

Boron: Type IV Cracking

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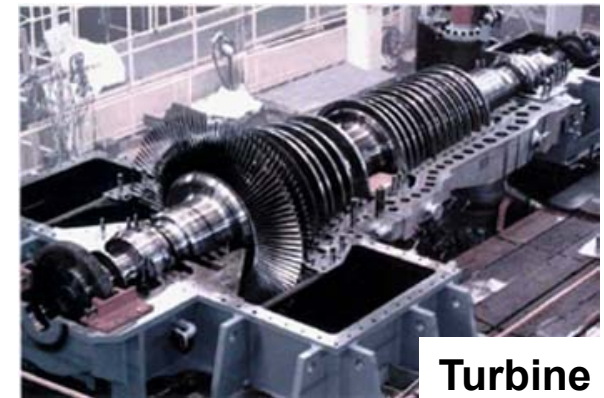
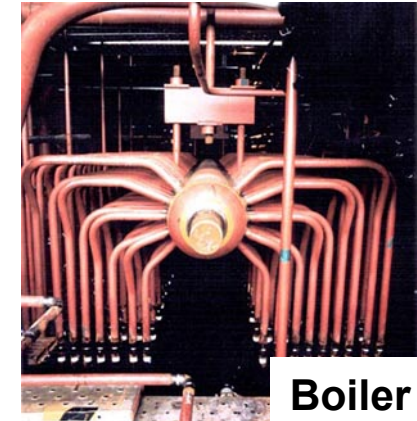
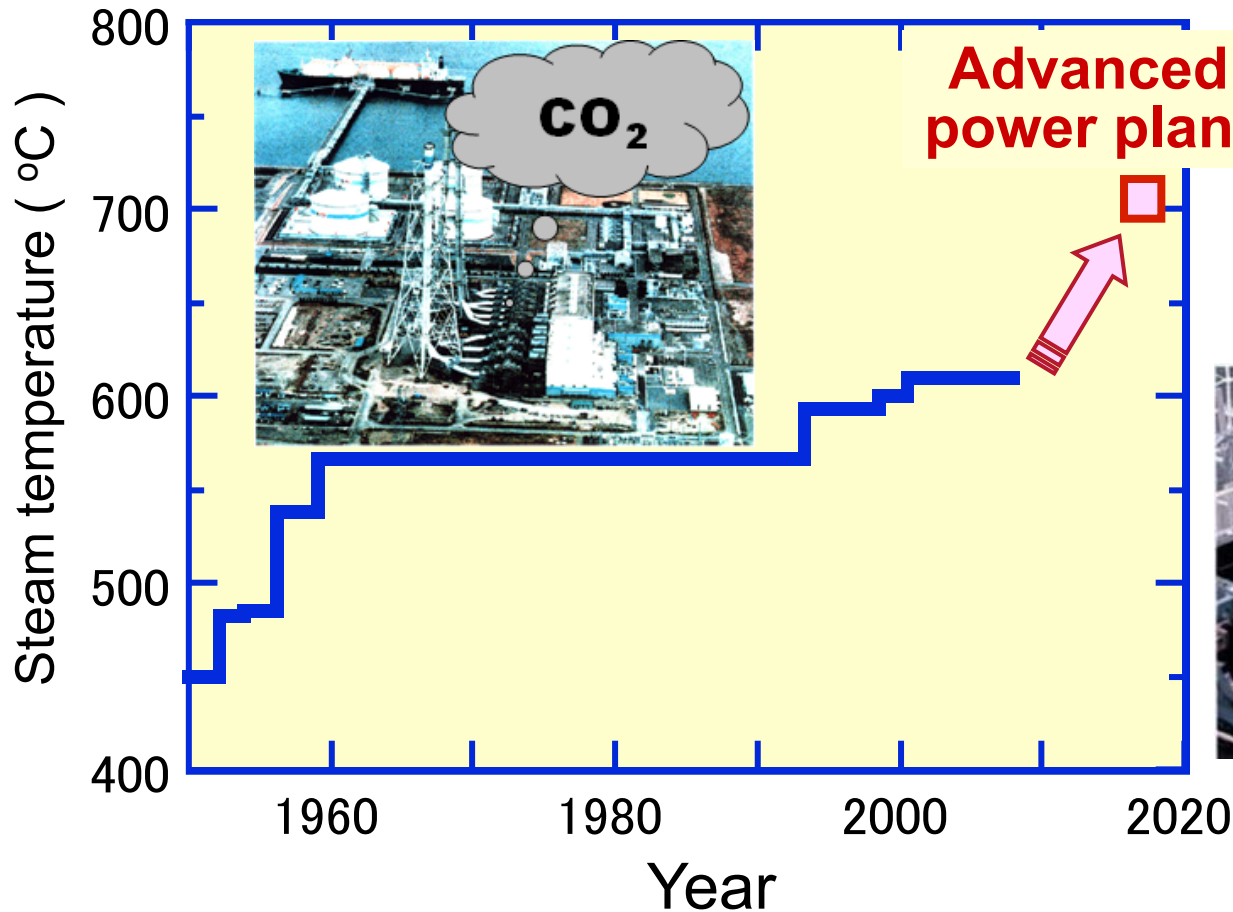


Contents

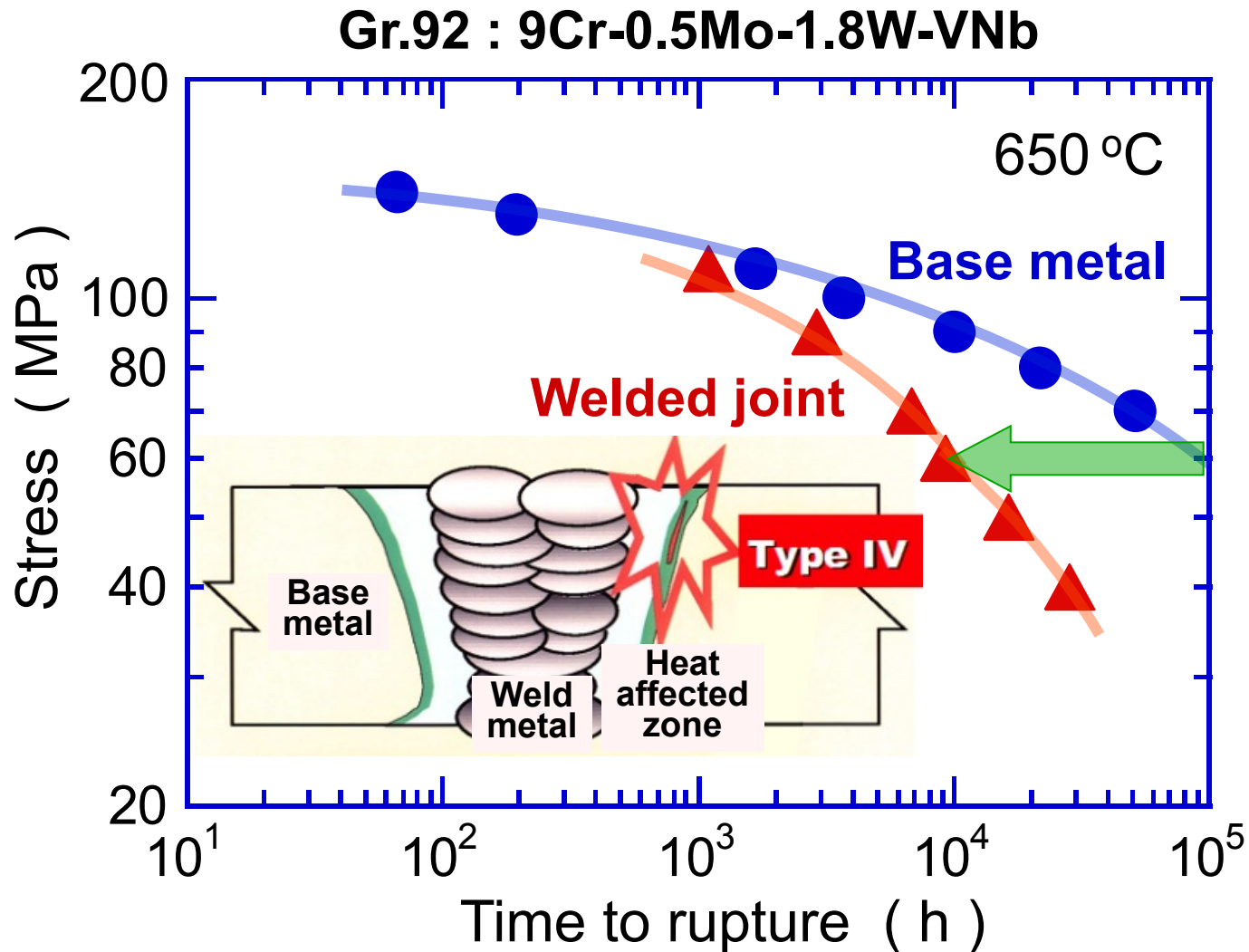
- 1. Background and objectives**
- 2. Microstructure and creep strength of heat-affected-zone (HAZ) of welded joint**
- 3. Suppression of Type IV cracking by boron**
- 4. Alloy design of 9Cr steel with high creep strength and no Type IV cracking at 650 °C**
- 5. Summary**

Higher steam temperature is strongly desired

Coal-fired power plants in Japan



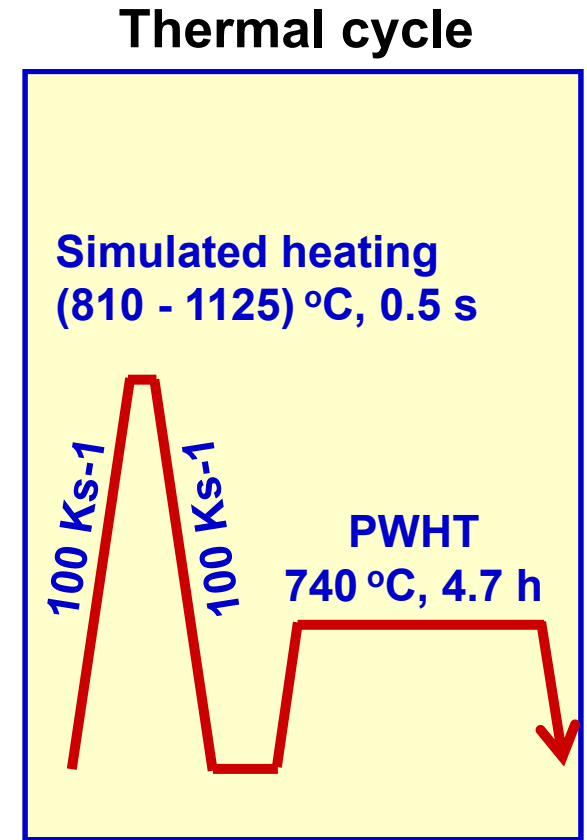
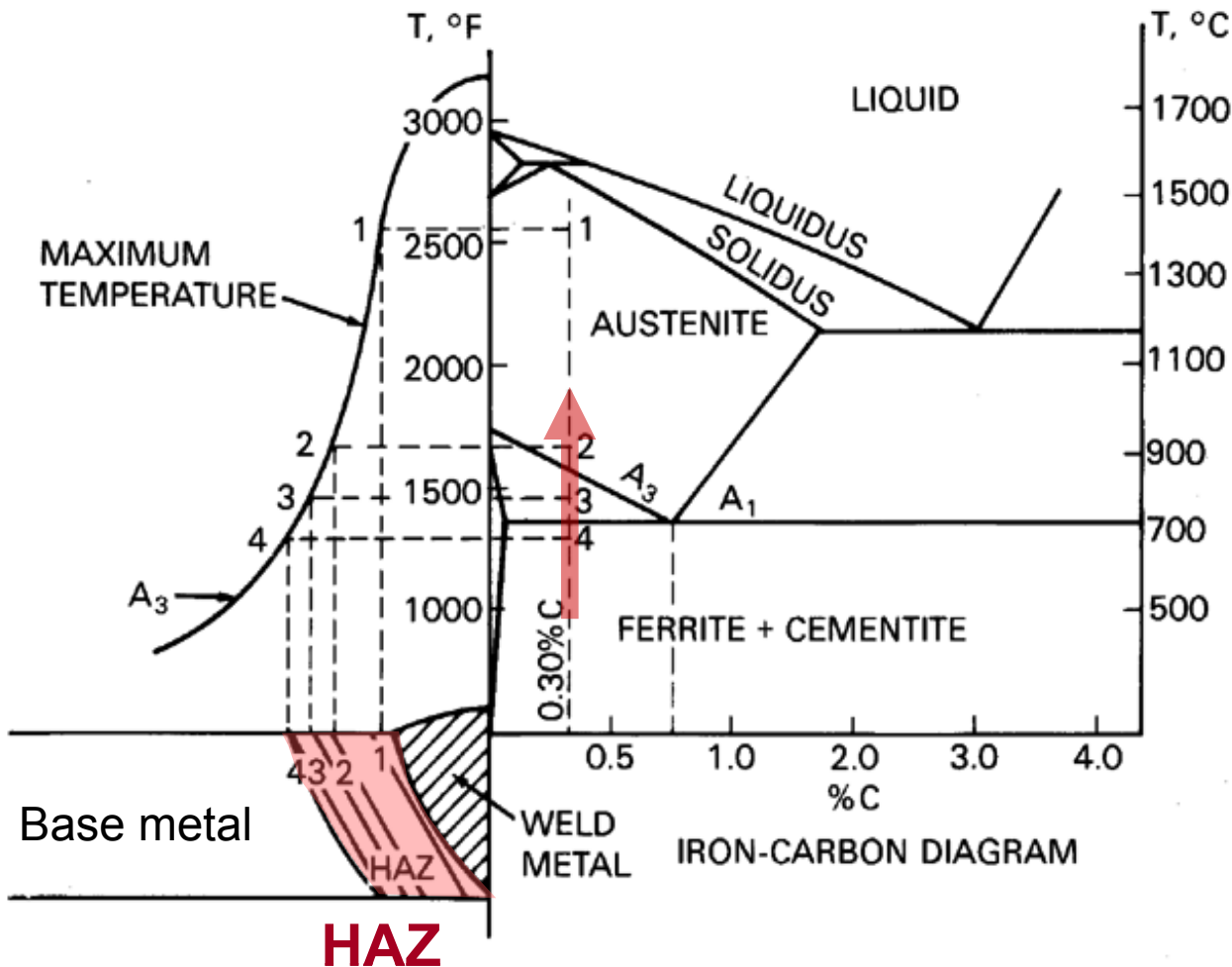
Conventional 9% Cr steel (Gr.92) shows significant degradation in welded joints



