

Nanostructured Steel Industrialization: a plausible reality

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CENTRO NACIONAL DE INVESTIGACIONES
METALÚRGICAS (CENIM)

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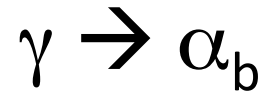


Nanobain

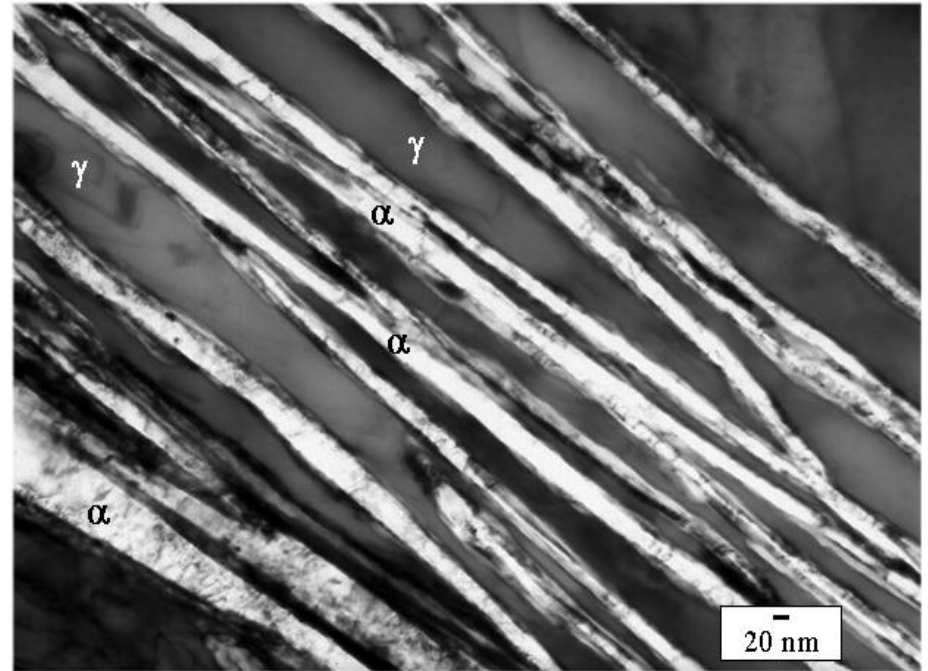
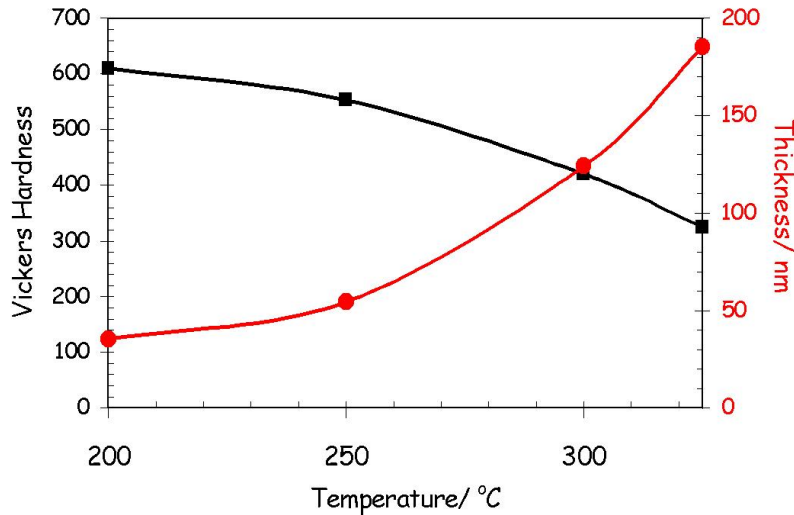
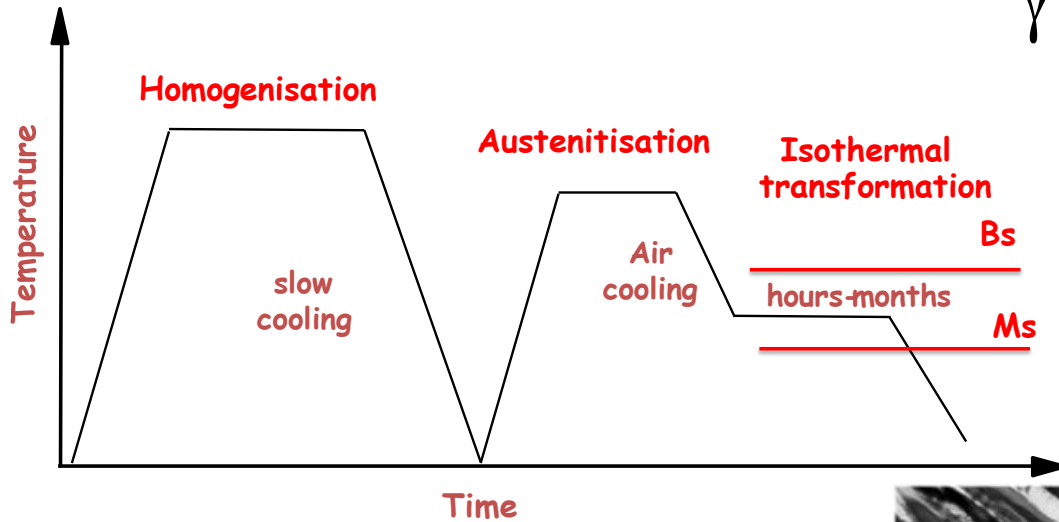
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CENIM-CSIC
ASCOMETAL
ALD
BOSCH
GERDAU
LTU
METSO

Bhadeshia's bainite phase transformation theory



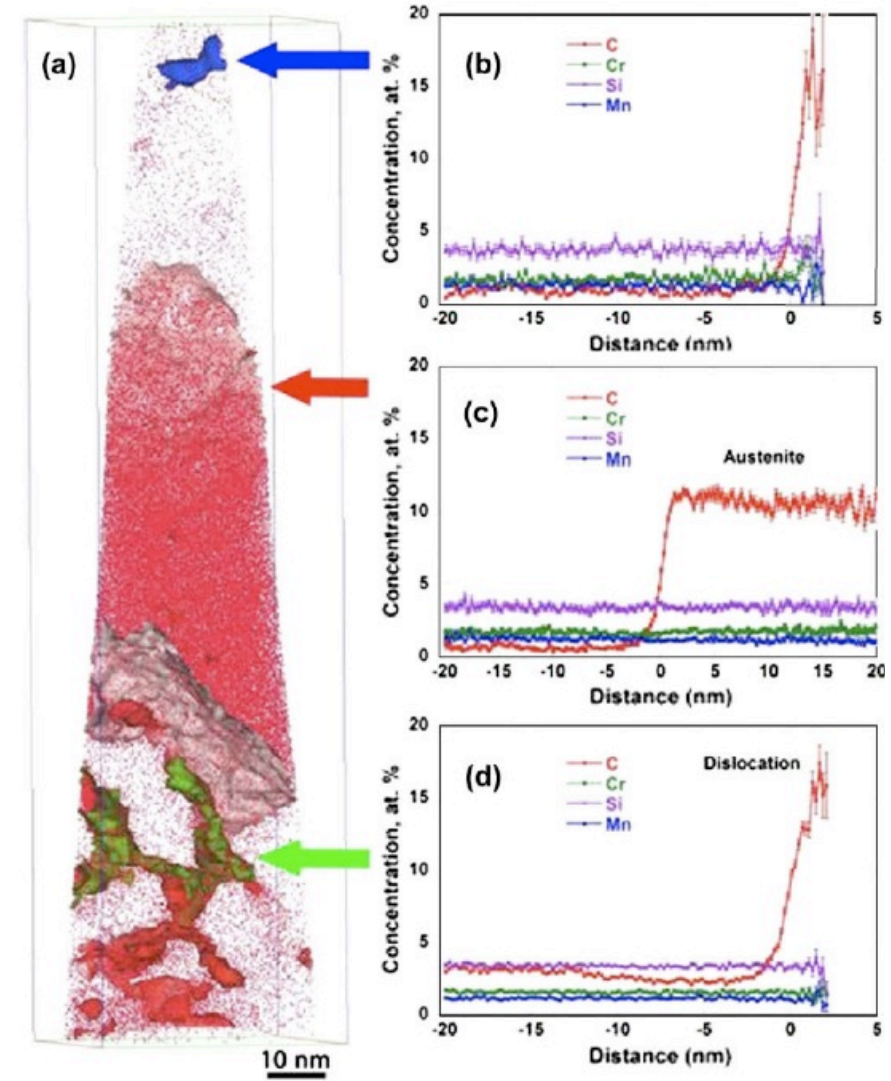
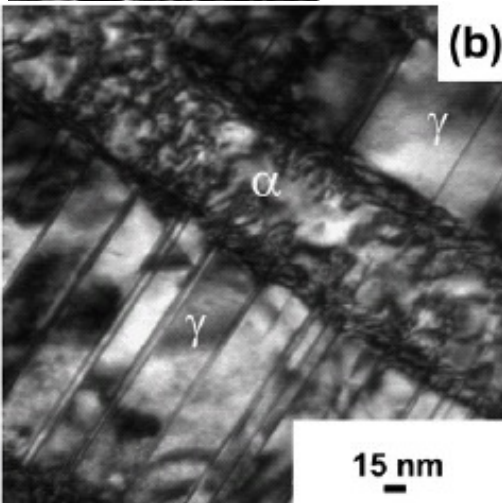
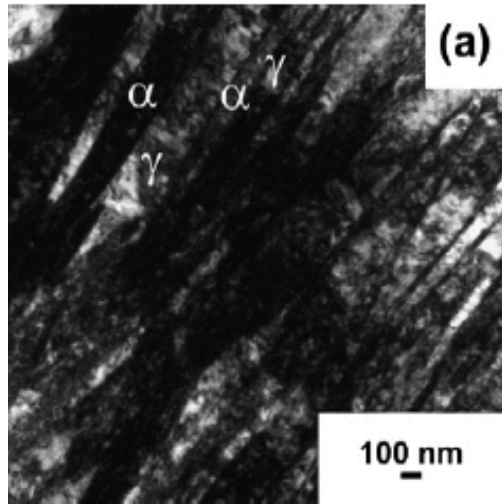
Alloy Design process



