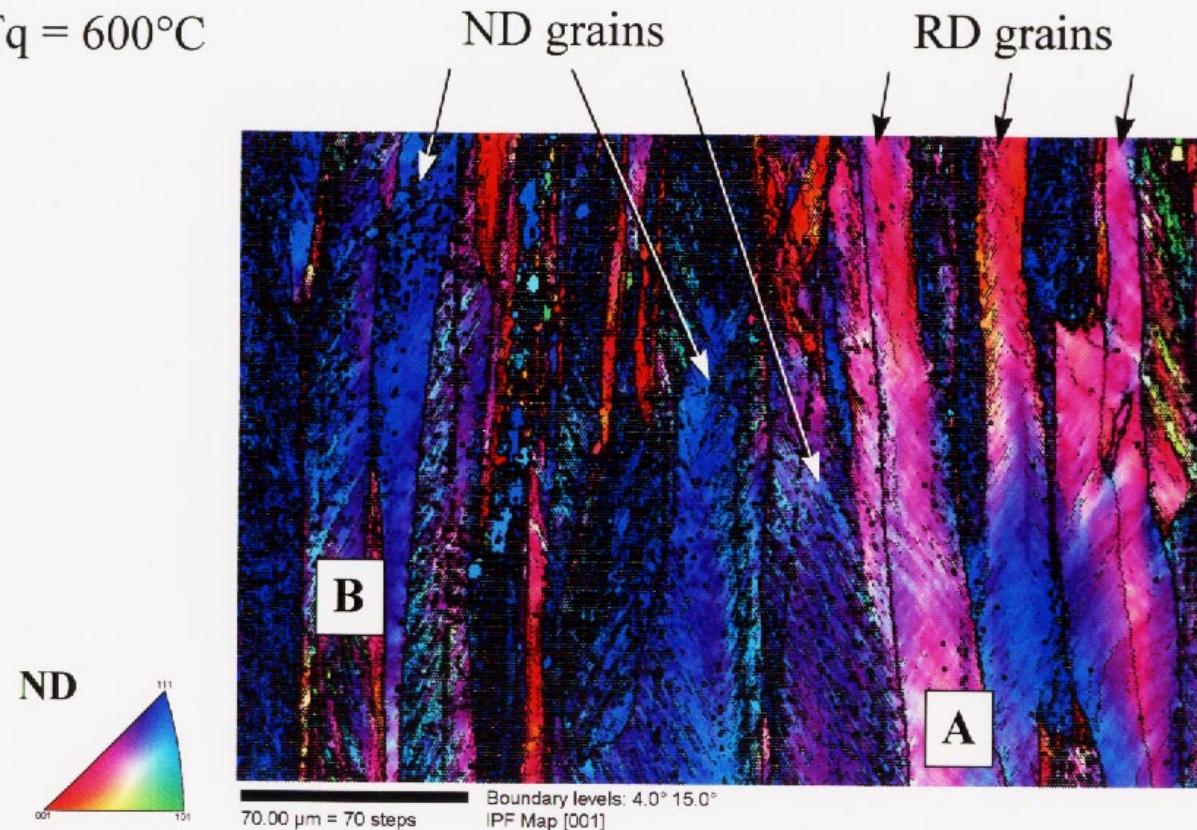
Deformation (sub-)structure : 45% cold rolled steel sheet