Objectives

- **to make clear mechanisms for lower creep strength of HAZ than base metal**
Fine grains?
- **to make clear boron effect on suppression of Type IV cracking in HAZ**
- **to establish alloy-design of 9Cr steel with high strength and no Type IV cracking**

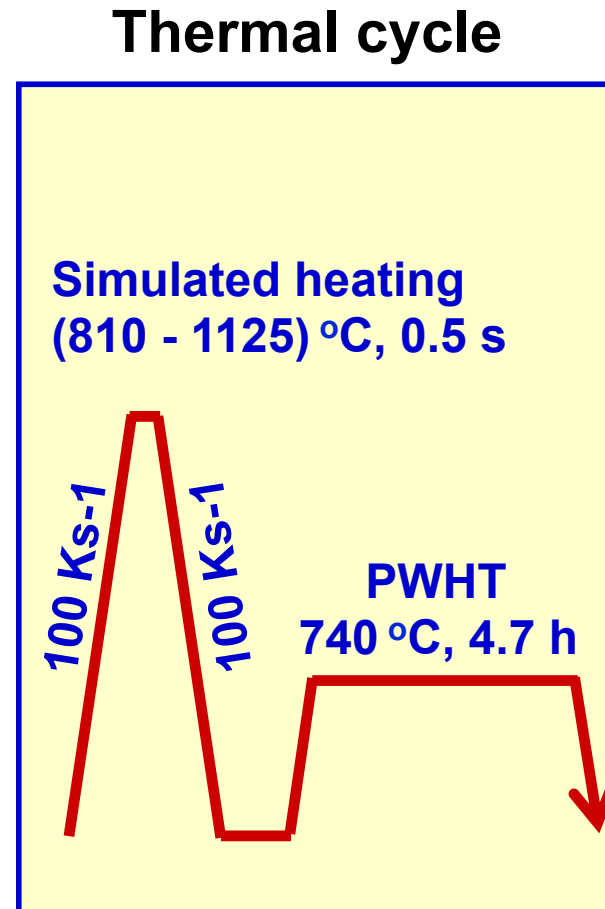
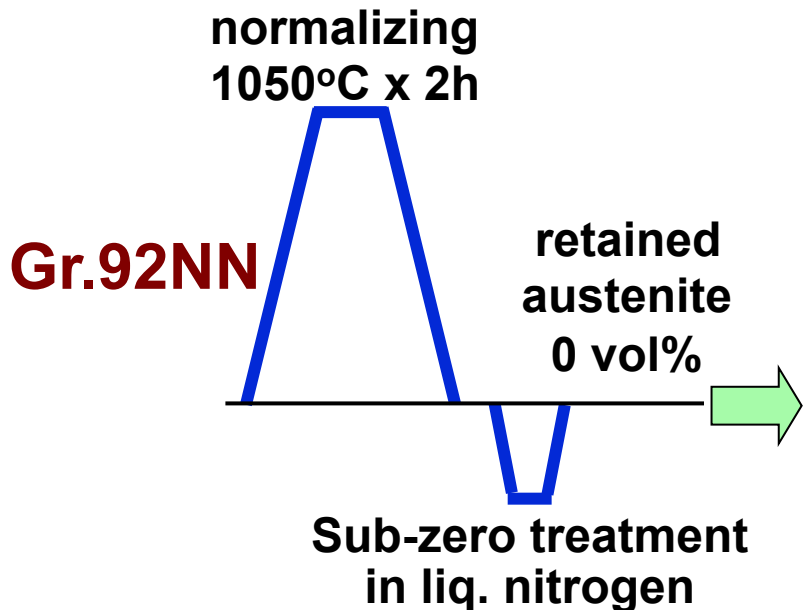
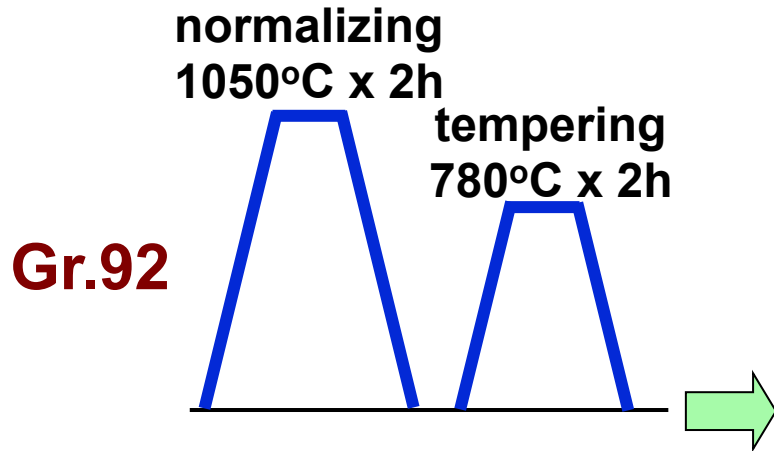
Simulated heat-affected-zone (HAZ) specimens



M. Santella et al. (2006)

Same grain size, Different amount of GB $M_{23}C_6$ carbides

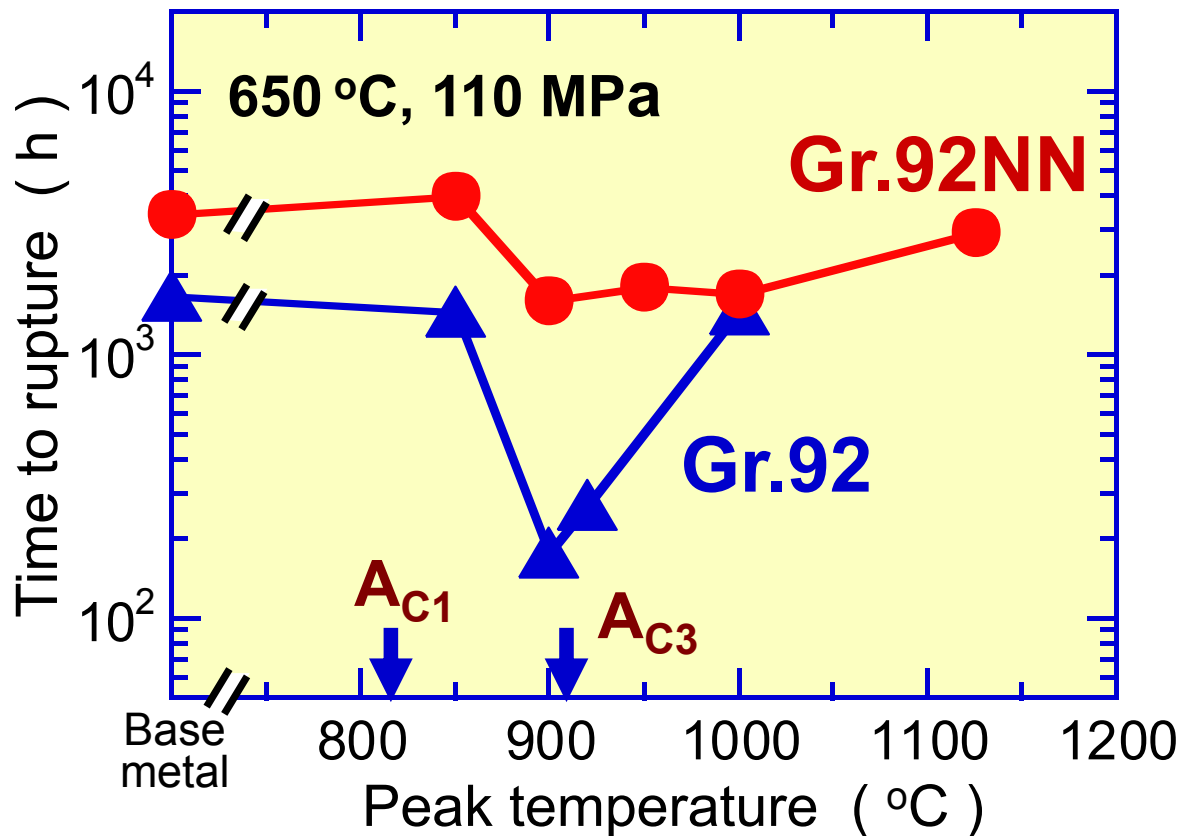
Gr.92 : 9Cr-0.5Mo-1.8W-0.2V-0.05Nb



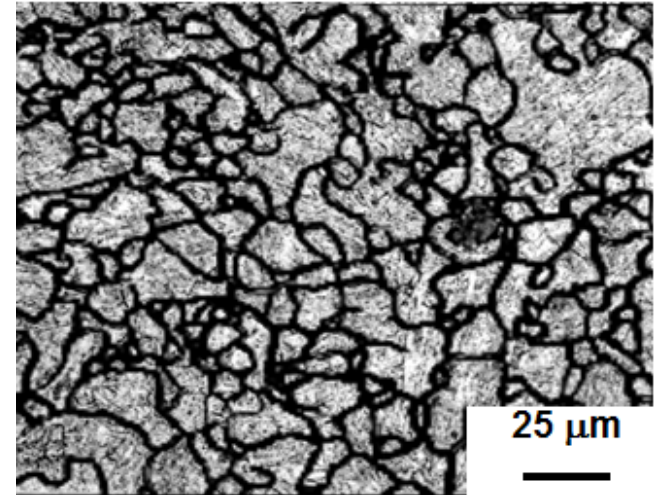
Creep test
650 °C
tr < 10,000 h

Grain refinement is not a main reason for degradation

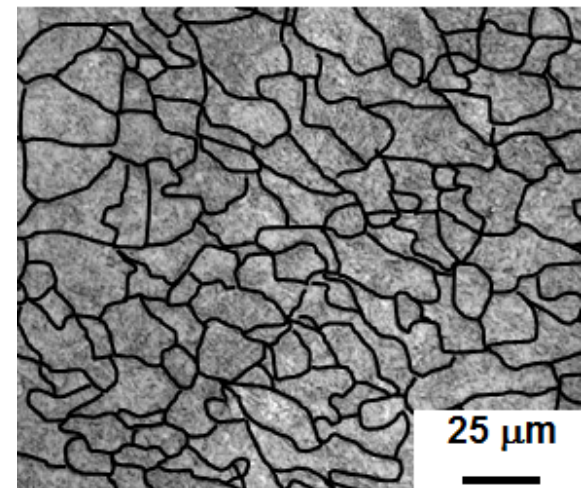
Simulated-HAZ specimens



Gr.92
after A_{C3} thermal cycle



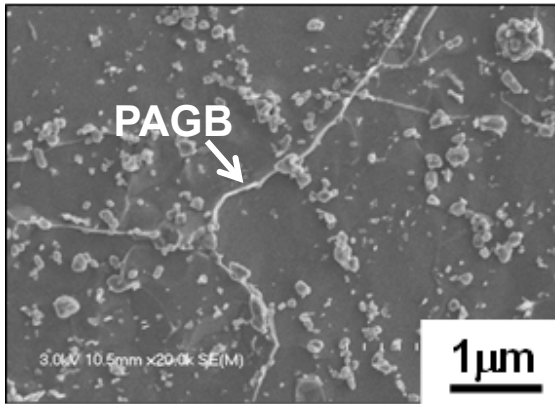
Gr.92NN
after A_{C3} thermal cycle



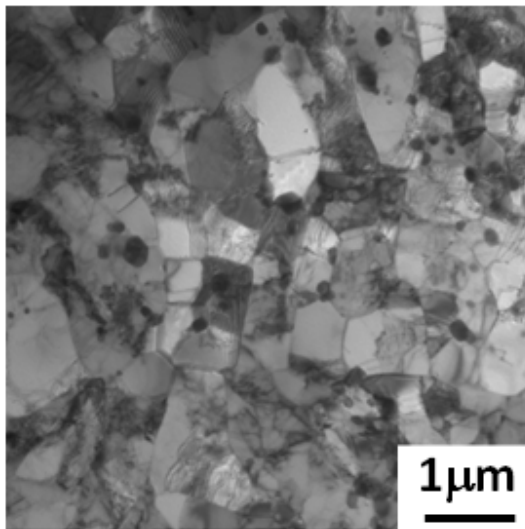
Microstructure after A_{C3} thermal cycle