200 $^{\circ}\text{C}$ 10d

High density strong interfaces *nano* $\alpha+\gamma$

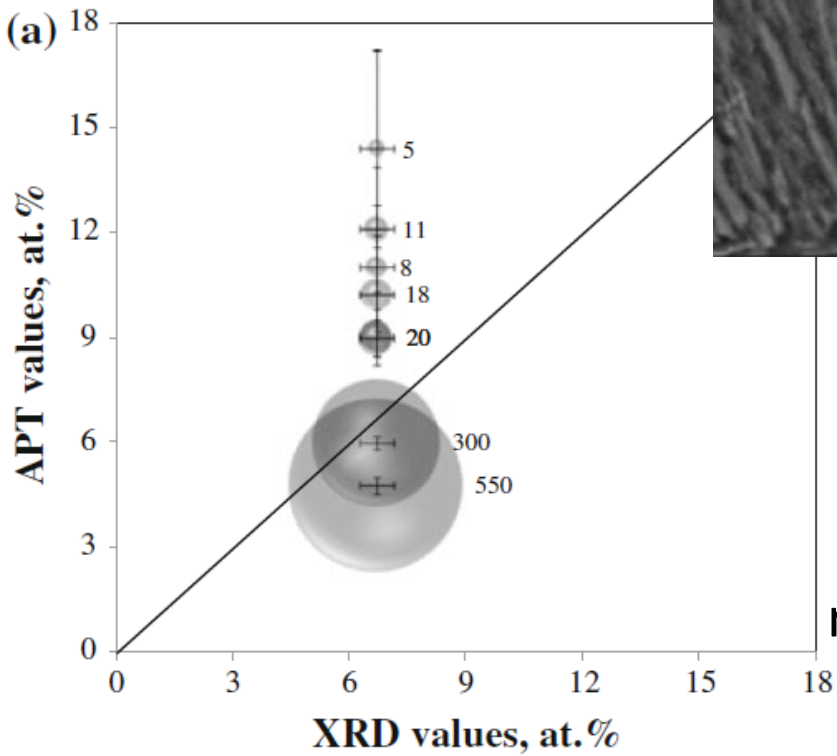
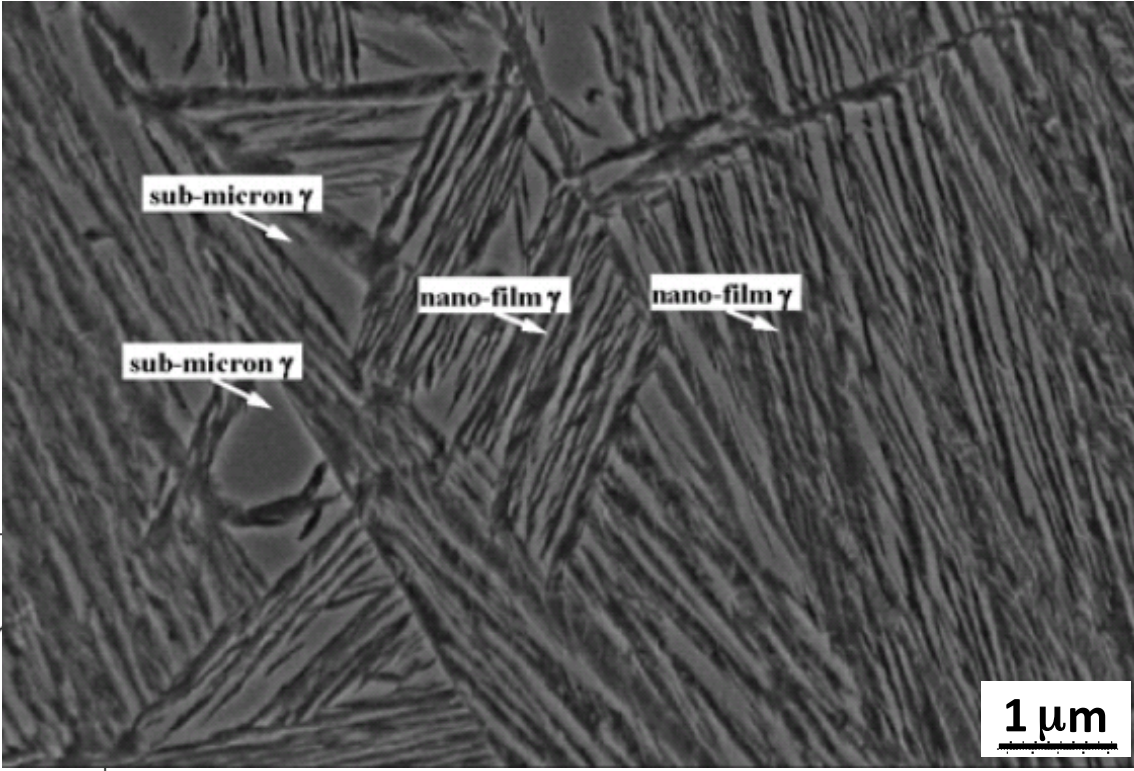
Displacive (*Dislocations, Nano-twins*) & Diffusionless



200 °C for 2 days.

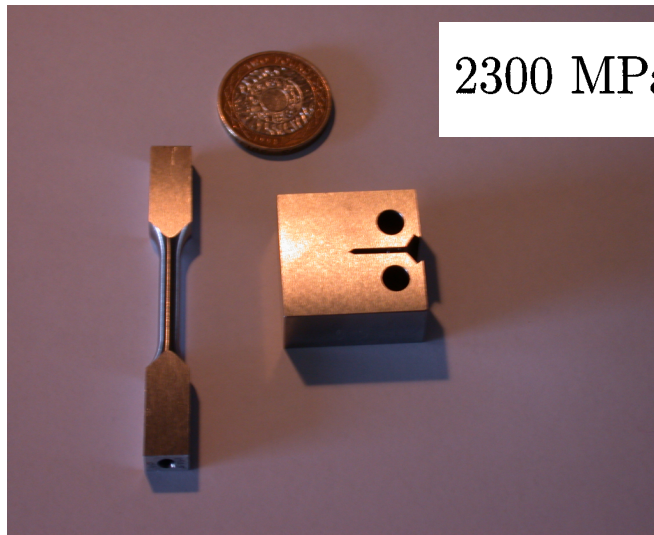
Caballero . Acta 2010 & 2011

Austenite morphologies

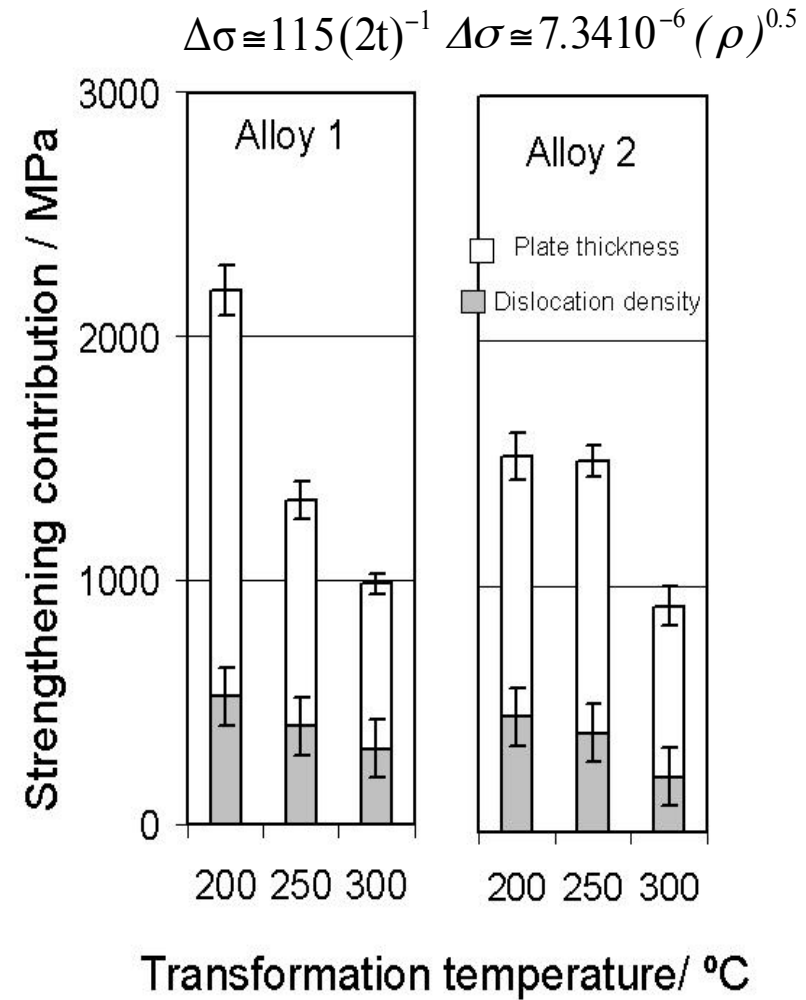
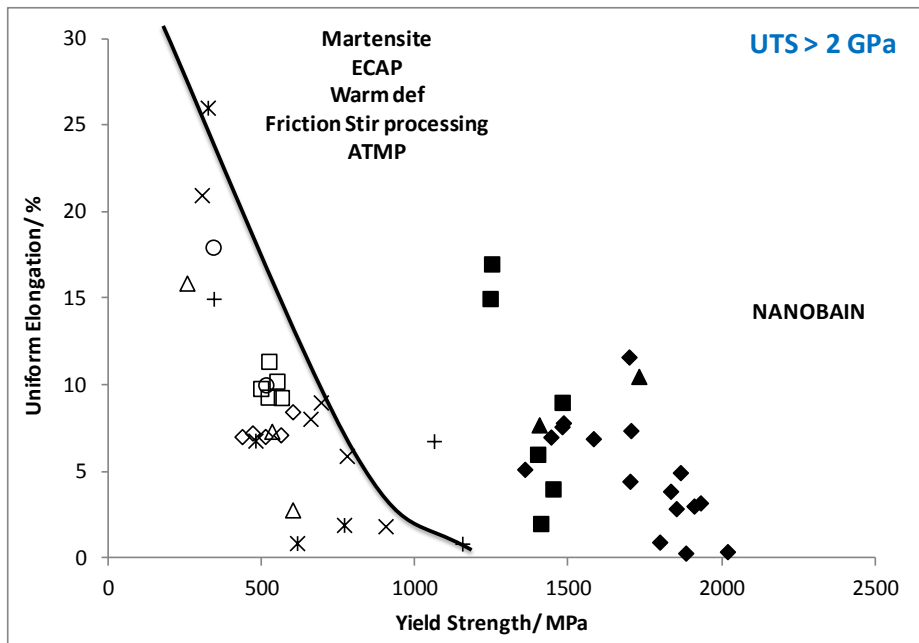