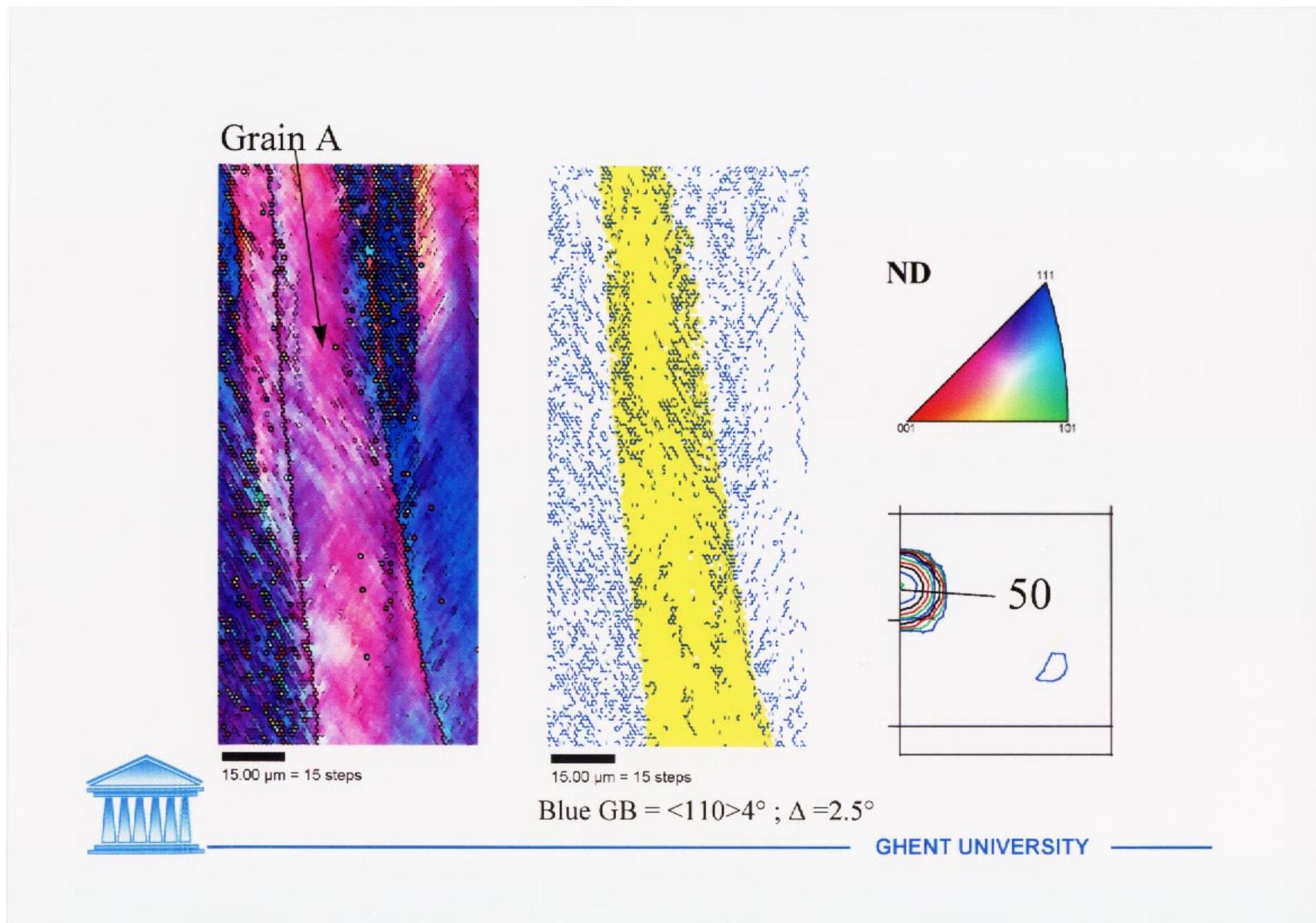
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Observation of in-grain shear bands in 70% CR + recovered structure

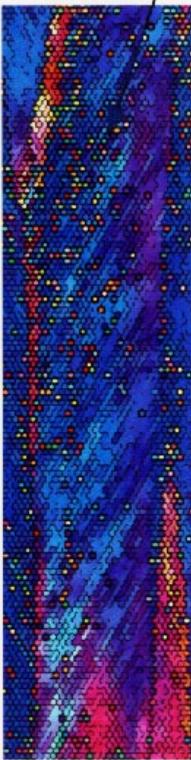
Tq = 600°C



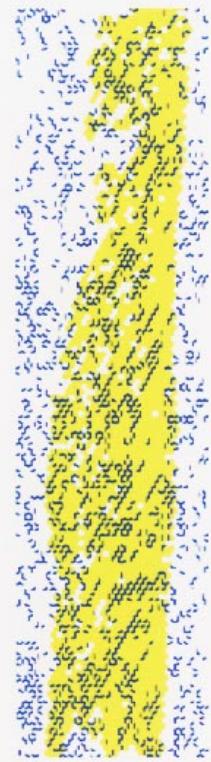
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Grain B

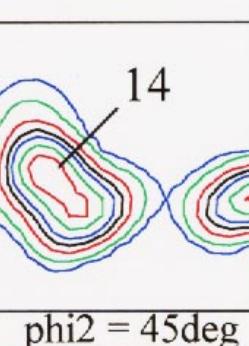
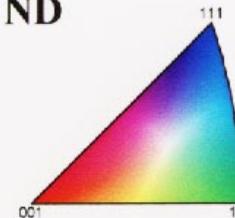