Gr.92

normalizing
→ tempering
→ A_{C3} thermal cycle

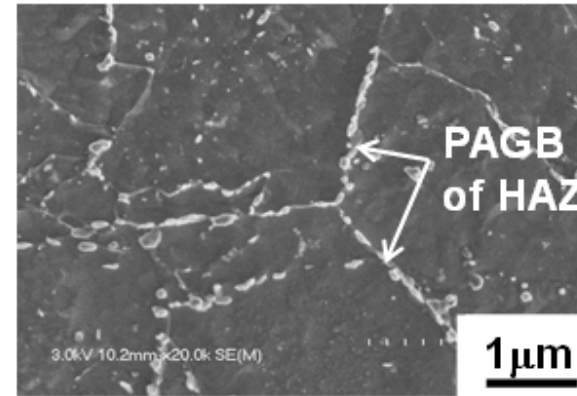


Poor $M_{23}C_6$ along GB and Sub-GB

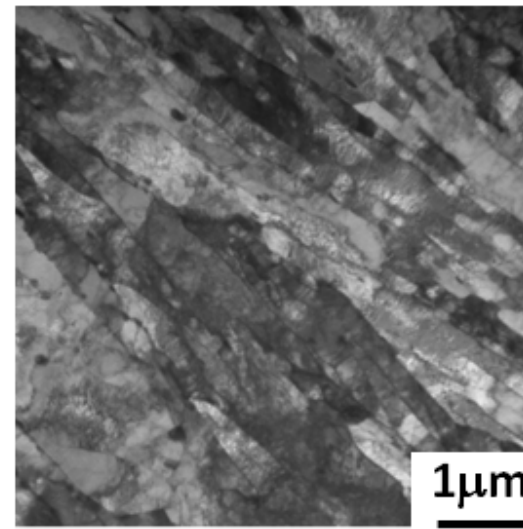


Gr.92NN

normalizing
→ sub-zero in liq. N
→ A_{C3} thermal cycle

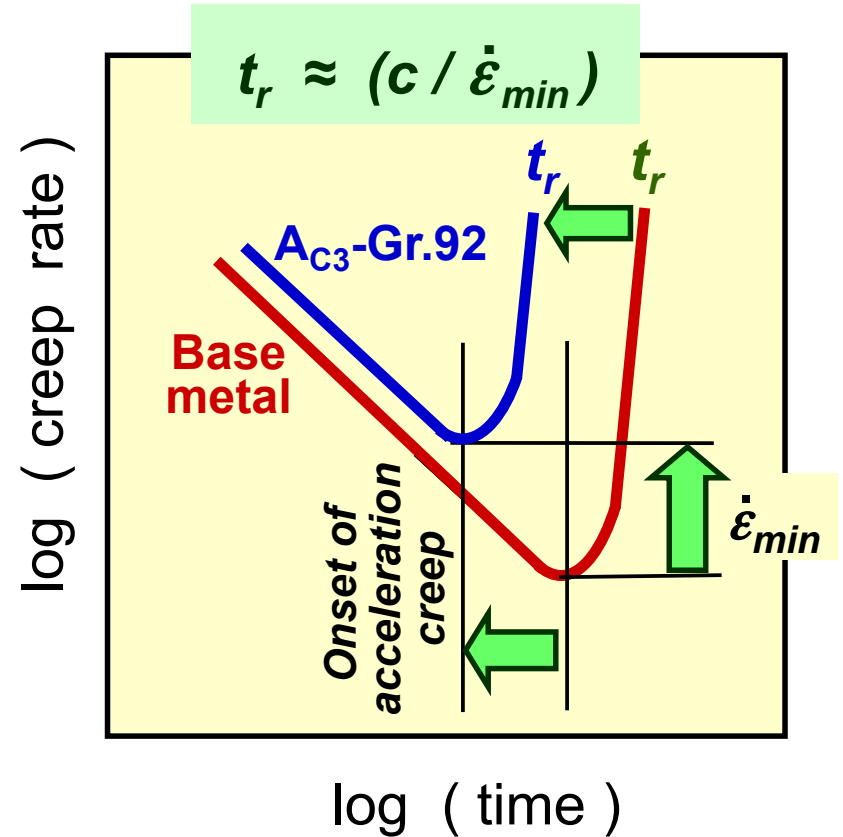
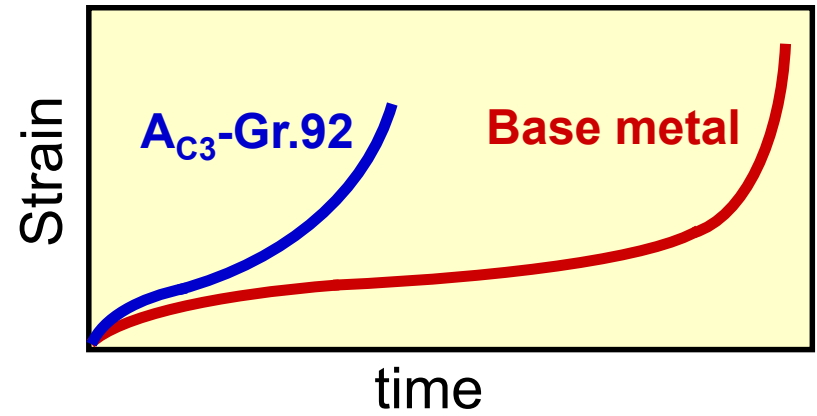
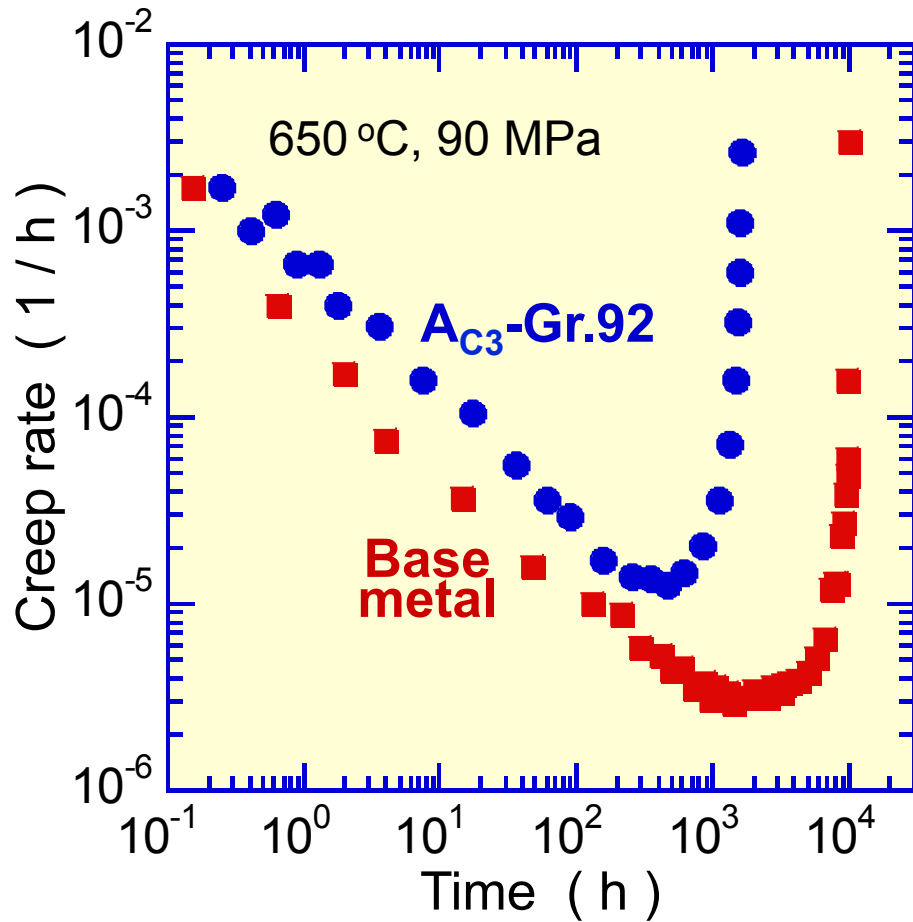