Very different C content
as size ↓ → ↑ C → better mechanical
response

Garcia-Mateo, et.al. *Journal of Materials Science* 2012,
Bhadeshia, et al. *Metal Science* 1983,



2300 MPa, 27 MPa m^{1/2}



Big potential foreseen

R+D



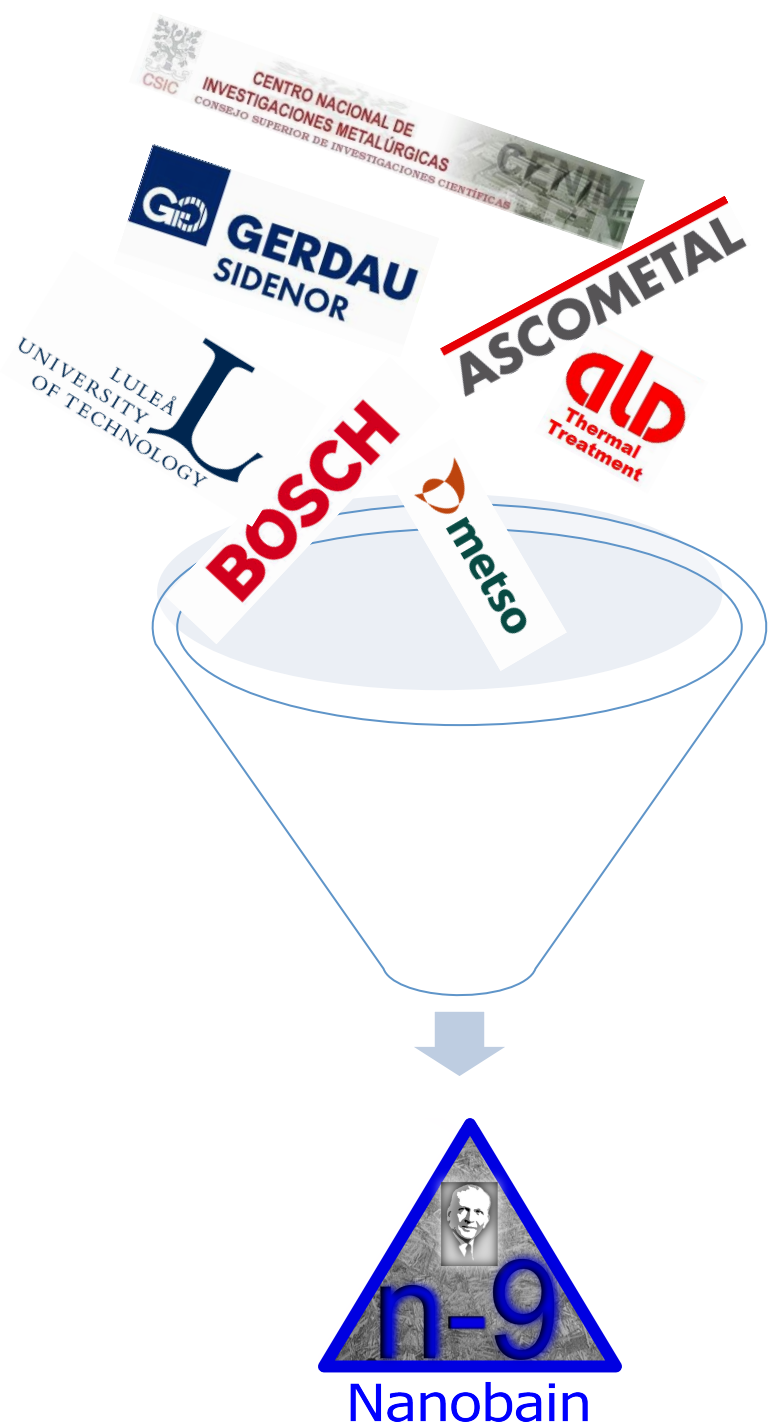
Steel
Makers



Industrialization
of NANOBAIN



Final
User



Fundamental & Industrial Design Considerations

Carbide free microstructure

Simple alloy system

Reasonable transformation time

Hardenability

700 x 400 x 250 mm³ → Salt bath

200 x 20 x 30 mm³ → DryBainTM

High fraction of bainitic ferrite

Low transformation temperatures

Approach

Addition of 1.5Si wt.%

Fe-C-Si-Mn-Cr

Reduction of PAGS

$$\Delta G^{\gamma \rightarrow \alpha}$$

$$\ln \left| \frac{\tau (\Delta F_m)^p}{T^z} \right| = \frac{Q'}{RT} + C_4$$

$$V_\alpha = f(\text{To line}, T)$$

B_s

$$\Delta G_m < G_N$$

$$\Delta G^{\gamma \rightarrow \alpha} < -G_{SB}$$

M_s

$$\Delta G^{\gamma \rightarrow \alpha} \{M_s\} < G_N^{\alpha'}$$

NANO BAIN steel

C	Si	Mn	Cr	Mo	Nb
0.99	1.58	0.76	0.45		
1.00	1.53	0.75	0.51		0.02
1.01	1.51	0.82	0.46	0.096	
0.98	2.90	0.77	0.45		
0.88	1.54	0.69	0.50		
0.67	1.6	1.25	1.50		
0.61	1.45	0.76	2.42		
0.64	1.60	1.27	1.5		0.03
0.58	1.63	1.29	1.43	0.1	