5.00 $\mu\text{m} = 5$ steps



5.00 $\mu\text{m} = 5$ steps

ND



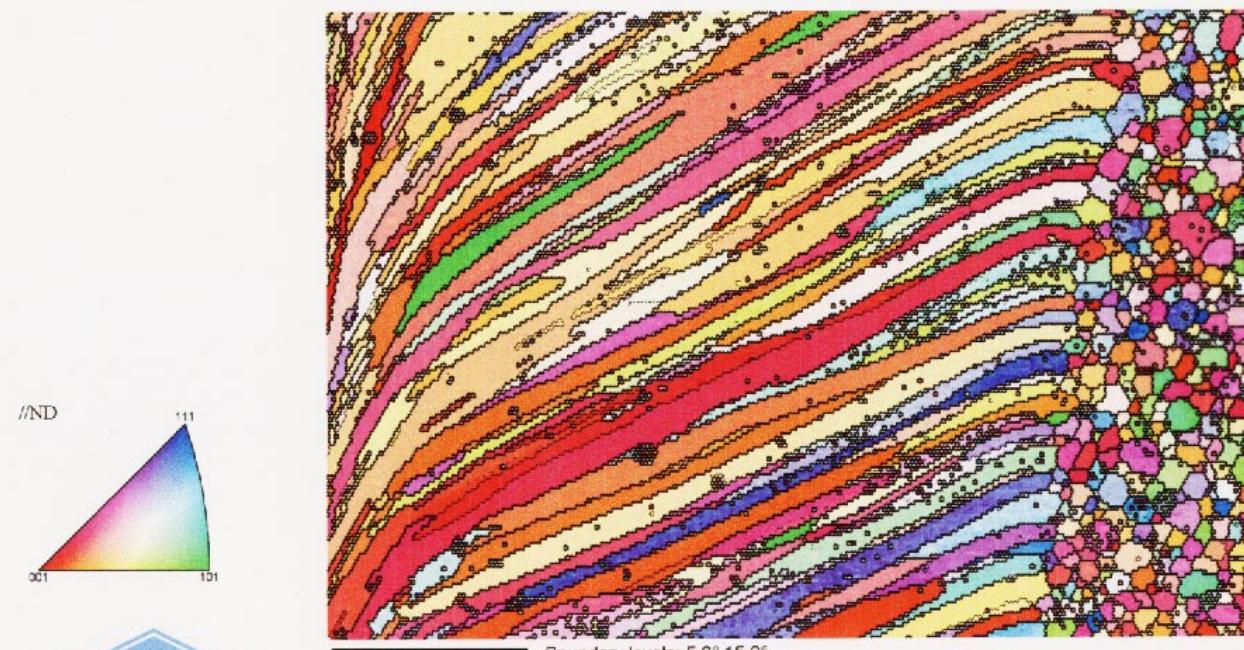
$\text{phi}2 = 45\text{deg}$



Blue GBs = $<110>6\text{deg}$
 $\Delta = 6\text{deg}$

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Electron Beam Weld in Beryllium



(courtesy: TexSem Ltd.)



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Conclusion

- Originally the *Electron BackScattering Diffraction* technique was mainly used for local orientation studies in texture related research.
- *Orientation Imaging Microscopy* has fully developed into a new branch of microscopy.
- The main advantages of this technique are:
 - ☺ allows to produce microscope images based on crystallographic contrast
 - ☺ wide variety of materials
 - ☺ relatively friendly operation mode.
- Thus, potential of scanning electron microscopy is extend into a field which was hitherto mainly accessible by X-ray techniques.



ESEM studiedag

ORIENTATION IMAGING MICROSCOPY

Leo Kestens

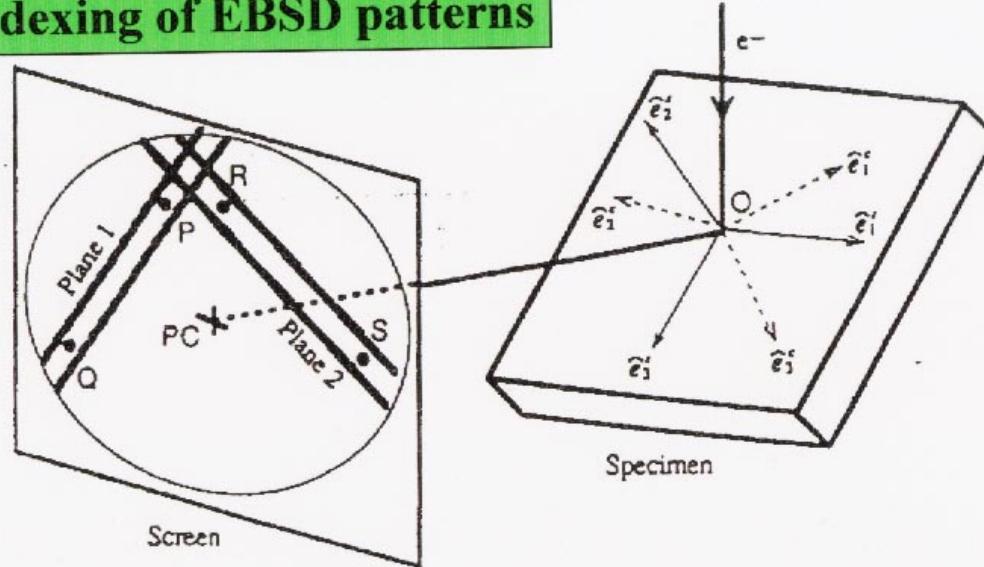
*Laboratorium voor Algemene Metallurgie, Siderurgie
en Fysische Metaalkunde, Universiteit Gent*

