



Solution to the Bagaryatskii and Isaichev ferrite–cementite orientation relationship problem

H. K. D. H. Bhadeshia

Department of Materials Science and Metallurgy, University of Cambridge, Cambridge, UK

ABSTRACT

The Bagaryatskii and Isaichev orientation relationships between cementite and ferrite are closely related but not identical. They cannot easily be distinguished using ordinary electron diffraction methods and precise methods indicate that the Bagaryatski orientation does not exist. The issue is important when considering the mechanism by which cementite forms during the tempering of martensite or the formation of lower bainite, where the iron and substitutional solutes are unable to diffuse during the course of precipitation. It is demonstrated here that just one of the orientation relationships is consistent with the mechanism of precipitation at low temperatures, and is associated with much smaller deformations than the other.

ARTICLE HISTORY

Received 15 April 2018
Revised 18 April 2018
Accepted 20 April 2018

KEYWORDS

Cementite; ferrite;
orientation relationship;
displacive transformation;
Bagaryatskii; Isaichev

Introduction

There are inconsistencies between orientation relationships and the atomic mechanism of transformation when cementite (θ) precipitates from supersaturated ferrite (α , e.g. bainite or martensite) at temperatures that are too low to sustain the diffusion of iron or substitutional solutes. The details are described in the next section, but the key issues can be summarised as follows:

- The popular α/θ Bagaryatskii orientation relationship may not in fact exist. It could simply be an imprecise representation of the closely related Isaichev orientation [1,2]. If this is the case, then what justifies the existence of the Isaichev relation?
- If cementite is to form without the diffusion of substitutional atoms, then is the observed orientation relationship consistent with existence of a glissile transformation interface?

The carbon concentration of cementite is large so any displacive transformation mechanism would involve paraequilibrium at the transformation front, with the growth rate controlled by the diffusion of carbon towards the growing cementite particle. This is akin to the displacive, paraequilibrium growth of Widmanstätten ferrite in steel [3] and to the precipitation of β -vanadium hydride [4]; in both cases, the change in crystal structure is achieved by a deformation of the parent lattice into that of the product, but at a rate dependent on the diffusion of interstitial solute.

Crystallographic analysis is presented here which we believe resolves the issues listed above. We note that the orthorhombic crystal structure of cementite has been represented in two ways [5]. The space group is $Pbnm$ when the lattice parameters $a_\theta < b_\theta < c_\theta$, and $Pnma$ when $b_\theta > a_\theta > c_\theta$. The latter corresponds to the original solution by Lipson and Petch [6] and is used across the disciplines; it is also the most abundant space group of known inorganic crystals and minerals [7]. Therefore, the $Pnma$ convention is used consistently throughout this paper.

Analysis

The Bagaryatskii orientation relationship is given by [8]

$$\begin{aligned} [1\ 0\ 0]_\theta &\parallel [1\ \bar{1}\ \bar{1}]_\alpha \parallel \mathbf{z}_1 \\ [0\ 1\ 0]_\theta &\parallel [2\ 1\ 1]_\alpha \parallel \mathbf{z}_2 \\ [0\ 0\ 1]_\theta &\parallel [0\ \bar{1}\ 1]_\alpha \parallel \mathbf{z}_3 \end{aligned} \quad (1)$$

The orthonormal basis 'Z' is defined for the calculations that follow later, formed by the unit vectors \mathbf{z}_1 , \mathbf{z}_2 and \mathbf{z}_3 .

Andrews' model [9] on the displacive transformation of ferrite to cementite begins with the observed Bagaryatskii orientation relationship and proposes a deformation in which the orthogonal vectors listed in the identities (1) are either contracted or expanded but not rotated. This is a pure deformation $(Z\ S\ Z)^1$ which would convert the ferrite to the cementite cell, although as Andrews pointed out, the deformation would be

CONTACT H. K. D. H. Bhadeshia  hkdb@cam.ac.uk  Department of Materials Science and Metallurgy, University of Cambridge, Cambridge, UK

¹ This notation is due to Bowles and MacKenzie [5,10] to indicate that the deformation 'S' is defined relative to the basis 'Z'.

accompanied by the shuffle of atoms within the unit cell to recover the correct structure, and by the necessary diffusion of carbon. Referring the deformation to the orthonormal basis Z (identities (1)):

$$(Z S Z) = \begin{pmatrix} k & 0 & 0 \\ 0 & g & 0 \\ 0 & 0 & m \end{pmatrix} \quad (2)$$