Enough $M_{23}C_6$ along GB and LB



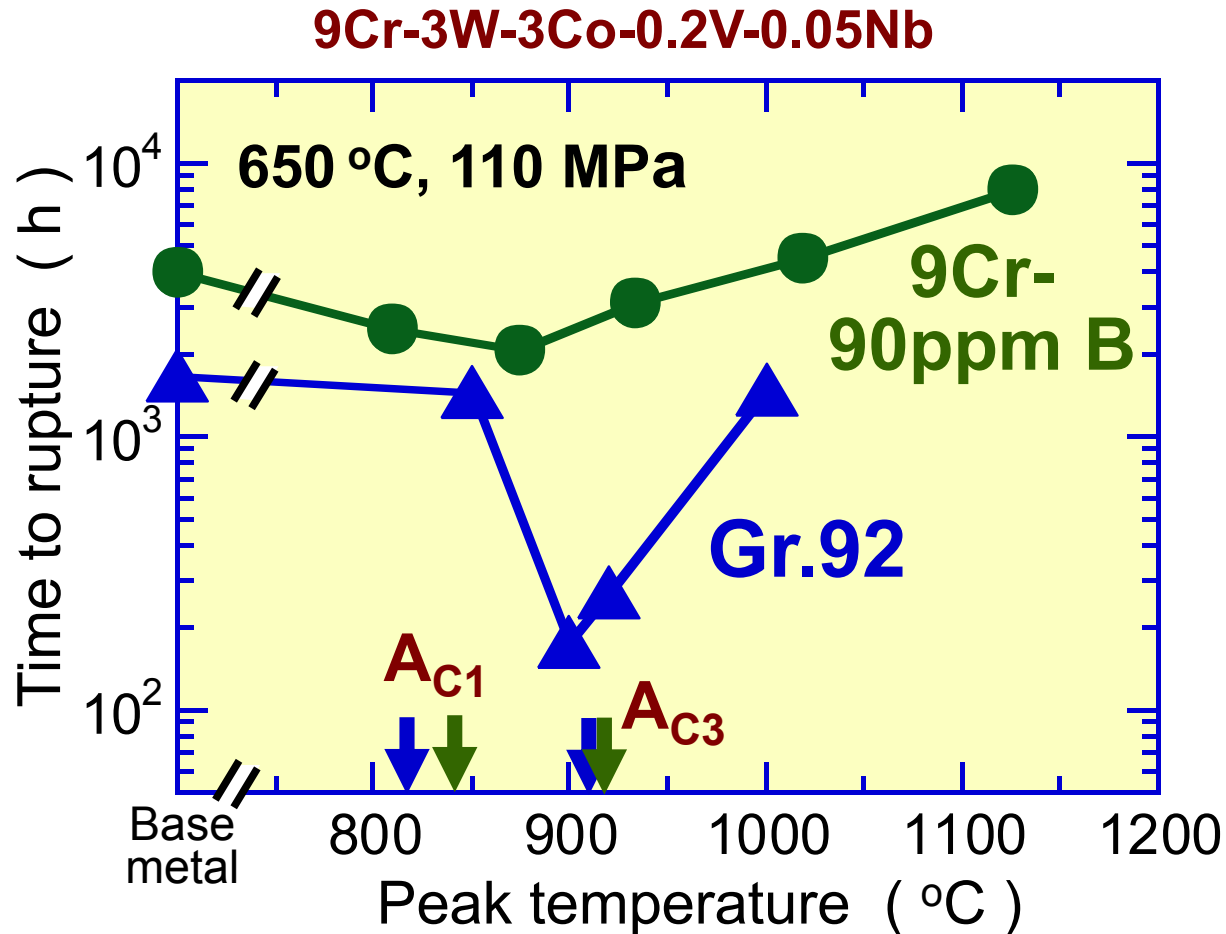
LB :
lath boundary

Poor GB $M_{23}C_6$ promotes microstructure recovery



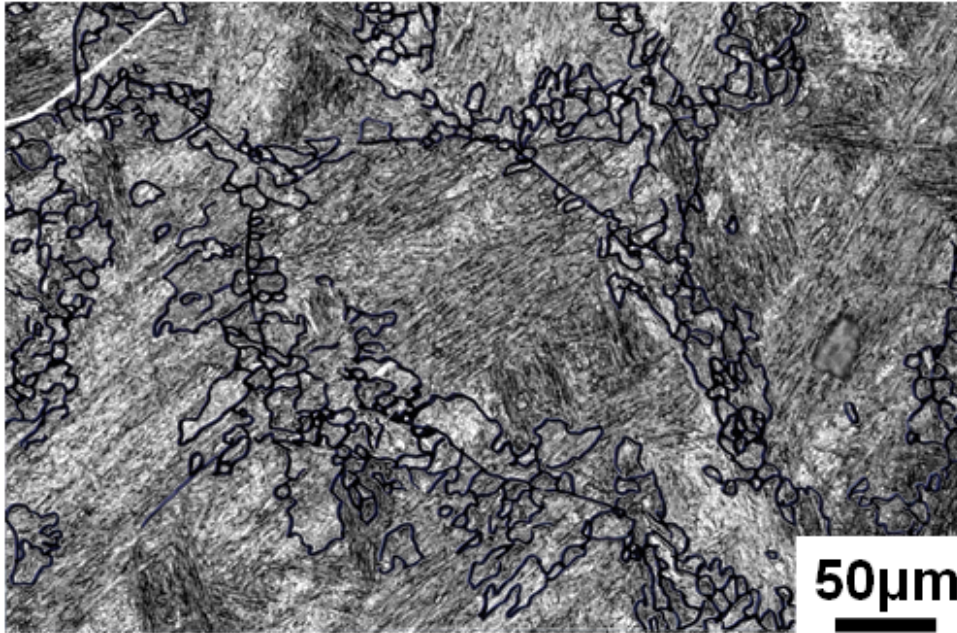
9Cr-90ppm boron steel

No degradation even after A_{C3} thermal cycle

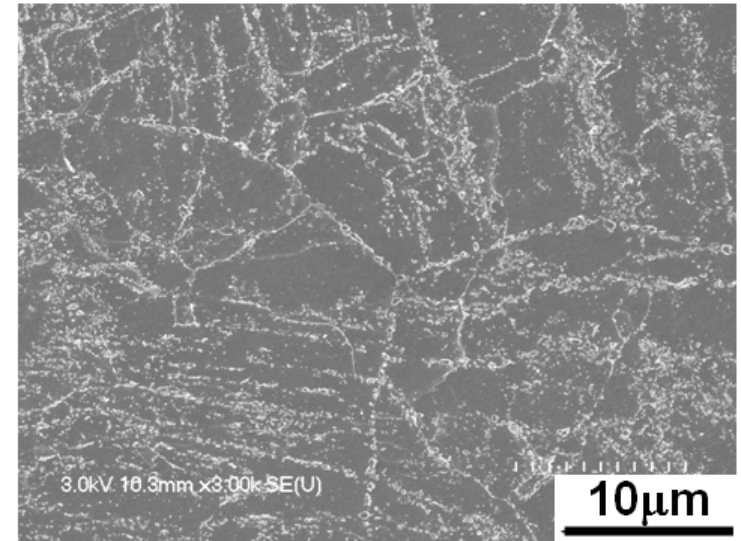


9Cr-90ppm B after A_{C3} thermal cycle

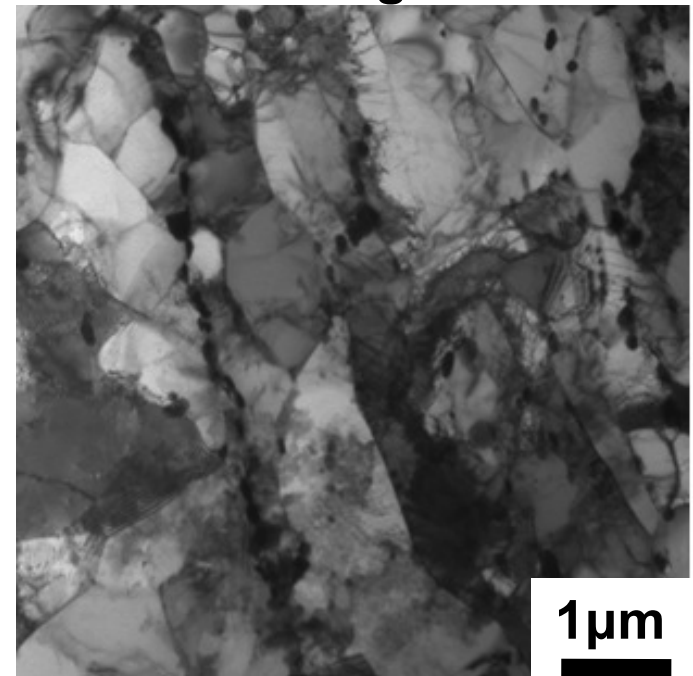
Enough $M_{23}C_6$ carbides along
PAGBs and block boundaries



Near PAGBs

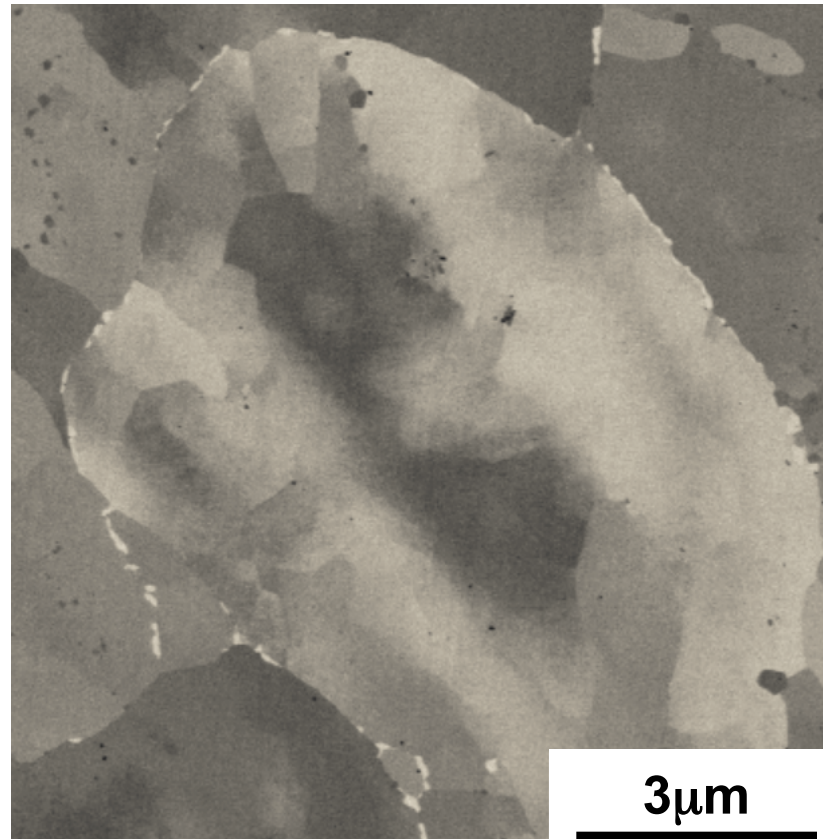


Inside grain



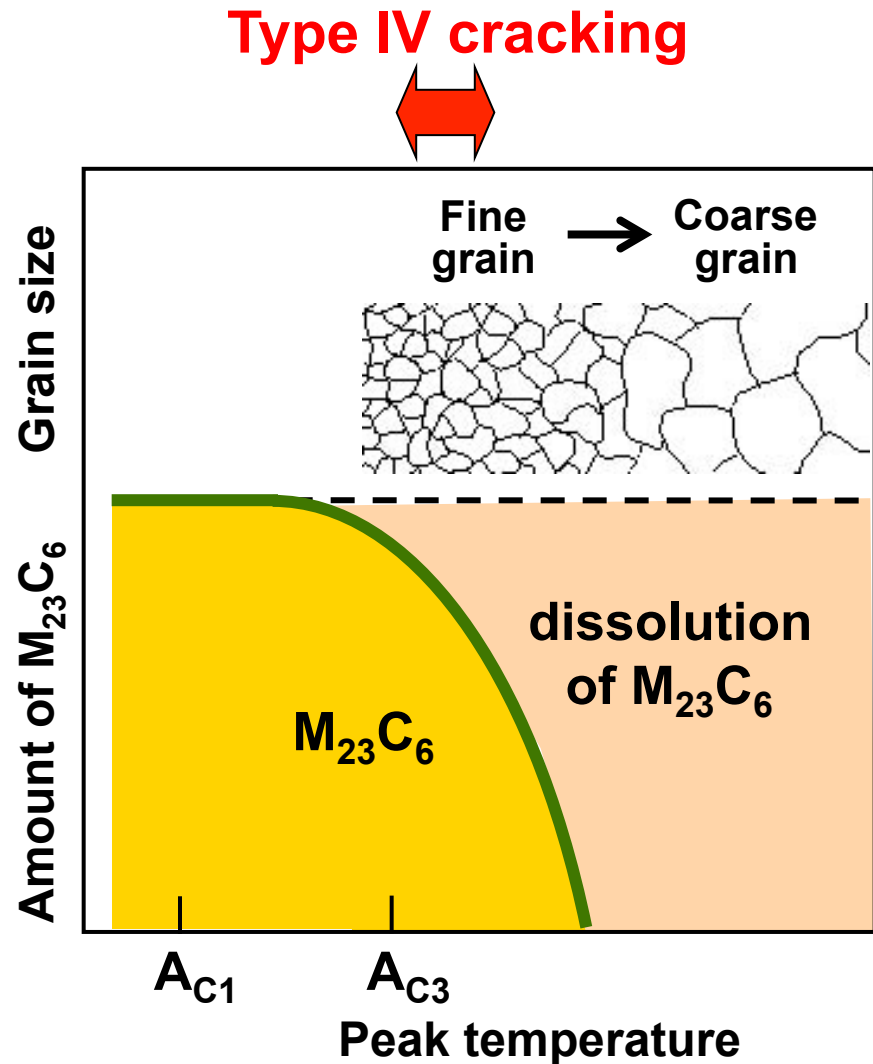
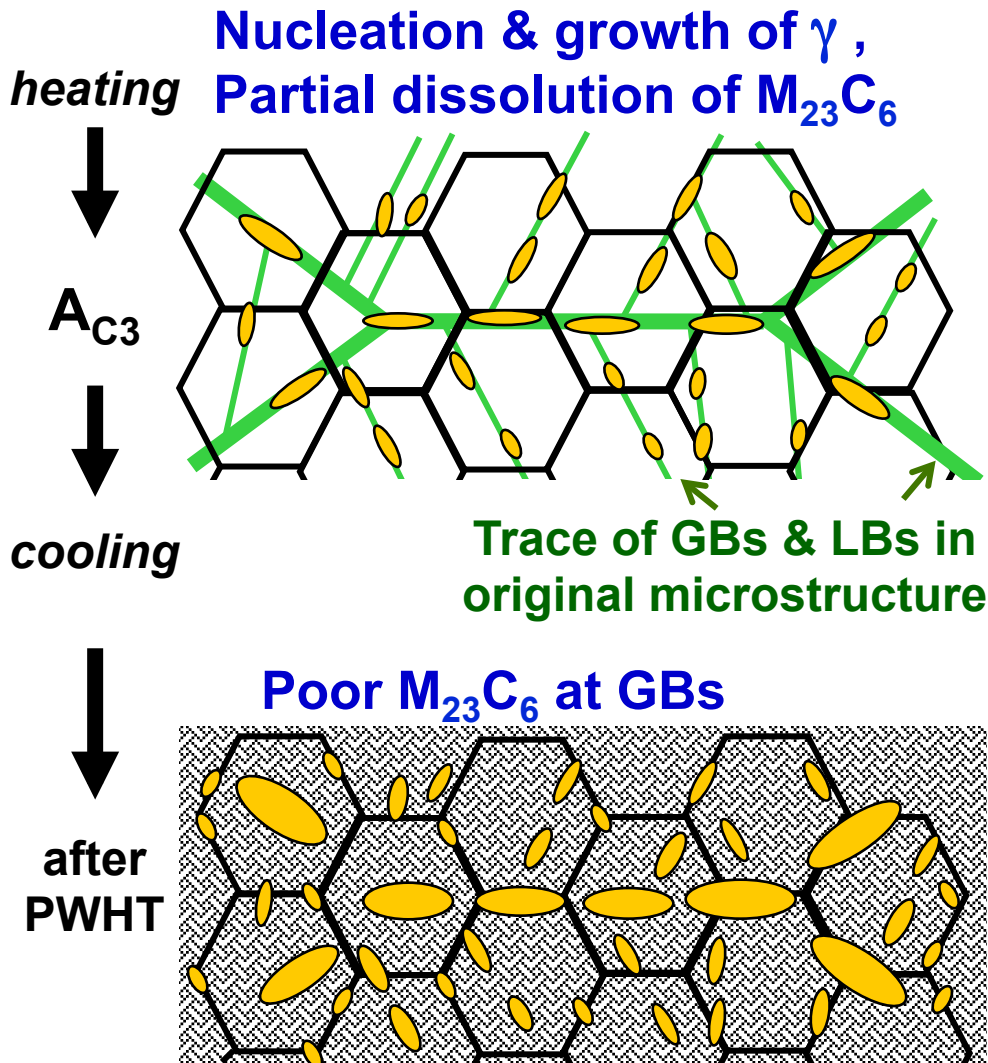
9Cr-90ppm B after A_{C3} thermal cycle