14/10/1998



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Indexing of EBSD patterns

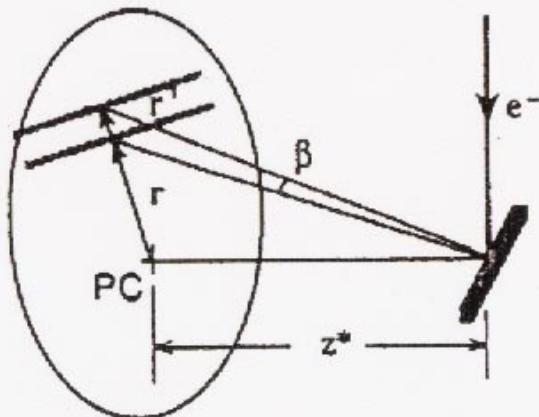


Interplanar angle between planes: $\cos \gamma = |\bar{n}_1 \cdot \bar{n}_2|$

$$\bar{n}_1 = \frac{\overline{OP}x\overline{OQ}}{|\overline{OP}x\overline{OQ}|} \quad \bar{n}_2 = \frac{\overline{OR}x\overline{OS}}{|\overline{OR}x\overline{OS}|}$$

$$\cos \gamma = \frac{h_1h_2 + k_1k_2 + l_1l_2}{(h_1^2 + k_1^2 + l_1^2)^{1/2}(h_2^2 + k_2^2 + l_2^2)^{1/2}}$$

Bandwidth angle β



$$\beta = \tan^{-1}\left(\frac{r'}{z^*}\right) - \tan^{-1}\left(\frac{r}{z^*}\right)$$

$$\beta = 2 \sin^{-1}\left(\frac{\lambda}{2d_{hkl}}\right)$$

Values for γ and β are compared with theoretical values



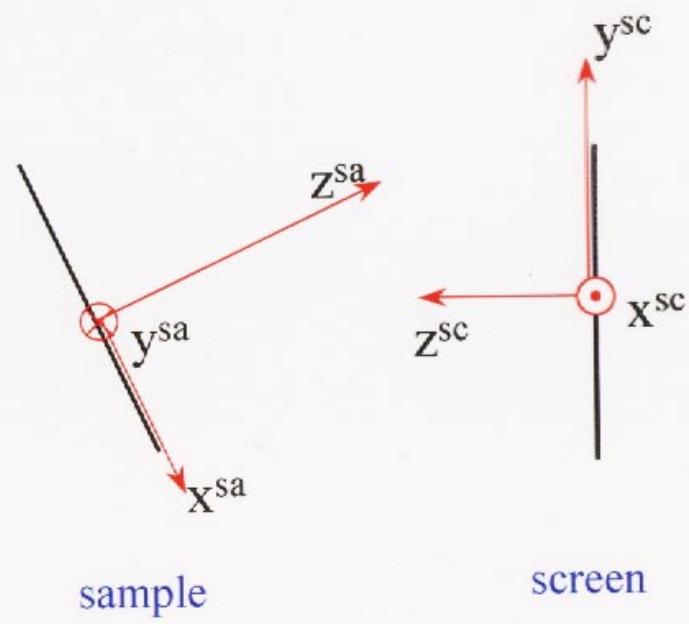
selection is made on basis of closest match



Calculation of Crystal Orientation

Following reference systems are involved:

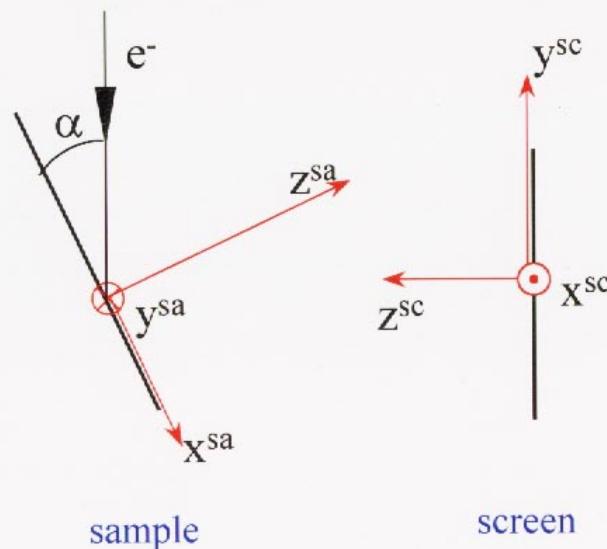
1. Crystal reference system: \bar{e}_i^c
($<100>$ axes for a cubic crystal)
2. Sample reference system: \bar{e}_i^{sa}
(RD, TD, ND axes for rolled sheet)
3. Screen reference system: \bar{e}_i^{sc}
4. Pattern reference system \bar{e}_i^p



Transformation from pattern to screen coordinates:

$$g'_{ij} = \bar{e}_i^p \cdot \bar{e}_j^{sc}$$

Transformation from screen to sample coordinates: $g''_{ij} = \bar{e}_i^{sc} \cdot \bar{e}_j^{sa}$



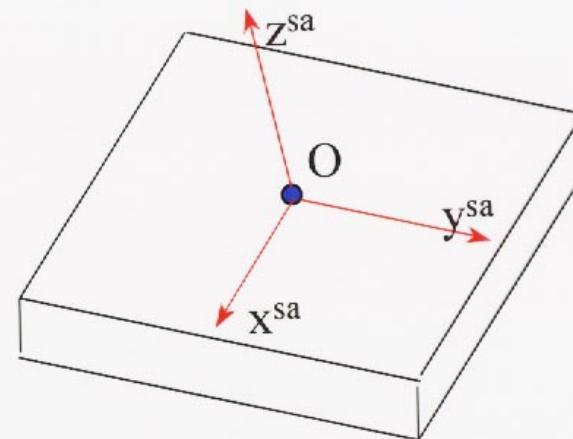
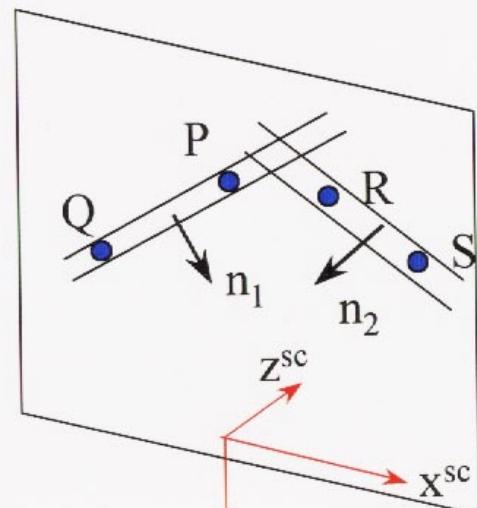
$$\bar{e}_1^{sc} = -\bar{e}_2^{sa}$$

$$\bar{e}_2^{sc} = \bar{e}_1^{sa} \sin \alpha - \bar{e}_3^{sa} \cos \alpha$$

$$\bar{e}_3^{sc} = -(\bar{e}_1^{sa} \sin \alpha + \bar{e}_3^{sa} \cos \alpha)$$

Construction of pattern reference system

base: \bar{e}_i^p



$$\bar{n}_1 = \frac{\overline{OP} \times \overline{OQ}}{|\overline{OP} \times \overline{OQ}|}$$

$$\bar{e}_1^p = \bar{n}_1$$

$$\bar{e}_2^p = \bar{n}_1 x \bar{n}_2$$

$$\bar{n}_2 = \frac{\overline{OR} \times \overline{OS}}{|\overline{OR} \times \overline{OS}|}$$

$$\bar{e}_3^p = \bar{e}_1^p x \bar{e}_2^p$$

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Transformation from crystal to pattern coordinates:

\bar{e}_i^p can be expressed in terms of crystal coordinates

$$\bar{e}_1^p = \frac{(hkl)_1}{|(hkl)_1|}$$

$$\bar{e}_2^p = \frac{(hkl)_1}{|(hkl)_1|} \times \frac{(hkl)_2}{|(hkl)_2|}$$

$$\bar{e}_3^p = \bar{e}_1^p \times \bar{e}_2^p$$

$$g'''_{ij} = \bar{e}_i^c \cdot \bar{e}_j^p$$

pattern to
screen

crystal to
pattern

screen to
sample

$$g_{ij} = g'''_{ik} \quad g'_{kl} \quad g''_{lj}$$



Orientation matrix:

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