where the principal deformations $k > 1$, $g < 1$ and $m > 1$ are given by

$$\begin{aligned} k &= \frac{|[1\ 0\ 0]_{\theta}|}{|[1\ \bar{1}\ \bar{1}]_{\alpha}|} = \frac{a_{\theta}}{a_{\alpha}\sqrt{3}} = 1.024957 \\ g &= \frac{|[0\ 1\ 0]_{\theta}|}{|[2\ 1\ 1]_{\alpha}|} = \frac{b_{\theta}}{a_{\alpha}\sqrt{6}} = 0.960242 \\ m &= \frac{|[0\ 0\ 1]_{\theta}|}{|[0\ \bar{1}\ 1]_{\alpha}|} = \frac{c_{\theta}}{a_{\alpha}\sqrt{2}} = 1.116120 \end{aligned} \quad (3)$$

assuming that the lattice parameters $a_{\alpha} = 0.28662$ nm and $a_{\theta} = 0.50883$ nm, $b_{\theta} = 0.67416$ nm and $c_{\theta} = 0.45241$ nm. Since two of these deformations are expansions and the third a contraction, it is not possible to find an invariant line between the two lattices without adding a rigid body rotation as an additional deformation. However, any such rigid body rotation would alter the orientation relationship from the observed Bagaryatskii relation. It follows that the Andrews deformation cannot lead to a glissile interface between the ferrite and cementite, a fundamental requirement for displacive transformation.

Using rational indices, the Isaichev orientation relationship [11] is given by

$$\begin{aligned} [1\ 0\ 0]_{\theta} \parallel [1\ \bar{1}\ \bar{1}]_{\alpha} \\ (0\ 3\ 1)_{\theta} \approx \parallel (1\ 0\ 1)_{\alpha} \end{aligned} \quad (4)$$

The approximation sign is omitted in most publications but Isaichev indicated that $\{0\ 3\ 1\}_{\theta}$ and $\{1\ 0\ 1\}_{\alpha}$ are not exactly parallel, some $1.5\text{--}2^{\circ}$ apart. Modern literature states this angle to be larger, at 3.8° [12–14], although the same publications use $\{0\ 3\ 1\}_{\theta} \parallel \{1\ 0\ 1\}_{\alpha}$ when quoting the orientation relationship.

The Isaichev orientation relationship is close to that of Bagaryatskii making them difficult to distinguish using conventional electron diffraction. As already pointed out, it deviates from Bagaryatskii by a rotation of 3.8° about the a -axis of the cementite [14]. Accurate measurements on tempered martensite have repeatedly identified the Isaichev orientation relationship and this has led to the suggestion that the Bagaryatskii orientation does not exist [1,2]. In some cases, electron diffraction patterns interpreted to show the Bagaryatskii orientation for tempered martensite [15] have been shown to be more consistent with the Isaichev relationship [2].

It turns out the deformation described in Equation (2), when combined with a rigid body rotation, that converts the Bagaryatskii orientation in that

of Isaichev, renders the combination an invariant-line strain. The matrix representing the rigid body rotation is obtained by substituting the angle-axis pair of 3.8° about the a -axis into, for example, Equation 7.9 of [5]:

$$\begin{aligned} (Z S_I Z) &= \underbrace{\begin{pmatrix} 1.024957 & 0 & 0 \\ 0 & 0.960242 & 0 \\ 0 & 0 & 1.116120 \end{pmatrix}}_{\text{'Bain strain, Bagaryatski orientation'}} \\ &\times \underbrace{\begin{pmatrix} 1 & 0 & 0 \\ 0 & 0.9978 & -0.0663 \\ 0 & 0.0663 & 0.9978 \end{pmatrix}}_{\text{'rigid body rotation'}} \\ &= \underbrace{\begin{pmatrix} 1.02496 & 0 & 0 \\ 0 & 0.958129 & -0.063664 \\ 0 & 0.073999 & 1.11366 \end{pmatrix}}_{\text{'Isaichev orientation, invariant-line strain'}} \end{aligned} \quad (5)$$

The eigenvectors and eigenvalues (λ_i) for $(Z S_I Z)$ are

$$\begin{aligned} [0\ -0.486527\ 0.873665]_Z, \quad \lambda_1 &= 1.07245 \\ [1\ 0\ 0]_Z, \quad \lambda_2 &= 1.02496 \\ [0\ 0.839485\ -0.543383]_Z, \quad \lambda_3 &= 0.999338 \end{aligned} \quad (6)$$

The third eigenvector is invariant because its magnitude is essentially unchanged; it is also noteworthy that the maximum elongation has been reduced to 7.2% compared with the 11.6% associated with the $(Z S Z)$ Bagaryatskii orientation.

The process described above for cementite is analogous to the martensitic transformation of austenite, where the Bain strain [16] changes the lattice but does not leave any line invariant, and the orientation relationship implied by the Bain strain is not that observed. The correct irrational orientation relationship that is observed is obtained by adding a precise rigid body rotation that in combination with the Bain strain becomes an invariant-line strain.

Summary

The Bagaryatskii deformation as described by Andrews does not leave any vector invariant. It has been discovered here that when the Bagaryatskii deformation is combined with a rigid body rotation that generates the Isaichev orientation, the resulting total deformation is an invariant-line strain. Furthermore, the principal deformations associated with this invariant-line strain are substantially smaller than those of the Bagaryatskii deformation. This explains the occurrence of the Isaichev orientation relationship.