Precipitation of Fe_7W_6 - μ phase at GBs of fine grains



Gr.92 after A_{C3} thermal cycle

Fine-grains & poor $M_{23}C_6$ along GBs



A_{C3} - HAZ of 9Cr-boron steel

Specimen surface (130 ppm B)

heating



A_{C3}

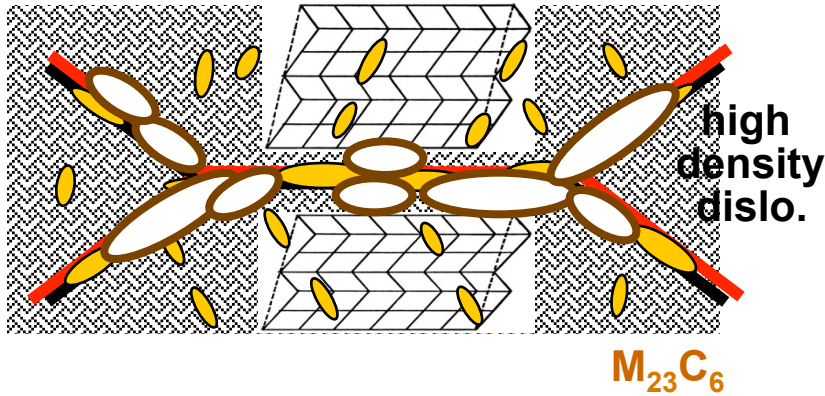


cooling

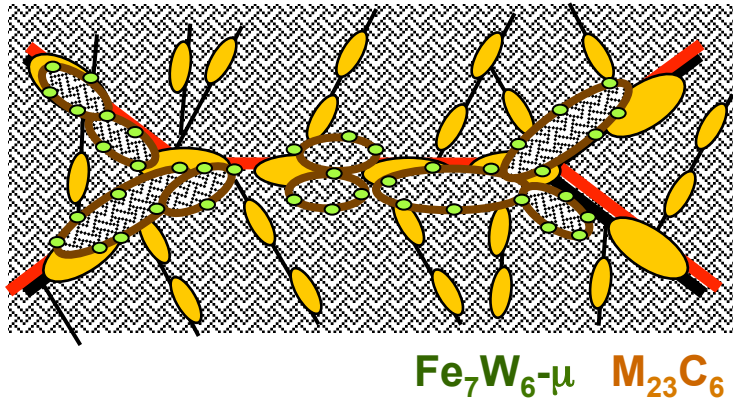


after
PWHT

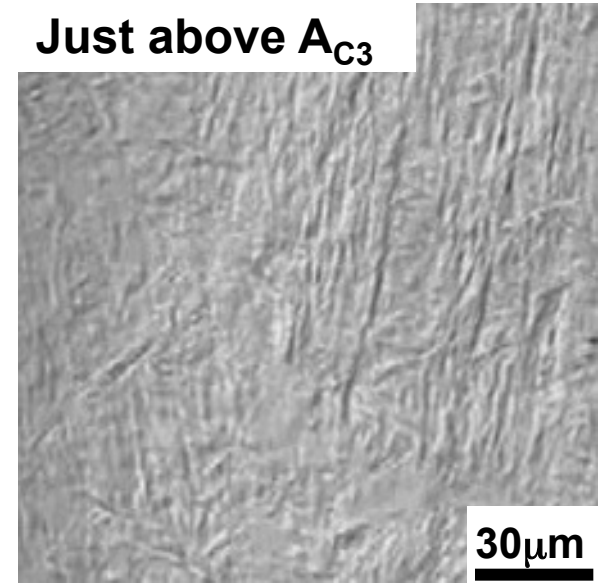
Coarse grains with
high dislocation density



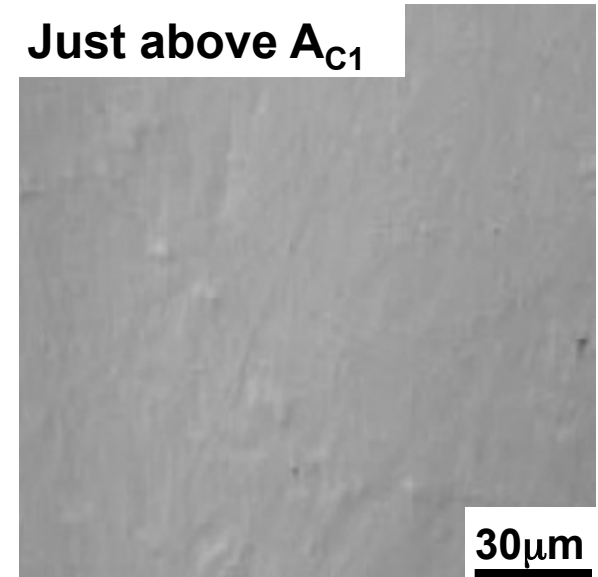
Un-recrystallized γ transforms
to martensite during cooling



Just above A_{C3}



Just above A_{C1}

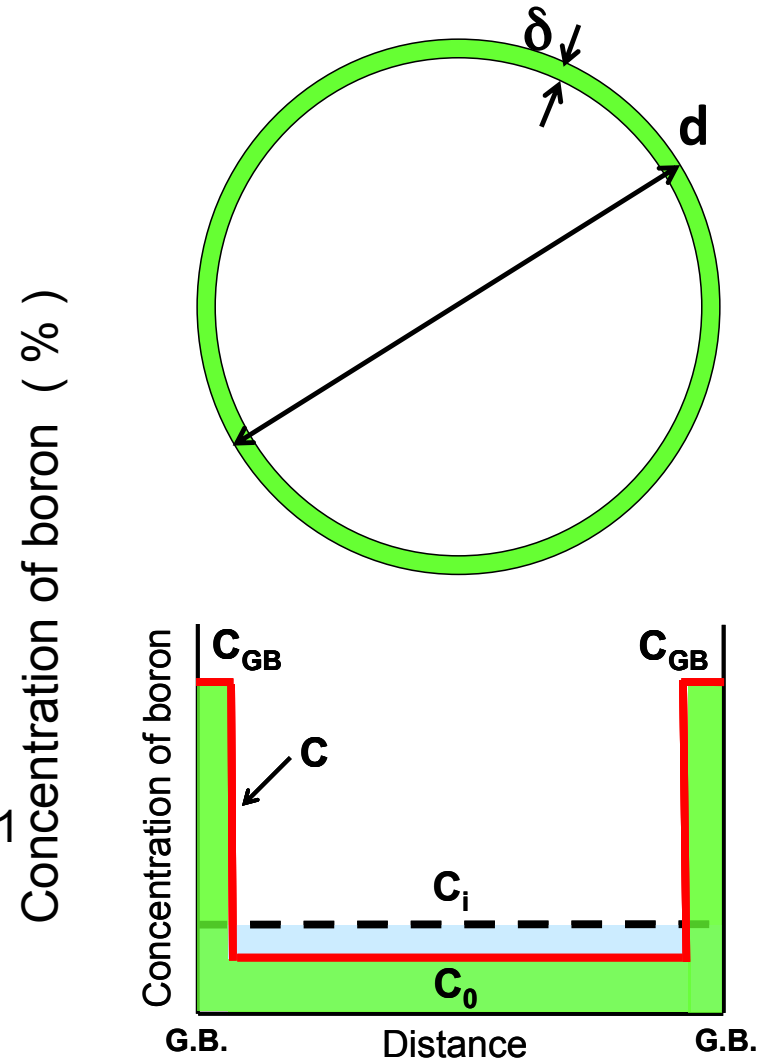
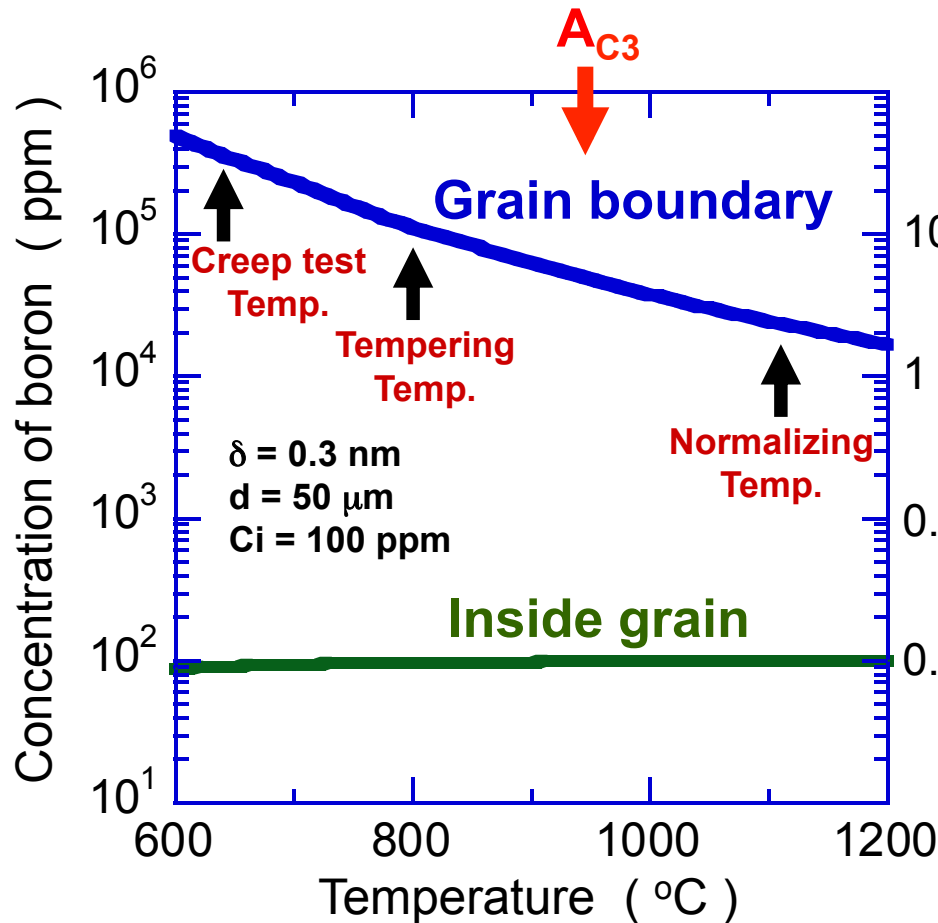