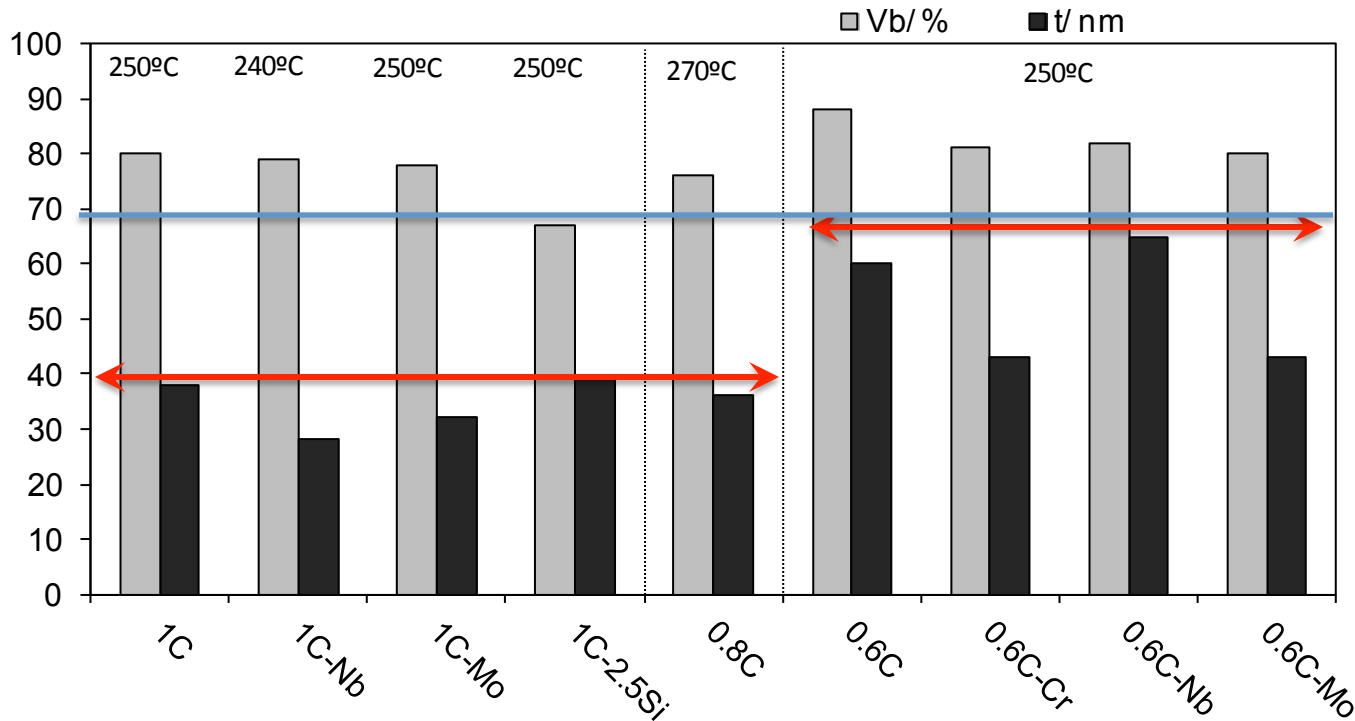
Simple alloy system ✓

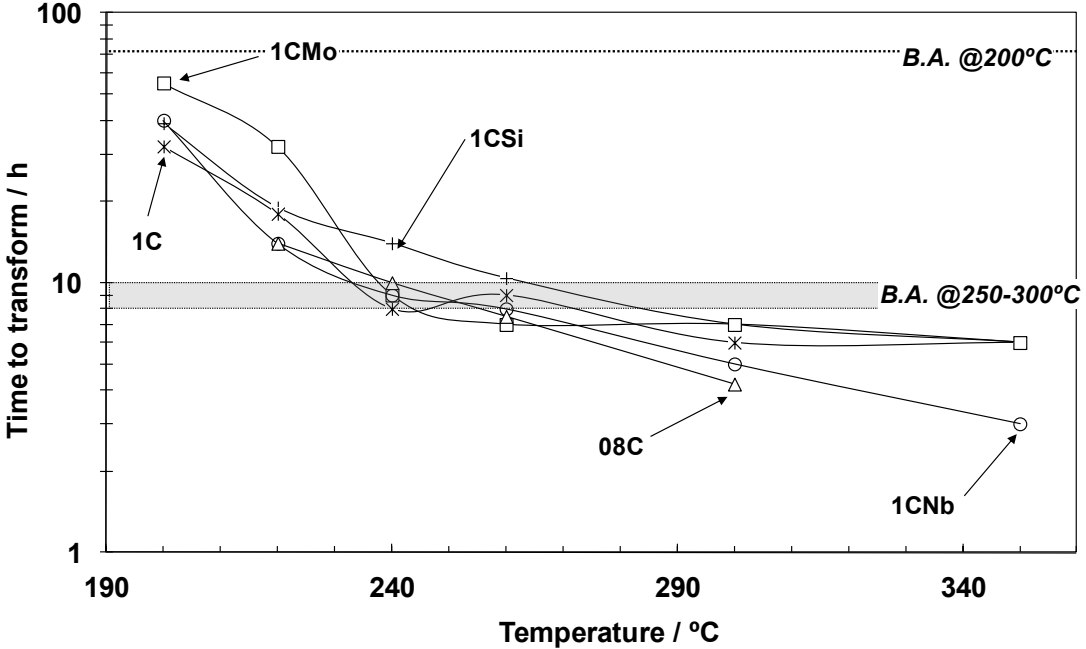
Low transformation temperatures ✓

(220°C-350°C)

NANO (plate thickness t) ✓

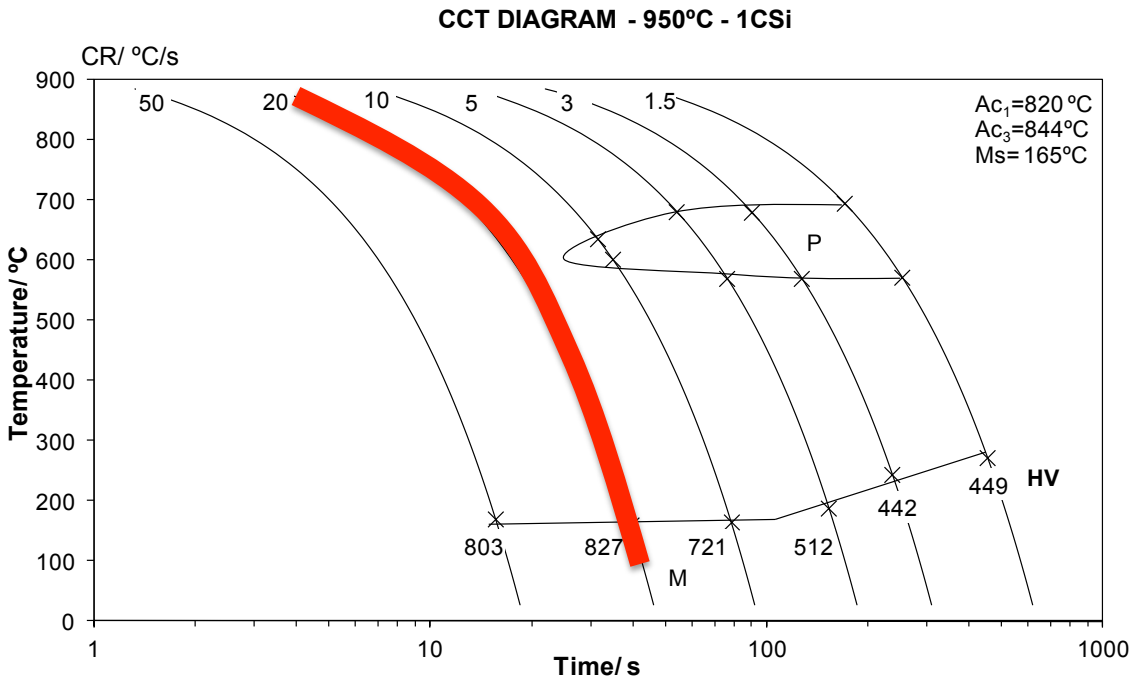
High fraction of bainitic ferrite (V_b) ✓