The analogy with the martensitic transformation of austenite (γ) is clear; the η_i are the principal distortions:

Transformation	Pure deformation	Pure deformation + rigid body rotation	Final orientation
$\gamma \rightarrow \alpha$	Bain strain $\eta_i = 1.136, 1.136, 0.803$	Invariant-line strain $\eta_i = 1.124, 1, 0.922$	Kurdjumov–Sachs type
$\alpha \rightarrow \theta$	Bagaryatskii $\eta_i = 1.116, 1.024, 0.960$	Invariant-line strain $\eta_i = 1.073, 1.025, 1$	Isaichev

The calculations will depend on the lattice parameters of cementite and ferrite, but as long as the parameters are known as a function of temperature and composition, they are straightforward to repeat.

As a corollary, the following observations now are compatible with the paraequilibrium, displacive precipitation of cementite supersaturated ferrite at low temperatures:

- It is possible to define a homogeneous deformation which is an invariant-line strain for the $\alpha \rightarrow \theta$ transformation. This is a minimum condition for the existence of a glissile interface between the parent and product lattices.
- Cementite variant selection occurs when elastically-loaded martensite is tempered [17]. Such selection is characteristic of a strong interaction of the shape deformation accompanying transformation, with the applied stress.
- The displacive precipitation of cementite would require the diffusion of carbon. However, the iron to substitutional solute ratio must remain constant. This has been verified using the atom-probe technique [18].

Acknowledgments

I would like to thank Professor Sergey Kaloshkin of the National University of Science and Technology (NUST-MISIS) in Moscow for sending me a copy of Isaichev's original manuscript. Dr Yurii P. Ivanov in Cambridge kindly assisted in its translation. I also thank Dr Saurabh Kundu and Professor Pat Kelly for their comments on the manuscript.

Disclosure statement

No potential conflict of interest was reported by the authors.

References

- Zhang M-X, Kelly PM. Determination of carbon content in bainitic ferrite and carbon distribution in austenite by using CBKLD. *Mater Charact.* **1998**;40:159–168.
- Zhang M-X, Kelly PM. Crystallography of spheroidite and tempered martensite. *Acta Mater.* **1998**;46:4081–4091.
- Bhadeshia HKDH. Rationalisation of shear transformations in steels. *Acta Metall.* **1981**;29:1117–1130.
- Bowles JS, Muddle BC, Wayman CM. The crystallography of the precipitation of beta vanadium hydride. *Acta Metall.* **1977**;25:513–520.
- Bhadeshia HKDH. *Geometry of crystals, polycrystals, and phase transformations.* Boca Raton, FL: CRC press; **2017**. ISBN 9781138070783.
- Lipson H, Petch NJ. The crystal structure of cementite Fe₃C. *J Iron Steel Inst.* **1940**;142:95P–103P.
- Urusov VS, Nadezhina TN. Frequency distribution and selection of space groups in inorganic crystal chemistry. *J Struct Chem.* **2009**;50:22–37.
- Bagaryatskii YA. Possible mechanism of martensite decomposition. *Dokl Akad Nauk SSSR.* **1950**;73:1161–1164.
- Andrews KW. The structure of cementite and its relation to ferrite. *Acta Metall.* **1963**;11:939–946.
- Bowles JS, Mackenzie JK. The crystallography of martensite transformations, part I. *Acta Metall.* **1954**;2:129–137.
- Isaichev IV. Orientatsiya tsementita v otpushchennoi uglerodistoi stali [Orientation of cementite in tempered carbon steel]. *Zhurnal Tekhnicheskoi Fiziki.* **1947**;17:835–838.
- Mangan MA, Shiflet GJ. The Pitsch–Petch orientation relationship in ferrous pearlite at small undercooling. *Metall Mater Trans A.* **1999**;30:2767–2781.
- Ohmori Y. Microstructural evolutions with precipitation of carbides in steels. *ISIJ Int.* **2001**;41:554–565.
- Wei FG, Tsuzaki K. Crystallography of [011]/54.7° lath boundary and cementite in tempered 0.2C steel. *Acta Mater.* **2005**;53:2419–2429.
- Ohmori Y, Davenport AT, Honeycombe RWK. Tempering of a high-C martensite. *Trans ISIJ.* **1972**;12:112–117.
- Bain EC. The nature of martensite. *Trans AIME.* **1924**;70:25–46.
- Stewart JW, Thomson RC, Bhadeshia HKDH. Cementite precipitation during tempering of martensite under the influence of an externally applied stress. *J Mater Sci.* **1994**;29:6079–6084.
- Babu SS, Hono K, Sakuri T. APFIM studies on martensite tempering of Fe–C–Si–Mn steel. *Appl Surface Sci.* **1993**;67:321–327.