Segregation of boron at G.B.

$$C_{GB} = C_0 \exp (B / RT)$$

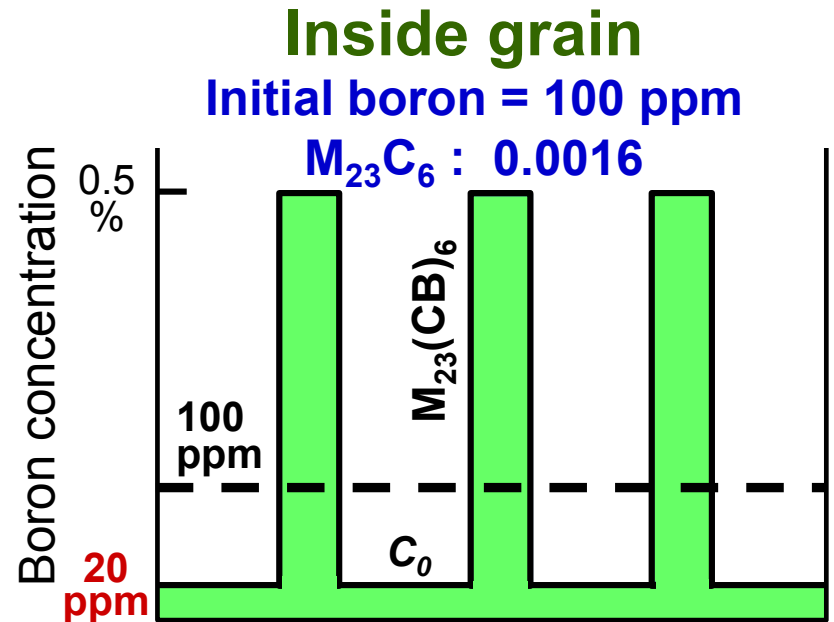
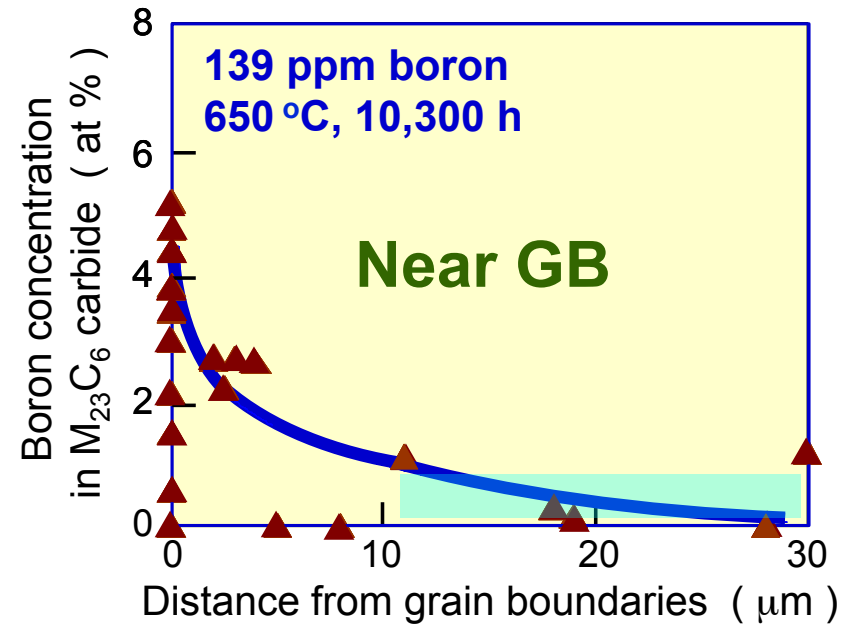
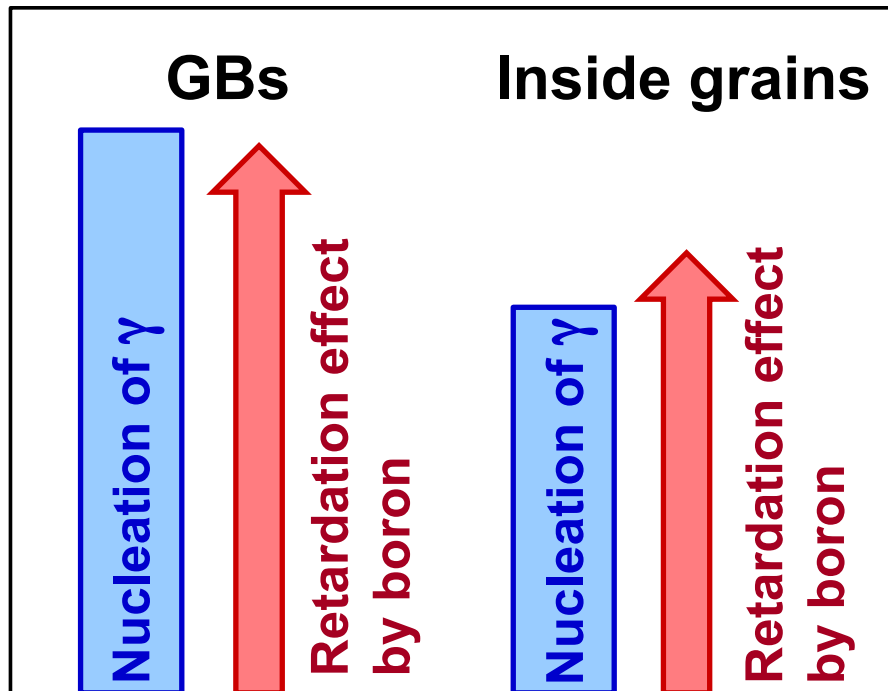
B = 62.7 kJ / mol (316ss)

L. Karlsson et al. (1988)



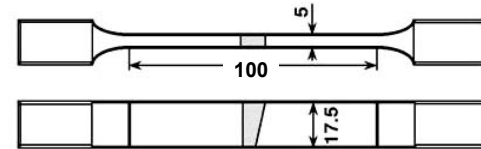
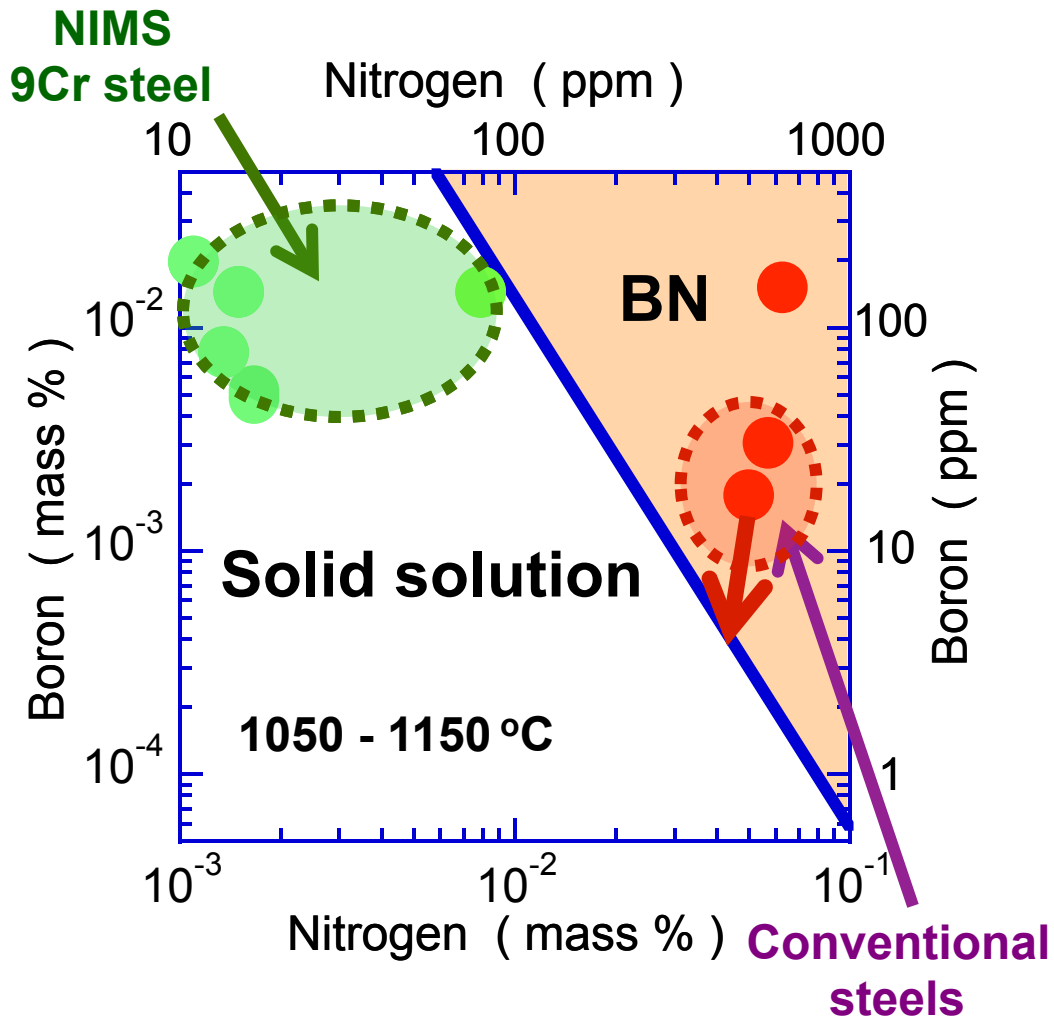
Competition : nucleation and retardation by boron

Driving force

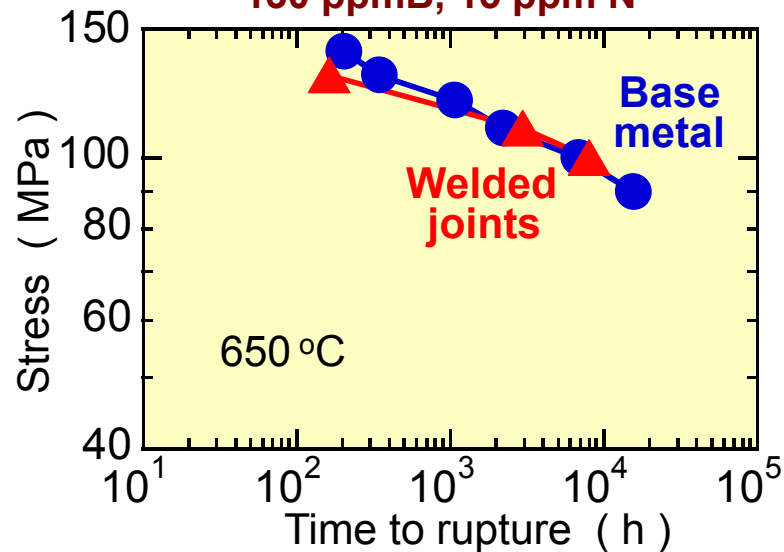


Soluble boron is essential for suppression of Type IV cracking

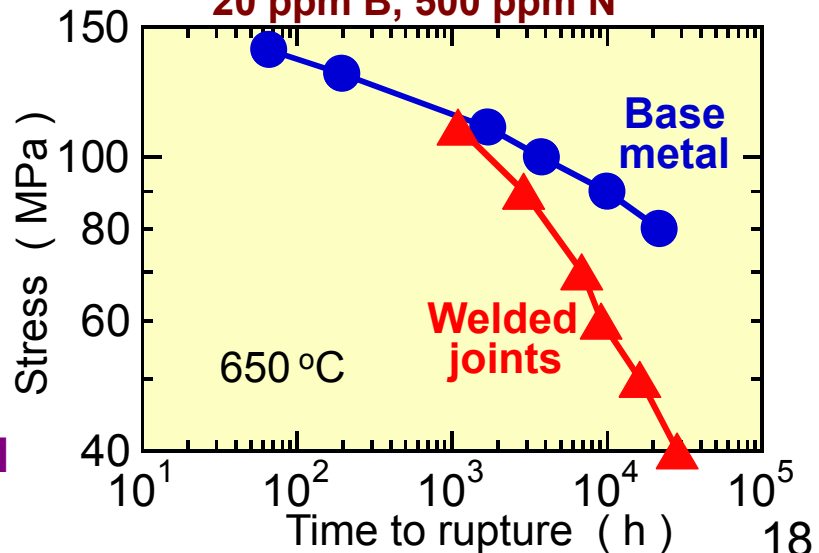
- No Type IV (No grain refinement)
- Type IV (Grain refinement)