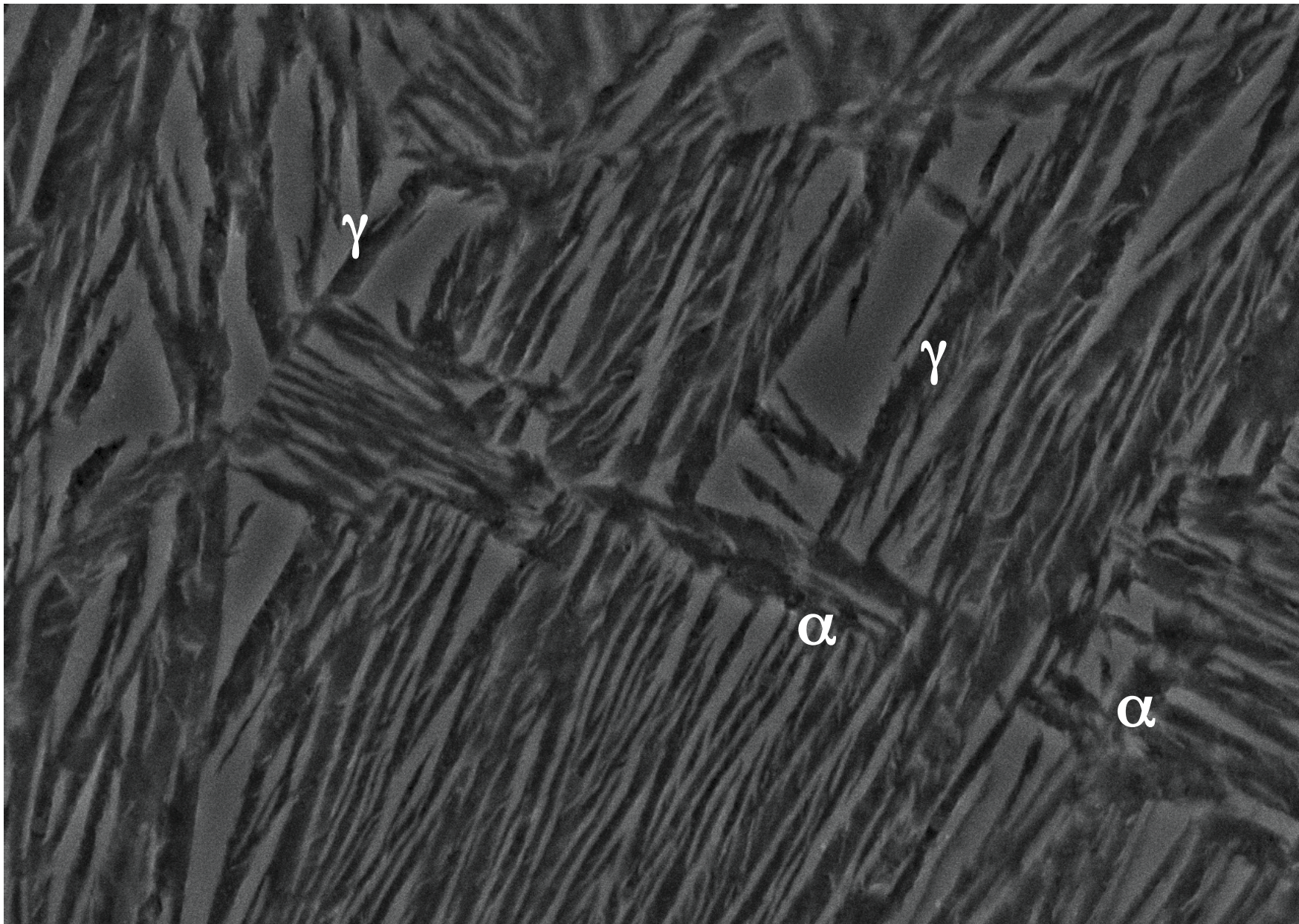




Faster transformation ✓

Suffice hardenability ✓





CENIM

SEI

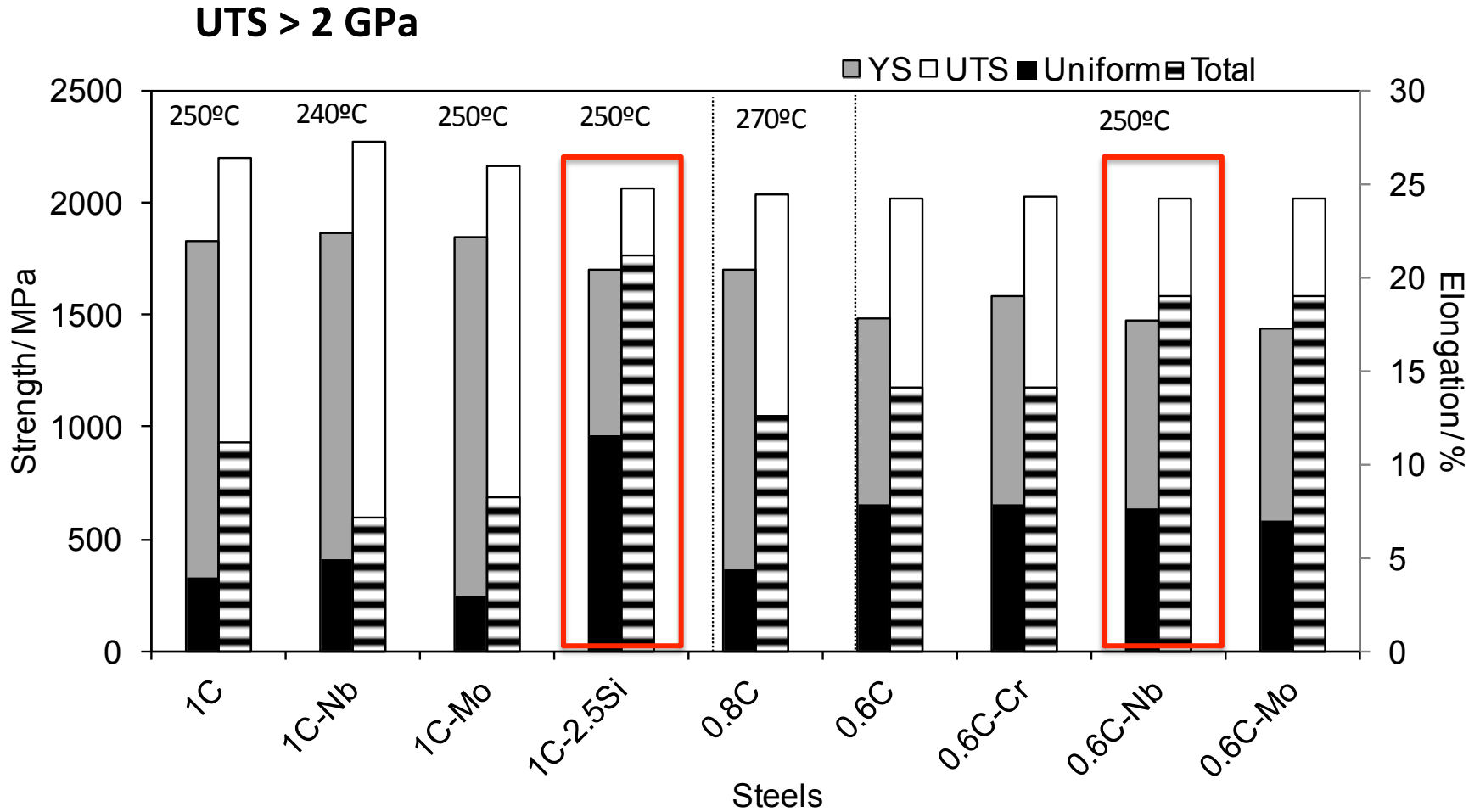
7.0kV

1 μ m

100 nm

1CSi 250°C 16 h

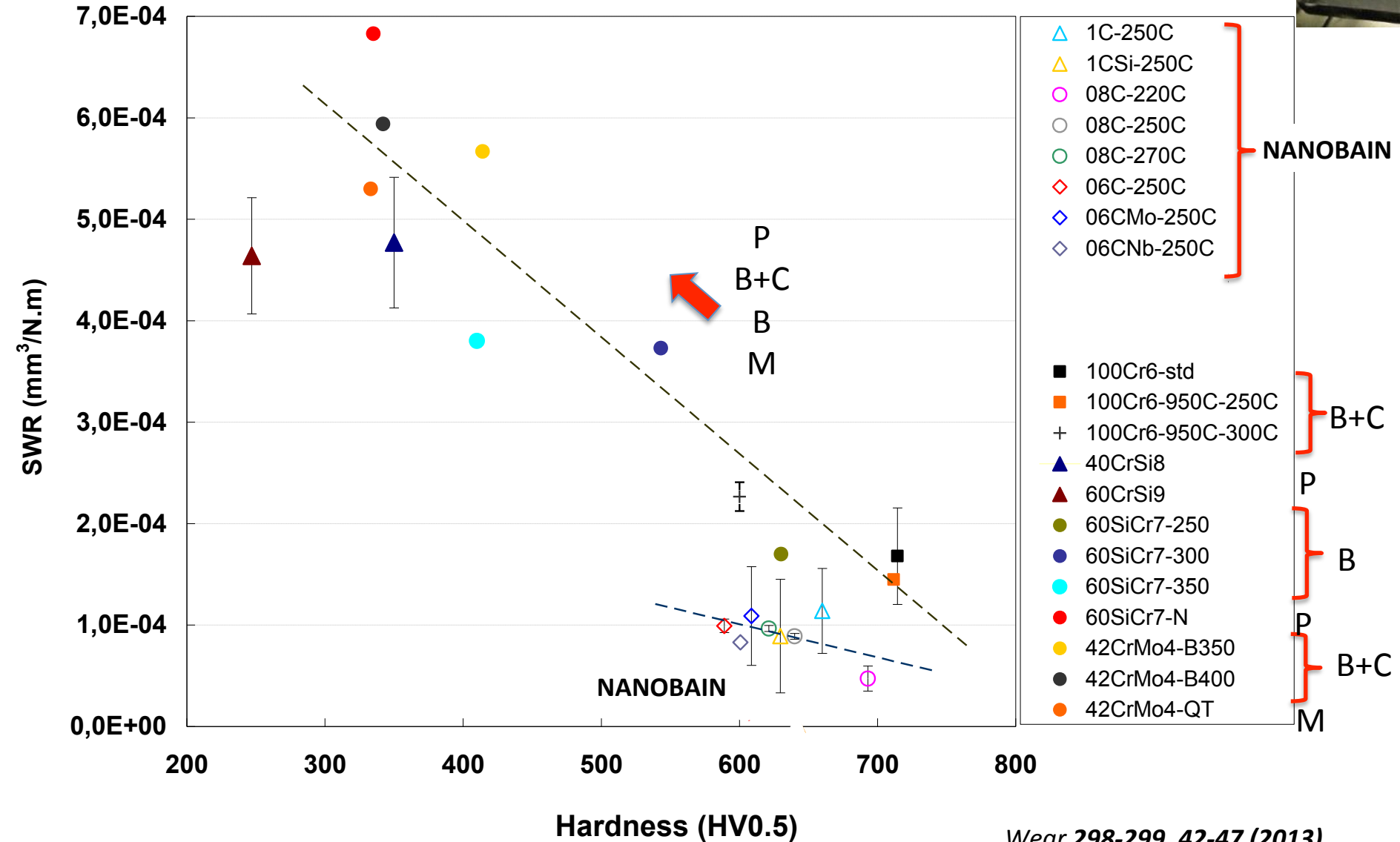
Strength / Ductility (Tensile test)



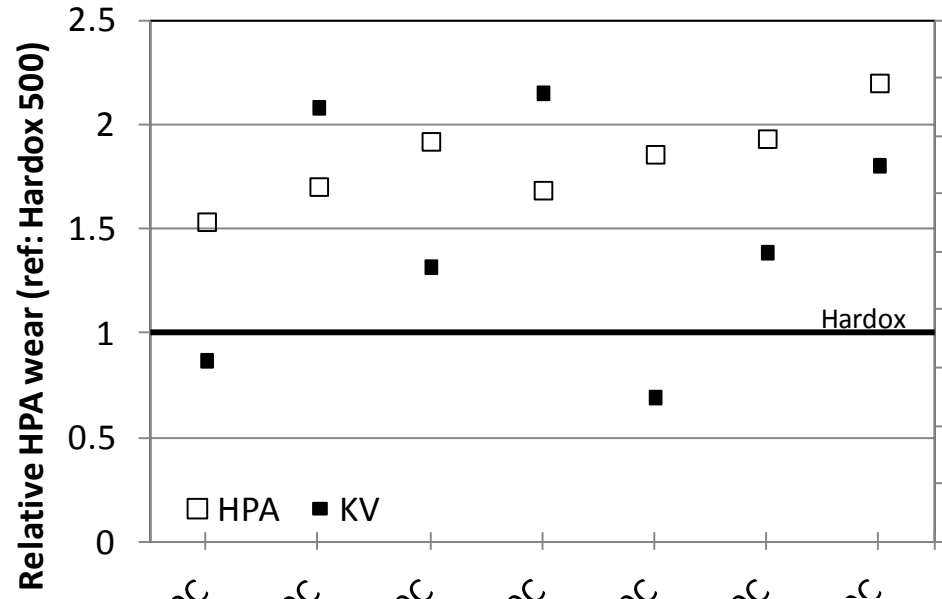
Wear properties : rolling-sliding wear tests



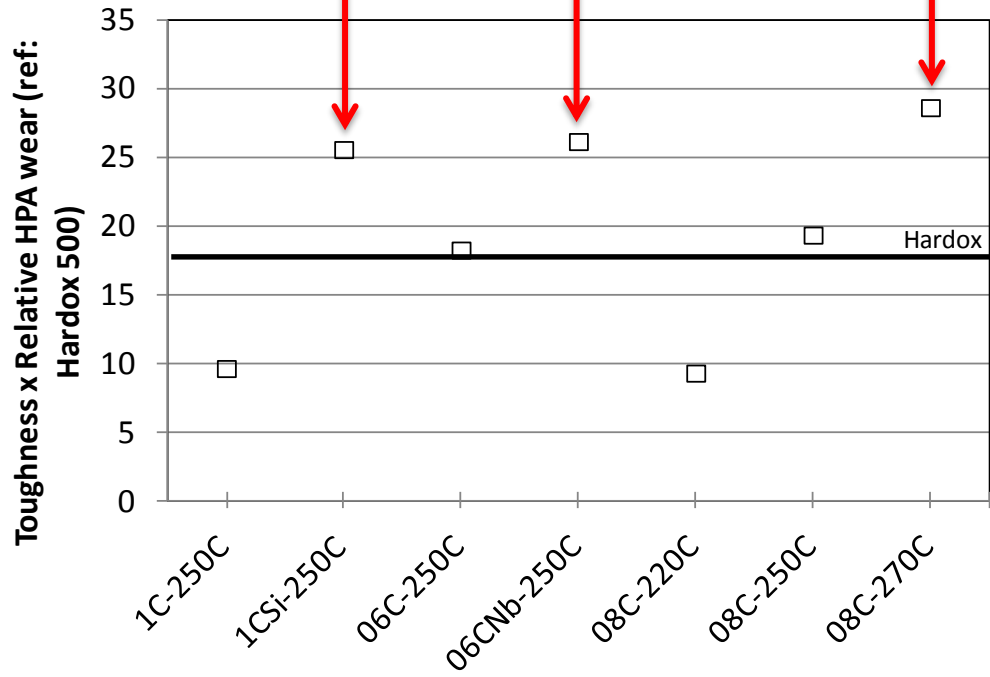
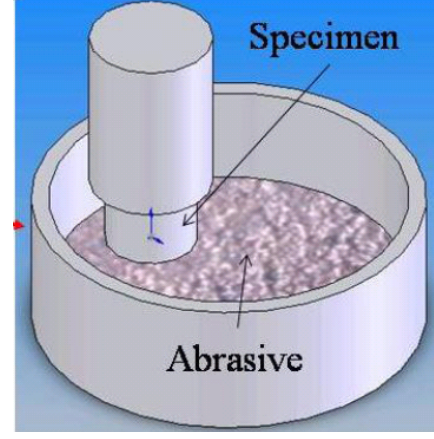
Twin discs wear test, 100 rpm, 5% slip, 300N load, 5h