9Cr-boron steel :
130 ppmB, 15 ppm N

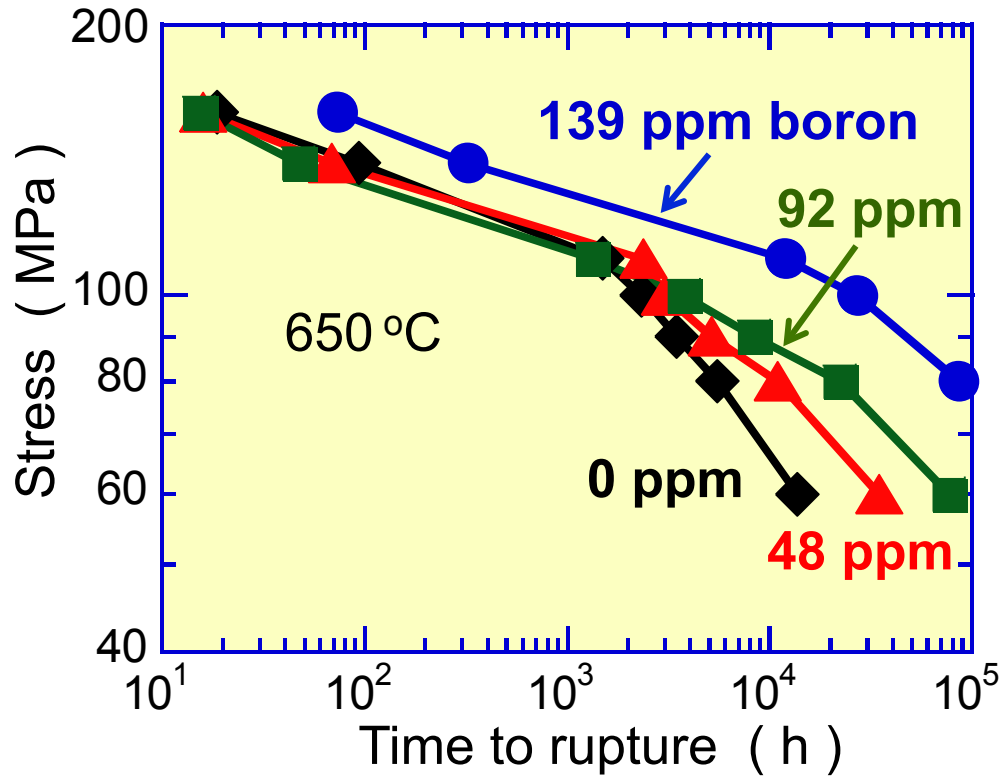


Conventional 9Cr steel (Gr.92) :
20 ppm B, 500 ppm N

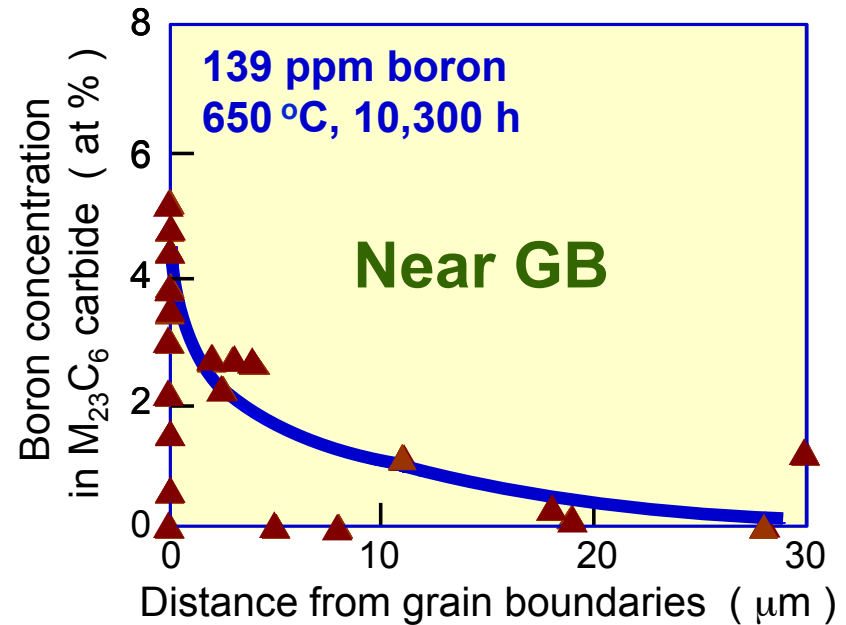
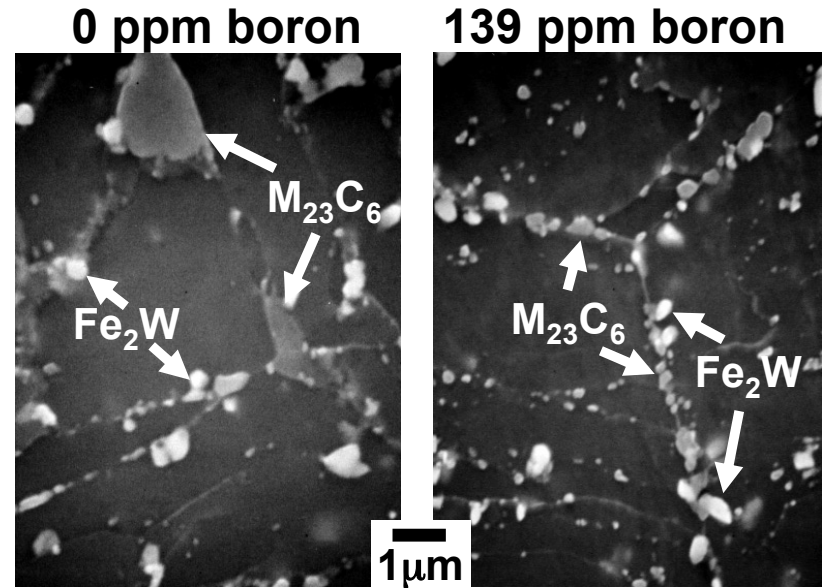


Base metal : boron improves creep strength

9Cr-3W-3Co-0.2V-0.05Nb steel

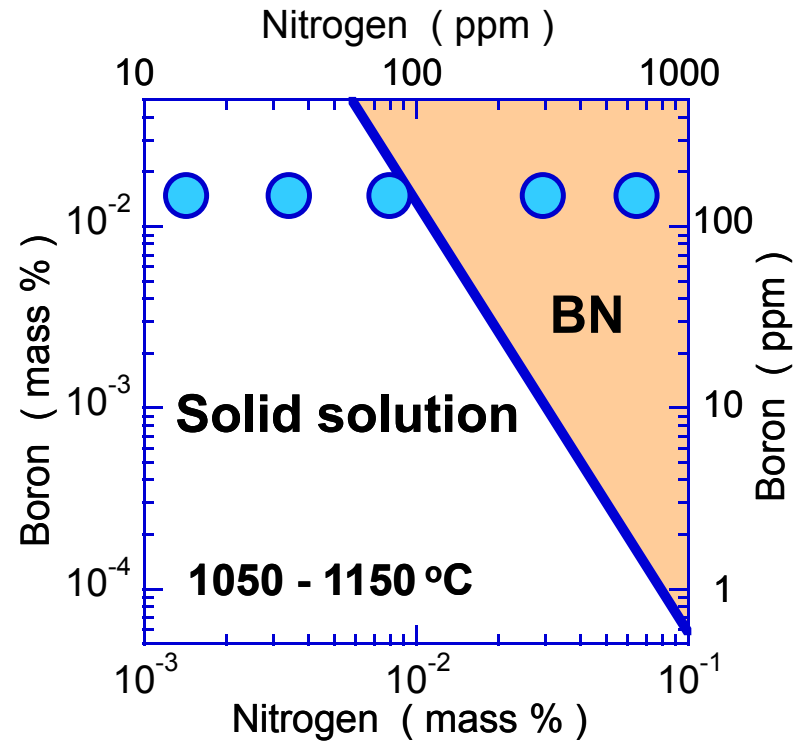
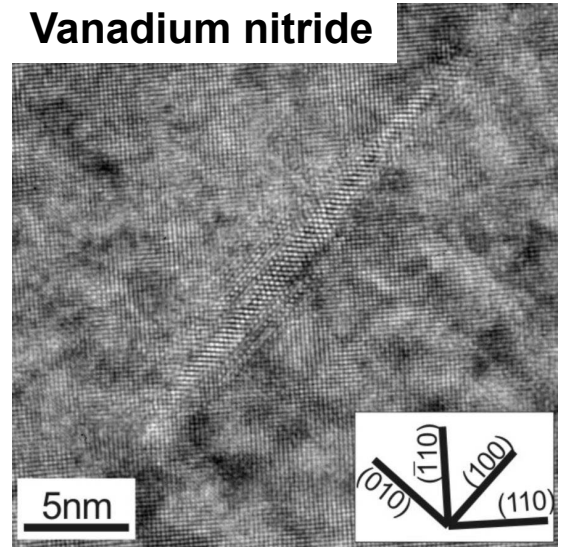
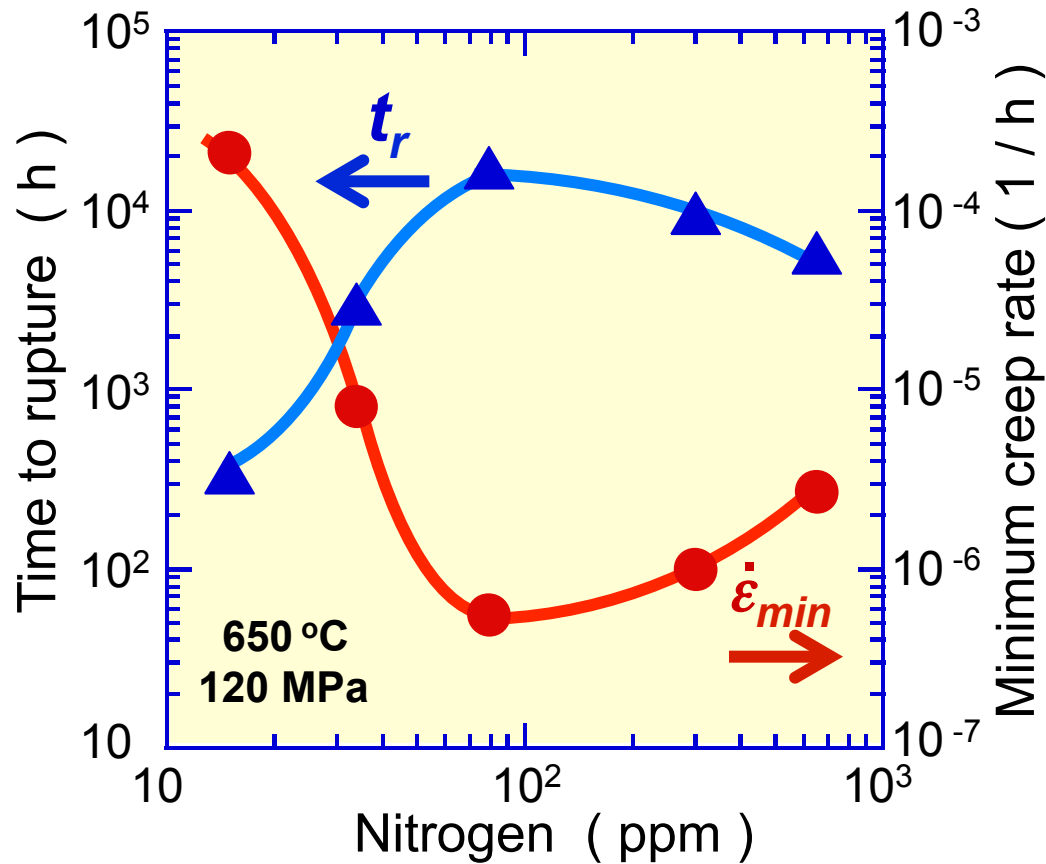


After aging
650 °C, 10,300 h



Additive strengthening due to boron and fine VN

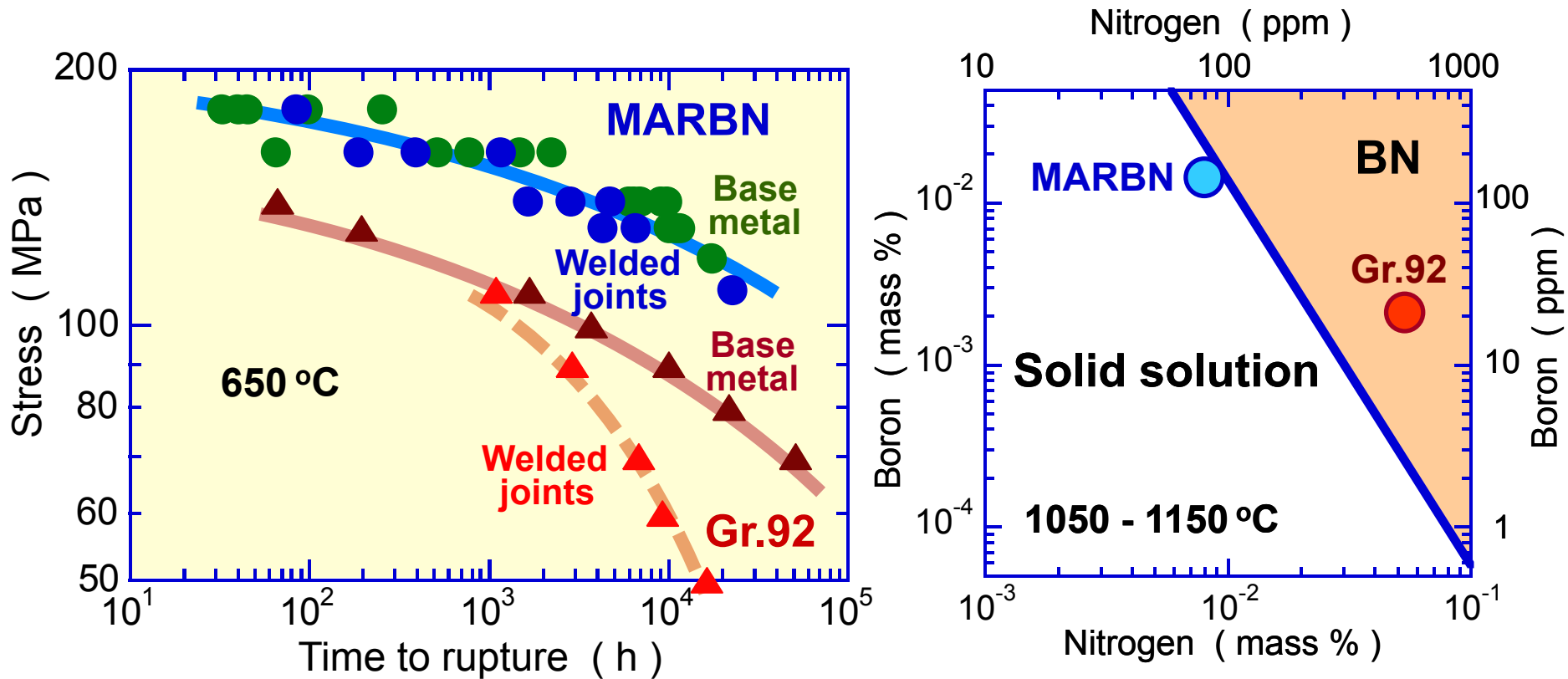
9Cr-3W-3Co-0.2V-0.05Nb steel
140 ppm boron



NIMS 9Cr steel : MARBN

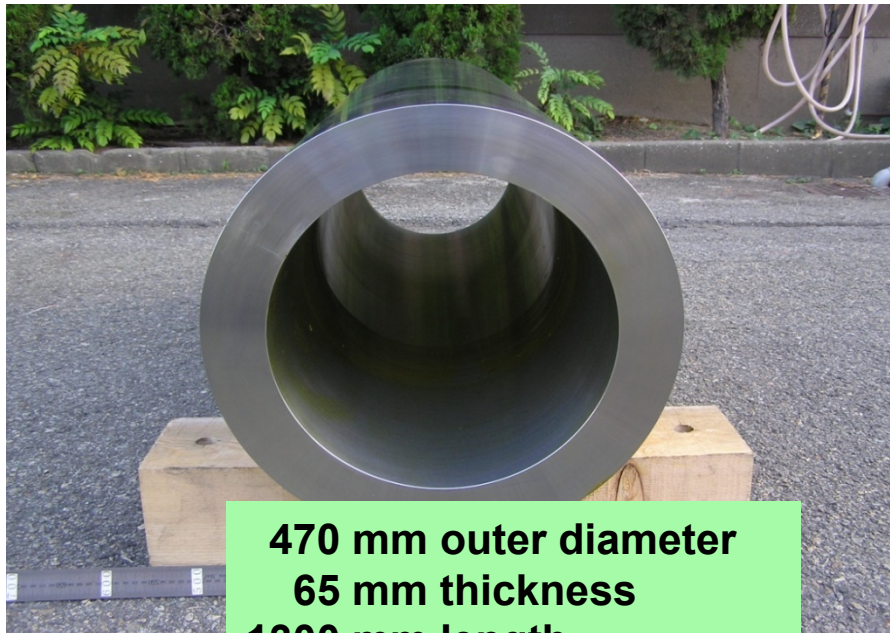
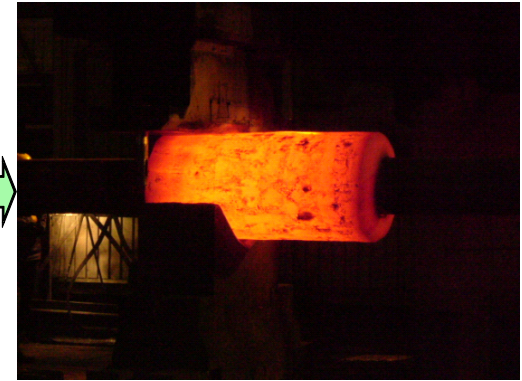
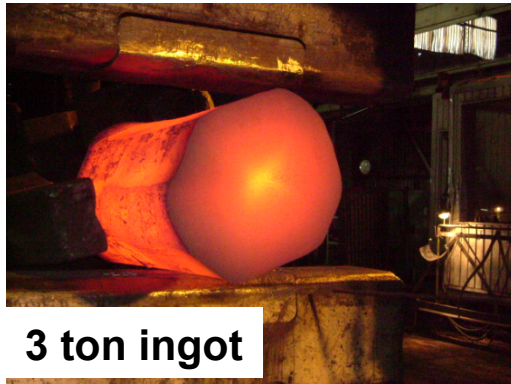
MARBN : MARTensitic 9Cr steel strengthened by Boron and MX Nitrides

120 - 150 ppm boron & 60 - 90 ppm nitrogen



Production of thick-walled MARBN pipe

MARBN (9Cr-2.8W-3Co-0.2V-0.05Nb-0.08C-0.008N-0.014B)



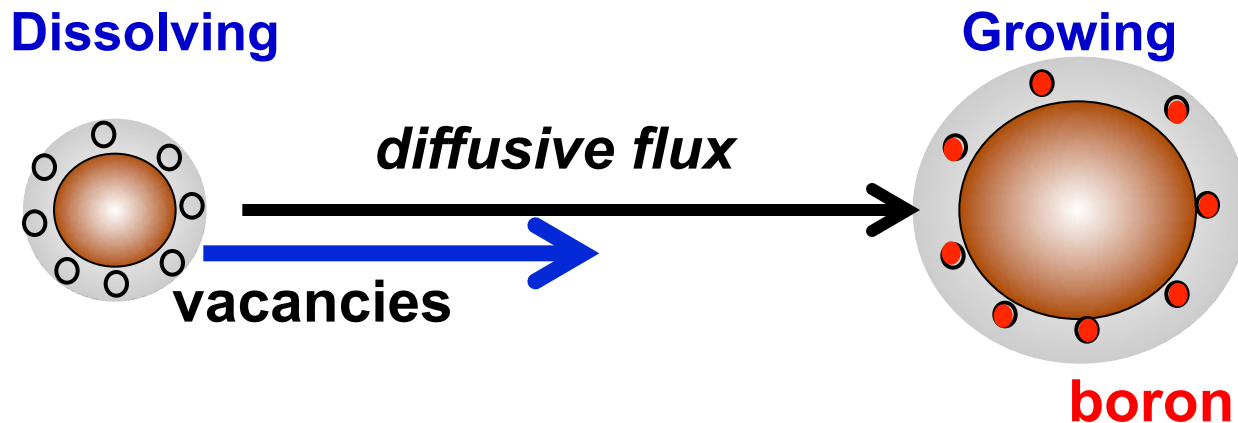
Summary

- (1) The degradation of creep strength in HAZ is not caused by grain refinement but reduction of boundary hardening is the most important.**
- (2) Soluble boron produces substantially same microstructure in HAZ as base metal, which suppresses Type IV cracking.**
- (3) NIMS 9Cr steel (MARBN) exhibits high creep strength and no Type IV cracking at 650 °C.**

Boron reduces coarsening rate of $M_{23}C_6$

Coarsening of carbide particle in solid matrix requires accommodation of local volume change.

If boron atoms occupy vacancies, local volume change cannot be accommodated.

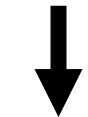


Gr.92NN after A_{C3} thermal cycle

Fine-grains & enough $M_{23}C_6$ along GBs

heating

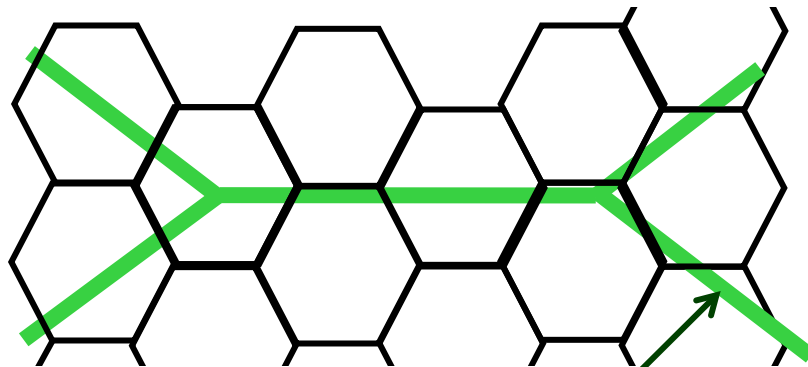
Fine γ grains



A_{C3}



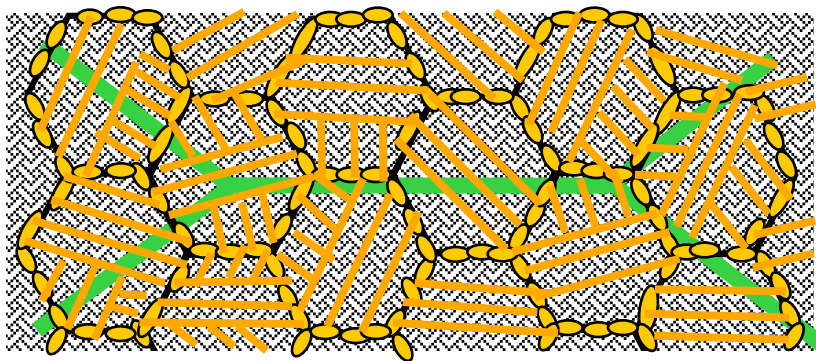
cooling



No $M_{23}C_6$

Trace of GBs in original microstructure

Enough $M_{23}C_6$ at GBs

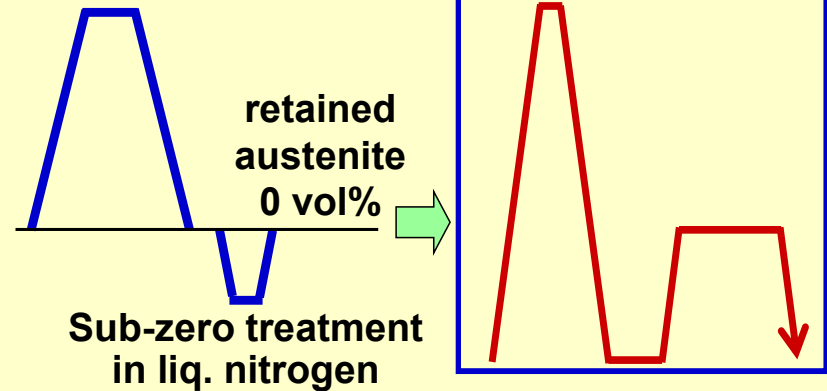


$M_{23}C_6$

after PWHT

Gr.92NN

normalizing
1050°C x 2h



Sub-zero treatment
in liq. nitrogen

No retained austenite

- No austenite memory effect
- Nucleation & growth of γ

No tempering

- PWHT corresponds to low temperature tempering