Wear properties : high pressure abrasion



200N
 $\Delta t = 2.5s$
 $t = 30 \text{ min}$



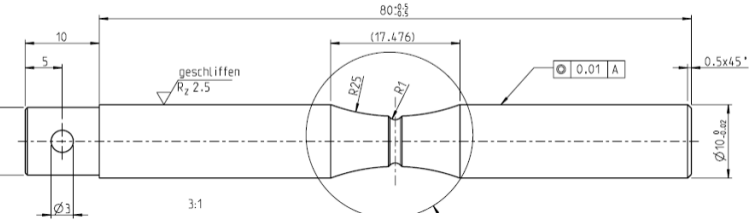
Hardox (SSA)

0.3C-0.7Si-1.6Mn-1.5Cr-1.5I

HV = 500 ; Tempered Marten

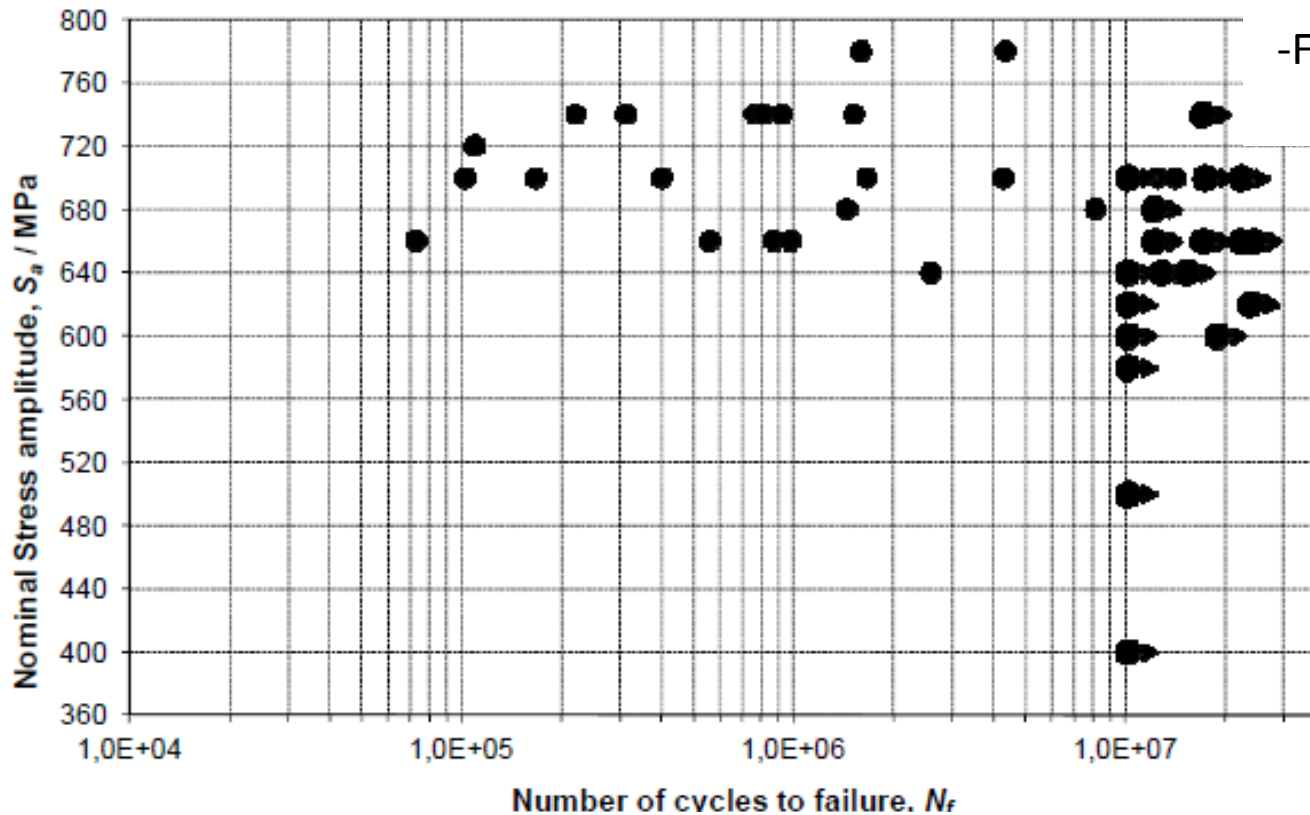
Fatigue properties : rotation bending on notched specimens.

Condition	UTS / MPa	Sa, 50% at 10^7 cycles / MPa for R = -1, Kt = 1,6 by rotation bending
1CNb-220-FAT	2073	430
06C-250-FAT	2023	665
08C-270-FAT	2036	440



Notch ?

-Application have stress concentrator



-Fatigue sensitive to austenite decomposition

0.6C @250°C

(Industrial) Material selection

Strength ductility balance → small component → 1CSi

HPA+Charpy performance → big component (hardenability) → 0.6 CNb

0.6 CNb ⇔ 0.6 CV

= microstructure, HV & Bs/ Ms + faster kinetics

	C	Mn	Si	Cr	Mo	V
1CSi-ind	1.00	0.75	2.50	1.00	0.03	0.00
0.6CV-ind	0.60	1.25	1.60	1.75	0.15	0.12

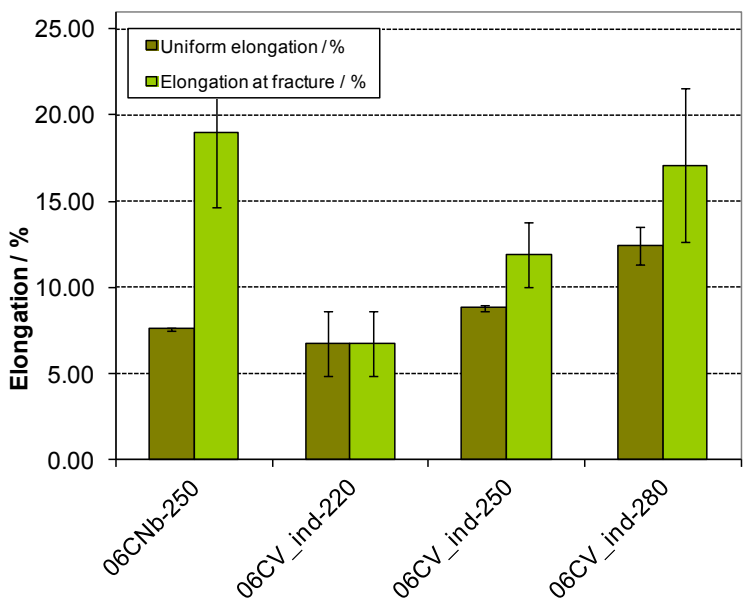
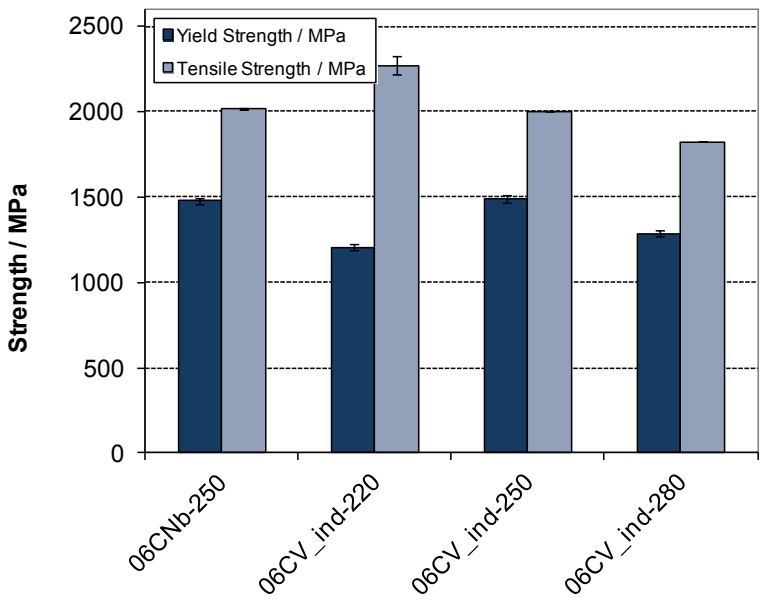
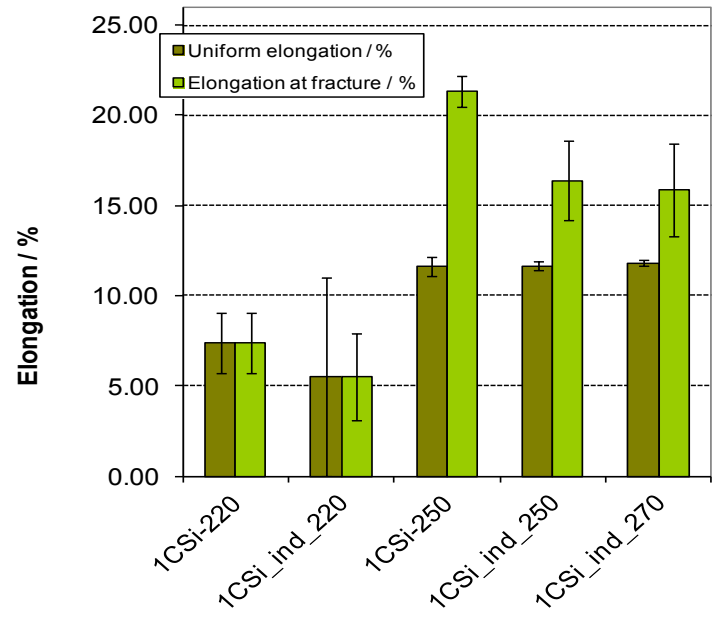
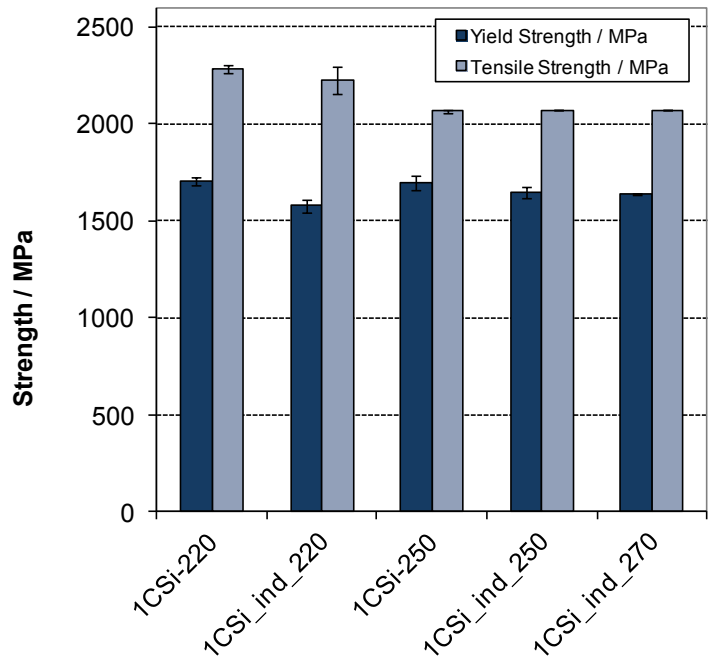
Industrial heats casted

Before fabricating a component or demonstrator

Battery of tests

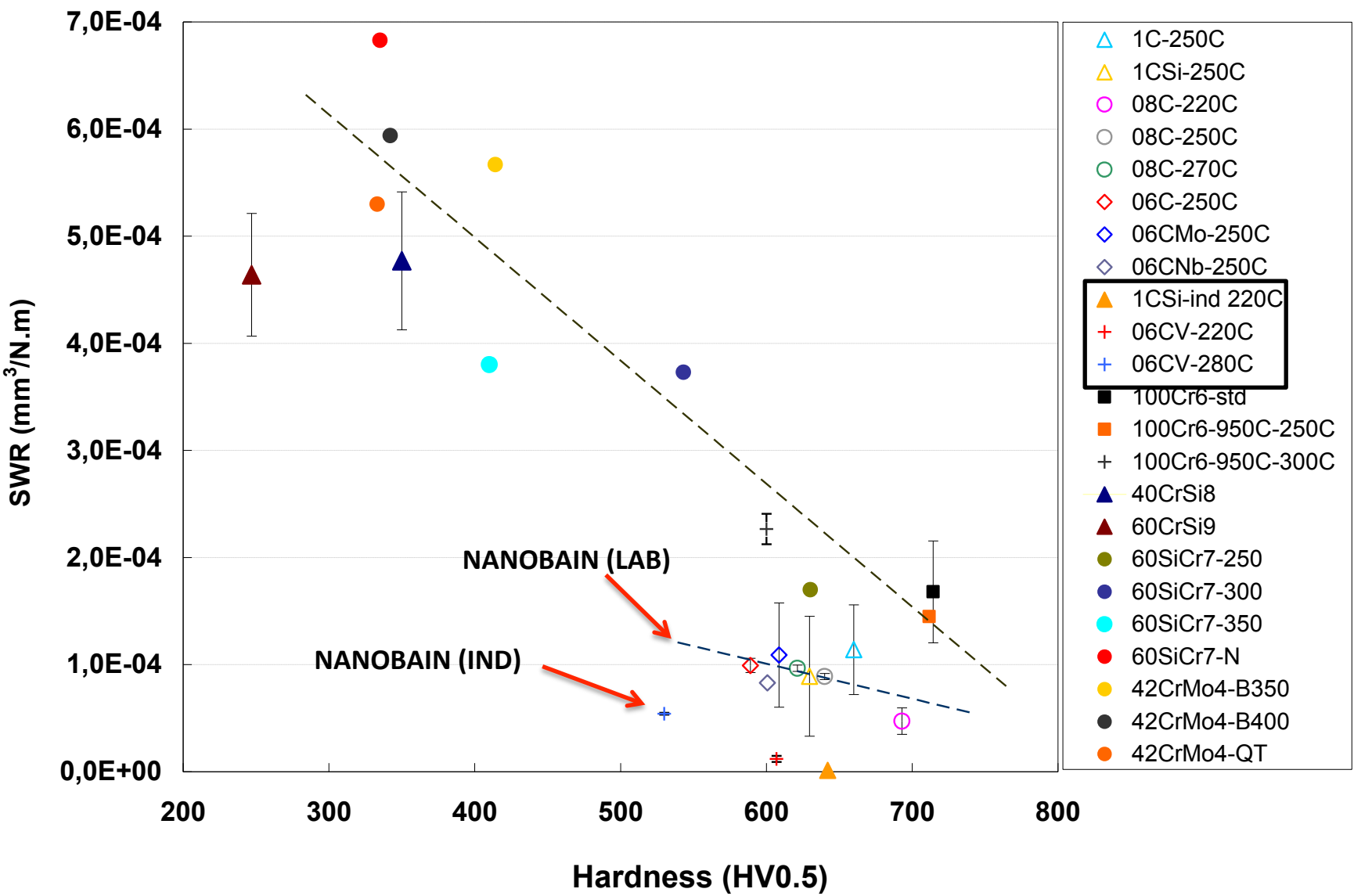


Tensile Industrial Material

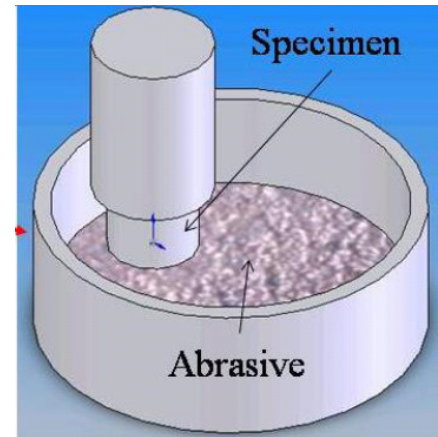
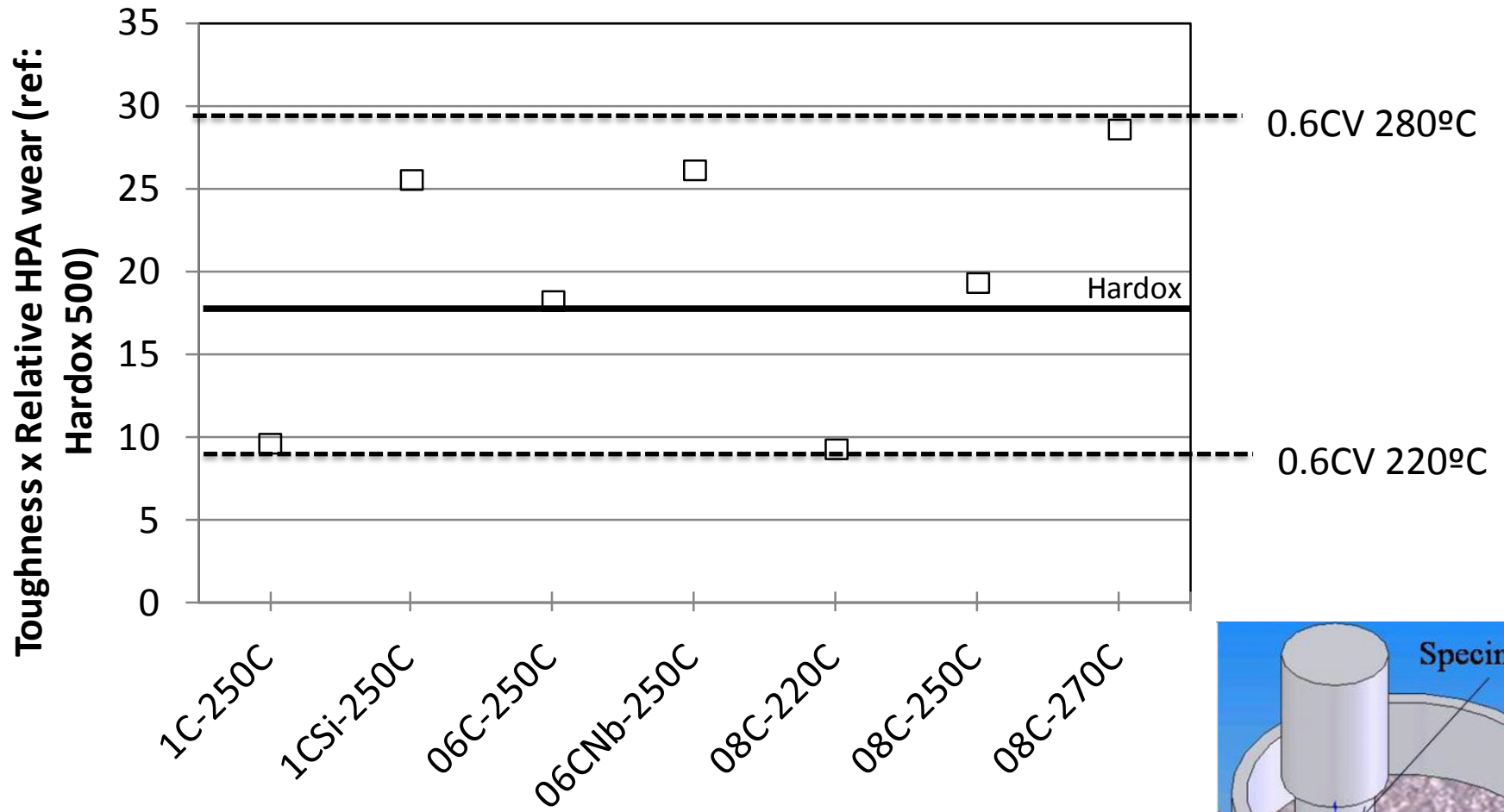


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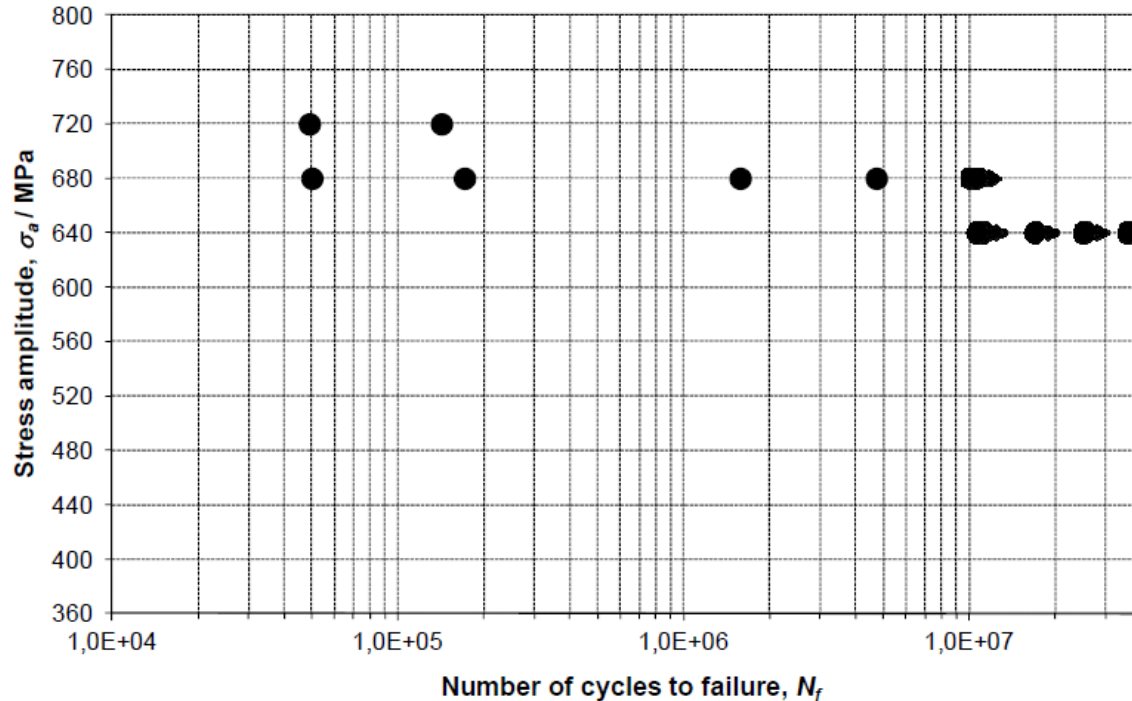
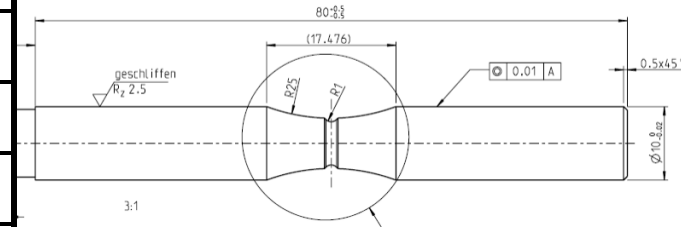


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Fatigue properties : rotation bending

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1CNb-220-FAT	2073	430
06C-250-FAT	2023	665
08C-270-FAT	2036	440
1CSi-ind-220-FAT	2224	550
1CSi-ind-250-FAT	2072	535
06CV-ind-250-FAT	2003	690
06CV-ind-270-FAT	1822	675

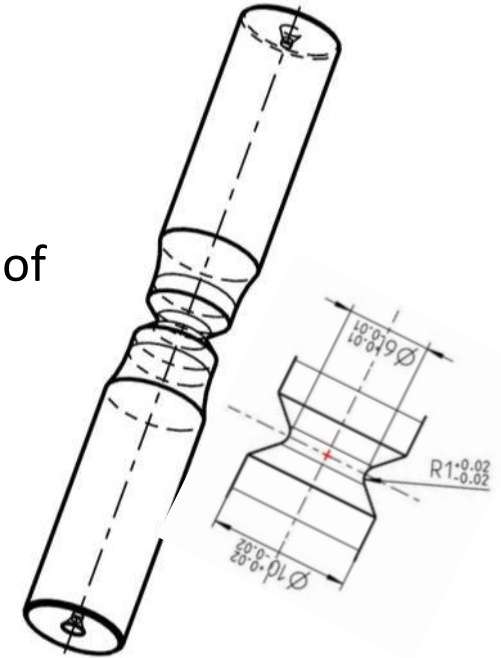


0.6CV @270°C

Fatigue properties : notched tension-tension specimens



Representative of the functioning of heavily loaded diesel engines



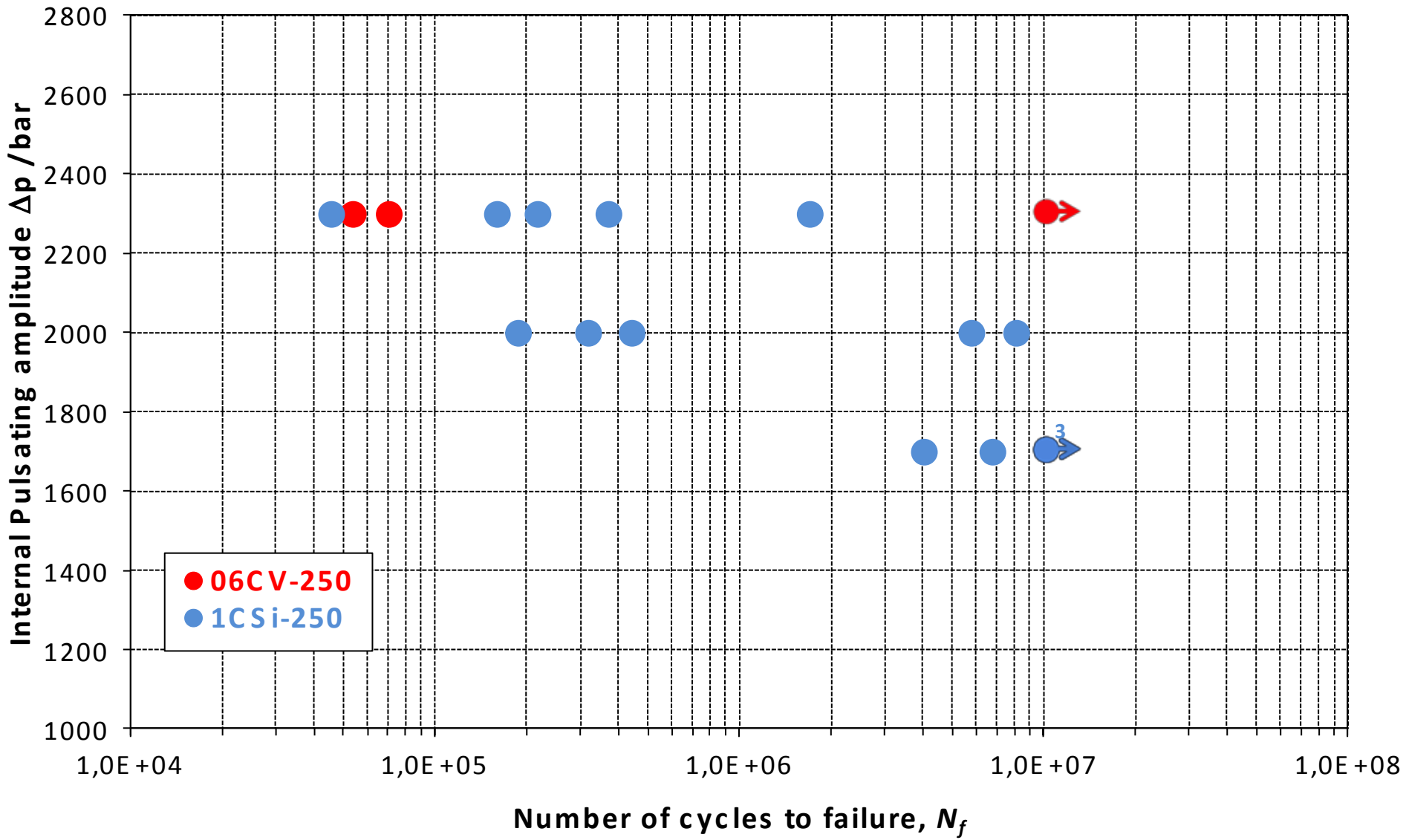
$$K_t = 2$$

$$R \text{ (stress ratio)} = 0.1$$

150Hz

	06CV-250	06CV-270	1CSi-220	1CSi-250	100Cr6 [TLi, 1998]	50CrMo4 [SSc, 2012]
UTS / MPa	2003	1822	2224	2072	2350	1180
$S_{a,50\%}$ at $N = 10^7$ cycles / MPa	350	330	205	245	355	305

Small component High pressure injection (demonstrator)



Big component Scrap shear blades (component)



0.6CV @ 280°C



524x80x200 mm³

	Surface		Core	
	Side	Middle	Side	Middle
Surface	545	535	542	541
Mid-thickness	541	535	542	543

30 kg load Vickers hardness for METSO's component. The hardness did not vary significantly with position.

Estimated potential gain 20%
due to the significantly cheaper material

Conclusions

- Design of 'cheap' steel grades adapted to low temperature bainitizing and application hardenability .
- Tensile properties : achieved excellent combinations of high strength and ductility. Unprecedented and unexpected 21% TE at UTS 2GPa.
- Fatigue tests, on par with 100Cr6 for one of the industrial grades, scope for improvement
- Large improvement in RS wear resistance, related to retained austenite decomposition



Achievement

In just over three years, the Nanobain project took the concept of nanostructured bainitic steels from a laboratory experiment to full scale industrial production and testing.

Solely by means of phase transformation theory.

$$\Delta G_m < G_N$$
$$\Delta G^{\gamma \rightarrow \alpha} < -G_{SB}$$

